

# Grain boundary diffusion cannot explain the W isotope heterogeneities of the deep mantle

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The low  $^{182}\text{W}/^{184}\text{W}$  and high  $^3\text{He}/^4\text{He}$  in some ocean island basalts compared to the bulk mantle values may derive from the Earth's core through long-term core-mantle interactions. It has been proposed that the grain boundary diffusion of siderophile elements is an efficient mechanism for core-mantle interaction and may effectively modify the W isotopic compositions of the plume-source mantle. In this study, we perform large-scale molecular dynamics simulations driven by machine learning potentials of ab initio quality to investigate the diffusion of W along ferropericlase grain boundaries and in (Mg,Fe)O liquid. Here we show that the diffusion of W is sluggish under core-mantle boundary conditions, and thus is unlikely to have observable impacts on the W isotopic compositions of terrestrial igneous rocks.

Ocean island basalts (OIBs) are volcanic rocks that may originate from rising plumes in the Earth's deep mantle, potentially sampling the materials near the core-mantle boundary (CMB)<sup>1</sup>. The isotopic analyses of OIBs reveal significant chemical heterogeneities in the Earth's deep interior that have long provided key insights into the evolution of our planet<sup>1</sup>. Recent studies have identified intriguing  $^{182}\text{W}$  deficits in some OIBs with  $\mu^{182}\text{W}$  (defined as the deviation of  $^{182}\text{W}/^{184}\text{W}$  from terrestrial standards in parts per million) as low as  $-23 \pm 4.5^{2-6}$ , while  $\mu^{182}\text{W}$  is around 0 in the modern upper mantle. The negative  $\mu^{182}\text{W}$  anomalies were produced by the depletion of the now extinct  $^{182}\text{Hf}$  with a half-life of 8.9 Myr<sup>7</sup>.

Various mechanisms have been proposed to account for the  $\mu^{182}\text{W}$  anomalies in OIBs. The element fractionation during the crystallization of putative magma oceans may lead to the formation of enriched deep mantle reservoirs or early protocrust with a low Hf/W ratio<sup>2,8,9</sup>, subsequently causing lower  $\mu^{182}\text{W}$ . However, subducted, early-formed crust may not possess sufficiently negative  $\mu^{182}\text{W}$  to account for OIBs<sup>5</sup>. In addition, the enriched deep mantle reservoirs and the protocrust would also be enriched in other incompatible elements, including U and Th. This enrichment leads to more radiogenic  $^4\text{He}$  and lower  $^3\text{He}/^4\text{He}$  ratio ( $^3\text{He}$  is primordial), which is inconsistent with the observed negative correlation between  $\mu^{182}\text{W}$  and  $^3\text{He}/^4\text{He}$ <sup>2,5</sup>.

As a moderately siderophile element, W primarily partitions into the core during core-mantle differentiation, while Hf remains in the

mantle due to its lithophile nature. The early core formation during the first ~30 Myr of solar system history, when  $^{182}\text{Hf}$  was still extant, resulted in the terrestrial mantle exhibiting a higher  $^{182}\text{W}/^{184}\text{W}$  ratio compared to bulk chondrites and the complementary reservoir, the core<sup>10-12</sup>. Therefore, the assimilation of the Earth's core materials or chondritic materials after core formation may significantly modify the  $\mu^{182}\text{W}$  characteristics of OIB source reservoirs. For example, it has been proposed that iron-rich liquids from the core may penetrate into the oxides in the lowermost mantle due to morphological instability<sup>13</sup> or capillarity phenomenon<sup>14</sup>. Since the core is believed to host a substantial amount of  $^3\text{He}$ <sup>15,16</sup>, it may also be a source of high  $^3\text{He}/^4\text{He}$  in OIBs<sup>17-20</sup>, simultaneously explaining the  $^{182}\text{W}$  depletion and  $^3\text{He}$  enrichment. However, the direct incorporation of outer core metal or late-accreted materials could lead to significant enrichment of highly siderophile elements (HSEs), which contradicts the observed absence of HSEs enrichments of OIB samples<sup>4,5</sup>. In addition, the extreme lithophile behavior of Ne under high pressures predicted by ab initio calculations likely results in a high  $^3\text{He}/^{22}\text{Ne}$  ratio of core materials, which is not observed in OIBs either<sup>21</sup>.

It is suggested that atomic diffusion of some siderophile elements along grain boundaries (GBs) may be an efficient mechanism for core-mantle interaction<sup>22</sup>. Recent experimental results also indicate high GB diffusivities of W in lower mantle phases with a strong temperature dependence<sup>23</sup>. Therefore, diffusive isotopic exchange at the CMB has

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been considered a viable mechanism to explain the  $^{182}\text{W}$  deficits in OIBs<sup>4,5,18,23</sup>. However, those experiments have puzzling aspects: First, W and Au exhibit diffusivity three orders of magnitude higher than other siderophile elements, despite having similar atomic masses, radii, and chemical properties<sup>22</sup> (Fig. 1). Second, extrapolating the W diffusivity to the CMB temperatures according to the experimentally determined temperature dependence yields unreasonably high results<sup>23</sup> (see Results and Discussion). Recently, a theoretical prediction suggests that the GB transport in MgO is considerably less efficient at CMB conditions than has been extrapolated from experimental results at lower pressures and temperatures<sup>24</sup>. Our recent findings also favor sluggish Fe diffusion along ferropericlase GBs, questioning the feasibility of GB diffusion as an effective core-mantle interaction mechanism<sup>25</sup>. Therefore, it is important to further quantify the efficiency of GB diffusion in exchanging W isotopes, especially at the temperature and pressure conditions of the CMB.

Performing diffusion experiments under high temperature and pressure is challenging<sup>26</sup>, and obtaining the GB diffusivity requires separating the contributions of lattice and GB diffusion to the bulk diffusion profiles (e.g., Yoshino et al.<sup>23</sup>), introducing additional errors. In this study, we examine the self-diffusion of W in ferropericlase GBs and (Mg,Fe)O liquid using molecular dynamics (MD) simulations under various pressure and temperature conditions. The recently developed machine learning potential (MLP) method<sup>27,28</sup> enables the simulations of large GB systems with high efficiency while maintaining ab initio accuracy. Considering the isotopic exchange process via GB diffusion, we evaluate the  $\mu^{182}\text{W}$  anomalies of the plume-source mantle that might have arisen over the past 4.54 billion years. Our findings imply that GB diffusion across the CMB cannot explain W isotope anomalies in the plume-source mantle, and other mechanisms are desired to resolve the puzzle of  $\mu^{182}\text{W}$  anomalies in mantle-derived rocks.

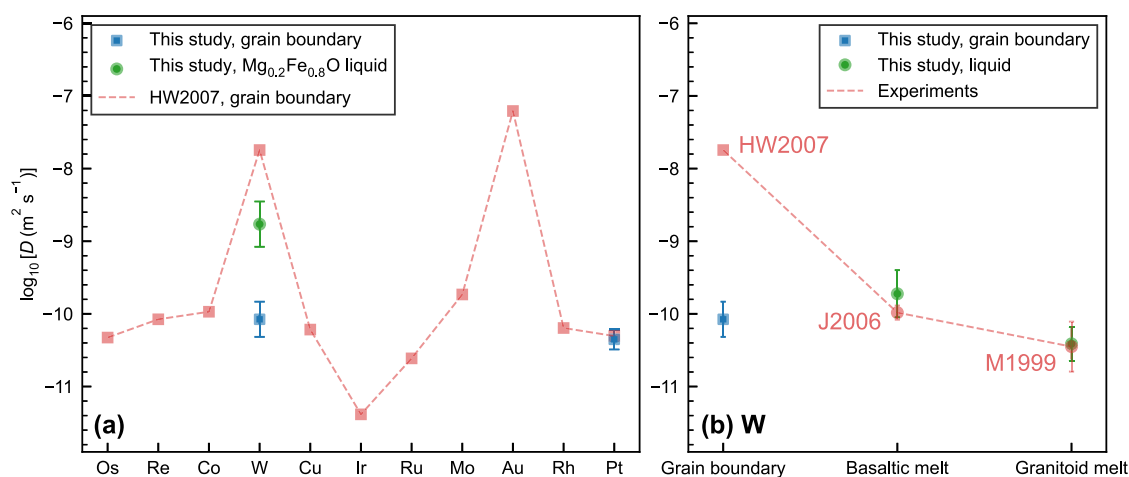
## Results and Discussion

### Grain boundary diffusion of W and Pt in ferropericlase

We developed an MLP for Mg-Fe-O-W system by training an artificial neural network model with the ab initio data (see Methods). In addition, a separate MLP for Mg-Fe-O-Pt system was also trained to compare the diffusion behaviors of different siderophile elements under the CMB conditions, since Pt is a representative and well-studied HSE

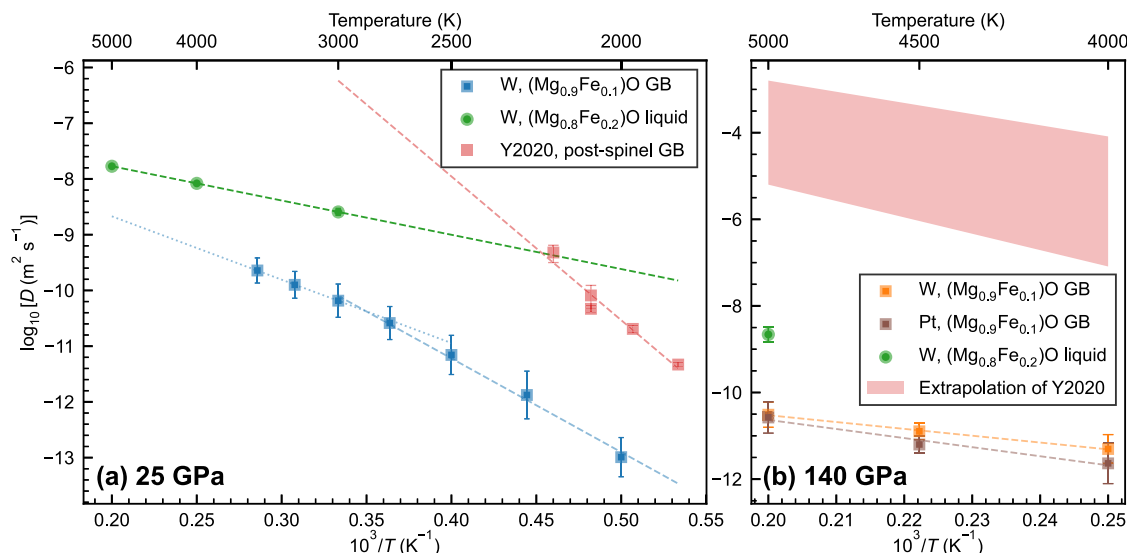
(e.g., Médard et al.<sup>29</sup>). The MLPs target temperatures up to 8000 K and pressures up to 200 GPa. We evaluated the reliability of the MLPs by comparing the energies, atomic forces, and stresses with those obtained from ab initio simulations based on density functional theory (DFT), for configurations not present in the training set (Methods; Supplementary Fig. 1). The diffusion coefficients were directly obtained from atomic trajectories of MD simulations (Eq. (2), Methods). The diffusion coefficients of Mg, Fe, O, and W calculated using MLPs align well with the DFT results for (Mg,Fe)O liquids (Supplementary Fig. 2). In addition, our DFT results show excellent consistency with experimentally measured W diffusivities in silicate melts, further validating the reliability of our methods (Fig. 1b). As shown in Supplementary Information, we also carefully investigated the factors affecting the diffusion of W including the exchange-correlation functional used in DFT calculations (Supplementary Fig. 3), system size (Supplementary Fig. 4), GB structure (Supplementary Fig. 5), vacancy concentration (Supplementary Fig. 6), valence state of W (Supplementary Fig. 7), and distribution of elements and vacancies (Supplementary Fig. 8), to ensure that diffusivity results are converged and robust<sup>25,30</sup>.

Figure 1a presents our results for the diffusion coefficients of W and Pt in periclase at 2.5 GPa and 1873 K. These two siderophile elements show similar diffusion behavior along GBs and exhibit similar diffusion coefficients (Fig. 1a). The GB diffusivity for W is approximately two orders of magnitude lower than its diffusivity in supercooled (Mg,Fe)O liquid, which aligns with previous findings of GB diffusion in olivine<sup>31</sup>. A previous experimental study, also plotted in Fig. 1a for comparison, reported GB diffusivities for multiple siderophile elements in MgO periclase under the same condition<sup>22</sup>. For Pt, our theoretical prediction closely matched the experimental result. However, experimental results indicated strong variations in GB diffusivities for different siderophile elements, spanning up to four orders of magnitude, with notably higher diffusivities for W and Au compared to other elements. This is perplexing considering the similarities of these two elements with other siderophile elements shown in Fig. 1a in terms of atomic mass and siderophilicity<sup>22</sup>. Furthermore, the experimental diffusivity for W is an order of magnitude higher than those in supercooled (Fe,Mg)O liquid, and over two orders of magnitude higher than W diffusivities in silicate melts under similar



**Fig. 1 | Comparisons of experimental and theoretical results under 0–2.5 GPa.** **a** Diffusivities of siderophile elements along (Mg<sub>0.9</sub>Fe<sub>0.1</sub>)O grain boundaries (blue) and in supercooled (Mg<sub>0.2</sub>Fe<sub>0.8</sub>)O liquid (green) obtained from molecular dynamics simulations under 2.5 GPa and 1873 K compared with previous experimental results (HW2007<sup>22</sup>) of grain boundary diffusivity in MgO under the same condition. **b** Diffusivities of W in different systems obtained from molecular dynamics simulations compared with experimental diffusivities in MgO grain boundaries

(HW2007<sup>22</sup>), basaltic melt (J2006<sup>33</sup>), and granitoid melt (HPG8 + 20 wt% Na<sub>2</sub>O, M1999<sup>32</sup>) under 0–2.5 GPa and 1873 K. The compositions of the simulated systems were made consistent with those of the experimental samples. The molecular dynamics simulations of the grain boundary systems and liquids were performed using machine learning potentials and ab initio calculations, respectively. Error bars indicate the standard deviations across different simulation runs. Source data are provided as a Source Data file.



**Fig. 2 | Diffusion coefficients of W and Pt under high pressures.** **a** Diffusion coefficients of W in  $(\text{Mg}_{0.9}\text{Fe}_{0.1})\text{O}$  grain boundary (GB) and  $(\text{Mg}_{0.8}\text{Fe}_{0.2})\text{O}$  liquid as a function of reciprocal temperature under 25 GPa. Previous experimental results for the post-spinel phase are shown for comparison<sup>23</sup>. **b** Diffusion coefficients of W and Pt as a function of reciprocal temperature under 140 GPa. The shading area represents the extrapolation of experimental results<sup>23</sup> assuming an activation

volume of  $0\text{--}2\text{ cm}^3\text{ mol}^{-1}$  (Eq. (4), Methods). Dashed and dotted blue lines are fitted Arrhenius equations (Eq. (3), Methods) over  $2000\text{--}2750\text{ K}$  and  $2750\text{--}3500\text{ K}$ , respectively. Other dashed lines are fitted Arrhenius equations from the data points with the same color. Error bars indicate the standard deviations across different simulation runs. Source data are provided as a Source Data file.

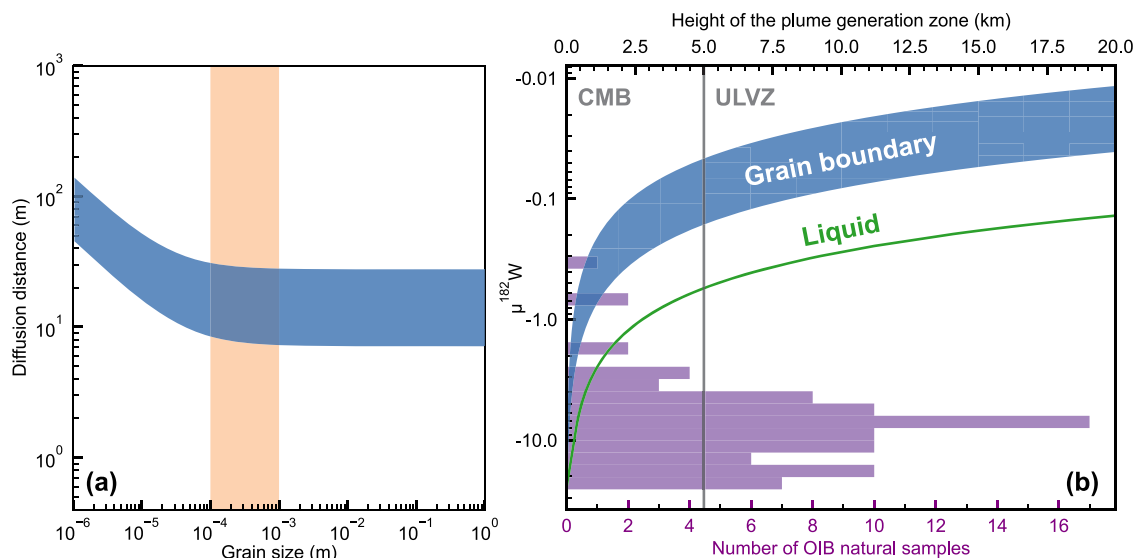
temperature and pressure conditions<sup>32,33</sup> (Fig. 1b). Admittedly, the GB features a relatively disordered atomic environment and exhibits higher diffusivities than the lattice counterpart<sup>34,35</sup>. However, regardless of the GB structure, GB diffusivity is always lower than that in a fully disordered liquid with the same chemical composition<sup>36,37</sup>. Therefore, the notably faster GB diffusion for W compared to liquid diffusion is unreasonable, suggesting that contributions to the large flux of W in the experiment may come from additional sources other than GB diffusion.

The high-pressure GB diffusion coefficients in  $(\text{Mg}_{0.9}\text{Fe}_{0.1})\text{O}$  are shown in Fig. 2. The temperature dependence of the GB diffusivities can be fitted using the Arrhenius equation (Eq. (3), Methods), denoted as the activation enthalpy ( $\Delta H$ ), which is proportional to the slope of the fitted lines. A recent experimental study reported the GB diffusivity of W in a post-spinel (bridgmanite + ferropericlasite) phase at 25 GPa<sup>23</sup>, indicating that W exhibits fast and strongly temperature-dependent GB diffusion, with an activation enthalpy of  $493\text{ kJ mol}^{-1}$ . To directly compare with these experimental results, we performed simulations at multiple temperatures under a pressure of 25 GPa and calculated the GB diffusivities of W (Fig. 2a). We observe a lower diffusivity and weaker temperature dependence of diffusivities compared to Yoshino et al.<sup>23</sup>, and the activation enthalpy is found to vary as a function of temperature ( $\Delta H = 322\text{ kJ mol}^{-1}$  for  $2000\text{--}2750\text{ K}$ ;  $\Delta H = 218\text{ kJ mol}^{-1}$  for  $2750\text{--}3500\text{ K}$ ). Note that our calculations pertain to self-diffusion without a W concentration gradient, which is different from the chemical diffusion measured in the experiments, even though the two are similar at very low W concentrations (Supporting Information). Moreover, W may oxidize to  $\text{WO}_3$  in experiments<sup>23</sup>, resulting in a significant drop in its melting point. This allowed the oxide melt to mechanically penetrate GBs due to surface tension, with a penetration rate likely exceeding the GB diffusion rate, potentially providing another mechanism for core-mantle interaction. This process, however, was absent in our simulations because we set a low W concentration to mimic the extremely low W content in Earth's mantle<sup>38</sup> (Methods), while previous experiments used metallic W foils as the diffusion source<sup>22,23</sup> and the W concentrations reached several tens of wt% in sink particles<sup>23</sup>. Additionally, the much lower valence state of Pt<sup>29</sup> compared to W<sup>39,40</sup> in silicate melts suggests Pt may be less likely to

form oxides and migrate via a similar mechanism. This difference likely explains the significantly lower flux of Pt compared to W observed by Hayden & Watson<sup>22</sup>.

The low and temperature-dependent  $\Delta H$  reported in this study can be justified from a microscopic perspective.  $\Delta H$  is defined as the energy difference between the system when the diffusing atom occupies the most energetically favorable positions (energy minima) and when it traverses the saddle point (energy barrier) connecting the energy minima<sup>41–43</sup>. For lattice diffusion, W atoms face a strong steric effect when migrating between lattice sites (deep energy minima) by passing through interstitial sites (high energy barrier), and thus show high activation enthalpy. The high  $\Delta H$  may be contributed by the increased strain energy caused by the distorted crystal lattice when W transitions at the saddle point<sup>43</sup>. In liquids, due to the disordered nature of the medium, W is likely to have a more flexible diffusion pathway that may be coupled with the free movements of neighboring atoms in the absence of lattice strain. Additionally, the larger volume and thus larger interatomic distances in liquids compared to solids may facilitate W migration, resulting in low  $\Delta H$ . GBs, as solid-solid interfaces, exhibit characteristics of an amorphous state and should therefore display diffusion behaviors that are intermediate between those of solids and liquids, which is supported by the observations that GB atoms do not have fixed lattice sites, yet there remains a significant steric effect on diffusion<sup>24,25</sup>. Thus, the  $\Delta H$  for GB diffusion of W should lie between the  $\Delta H$  for W diffusion in liquids ( $118\text{ kJ mol}^{-1}$ , green data in Fig. 2a) and in the single crystal ( $365\text{ kJ mol}^{-1}$  for olivine)<sup>40</sup>.

As the temperature increases, the disorder in the GB structure intensifies, and may trigger premelting (i.e., the GB transitions into a liquid-like, thermodynamically stable nanoscale film below melting temperature)<sup>44</sup>, as observed in ferropericlasite<sup>25</sup> and many other materials<sup>45–47</sup>. The incorporation of  $\text{W}^{6+}$  may reduce the melting point of ferropericlasite and facilitate this premelting process. This order-disorder transition of GB structures results in the temperature-dependent activation enthalpy, represented by solid-like and high  $\Delta H$  at low temperature and liquid-like and low  $\Delta H$  at high temperature, which has been observed in GB diffusion in  $\text{MgO}$ <sup>24</sup>. Consequently, GB diffusivities gradually increase towards liquid diffusivities with increasing temperature (Fig. 2a).



**Fig. 3 | The efficiency of grain boundary diffusion in transporting W isotope signals across the core-mantle boundary.** **a** The effective diffusion distance of W as a function of grain size over 4.54 Gyr, calculated from effective diffusivities under 4000–5000 K and 140 GPa. The orange shading corresponds to the grain size predicted for the average lower mantle<sup>94</sup>. **b** The  $\mu^{182}\text{W}$  value of OIB samples as a function of the height of the plume generation zone after the diffusive isotopic exchange over 4.54 Gyr. The blue shading represents the values calculated from effective diffusion coefficients ( $5.0 \times 10^{-16} \text{ m}^2 \text{ s}^{-1}$  to  $6.2 \times 10^{-15} \text{ m}^2 \text{ s}^{-1}$ , calculated from

Eq. (5) in Methods) under 4000–5000 K and 140 GPa. The green curve is calculated from the effective diffusion coefficient ( $7.1 \times 10^{-14} \text{ m}^2 \text{ s}^{-1}$ ) assuming W diffusivity in (Mg,Fe)O liquid at 5000 K is the upper bound of grain boundary diffusivity. The grain size of 0.1 mm is considered. The lowest height of observable ultra-low velocity zones (ULVZs) is shown for reference<sup>62</sup>. The histogram showing the frequency distribution of previously reported  $\mu^{182}\text{W}$  measured in OIB natural samples<sup>5</sup> is shown for comparison. Source data are provided as a Source Data file.

Previous studies have attempted to extrapolate the experimentally measured GB diffusivities<sup>23</sup> to high pressure and high temperature conditions, in order to explain the W isotope anomalies in the plume-source mantle<sup>18,23</sup>. Due to the temperature-dependent activation enthalpy described above, such extrapolation is likely unreliable. We now evaluate the feasibility of these extrapolations. At 25 GPa, the extrapolated W diffusivity at 3000 K is  $6 \times 10^{-4} \text{ m}^2 \text{ s}^{-1}$ , over two orders of magnitude higher than the W diffusivity in (Mg<sub>0.9</sub>Fe<sub>0.1</sub>)O liquid under corresponding conditions calculated in this study (Fig. 2a). It is also orders of magnitude greater than the diffusivities for almost all elements in all kinds of silicate melts measured under ambient pressure, if we extrapolate those results to 3000 K<sup>48</sup>. Note that we have not considered the diminishing effect of pressure on diffusion in silicate melts; otherwise, the gap would be even more significant. In addition, if we assume an activation volume of  $2 \text{ cm}^3 \text{ mol}^{-1}$ <sup>18,23</sup> to incorporate the pressure effect (Eq. (4), Methods), the extrapolated value under CMB conditions also reaches  $10^{-6} \text{ m}^2 \text{ s}^{-1}$ , about three orders of magnitude higher than the (Mg<sub>0.9</sub>Fe<sub>0.1</sub>)O liquid. Therefore, extrapolating the GB diffusivities to CMB conditions using the previously reported low-temperature activation enthalpy is unreasonable, and we believe that the GB diffusivities and  $\Delta H$  may have been significantly overestimated in experiments, potentially indicating additional physical/chemical mechanisms beyond GB diffusion (e.g., melting and mechanical penetration of W oxides) that have transported a significant amount of W under those experimental conditions.

Under the CMB pressure, we reported the GB diffusivities of W and Pt at temperatures from 4000 to 5000 K (Fig. 2b), which covers the upper limit of the estimated CMB temperatures throughout the Earth's thermal evolution<sup>49</sup>. Similar to low-pressure results (Fig. 1), W and Pt exhibit very close GB diffusivities, which are essentially identical within the uncertainties. Additionally, due to the enhancement effect of pressure on activation enthalpy, the GB diffusivities at 140 GPa show a higher temperature dependence than those at 25 GPa, characterized by  $\Delta H = 371 \text{ kJ mol}^{-1}$  for W. The activation enthalpy of GB diffusion for W in ferropericlase is similar to that of Fe ( $328 \text{ kJ mol}^{-1}$ )<sup>25</sup> at 140 GPa, which may be related to the similar ionic radii of  $\text{W}^{6+}$  and  $\text{Fe}^{2+}$  ( $\text{W}^{6+}$ : 74

pm;  $\text{Fe}^{2+}$ : 75 pm)<sup>50</sup>. Although  $\text{W}^{6+}$  has a slightly lower mass and a much smaller radius than  $\text{Pt}^0$  (135 pm)<sup>50</sup>, its significantly lower electronegativity<sup>51</sup> and higher valence state<sup>29,39,40</sup> compared to  $\text{Pt}^0$  under high pressure may cause W to form strong chemical bonds with undercoordinated O atoms in ferropericlase GBs<sup>24,25</sup>, leading to similar diffusion behaviors for W and Pt. Our diffusivity results are more than four orders of magnitude lower than the extrapolated experimental results (Fig. 2b), leading to markedly different geochemical implications.

### Geochemical Implications

As a potentially significant reservoir of W (with low  $\mu^{182}\text{W}$ ) and He (with high  $^3\text{He}/^4\text{He}$ ), the core has been proposed to be a source of the negative correlation between  $\mu^{182}\text{W}$  and  $^3\text{He}/^4\text{He}$  anomalies in OIBs<sup>4,5,18,19</sup>. The isotopic exchange between the core and the mantle through GB diffusion has been proposed to be the underlying mechanism<sup>18,23</sup>. Here, we used the diffusion coefficients of W, directly obtained under the high temperature and pressure conditions of the CMB without any extrapolation, to reassess the reliability of this mechanism.

Due to its siderophile nature, W is abundant in the Earth's core ( $\sim 500 \text{ ppb}$ )<sup>52</sup> and depleted in the mantle (27–31 ppb)<sup>38</sup>, which enables a continuous supply of core-like W across the CMB, maintaining a constant, core-equilibrated  $\mu^{182}\text{W}$  value of about  $\sim 220$  at the very bottom of the mantle. Subsequently, the core-derived W may undergo isotopic exchange with the ambient background mantle ( $\mu^{182}\text{W} \approx 0$ ), mediated by self-diffusion in mantle rocks, gradually modifying the W isotopic characteristics of the lowermost mantle. Given that the cation diffusion in ferropericlase is significantly faster than that in bridgmanite and post-perovskite<sup>23,42,53,54</sup>, we calculated the effective diffusion coefficient of W ( $D_{\text{W}}$ , Eq. (5), Methods) using its GB diffusivity in ferropericlase to constrain the upper limit of this isotopic exchange efficiency. We first consider a diffusion distance,  $L = \sqrt{D_{\text{W}}t}$ , to assess the characteristic spatial scale  $L$  over which this mechanism can impact within a certain time scale  $t$ . We find that even if this diffusive isotope exchange remains undisturbed over the age of the Earth (4.54 Gyr)<sup>55</sup>,



the maximum diffusion distance is only about 30 meters (Fig. 3a), far lower than the values obtained by extrapolating low-temperature experimental results<sup>18,23</sup>.

Furthermore, we modeled the possible  $\mu^{182}\text{W}$  anomalies in OIBs that result from this diffusive isotopic exchange based on the diffusion coefficients derived above. OIBs with W or He isotope anomalies may originate from deep-rooted mantle plumes<sup>56</sup>. Seismic observations suggest a geographical correlation between the generation zones of deep-rooted mantle plumes and the large low-shear velocity provinces (LLSVPs), indicating that these plumes are likely generated at the thin margins of LLSVPs<sup>57,58</sup>. Specifically, OIBs with low  $\mu^{182}\text{W}$  values and high  $^3\text{He}/^4\text{He}$  ratios usually lie above ultra-low velocity zones (ULVZs) at the margins of LLSVPs<sup>2,59,60</sup>. Therefore, LLSVPs<sup>5,18,61</sup> and ULVZs<sup>2,5,23</sup> may be the potential source reservoirs for those isotopic anomalies. Using the effective diffusion coefficients, we calculated the vertical composition profile of W isotopes above the CMB after 4.54 Gyr of diffusive isotopic exchange (Eq. (7), Methods; Supplementary Fig. 9). We then calculated the  $\mu^{182}\text{W}$  composition of a plume generation zone (PGZ) based on the average composition of the mantle between the CMB and the top of PGZ (Eq. (8), Methods). We consider the degree of entrainment—the ratio of dense material carried by the hot plume to the total upwelling material—to calculate the final  $\mu^{182}\text{W}$  value of OIBs reaching the Earth's surface<sup>18</sup> (see Methods). The results are shown in Fig. 3b. If we use the observed minimum height of ULVZs (~5 km, limited by the seismic resolution)<sup>62</sup> as the lower limit for the height of PGZ, our results indicate that the  $^{182}\text{W}$  deficit caused by GB diffusion would be within 0.2 in  $\mu^{182}\text{W}$  towards negative. Even if we use the W diffusivity in super-cooled (Mg,Fe)O liquid at 5000 K instead of GB diffusivity to calculate the effective diffusion coefficient, which constrains the maximum possible effect on mantle W isotopic composition by GB diffusion, the change in  $\mu^{182}\text{W}$  is still within 0.55 towards negative. This value is around an order of magnitude smaller than those of many OIB samples (Fig. 3b)<sup>5</sup>. Therefore, we conclude that diffusive isotopic exchange cannot be the source of W isotope anomalies in OIBs or any other mantle-derived rocks. Note that even if we further reduce the height of the PGZ to 1 km, this mechanism still cannot explain the significant  $\mu^{182}\text{W}$  anomalies observed in a large number of natural samples. The model discussed above addresses the isotopic exchange in the absence of a concentration gradient of W in the mantle. We also modeled the chemical diffusion of W from the core to the mantle considering a net mass flux of W given the possible enhanced lithophility of W at the CMB conditions. The results show that chemical diffusion fails to explain the observed  $^{182}\text{W}$  anomalies in OIBs (see Supplementary Information). Although small-scale convection within LLSVPs may enhance the transport of core-affected material within the mantle<sup>63</sup>, the amount of core-derived W sufficient for the observed  $\mu^{182}\text{W}$  in OIBs requires a W diffusion coefficient of over  $10^{-11} \text{ m}^2 \text{ s}^{-1}$ <sup>63</sup>, which is several orders of magnitude higher than the effective diffusion coefficient calculated in this study (see caption of Fig. 3).

Although diffusion in solids is excessively sluggish, our results indicate that the diffusivity for W in (Mg,Fe)O liquid can reach  $2.19 \times 10^{-9} \text{ m}^2 \text{ s}^{-1}$  at 5000 K and 140 GPa, corresponding to a diffusion distance of ~18 km over a time scale of 4.5 Gyr. Therefore, partial melts, if present in the lowermost mantle<sup>64–66</sup>, may accelerate the equilibration process of W isotopes between the plume source and core<sup>5</sup>. However, the temporal and spatial extent of these partial melts remains highly uncertain. In addition, the expected downward percolation of dense partial melts can cause drainage and accumulation, forming a melt-rich layer at the CMB, which may be inconsistent with geophysical constraints<sup>67</sup>. Although the melts may remain in suspension within ULVZs due to the viscous stirring, numerical simulations suggest that the bottom of ULVZs is relatively depleted in liquids<sup>68</sup>, which limits the efficiency of the diffusive isotopic equilibration. Therefore, atomic diffusion, especially the GB diffusion alone, is unlikely to be a viable mechanism to modify the  $\mu^{182}\text{W}$  values of the plume-

source mantle. Other mechanisms, such as the addition of core-exsolved oxides that are potentially enriched in W during the secular cooling (e.g., Rizo et al.<sup>4</sup>, Deng & Du<sup>19</sup>), are necessary to transport the core-like signature into the Earth's mantle.

## Methods

### Development of Machine Learning Potentials

Recently, we developed an MLP for Mg-O-Fe system covering a wide compositional range of various Mg:O:Fe ratios at pressure up to 200 GPa and temperature up to 8000 K<sup>69</sup>. Here, we further extended the compositional space of this MLP to include one more element  $X$  ( $X$  is either Pt or W) following the same procedure outlined at Peng & Deng<sup>30</sup>. Two MLPs were built with training sets inherited from the Mg-O-Fe MLP<sup>69</sup> and additional ones that entail Mg-O-Fe- $X$  mixtures. Specifically, we added configuration with bulk compositions including  $32\text{MgO} + aX$  with  $a = 2\text{--}16$ ,  $64\text{Fe} + bX$  with  $b = 2\text{--}16$ ,  $c\text{MgO} + d\text{Fe} + eX + f\text{O}$  with  $c = 32\text{--}64$ ,  $d = 48\text{--}144$ ,  $e = 2\text{--}8$ ,  $f = 0\text{--}48$ .

To briefly summarize the training process, configurations (structures of a given atomic arrangement) of training and test sets were drawn from multithermal and multibaric simulations<sup>70</sup>, which were used to efficiently sample the multi-phase configurational space. We used the structure factor of B1 MgO as the collective variable to drive the sampling. High-accuracy ab initio calculations were performed on the selected configurations to derive the corresponding energies, atomic forces, and stresses. The DeepMD approach was employed to train an MLP that takes a configuration and predicts its energy, atomic forces, and stresses without iterating through the time-consuming self-consistent field calculation<sup>27,28</sup>. The loss function is defined as

$$L(p_e, p_f, p_\xi) = p_e \Delta e^2 + \frac{p_f}{3N} \sum_i |\Delta \mathbf{F}_i|^2 + \frac{p_\xi}{9} \|\Delta \xi\|^2, \quad (1)$$

where  $p_e$ ,  $p_f$ ,  $p_\xi$  are tunable prefactors for the difference between the MLP prediction and training data.  $e$  is the energy per atom;  $\mathbf{F}_i$  is the atomic force of atom  $i$ ;  $\xi$  is the virial tensor divided by  $N$ ;  $N$  is the number of atoms. We adopted the conventional setting of increasing both  $p_e$  and  $p_\xi$  from 0.02 to 1 while decreasing  $p_f$  from 1000 to 1 over the course of training. A three-layer fitting network with 240 nodes in each layer was used to determine the form of the potential energy surface following our previous studies<sup>30,69,71</sup>.

An iterative training scheme<sup>71</sup> was employed throughout the training process. Principal component analysis was used to identify the most representative frames from MD trajectories for high-accuracy DFT recalculations. These recalculated results were compared with the predictions of the current model to identify frames that were poorly described by the MLP, which were then added to the training set. This iterative approach ensures that the training set remains concise and well-balanced. The final training sets consist of 6245 and 7557 configurations for Mg-O-Fe-W and Mg-O-Fe-Pt systems, respectively. We compared the energies, atomic forces, and stresses from MLPs to those from DFT calculations for 16579 and 16956 configurations of test sets that are not included in the training set for Mg-O-Fe-W and Mg-O-Fe-Pt systems, respectively (Supplementary Fig. 1). For Mg-O-Fe-W system, the root-mean-square errors (RMSEs) of the energies, atomic forces, and stresses are 6.76 meV atom<sup>-1</sup>, 0.32 eV Å<sup>-1</sup>, and 0.53 GPa, respectively. For Mg-O-Fe-Pt, RMSEs are 6.69 meV atom<sup>-1</sup>, 0.30 eV Å<sup>-1</sup>, and 0.51 GPa, respectively. The RMSEs of energy falls below the commonly accepted threshold of 10 meV/atom; the RMSEs of stress is at least an order of magnitude lower than the discrepancies between the MgO equation of state obtained from DFT calculations and experimental measurements<sup>72</sup>.

### Ab Initio Calculations

Ab initio calculations were performed using Vienna Ab Initio Simulation Package (VASP)<sup>73</sup>. We used the projector augmented wave (PAW)

method<sup>74</sup> as implemented in VASP<sup>73</sup> and PBEsol approximation<sup>75</sup>. PAW potentials are widely utilized in DFT calculations for planetary interior conditions and, when combined with the PBEsol approximation, demonstrate excellent agreement with experimental measurements of the physical properties of silicates and oxides<sup>9,76,77</sup>. The core radii are O: 0.820 Å (2s<sup>2</sup>2p<sup>4</sup>), Mg: 1.058 Å (2p<sup>6</sup>3s<sup>2</sup>), Fe: 1.164 Å (3p<sup>6</sup>3d<sup>7</sup>4s<sup>1</sup>), W: 1.455 Å (5p<sup>6</sup>5s<sup>2</sup>5d<sup>4</sup>), Pt: 1.455 Å (6s<sup>1</sup>5d<sup>9</sup>). High-precision ab initio calculations were performed for training and test configurations used to construct MLPs with an energy cutoff of 800 eV, energy tolerance of 10<sup>-6</sup> eV and 2 × 2 × 2 Monkhorst-Pack mesh.

### Construction of the Grain Boundary Structure

The GBs of ferropericase are symmetric tilt GBs constructed with AtomsK<sup>78</sup> in our previous study<sup>25</sup>. MgO bicrystals with two misorientation angles were considered:  $\alpha = 36.8^\circ$  where the two crystals meet with {310} planes, and  $\alpha = 29.5^\circ$  where the GB plane does not correspond to any high-symmetry crystal plane (see Supplementary Information). The compact GB structures at energy minima favored under mantle pressures were identified<sup>79</sup>, which can further interact with vacancy defects and transform the atomic configuration<sup>80</sup>. Various amounts of Mg-O vacancy pairs were randomly introduced within a 1-nm-wide region at the GB, and the convergence of GB diffusivities as a function of the vacancy concentration was tested<sup>25</sup> (see Supplementary Information). Then, the structures of (Mg<sub>0.9</sub>Fe<sub>0.1</sub>)O ferropericase were generated by the random substitution of Mg with 10 mol% of Fe in the MgO bicrystals. We tested the potential impact of these random factors on the W diffusivities (see Supplementary Information). Ferropericase bicrystals contain over 50,000 atoms, of which ~2000 are located within the 1-nm-wide GB regions, resulting in an average distance of over 20 nm between the centers of adjacent grains. Twenty W or Pt atoms were randomly introduced into the GB region. Based on previous experimental results, we set the valence state of W to +6<sup>23,39,40</sup> and that of Pt to 0<sup>29</sup> for both GB and liquid systems. The valence states were controlled by removing Mg atoms near W. The effects of the valence state on the W diffusivity were tested (see Supplementary Information).

### Molecular Dynamics Simulations

Ab initio MD simulations were performed using VASP<sup>73</sup> under the canonical ensemble (NVT) with the Nosé-Hoover thermostat<sup>81</sup>. Our ab initio MD simulations run for 10–40 ps, sampling the Brillouin zone at the  $\Gamma$  point with energy cutoffs of 500 eV and an energy tolerance of 10<sup>-4</sup> eV. MD simulations driven by MLPs were conducted by LAMMPS<sup>82</sup> and DeePMD-kit<sup>28</sup>. The GB systems were first equilibrated in isothermal-isobaric conditions (NPT) for 100 ps under the target temperatures and pressures. The resulting structures were then used as the initial configurations to set up the MD simulations for 2 ns under an NVT ensemble, in order to evolve the GB structure to reach a steady state. Subsequently, we performed long NVT simulations for 2 ns, from which the GB diffusivity is calculated. (Mg,Fe)O liquids with different Fe contents were obtained by melting ferropericase at high temperatures and gradually cooling, followed by NPT equilibration for 10 ps. The diffusivity was obtained from NVT simulations for 40 ps. The timestep of all MD simulations is 1 fs. All NVT simulations were repeated 4–20 times to report the standard deviation (error bars in all figures) of the diffusion coefficients.

### Diffusion Coefficients

Diffusion coefficients were calculated as the slope of mean square displacements (MSDs) by,

$$D = \lim_{t \rightarrow \infty} \frac{\text{MSD}}{6t} = \lim_{t \rightarrow \infty} \frac{\left\langle \left[ \vec{r}(t+t_0) - \vec{r}(t_0) \right]^2 \right\rangle_H}{6t}, \quad (2)$$

where  $\vec{r}(t)$  is the particle trajectories in Cartesian space, and  $\langle \dots \rangle$  represents an average over all atoms of the target element and over time with different origins<sup>83</sup>. The diffusivity as a function of temperature obtained from our simulations was fitted by the Arrhenius equation,

$$D = D_0 \exp\left(-\frac{\Delta H}{RT}\right), \quad (3)$$

where  $D_0$  is the pre-exponential factor,  $R$  is the ideal gas constant,  $T$  is the temperature in K, and  $\Delta H$  is the activation enthalpy. Extrapolating experimental results to higher pressures requires an additional parameter to describe the pressure dependence of the diffusivity, then the Arrhenius equation becomes

$$D = D_0 \exp\left(-\frac{\Delta E + P\Delta V}{RT}\right), \quad (4)$$

where  $\Delta E$  is the activation energy and  $\Delta V$  is the activation volume.

The effective diffusion coefficient of W ( $D_W$ ) can be estimated by the combination of lattice diffusivity ( $D_W^{\text{lattice}}$ ) and GB diffusivity ( $D_W^{\text{GB}}$ )<sup>18,23,84</sup>:

$$D_W = D_W^{\text{lattice}} + \frac{3\delta}{d} D_W^{\text{GB}}, \quad (5)$$

where  $\delta$  is the GB width (~1 nm) and  $d$  is the grain size (0.1–10 mm in the lower mantle)<sup>85</sup>. Since there is no available  $D_W^{\text{lattice}}$  data in lower mantle minerals, we assume the lattice diffusivity of W is the same as that of Si<sup>86</sup> following Yoshino et al.<sup>23</sup> due to their similarly high valence state<sup>23,40</sup>. Since the available experimental data are limited to 25 GPa<sup>86</sup>, a small activation volume of 1 cm<sup>3</sup> mol<sup>-1</sup> is assumed to estimate the upper bound of the effective diffusion coefficient.

### Modeling $\mu^{182}\text{W}$ anomalies in OIBs

We calculated the vertical composition profile of W isotopes above CMB (Supplementary Fig. 9) considering Fick's second law of diffusion

$$\frac{\partial C}{\partial t} = -D_W \nabla^2 C \quad (6)$$

Since the bottom of the mantle may have a fixed  $\mu^{182}\text{W}$  value of -220 and the diffusion distance of W is within 1% of the core radius, the solution to Eq. (6) can be well approximated by the diffusion in a half-space with a constant surface concentration<sup>87</sup>:

$$C = C_\infty + (C_0 - C_\infty) \text{erfc}\left(\frac{y}{\sqrt{4Dt}}\right) \quad (7)$$

where  $C_0$  is the core-equilibrated  $^{182}\text{W}/^{184}\text{W}$  ( $\mu^{182}\text{W} = -220$ ),  $C_\infty$  is the  $^{182}\text{W}/^{184}\text{W}$  in the ambient mantle ( $\mu^{182}\text{W} = 0$ ),  $y$  is the vertical distance above the CMB,  $t$  is the diffusion time of 4.54 Gyr, and erfc is the complementary error function. The  $^{182}\text{W}/^{184}\text{W}$  of the PGZ is taken as the integral average of Eq. (7):

$$C_{\text{PGZ}} = \frac{1}{h} \int_0^h C(y) dy \quad (8)$$

where  $h$  is the height of the PGZ.

The mixing ratio between PGZ material and background mantle material in OIBs is characterized by the degree of entrainment of dense

material neighboring hot upwelling mantle<sup>18,88</sup>

$$R_{\max} = \left( \frac{\Delta\rho_T}{\Delta\rho_{\text{Ch}}} \right)^2 \quad (9)$$

where  $\Delta\rho_T$  is the thermal density anomaly in the hot mantle plume,  $\Delta\rho_{\text{Ch}}$  is the chemical density anomaly in the PGZ, and  $R_{\max}$  is the maximum ratio of entrained dense material to total material. Considering that the estimated density anomaly of ULVZs is greater than that of LLSVPs<sup>89–91</sup>, we took the  $\Delta\rho_{\text{Ch}}$  of LLSVPs following Ferrick & Korenaga<sup>18</sup> to constrain the maximum proportion of plume-source material in OIBs. All parameters and the  $R_{\max}$  result (11%) are consistent with those reported in Ferrick & Korenaga<sup>18</sup>. The  $\mu^{182}\text{W}$  in OIBs as a function of the height of PGZ was obtained by mixing the plume source (Eq. (8)) and the background mantle ( $\mu^{182}\text{W} = 0$ ) with a mixing ratio of 1:8.

## Data availability

All the data used in this study have been deposited in the Open Science Framework via <https://doi.org/10.17605/OSF.IO/F4PCY92>. Source data are provided with this paper.

## Code availability

For the software packages used in this study, VASP is a commercial code available at <https://www.vasp.at>; DeepMD-kit is developed openly at <https://github.com/deepmodeling/deepmd-kit>; LAMMPS is developed openly at <https://github.com/lammps/lammps> and available in Zenodo (<https://doi.org/10.5281/zenodo.6386596>)<sup>93</sup>; PLUMED 2 is developed openly at <https://github.com/plumed/plumed2>.

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## Author contributions

Y.P. and J.D. carried out the simulations. J.D. supervised the project. Y.P., T.Y., and J.D. all contributed to data analysis and writing the paper.

## Competing interests

The authors declare no competing interests.

## Additional information

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