

REVIEW ARTICLE

Calcium phosphate cements for bone engineering and their biological properties

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Calcium phosphate cements (CPCs) are frequently used to repair bone defects. Since their discovery in the 1980s, extensive research has been conducted to improve their properties, and emerging evidence supports their increased application in bone tissue engineering. Much effort has been made to enhance the biological performance of CPCs, including their biocompatibility, osteoconductivity, osteoinductivity, biodegradability, bioactivity, and interactions with cells. This review article focuses on the major recent developments in CPCs, including 3D printing, injectability, stem cell delivery, growth factor and drug delivery, and pre-vascularization of CPC scaffolds via co-culture and tri-culture techniques to enhance angiogenesis and osteogenesis.

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INTRODUCTION

There has been a continuous and fast-paced emergence of new synthetic biomaterials developed for bone repair and regeneration over the past several decades. These biomaterials include metals, polymers, ceramics, bioactive glasses, calcium sulfates, calcium carbonates and calcium phosphates (CaPs). Among them, calcium phosphate cements (CPCs) are promising for clinical applications due to their advantageous properties including bioactivity, osteoconductivity, injectability and moldability. The discovery of the first CPC occurred inadvertently via the observation of calcium phosphate solubility behavior.^{1–3} Brown and Chow found that the solubilities of tetracalcium phosphate [TTCP: $\text{Ca}_4(\text{PO}_4)_2\text{O}$], dicalcium phosphate (DCPA: CaHPO_4) and dicalcium phosphate dehydrate (DCPD: $\text{CaHPO}_4 \cdot 2\text{H}_2\text{O}$) were much greater than that of hydroxyapatite (HA) under neutral pH conditions.⁴ A slurry

containing appropriate amounts of TTCP and DCPD (or DCPA) led to HA precipitation as an end product and was capable of self-setting to form a hard mass.^{2–3} In the decade following this first discovery, CPCs were approved by the Food and Drug Administration (FDA) and were introduced into clinical practice for the treatment of craniofacial defects⁵ and bone fractures.⁶ Since then, other CPC formulations have been developed, and a large amount of research has been conducted.^{7–18} Currently, CPCs are defined as a combination of one or more calcium phosphate powders which, upon mixing with a liquid phase, form a paste able to self-set and harden *in situ* in the bone defect site to form a scaffold.¹⁹

One of the most important characteristics of CPCs is their ability to form *in situ* through a body-temperature dissolution-precipitation reaction.¹⁹ This feature gives rise

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to other beneficial properties such as molding capability upon mixing,²⁰ injectability that enables minimally invasive application,²¹ and the ability to serve as a carrier for drug and biological molecule delivery.²² Early research on CPCs primarily focused on improved setting, handling and mechanical properties of CPCs through the tailoring of many processing parameters such as cement composition, additives, porogens, and particle size.^{23–28} In recent years, in addition to the development of new processing technologies in CPC manufacturing, the paradigm has shifted toward biological responses by emphasizing the enhancement of biological interactions of CPCs with cells and tissues as well as their applications in bone tissue engineering.^{29–33} Biological responses of scaffolds are a key factor in the translational application of biomaterials and their commercialization for clinic applications. Several meritorious reviews on CPCs have described their mechanical properties,^{34–36} processing approaches,^{37–38} drug delivery,^{19,22,39–40} and functional enhancement by polymeric additives,⁴¹ which will not be repeated here. The present article reviews the major new developments in CPC processing technologies in recent years and focuses on novel biological interactions of CPCs, particularly in the context of stem cell responses and delivery as well as *in vivo* bone regeneration. The various CPC categories described in this article and their major biological properties are summarized in the diagram in Figure 1.

PRE-FABRICATED CPC SCAFFOLDS AND 3D PRINTING

Although injectability is one of the advantages of CPCs, pre-fabricated CPC scaffolds are often prepared for two reasons: (1) To ensure a complete setting reaction because only fully set CPCs demonstrate excellent tissue responses. When CPCs fail to set, they cause inflammatory reactions.⁴² Therefore, manufacturing pre-fabricated CPCs ensures complete setting prior to *in vivo* application. (2) To facilitate the creation of interconnected macroporous structures into CPCs. Self-setting CPC scaffolds without any modification are microporous but not macroporous and have limited pore interconnections.⁴³ To promote tissue in-growth and accelerate the CPC degradation rate and subsequent replacement by bone, macropores were incorporated into CPCs via two methods: particle leaching (the addition of water-soluble particles, such as sodium bicarbonate, mannitol, salt or glucose, that dissolve or degrade after setting) and gas-foaming (the formation of air bubbles during the setting period).^{37,44} *In situ* setting with particle leaching has several disadvantages. First, because the porogens inside the cement have limited exposure to body fluids, the degradation or solubility of the particles may be compromised, which leads to limited porosity.⁴⁵ Second, the *in vivo* dissolution of some particles may result in hyperosmosis.⁴⁶ Third, some porogens may increase the paste viscosity and impede the injectability of CPC. The major drawback of *in situ* application of the gas-foaming method is the risk of air emboli or emphysema. Therefore, pre-fabricated CPC scaffolds have been developed to

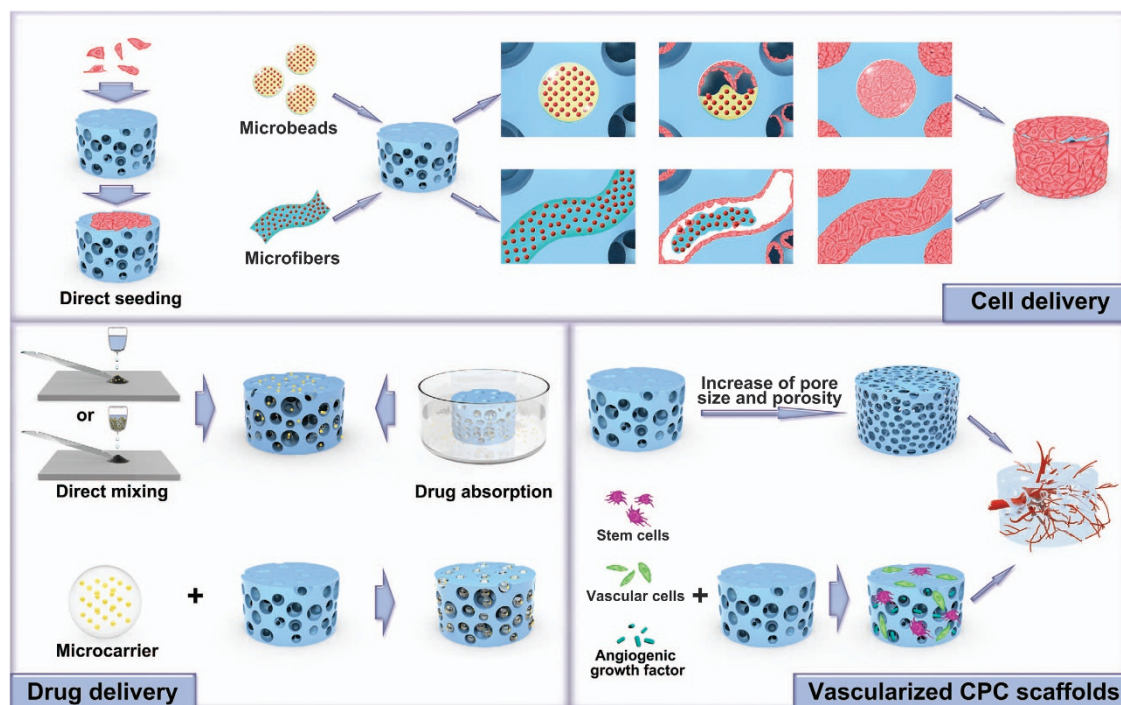


Figure 1. Schematic diagram summarizing the various CPC categories described in this article and their major biological properties.

allow more delicate control of the setting process and macroporous architecture of the scaffolds before *in vivo* implantation.

Recently, three-dimensional (3D) printing has rapidly developed to allow the fabrication of pre-set CPC scaffolds. 3D printing is an additive manufacturing process in which geometrical data are used to produce 3D structures by depositing materials layer by layer.⁴⁷ 3D-printed CPC scaffolds are favored over customization to meet the specific needs of each patient/defect. The benefits for clinical applications include easy adaptation and fixation, reduced surgical time, favorable esthetic results and minimal waste products. There are several different techniques for 3D printing, including direct 3D printing (direct ink writing), fused deposition modeling (FDM), stereolithography (SLA), and selective laser sintering (SLS). For a detailed description of each technique, readers are encouraged to read previous review papers on this topic.^{48–49} For CPC scaffolds, binder jetting is the most commonly employed 3D printing technique.⁵⁰ Briefly, one or several print heads spray a binder solution (for example, an aqueous solution) precisely onto a bed layer of CPC powder. The binder locally joins adjacent powder particles together and hardens the wetted areas through the dissolution-precipitation reaction. The process repeats by spreading another layer of powder and ejecting binders according to a pass designed by the computer. This continues until the complete 3D structure is formed.⁴⁸ The printability of the material is related to many parameters such as particle size and size distribution, morphology and surface area of the powder, roughness and flowability of the powders, the solubility/wettability/reactivity of the powder with the binder, and binder drop size.⁵¹ A study investigating beta-tricalcium phosphate powder suggested that 3D printing was not feasible with particles either too small (with a mean particle size of 7 μm) or too large (with a mean particle size of 51 μm), while mean particle sizes in the range of 20–35 μm resulted in good printing accuracy.⁵¹ Small particles tend to agglomerate under the influence of van der Waals forces. Very fine or porous particles exhibit low flowability and high surface roughness. Therefore, these factors greatly affect the smoothness and homogeneity of the powder bed, resulting in smearing and poor resolution.⁵¹ However, although large particles have better flowability, they tend to yield layer displacements due to low powder bed stability and low accuracy because the resolution is at least twice the particle size.⁵² Flowability was shown to be significantly reduced by decreasing the HA granule size.⁵³ To work with small particle sizes to achieve a high resolution, strategies such as plasma coating⁵¹ and moisture application⁵⁴ were attempted to stabilize the top layer surface and allow particle rearrangement and

wetting while avoiding particle ejection out of the powder bed. Furthermore, by adding reactive minerals such as calcium sulfates into calcium phosphate, significant improvements to 3D printing parameters are achieved.⁵⁵ The dimensional accuracy of printed CPC scaffolds (powder: α -TCP; liquid: Na_2HPO_4) is $\sim 200\text{-}\mu\text{m}$, which indicates a good degree of fitting to craniofacial defects in anatomical models.⁵⁶ A critical step for powder-based 3D printing is the removal of the loose powder inside the pores of the printed scaffold after printing, a process known as depowdering. Depowdering is especially challenging when the pores and pore interconnections are small and found in the innermost parts of the scaffolds with large dimensions. One possible solution may be the use of depowdering-friendly designs with large windows and free-to-move fillers.⁵⁷ In addition, layer thickness and printing orientations (parallel to the X, Y and Z directions) are important for depowdering.⁵⁸ Shear forces at the powder bed increase with reduced layer thickness, which leads to the deterioration of the final printed samples upon depowdering. Depowdering is easier in scaffolds printed in the X and Y directions than that in scaffolds printed in the Z direction because of the distortion in samples printed in the Z direction.⁵⁸ However, the relationship between 3D printing parameters and CPC scaffold quality and performance has yet to be established and warrants further study.

3D plotting (direct ink writing, direct write assembly, material extrusion) is another common technique for CPC 3D printing.⁵⁹ This is an extrusion-based printing technology in which a paste or viscous materials, instead of powders, are used as the starting form and deposited as strands via a nozzle in a layer-by-layer fashion based on predesigned structures.⁶⁰ For 3D plotting, the printability is dependent on even dispersion, viscosity, fluidity, extrusion performance, setting time of the paste, and the shape stability of the printed strands to withstand the weight of the structure during assembly. The setting time for CPCs plays an important role in controlling the printable time period of the paste. One study reported the printable time of a CPC (powder: TECP:DCPA = 1:1 molar ratio, liquid/binder: polyvinyl alcohol) as only 10-min, which makes printing difficult.⁶¹ With the addition of a mesoporous calcium silicate, the printable time was increased to approximately 120-min.⁶¹ Other optimizations of the direct printing ink formulation have included the addition of gelatin to introduce an induction time for the onset of the CPC setting reaction.⁶² Specifically, this formula includes Targon 1128 as the dispersant, hydroxypropyl methylcellulose (HPMC) as the thickening agent, polyethylenimine (PEI) as the jellifying agent,⁶³ and a ready-to-use oil-based CPC paste that sets only upon contact with water and thus has no time limit for printing.⁵⁹

A critical issue for printing resolution is nozzle diameter and the stability of the extruded strands.⁵⁰ 3D plotting has two advantages: (1) it enables easy printing of a combination of different materials,⁶⁴ and (2) due to the

mild conditions, it allows simultaneous cell or growth factor plotting, known as bioprinting.^{64–65} Using a two-channel plotting method, a scaffold with the combination of an oil-based CPC and an alginate-gellan

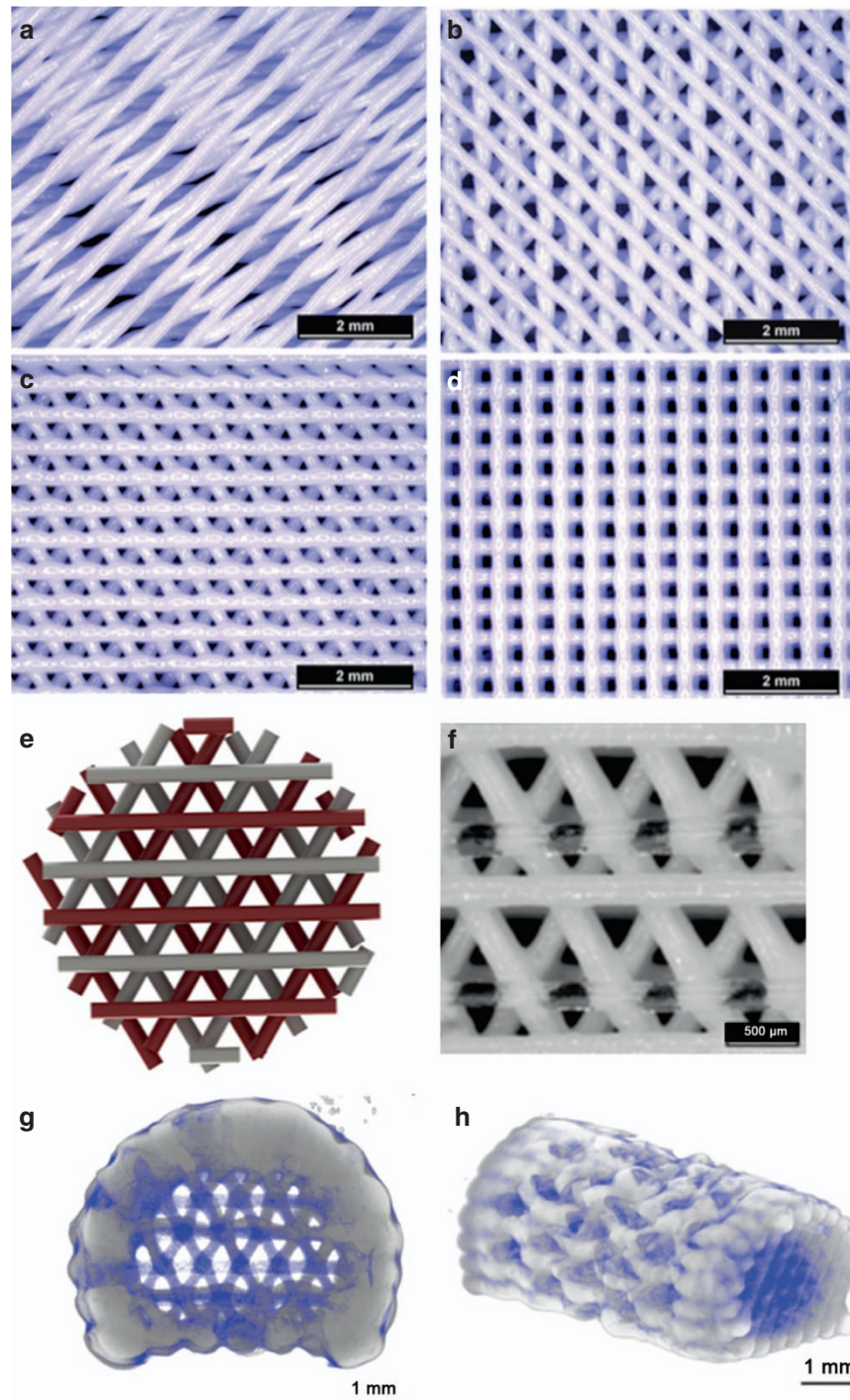


Figure 2. Highly sophisticated CPC scaffold structures via 3D plotting. Stereomicroscopic images of CPC scaffolds plotted with 15° (a), 45° (b), 60° (c) and 90° (d) configurations (change in orientation relative to the layer underneath). Design and printing of a CPC-hydrogel biphasic scaffold: model of biphasic scaffolds with CPC (white) and a growth factor-loaded hydrogel (red) (e); the printed scaffold (f); 3D reconstructions from micro-CT data of the biphasic scaffold (g, h). CPC is grayish white. Alginate-gellan hydrogel is blue. (Adapted from Ahlfeld *et al.*⁶⁴ with permission.)

hydrogel was fabricated and laden with growth factor VEGF, involving a highly sophisticated strand arrangement, pore structure and geometry (Figure 2).⁶⁴ In another study, a bone morphogenetic protein 2 (BMP2)-loaded mesoporous silica/CPC porous scaffold was 3D-plotted and tested in *in vitro* cell culture and in a rabbit femur defect model.⁶⁶ The scaffold promoted the osteogenic differentiation of human bone marrow stromal cells (hBMSCs) and enhanced vascularization and osteogenesis compared to the CPC control.⁶⁶ In terms of cell-containing bioprinting, hydrogels such as alginate,⁶⁷ collagen,⁶⁸ synthetic polymers such as PLGA, and PEG⁶⁹ are primarily used as bioinks due to their resemblance to the extracellular matrix (ECM) and good printability. In some cases, calcium phosphates are added to enhance cell attachment and osteogenic differentiation, thus favoring the use of bioink for bone tissue engineering applications.⁶⁷

In general, due to the incremental addition of materials, 3D printing allows for not only the easy control of scaffold shape and geometry but also the control of fine features such as interconnected porosity, pore size and distribution, and complex spatial heterogeneity, which are not achievable with traditional strategies.⁵⁰ The possibility of manufacturing customized implants with almost no design limitations makes 3D printing highly valuable in reconstructive surgery. However, more extensive research is needed to optimize the key parameters for successful 3D printing of CPC scaffolds.

INJECTABLE CPC SCAFFOLDS

Traditional bone grafting requires an open surgical approach to graft application sites and may be associated with complications such as a large surgical scar, increased pain and a longer post-operative recovery. To overcome these drawbacks, injectable bone graft substitutes are used for minimally invasive surgery. Two main obstacles that inhibit CPC injectability are liquid-solid phase separation during injection⁷⁰ and paste disintegration upon contact with blood or body fluids.⁷¹ Phase separation leads to not only the presence of non-extrudable paste left in the syringe but also extravasation at the injection site and a decrease in the viscosity and mechanical strength of CPCs. The disintegration of CPCs in the body causes inflammatory responses and even severe consequences such as cement embolism and cardiovascular deterioration by simulating blood coagulation.⁷² Therefore, efforts have been made to improve CPC injectability. These strategies include the following: (1) increasing the viscosity of the liquid phase by adding viscous binders such as chitosan,⁷⁴ gelatin,⁷³ hyaluronic acid,⁷⁴ methylcellulose,⁷⁵ and others; (2)

optimizing the CPC powder in terms of the particle size, particle size distribution, particle shape, and particle-particle interactions;⁷⁶ (3) regulating the setting reaction;⁷⁷ and (4) modifying the extrusion parameters such as CPC mixing and the sizes of the syringes and/or needles.⁷⁸ All of these factors were discussed in detail in a recent review on CPC injectability.⁷⁰

Recently, many studies have applied various injectable CPC formulations into animal models for bone regeneration.^{79–80} Injectable CPCs containing 50% (volume ratio) microspheres (poly(lactic-co-glycolic acid) (PLGA), gelatin (GEL) or poly(trimethylene carbonate) (PTMC)) were implanted into rabbit femoral bone defects. CPC/GEL had a significantly lower score than all other groups at the cement-bone interface. Both CPC and CPC/PLGA showed a better response than CPC/PTMC at 4 weeks, but there were no significant differences among these three groups at 8 and 12 weeks.⁷⁹ A recent study applied a commercially injectable CPC (Calcibon) with platelet lysates in bilateral calvarial defects in rats.⁸¹ The delivery of the platelet lysate enhanced bone healing with an injectable CPC at early healing times. In large animal models, injectable CPCs have also shown promise for bone regeneration. For example, injectable CPC/PLGA composites demonstrated biocompatibility and direct bone contact for sinus floor augmentation procedures in a sheep model.⁸² Another study evaluated the efficiency of local bisphosphonate delivery via injectable CPC in vertebral bodies of the lumbar spine of an osteoporotic sheep model where the consequences of osteoporotic fractures were highly deleterious in patients. The bisphosphonate-combined cement in vertebral body bone defects had a beneficial impact on both bone content and the micro-architectural properties of the trabecular bone surrounding the implant.⁸³ These animal studies demonstrated the promise of using injectable CPCs for bone repair and regeneration.

Indeed, CPCs have gained clinical acceptance as valuable bone substitution biomaterials for over 20 years, and several CPCs are commercially available. Injectable CPCs were used to repair human periodontal intrabony defects and showed favorable radiographic results.⁸⁴ CPCs were also used in young patients for balloon kyphoplasty instead of polymethylmethacrylate cement. In most cases, good integration of CPCs in the vertebra was observed with no radiological signs of osteolysis or osteonecrosis. Only a few patients showed demineralization in follow-up CT scans.⁸⁵ Several papers reviewing the properties of injectable CPCs are available for readers who want additional detail.^{86–88} The present review focuses on new developments in CPCs with an emphasis on their biological interactions and cell delivery as detailed in subsequent sections.

BIOLOGICAL REQUIREMENTS AND BIOLOGICAL RESPONSES OF CPCs

Biocompatibility

Biocompatibility is defined as the property of a material being compatible with living tissues. Biocompatible materials do not induce a toxic response when implanted in the body.⁸⁹ Biocompatibility is an essential requirement for tissue-engineered products to support cellular activities and optimize tissue regeneration without eliciting a cytotoxic effect in those cells or causing undesirable local or systemic responses in the host. The end products of the dissolution-precipitation reactions for CPCs include brushite (DCPD) and apatite (HA or calcium deficient HA (CDHA)), which are known to be biocompatible.⁹⁰ Pre-set CPCs exhibit favorable short-term and long-term biocompatibility, as evidenced by many studies evaluating tissue responses in rats,^{91–92} rabbits,⁹³ dogs,⁹⁴ sheep,^{16,32} and goats,⁹⁵ as well as various types of cultured cells.^{24,93,96} However, injectable CPCs require the completion of the setting reaction to avoid cytotoxicity, as unset or disintegrated CPCs cause severe inflammatory responses, blood clotting, and cement embolism.^{72,97} Incorporating polymers into CPCs is a strategy used to improve CPC properties.⁴¹ In a recent study, an injectable macroporous CPC was prepared by the syringe-foaming method using a hydrophilic viscous polymeric solution known as silanized-hydroxypropyl methylcellulose (Si-HPMC).⁹⁸ Si-HPMC not only acts as a foaming agent to create macroporous structures inside CPCs but also endows the CPC paste with an appealing rheological behavior at the early stage of setting due to its self-crosslinking properties, thus improving its injectability and cohesion.⁹⁸ Indeed, when this CPC was injected into defective rabbit femurs, no adverse foreign body reaction was observed at 1 week and 6 weeks post-implantation.⁹⁸

Bioactivity

Bioactivity refers to the ability of bone scaffolds to bind directly to the surrounding bone without the formation of fibrous tissue.⁹⁹ Bioactivity is often evaluated by examining the ability to form apatite on the biomaterial in a simulated body fluid (SBF) with ion concentrations close to those in human blood plasma.¹⁰⁰ A bioactive material is defined as one that accelerates apatite crystallization in a solution supersaturated with respect to hydroxyapatite.¹⁰⁰ However, the validity of using an *in vitro* SBF test to predict the *in vivo* bioactivity of a material has been questioned.¹⁰¹ For example, Bohner and Lemaître showed that a bioactivity test with SBF may not only give false-positive results but also false-negative results.¹⁰¹ The authors concluded that “*in vitro* bioactivity tests in SBF solutions cannot be used

to predict the *in vivo* bone bonding ability of a material”. With some improvements to the protocol, these tests may be used for initial screening. However, the most reliable evaluation method remains *in vivo* implantation in a bone defect.

Bioactivity is one of the most important properties of CPCs.¹⁹ To further enhance CPC bioactivity, bioactive glass, which is known for its bioactivity, was incorporated into CPCs.^{102–103} The bioactive glass acted as a source of calcium and phosphate ions in the cement setting reaction. With this addition, increasing apatite formation was detected on the surface of the CaP compound after soaking in SBF for 7 days.¹⁰³ *In vivo* examination of samples implanted into rabbit femoral bones indeed showed a better healing process and more bone growth with the addition of bioactive glass.¹⁰³

Osteoconductivity

Osteoconductivity is defined as a biomaterial property that facilitates the in-growth of new bone into a surface or a volume in which the biomaterial serves as a scaffold to guide new bone formation.¹⁰⁴ CPCs are osteoconductive because they permit the attachment, proliferation, migration and phenotypic expression of bone cells, leading to the formation of new bone.^{105–106} Osteoconduction is related to the architectural geometry of the scaffold.¹⁰⁶ Intimate adaptation, fixation and stability of the implant to the defect site are of critical importance to facilitate the in-growth of bone tissue. In addition, the scaffold should have high porosity and interconnectivity with optimal pore sizes to ensure cell penetration, nutrient exchange and waste elimination. For bone tissue engineering, an ideal scaffold should have 60%–80% interconnected porosity with pore sizes ranging from 150 to 500 μm .¹⁰⁷

Osteoconduction also depends on the chemical composition of the scaffold. The incorporation of several types of ions benefit CPC osteoconductivity. For instance, a silicon CPC (Si-CPC) was developed,¹⁰⁸ and the cytocompatibility of the Si-doped cement was tested with a human osteoblast-like cell line (MG-63), which showed enhanced cell proliferation (up to threefold) over that without Si. When implanted in a rabbit parietal bone defect model, significantly greater amounts of new bone were detected in the 10% Si-CPC group compared to that in the CPC control group.¹⁰⁸ In another study, strontium was incorporated into CPC (Sr-CPC) to enhance its osteoconductivity and accelerate its degradation.¹⁰⁹ *In vitro* studies showed higher osteoblastic cell proliferation rates in Sr-CPC groups. *In vivo* studies demonstrated more rapid degradation and advanced osteoconductivity in the 10% Sr-CPC group compared to those in the CPC control at 2, 4, 8, 16, and 32 weeks after the operation.¹⁰⁹

Osteoinductivity

Osteoinduction is defined as the recruitment and stimulation of progenitor cells to differentiate toward the osteoblastic lineage.¹⁰⁴ CPCs are generally osteoconductive but not osteoinductive.²⁰ However, several CPCs reportedly have the ability to form bone in nonosseous sites *in vivo* without the addition of osteogenic factors.¹¹⁰ Since this osteoinductive property is observed for some CPCs but not others, these materials are described as having “intrinsic” osteoinductivity.¹¹¹ This inductive phenomenon is likely attributable to the combined effects of topography, composition, and micro and macroporosity of the CPC scaffolds.¹¹¹ It is likely that the intricate architecture of the scaffold permits the entrapment and concentration of circulating growth factors, such as BMPs and osteoprogenitor cells, *in vivo* thus conferring osteoinduction capability upon the CPCs.¹¹¹ In addition, CPCs serve as calcium and phosphate ion sources *in vivo*. Ca^{2+} , PO_4^{3-} and HPO_4^{2-} ions are released into the surrounding tissues, regulate osteoblast functions¹¹² and induce localized ion supersaturation, which causes the reprecipitation of carbonated apatite on the scaffold.^{113–114} A previous study proposed a new strategy to regulate bone marrow mesenchymal stem cell (BMSC) adhesion and osteogenic differentiation by adding magnesium into the CPC, thus improving its osteoinductivity.¹¹⁵ A CPC containing 5 wt% and 10 wt% magnesium not only enhanced BMSC adhesion but also upregulated osteogenic gene and protein expression *in vitro*. An *in vivo* study demonstrated that CPC with 5 wt% magnesium achieved the greatest bone volume at 2 and 8 weeks, confirming its beneficial osteogenesis effects via the addition of magnesium.¹¹⁵ To gain or enhance CPC osteoinductivity, novel strategies such as the addition of osteoprogenitor cells,^{116–117} growth factors,^{118–119} bioactive proteins^{120–121} or peptides^{122–123} into CPCs have exhibited favorable effects. Therefore, novel CPC compositions with intrinsic and engineered osteoinductivity are highly promising to enhance bone regeneration.

Biodegradability

Ideally, a CPC scaffold should degrade at the same rate that new bone forms. CPCs biodegrade primarily via two mechanisms: a passive resorption process via chemical dissolution and an active resorption through a cell-mediated process.¹²⁴ The degradation of CPCs is tailored by controlling several factors: (1) physical factors such as the physical form of the CPC (particulate or bulk), porosity, surface area, and crystallinity (crystal size, crystal perfection, and grain size), and so on; (2) chemical factors such as the composition and ionic substitutions; and (3) biological factors such as the activation of macrophages

or osteoclasts.¹²⁵ Enhancing CPC degradation is achieved by adding rapidly degradable porogens such as PLGA to generate macropores upon PLGA degradation. PLGA degrades hydrolytically, leading to the production of lactic and glycolic acid monomers. The acidic nature of the resulting byproducts is an additional advantage of PLGAs in combination with poorly degradable CPCs because CPCs degrade by acid dissolution.¹²⁶ After being injected into a rabbit femoral bone defect model, CPC-PLGA exhibited favorable bone responses with >55% degradation and >13% bone formation at 6 weeks and >90% degradation and >40% bone formation at 26 weeks postoperation.¹²⁷ Based on this same mechanism, glucono delta-lactone (GDL), which has a faster degradation rate than PLGA, was incorporated into CPCs as acid-producing microparticles to accelerate CPC degradation.¹²⁸ Indeed, histomorphometrical evaluation revealed that CPCs containing 10% of GDL degraded more rapidly and were replaced by more bone tissue (32.8%) than CPC-PLGA at 2 weeks after implantation in a rabbit femoral bone defect.¹²⁸

CPC SCAFFOLD CONSTRUCTS FOR BONE TISSUE ENGINEERING

Cell delivery

Recent advancements in tissue engineering and regenerative medicine have indicated that cell-based therapeutics achieve robust regeneration with greater efficacy and better predictability than methods that do not involve cell seeding.¹²⁹ These novel approaches employ scaffold constructs in combination with living cells to generate cell-driven, functional tissue rather than filling a defect with a nonliving scaffold. A tissue-engineered construct acts both as a scaffold to bridge the defect and as a cell delivery vehicle. The biomaterial-cell interactions of CPCs with various types of stem cells, such as BMSCs, umbilical cord mesenchymal stem cells (UCMSCs), embryonic stem cells (ESCs), induced pluripotent stem cells (iPSCs), were previously reviewed.^{130–131} The present article specifically explores recent advances in strategies for cell delivery, specifically highlighting the design of CPC-based scaffolds.

Direct cell seeding onto the porous surfaces of preformed CPC scaffolds is a common approach due to its simplicity. However, this type of static cell seeding has limitations, including low seeding efficiency and minimal cell penetration into the scaffold, leading to non-uniform cell distribution.¹³² It is not feasible to directly mix cells into the CPC paste because the mixing forces, ionic exchanges and pH fluctuation during CPC setting are detrimental to cell viability. To address this problem, cell encapsulation has been proposed to protect cells during CPC mixing and injection (Figure 3). In a recent study,

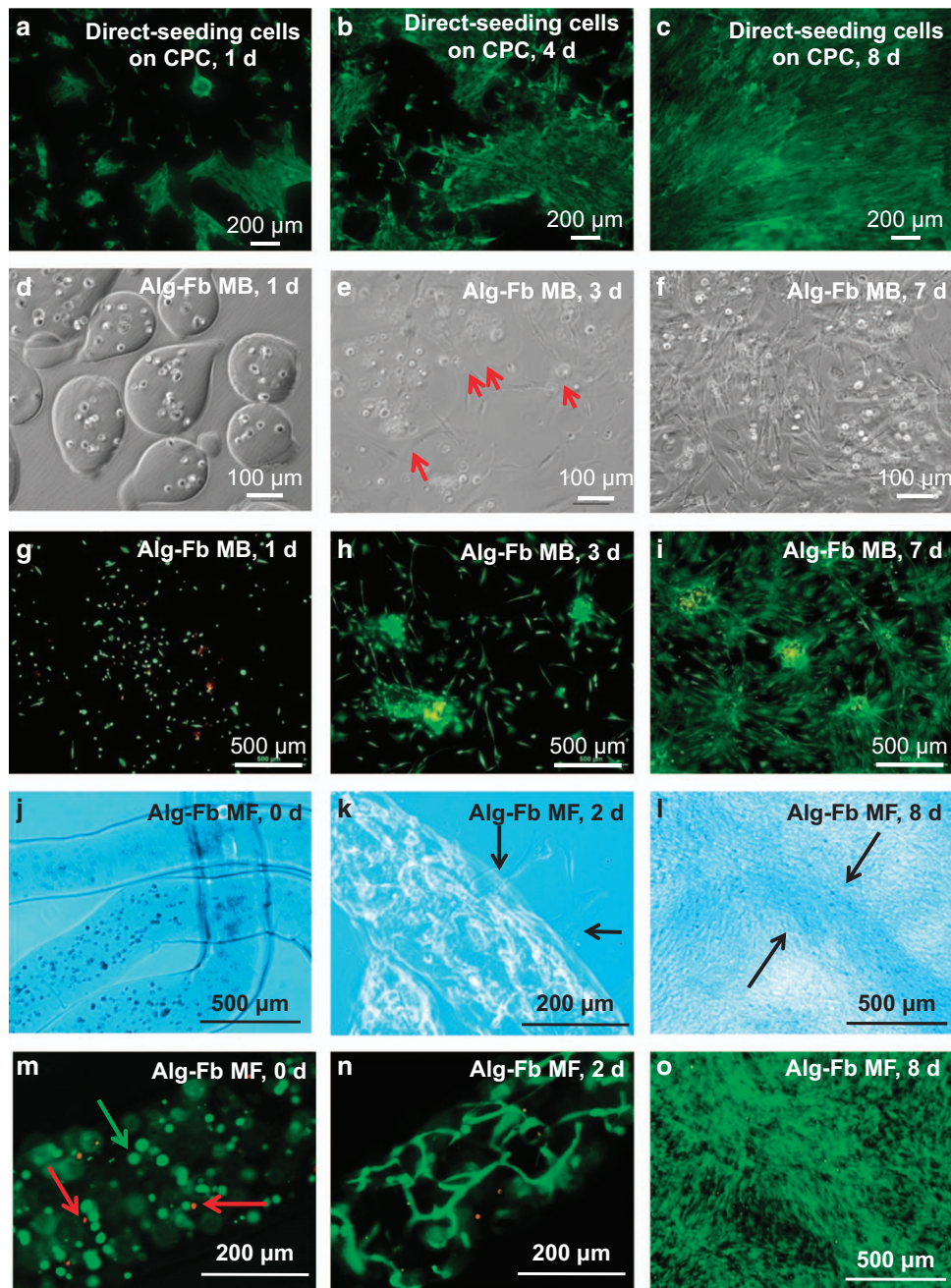


Figure 3. Methods of cell delivery via CPCs. Live-dead staining of (1) direct cell seeding on CPC surfaces (a–c); (2) cell encapsulation in alginate-fibrin microbeads (Alg-Fb MB) (d–f); (3) cell encapsulation in alginate-fibrin microfibers (Alg-Fb MaF) (j–o). (Adapted from Wang *et al.*¹³³ and Song *et al.*¹³⁹ with permission.)

human iPSC-derived MSCs (hiPSC-MSCs) were either pre-osteinduced for 2 weeks (OS-hiPSC-MSCs) or transduced with BMP2 (BMP2-hiPSC-MSCs) to enhance their osteogenic capacity.¹³³ The cells were then encapsulated in rapidly degradable alginate microbeads. The microbeads were mixed with CPC paste at a ratio of 1:1 and filled into cranial defects in nude rats.¹³³ The results showed that the cells maintained good viability inside the microbeads after

injection. Once the CPC set to form a scaffold, the cells were released as early as 3 days and demonstrated the up-regulation of osteogenic markers and bone mineral deposition. Cell-encapsulated groups produced greater amounts of new bone area *in vivo*, with $22.5\% \pm 7.6\%$, $38.9\% \pm 18.4\%$, and $44.7\% \pm 22.8\%$ for the CPC-hiPSC-MSC, CPC-OS-hiPSC-MSC, and CPC-BMP2-hiPSC-MSC groups, respectively, compared to that for the non-cell CPC

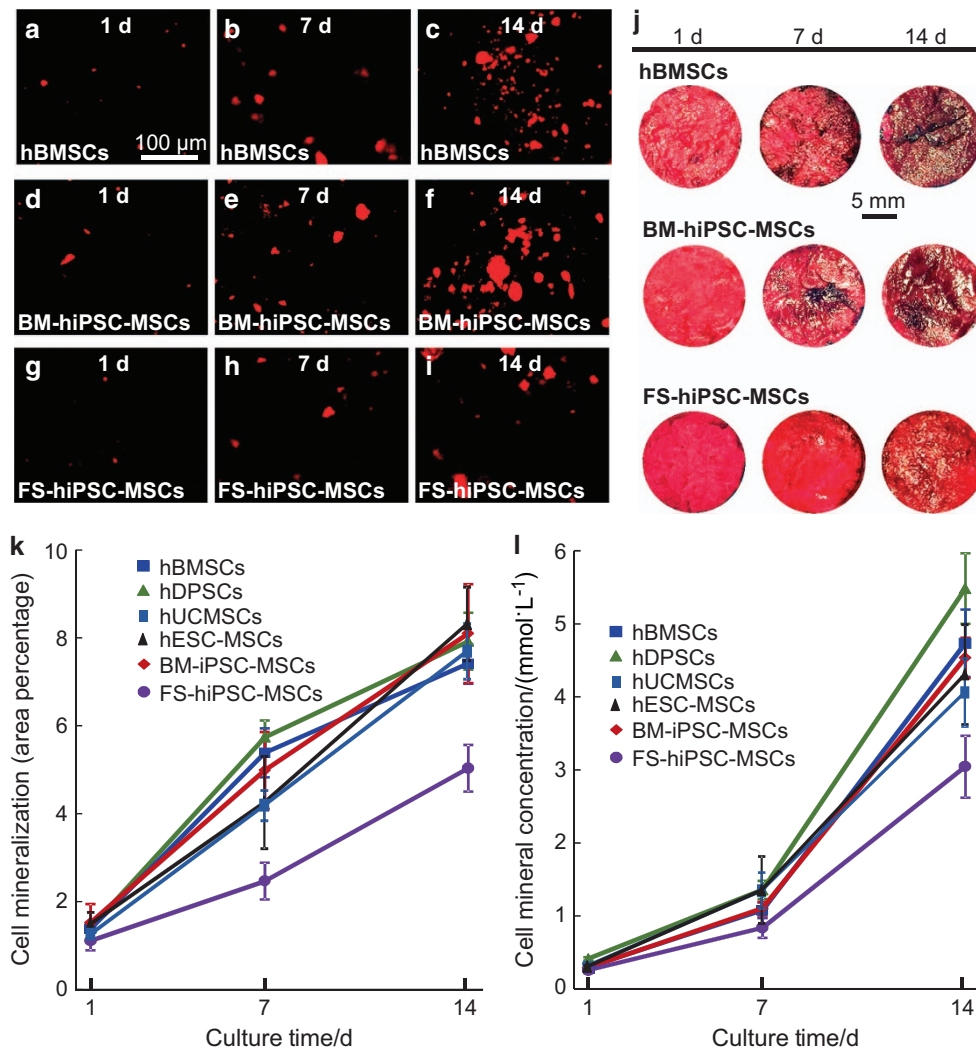


Figure 4. Synthesis of bone minerals by encapsulated stem cells. Images of (a–c) hBMSCs, (d–f) BM-hiPSC-MSCs, and (g–i) FS-hiPSC-MSCs stained with Xylenol orange (images of hESC-MSCs, hUCMSCs, and hDPSCs are similar to those of hBMSCs). (j) ARS staining of hBMSCs, BM-hiPSC-MSCs and FS-hiPSC-MSCs in CPC-CAF (images of hESC-MSCs, hUCMSCs, hDPSCs are similar to those of hBMSCs). (k) Xylenol orange mineral staining area (mean \pm s.d.; $n=6$). (l) ARS mineral concentration synthesized by cells in CPC-CAF (mean \pm s.d.; $n=6$). ARS: Alizarin red S, CAF: cell-encapsulating alginate–fibrin fibers. (Adapted from Wang *et al.*^{137–138} with permission.)

control group ($15.6\% \pm 11.2\%$) at 12 weeks.¹³³ Furthermore, the incorporation of cells accelerated the resorption of the CPC scaffold. The amount of residual CPC in the CPC-BMP2-hiPSC-MSC group was sevenfold less than that in the CPC control.¹³³

Recently, rapidly degradable hydrogel fibers were developed for cell encapsulation and delivery.¹³⁴ Encapsulation of cells inside microfibers possesses several advantages over microbeads. (1) Microfibers are easily fabricated by using a simple needle extrusion/external gelation method. To generate microbeads, air injection and electronic injection are needed to break up alginate droplets to form microbeads in sizes of several hundred microns.¹³⁵ The air flow or electrostatic force during

microbead formation may impose harsh shearing forces on the cells. Furthermore, the air flow forms “tails” on the microbeads, which may cause an immune response *in vivo*.¹³⁵ (2) Microfibers with diameters of several hundred microns and millimeter-scale lengths are relatively easy to handle. (3) Microfibers provide more space for cellular self-assembly, through which living cells organize into functional units, allowing cells to grow, migrate and differentiate in the extracellular matrix.¹³⁶ (4) Long microfibers form long macroporous channels with interconnectivity upon alginate degradation inside CPCs, while microbeads only form spherical pores with limited interconnectivity. These long channels improve osteoconductivity and nutrient and waste exchange of the scaffold. (5) Long microfibers

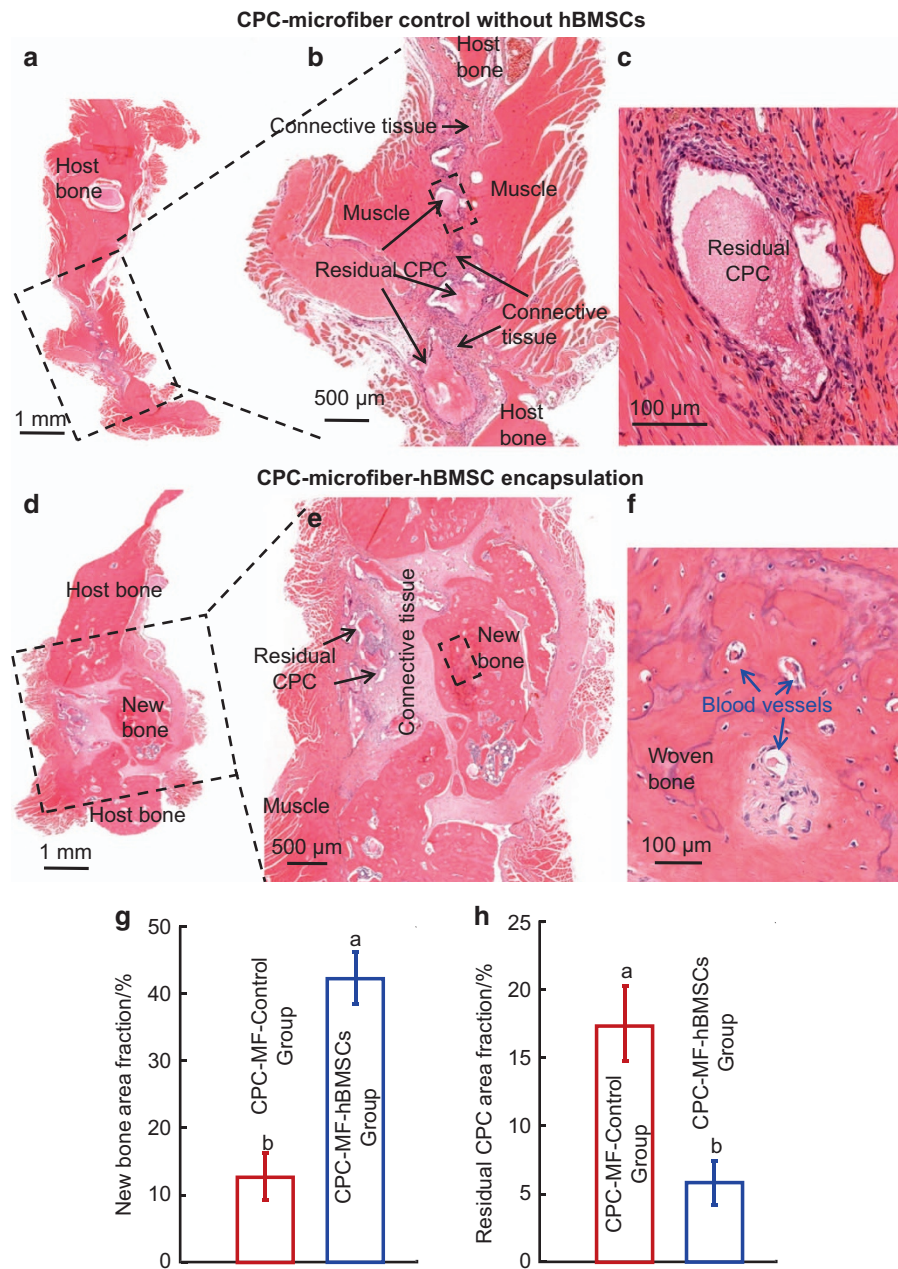


Figure 5. Representative h&e images at 12 weeks after surgery with the CPC-microfiber control group (a–c) and the CPC-microfiber-hBMSCs group (d–f) as well as quantification of the new bone area fraction (g) and residual CPC area fraction (h). Bone bridging was achieved in rat critical-sized mandibular defects in the CPC-microfiber-hBMSC group. The defect was closed with newly formed bone. (b) and (c), (e) and (f) are high-magnification images. Bars with dissimilar letters indicate significantly different values ($P < 0.05$). Each value is the mean \pm SD ($n = 6$). MF: microfibers. (Adapted from Song *et al.*¹³⁹ with permission.)

potentially facilitate the formation of blood vessels in CPCs for bone engineering via co-seeding of endothelial cells and osteoblasts.

Recent studies have encapsulated six types of stem cells, specifically hBMSCs, human dental pulp stem cells (hDPSCs), hUCMSCs, hESC-MSCs, and hiPSC-MSCs derived from bone marrow (BM-hiPSC-MSCs) and foreskin (FS-hiPSC-MSCs), in hydrogel microfibers and then delivered

them inside an injectable CPC.^{126–127} The CPC paste encapsulating the stem cells was fully injectable under a small injection force, and the injection exerted no harmful effects on cell viability.¹³⁷ The porosity of the microfiber-CPC construct was 62%.¹³⁸ All six types of cells proliferated well and differentiated down the osteogenic lineage. hUCMSCs, hESC-MSCs, hDPSCs, BM-hiPSC-MSCs and hBMSCs exhibited high ALP, RUNX2, COL1A1, and OC

gene expression. Cell-synthesized bone minerals increased with time, with no significant differences among hUCMSCs, hESC-MSCs, hDPSCs, BM-hiPSC-MSCs and hBMSCs, indicating good bone regeneration potential similar to gold-standard hBMSCs.^{137–138} However, FS-hiPSC-MSCs were inferior in terms of osteogenic differentiation compared to other cell types (Figure 4).¹³⁸ In another *in vivo* study, an hBMSC-encapsulated microfiber-CPC paste was applied to repair rat cranial defects,¹³⁸ and the hBMSC-encapsulated microfiber-CPC tissue engineering construct exhibited a robust capacity for bone regeneration. At 12 weeks, an osseous bridge in the rat mandibular defect was observed in the CPC-microfiber-hBMSCs group with a new bone area fraction of $42.1\% \pm 7.8\%$, which was threefold greater than that of the control group (Figure 5).¹³⁹ Therefore, these results demonstrate that injectable hydrogel microfiber-CPC paste is a promising carrier for cell delivery and greatly enhances bone regeneration *in vivo*.

Drug delivery

The non-exothermic setting reaction and the intrinsic porosity of CPCs allow the incorporation of drugs and biologically active molecules with low risk of thermal denaturalization or loss of activity during preparation or implantation.¹⁹ For drug incorporation into CPCs, the drug is simply mixed with either the liquid or solid components of the cement.¹⁴⁰ Alternatively, it is added by adsorption onto the pre-set scaffold¹⁴¹ or incorporated into polymeric microspheres or microfibers before blending with CPC paste.¹⁴² Several factors influence the loading and release of therapeutic substances. These include the microstructure, porosity and surface area of the CPCs, the way in which the drug is incorporated into the CPCs, and the interaction between the drug and the CPC matrix.^{19,143} CPCs have been used as drug carriers for antibiotics¹⁴⁴ as well as anti-cancer,¹⁴⁵ anti-inflammatory,¹⁴⁶ and anti-resorptive (anti-osteoporotic) drugs.¹⁴⁷ CPCs have also been used as drug carriers for therapeutically active proteins or growth factors that foster local bone generation.¹⁴⁸ Recently, ionically modified CPCs (for example, with Sr^{2+} , SiO_4^{4-} , Zn^{2+} , Mg^{2+}) with the capability of influencing bone modeling and remodeling processes were investigated.^{115,149–150} For additional details, readers are referred to a review on the use of CPCs for drug delivery.¹⁹ Of note, the incorporation of the second phase of a degradable carrier into CPCs for drug delivery is beneficial for a more sustained release than directly loading the drugs into CPCs.¹⁴⁸ For this purpose, gelatin microspheres,¹⁵¹ PLGA microparticles,¹⁵² bioactive glass,¹⁴⁸ and chitosan/dextran sulfate microparticles¹⁵³ have been used in CPCs to deliver drugs with tailored degradation rates to control the release profiles.

Vascularized CPC scaffolds

Adequate and rapid vascularization is essential for successful bone regeneration. Failure of the bone healing process, including delayed healing or non-unions, is often attributable to a lack of adequate vascularization.¹⁵⁴ Furthermore, vascularization is critical for the viability of seeded cells in the scaffold. If the distance between cells and the nearest capillary network is greater than 100–200- μm , which exceeds the diffusion or perfusion limits of nutrients and oxygen, the viability of the seeded cells is compromised.⁸⁹

Improvement in CPC vascularization is stimulated by modifications to the material itself. Physical features such as porosity and pore sizes are known to impact vascularization.^{155–156} To this end, a study fabricated a self-setting CPC composite with gelatin fibers to create interconnected hollow channels in the CPC after dissolution of the gelatin fibers.¹⁵⁷ *In vivo* subcutaneous implantation showed that the resulting channels in CPC indeed facilitated vascular infiltration into the construct.¹⁵⁷ In addition, different channel sizes induced different vascularization behaviors *in vivo*. Channels with a 250- μm diameter increased the expression of the representative angiogenic factors HIF1 α , PLGF and migration factor CXCR4, which induce the formation of small vessels. Channels with a larger diameter of 500 μm enhanced VEGF expression, which induces the development of large vessels. More HIF1 α -positive cells were found in the interconnected intersections of several channels, indicating high levels of sprouting and vasculogenesis potential under hypoxic conditions.¹⁵⁷ While the majority of research has focused on modifying the physical features of CPCs to improve vascularization, chemical features, such as the release of ionic calcium and phosphate, have also been suggested to play a role in regulating vascularization.¹⁵⁸ In a recent study, CPCs were coated with a graphene oxide-copper nanocomposite with the rationale that the oxygen-containing functional groups in graphene oxide would provide more binding sites for serum proteins and thereby enhance initial cell adhesion and other bioactivities.¹⁵⁹ When incubated with rat BMSCs, CPCs with the novel graphene oxide-copper nanocomposite coating activated Hif-1 α and further enhanced the expression of VEGF and BMP-2 via the Erk1/2 signaling pathway. Indeed, an *in vivo* study found more blood vessel volume and bone regeneration in the coated-CPC group.¹⁵⁹ However, the mechanism underlying vascularization and the impact on bone regeneration efficacy via CPCs require additional experiments, particularly *in vivo* studies.

From a biological point of view, angiogenic growth factors, stem cells and vessel-forming cells are highly promising approaches to promote vascularization. A recent study investigated the use of autologous BMSCs in

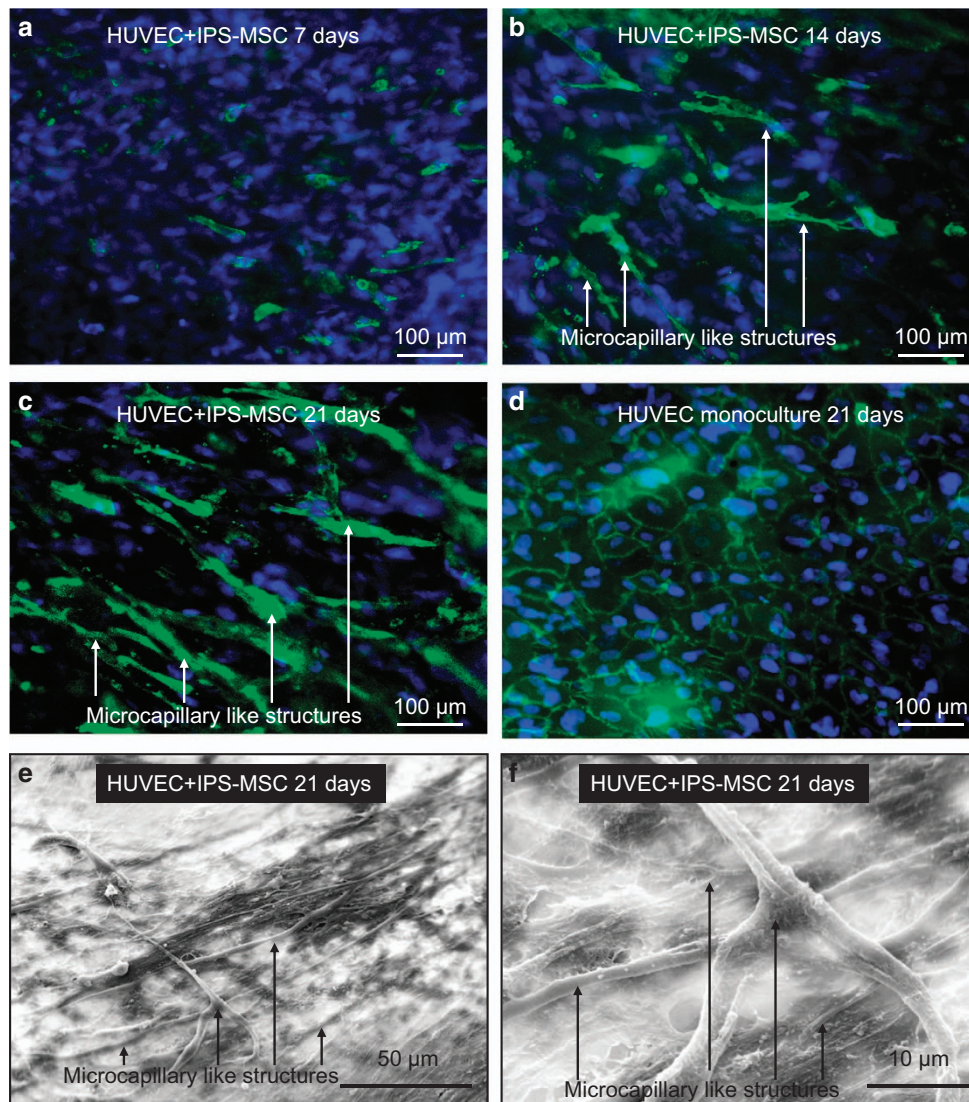


Figure 6. Formation of microcapillary-like structures by HUVECs and hiPSC-MSCs co-cultured on CPC scaffolds at 21 days (a-c). HUVECs were identified by immunostaining with the endothelial marker PECAM1 in green on the cell membrane, and nuclei were stained with DAPI in blue. hiPSC-MSCs were identified by nuclei counterstained with DAPI in blue but lacking green staining on the cell membrane. Microcapillary-like structures increased with culture time. d shows the HUVEC monoculture control group, which exhibited no evidence of vascular-like structures. Representative SEM images of microcapillary-like structures via the co-culture system (e,f). (f) A higher magnification image of the image in e. (Adapted from Liu *et al.*¹⁶⁵ with permission.)

combination with autologous platelet-rich plasma (PRP) delivered via a macroporous CPC to regenerate large bone defects in minipigs.¹⁶⁰ The CPC-BMSC-PRP group generated twofold more new bone and twofold higher blood vessel density compared to those of the macroporous CPC control at 12 weeks.¹⁶⁰ In addition, recombinant growth factors and cell signaling molecules are alternatives to autologous growth factors that provide more flexible and delicate control over the dose and factors to be incorporated. Several studies have loaded dual agents, specifically BMPs and VEGF, in a single CPC scaffold, which demonstrated excellent angiogenic

activity *in vitro* and *in vivo*.^{161–162} In addition to using growth factors, CPC pre-vascularization *in vitro* was investigated.¹⁶³ In this method, vessel-forming cells were co-seeded with bone-forming cells on the engineered tissue construct to form microvascular structures before implantation *in vivo*. The co-culture of human osteoblasts and human umbilical vein endothelial cells (HUVECs) on gas-foaming macroporous CPCs *in vitro* successfully generated microcapillary-like structures and elevated the expression of angiogenic and osteogenic markers.¹⁶³ Furthermore, the beneficial effects of co-culture were amplified by using an Arg-Gly-Asp (RGD)

modification for the CPC scaffold.¹⁶⁴ Similarly, the co-culture of hiPSC-MSCs and HUVECs on a macroporous CPC *in vitro* also generated microcapillary-like structures (Figure 6).¹⁶⁵ In an animal study, HUVECs were co-cultured with four types of stem cells, specifically hUCMSCs, hBMSCs, hiPSC-MSCs and hESC-MSCs, on CPCs and then implanted in an 8-mm critical cranial bone defect in rats for 12 weeks.¹⁶⁶ Microcapillary-like structures were successfully

formed on CPCs *in vitro* in all four co-culture groups. New bone formation and the blood vessel densities of the co-cultured groups *in vivo* were much greater than that of the CPC control without cell seeding or the CPC-BMSCs group without co-culture ($P < 0.05$).¹⁶⁶ These results demonstrated the promise of co-culture and CPC pre-vascularization to greatly enhance osteogenesis and angiogenesis *in vivo*.

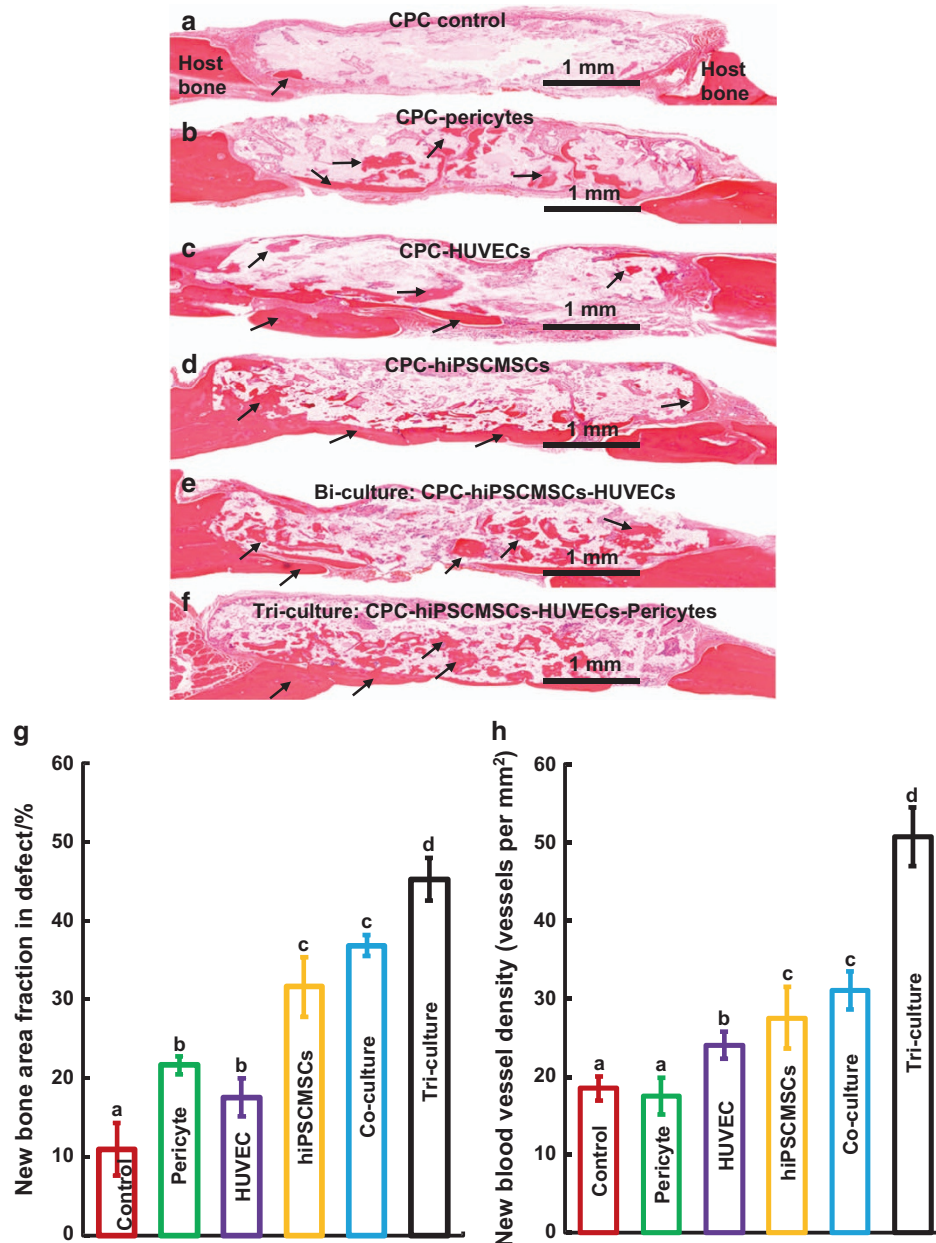


Figure 7. Representative h&e images at 12 weeks after the implantation of CPC scaffolds generated utilizing different pre-vascularization strategies in rat cranial bone defects. Mineralized new bone is stained in red (black arrows). The white area is attributable to slight detachment of the tissue. The dura is at the bottom. Cell-seeded groups had more new bone than the CPC control. Much higher amounts of new bone formed in the tri-culture group. Histomorphometric analysis of the fraction of new bone (g) and new blood vessel density (h). The tri-culture group had the greatest amount of new bone and new blood vessel density among all groups ($P < 0.05$). Each value represents the mean \pm sd ($n = 6$). Dissimilar letters indicated significantly different values ($P < 0.05$). (Adapted from Zhang *et al.*¹⁶⁹ with permission.)

For successful bone regeneration, it is important to establish vascularization in a timely manner, but the stabilization of such a vascular network is of similar importance, although it is often neglected. Angiogenesis without vessel maturation produces abnormal, defective blood vessels that are prone to regression.¹⁶⁷ Perivascular cells such as pericytes play important roles in the stabilization and maturation of blood vessels by guiding the developing vessels to respond to angiogenic stimuli.¹⁶⁸ Enlightened by this fact, further improvement of the pre-vascularization strategy with the addition of pericytes was attempted.¹⁶⁹ A tri-culture system comprising hiPSC-MSCs, HUVECs and pericytes was developed to pre-vascularize the CPC scaffolds.¹⁶⁹ Both the bi-culture and tri-culture groups exhibited the formation of vessel-like structures *in vitro*, greatly elevated levels of angiogenic and osteogenic markers, and bone matrix mineralization. After implantation in a rat model with a cranial bone defect for 12 weeks, the tri-culture group demonstrated much higher amounts of new bone than the bi-culture and monoculture groups and the CPC control (Figure 7).¹⁶⁹ The substantial increase in bone formation in the tri-culture group was likely related to enhanced vascularization and the stabilization and maturation of blood vessels.

In vivo pre-vascularization is also achieved using a surgical method involving the implantation of a scaffold into a well-vascularized and easily accessible body tissue such as a subcutaneous pocket or a muscle pouch. Microvascular structures are formed as a result of invasion and outgrowth of the surrounding host microvasculature.^{170–171} After the completion of pre-vascularization, the tissue construct is harvested and grafted into the defect site, where the preformed microvessels inside the construct inosculate and anastomose with the host blood vessels. The disadvantages of this approach are obvious: the invasive nature of the surgery, higher cost, and a relatively longer treatment process. Therefore, new tissue engineering methods utilizing CPC scaffolds with co-culture and tri-culture represent exciting alternative strategies that warrant further research for continued improvement to achieve wide clinical applications.

CONCLUSIONS

Due to their injectability, bioactivity and biocompatibility, CPCs are highly promising for bone tissue engineering applications and are used as scaffolds and carriers to deliver stem cells, drugs and growth factors. CPCs are either used as pre-set scaffolds or injectable pastes. 3D printing is a promising technology for fabricating CPC scaffolds with a high degree of accuracy and is used to develop intricately detailed biomimetic structures that are not achievable via traditional manufacturing methods. 3D

printing has the potential to facilitate the next generation of smart and functional CPCs. Furthermore, with recent advances in tissue engineering, a new emphasis on “tissue regeneration by natural tissues” instead of “tissue replacement by biomaterials” has been proposed. Thus, CPCs with excellent biological interactions, such as osteoconductivity, osteoinductivity, biodegradability and bioactivity, are promising to meet this need. CPC composite constructs and hybrid systems involving the incorporation of cells, growth factors, bioactive molecules, bioinorganics, polymers, and bioactive glass are likely to yield favorable bone regenerative outcomes and greatly widen the clinical applications of CPCs. In addition, the co-culture and tri-culture of various tailored cell types with CPC scaffolds offer exciting potential for vascularization in bone tissue regeneration, which is especially important for treating large-sized bone defects. Further studies are needed to realize these promises and understand the underlying mechanisms to further the development of tissue engineering and regenerative medicine.

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Competing interests

The authors declare no conflict of interest.

References

- 1 Chow LC, Takagi S. A natural bone cement-A laboratory novelty led to the development of revolutionary new biomaterials. *J Res Natl Inst Stand Technol* 2001; **106**: 1029–1033.
- 2 Brown WE, Chow LC. A new calcium phosphate, water setting cement. Brown PW (Ed). *Cements Res Progress*. Westerville: American Ceramic Society. 1986, 352–379.
- 3 Brown W, Chow LC. A new calcium phosphate setting cement. *J Dent Res* 1983; **63**: 672–679.
- 4 Brown WE. Solubilities of phosphates and other sparingly soluble compounds. Griffith EJ, Beeton A, Spencer JM *et al.* (eds). *Environmental phosphorus handbook*. New York: John Wiley & Sons. 1973, 203–239.
- 5 Friedman CD, Costantino PD, Takagi S *et al.* Bone Source hydroxyapatite cement: a novel biomaterial for craniofacial skeletal tissue engineering and reconstruction. *J Biomed Mater Res* 1998; **43**: 428–432.
- 6 Constantz BR, Ison IC, Fulmer MT *et al.* Skeletal repair by *in situ* formation of the mineral phase of bone. *Science* 1995; **267**: 1796–1799.
- 7 Ginebra MP, Fernandez E, De Maeyer EA *et al.* Setting reaction and hardening of an apatitic calcium phosphate cement. *J Dent Res* 1997; **76**: 905–912.
- 8 Constantz BR, Barr BM, Ison IC *et al.* Histological, chemical, and crystallographic analysis of four calcium phosphate cements in different rabbit osseous sites. *J Biomed Mater Res A* 1998; **43**: 451–461.

- 9 Knaack D, Goad MEP, Aiolo M *et al*. Resorbable calcium phosphate bone substitute. *J Biomed Mater Res A* 1998; **43**: 399–409.
- 10 Miyamoto Y, Ishikawa K, Takechi M *et al*. Histological and compositional evaluations of three types of calcium phosphate cements when implanted in subcutaneous tissue immediately after mixing. *J Biomed Mater Res A* 1999; **48**: 36–42.
- 11 Barralet JE, Gaunt T, Wright AJ *et al*. Effect of porosity reduction by compaction on compressive strength and microstructure of calcium phosphate cement. *J Biomed Mater Res* 2002; **63**: 1–9.
- 12 Yokoyama A, Yamamoto S, Kawasaki T *et al*. Development of calcium phosphate cement using chitosan and citric acid for bone substitute materials. *Biomaterials* 2002; **23**: 1091–1101.
- 13 Gisepp A, Wieling R, Bohner M *et al*. Resorption patterns of calcium-phosphate cements in bone. *J Biomed Mater Res A* 2003; **66**: 532–540.
- 14 Ehara A, Ogata K, Imazato S *et al*. Effects of alpha-TCP and TetCP on MC3T3-E1 proliferation, differentiation and mineralization. *Biomaterials* 2003; **24**: 831–836.
- 15 Yuasa T, Miyamoto Y, Ishikawa K *et al*. Effects of apatite cements on proliferation and differentiation of human osteoblasts *in vitro*. *Biomaterials* 2004; **25**: 1159–1166.
- 16 Apelt D, Theiss F, El-Warrak AO *et al*. *In vivo* behavior of three different injectable hydraulic calcium phosphate cements. *Biomaterials* 2004; **25**: 1439–1451.
- 17 Fernandez E, Vlad MD, Gel MM *et al*. Modulation of porosity in apatitic cements by the use of alpha-tricalcium phosphate-calcium sulphate dihydrate mixtures. *Biomaterials* 2005; **26**: 3395–3404.
- 18 del Real RP, Wolke JG, Vallet-Regi M, Jansen JA. A new method to produce macropores in calcium phosphate cements. *Biomaterials* 2002; **23**: 3673–3680.
- 19 Ginebra MP, Canal C, Espanol M *et al*. Calcium phosphate cements as drug delivery materials. *Adv Drug Deliv Rev* 2012; **64**: 1090–1110.
- 20 Chow LC. Calcium phosphate cements: chemistry, properties, and applications. *MRS Proc* 1999; **599**: 27.
- 21 Xu HH, Weir MD, Burguera EF *et al*. Injectable and macroporous calcium phosphate cement scaffold. *Biomaterials* 2006; **27**: 4279–4287.
- 22 Ginebra MP, Traykova T, Planell JA. Calcium phosphate cements as bone drug delivery systems: a review. *J Control Release* 2006; **113**: 102–110.
- 23 An J, Wolke JG, Jansen JA *et al*. Influence of polymeric additives on the cohesion and mechanical properties of calcium phosphate cements. *J Mater Sci Mater Med* 2016; **27**: 58.
- 24 Xu HH, Simon CG Jr. Fast setting calcium phosphate-chitosan scaffold: mechanical properties and biocompatibility. *Biomaterials* 2005; **26**: 1337–1348.
- 25 Cherng A, Takagi S, Chow LC. Effects of hydroxypropyl methylcellulose and other gelling agents on the handling properties of calcium phosphate cement. *J Biomed Mater Res A* 1997; **35**: 273–277.
- 26 Fukase Y, Eanes ED, Takagi S *et al*. Setting reactions and compressive strengths of calcium phosphate cements. *J Dent Res* 1990; **69**: 1852–1856.
- 27 Hesarakis S, Zamanian A, Moztarzadeh F. The influence of the acidic component of the gas-foaming porogen used in preparing an injectable porous calcium phosphate cement on its properties: acetic acid versus citric acid. *J Biomed Mater Res B Appl Biomater* 2008; **86**: 208–216.
- 28 Ginebra MP, Driessens FC, Planell JA. Effect of the particle size on the micro and nanostructural features of a calcium phosphate cement: a kinetic analysis. *Biomaterials* 2004; **25**: 3453–3462.
- 29 Weir MD, Xu HH. Osteoblastic induction on calcium phosphate cement-chitosan constructs for bone tissue engineering. *J Biomed Mater Res A* 2010; **94**: 223–233.
- 30 Mestres G, Le Van C, Ginebra MP. Silicon-stabilized α -tricalcium phosphate and its use in a calcium phosphate cement: characterization and cell response. *Acta Biomater* 2012; **8**: 1169–1179.
- 31 Lanao RPF, Leeuwenburgh SC, Wolke JG *et al*. Bone response to fast-degrading, injectable calcium phosphate cements containing PLGA microparticles. *Biomaterials* 2011; **32**: 8839–8847.
- 32 Theiss F, Apelt D, Brand B *et al*. Biocompatibility and resorption of a brushite calcium phosphate cement. *Biomaterials* 2005; **26**: 4383–4394.
- 33 Noetzel J, Özer K, Reissbauer B-H *et al*. Tissue responses to an experimental calcium phosphate cement and mineral trioxide aggregate as materials for furcation perforation repair: a histological study in dogs. *Clin Oral Investig* 2006; **10**: 77.
- 34 Ambard AJ, Mueninghoff L. Calcium phosphate cement: review of mechanical and biological properties. *J Prosthodont* 2006; **15**: 321–328.
- 35 Canal C, Ginebra MP. Fibre-reinforced calcium phosphate cements: a review. *J Mech Behav Biomed Mater* 2011; **4**: 1658–1671.
- 36 Zhang J, Liu W, Schnitzler V *et al*. Calcium phosphate cements for bone substitution: chemistry, handling and mechanical properties. *Acta Biomater* 2014; **10**: 1035–1049.
- 37 Ginebra MP, Espanol M, Montufar EB *et al*. New processing approaches in calcium phosphate cements and their applications in regenerative medicine. *Acta Biomater* 2010; **6**: 2863–2873.
- 38 Bohner M, Gbureck U, Barralet JE. Technological issues for the development of more efficient calcium phosphate bone cements: a critical assessment. *Biomaterials* 2005; **26**: 6423–6429.
- 39 Verron E, Khairoun I, Guicheux J *et al*. Calcium phosphate biomaterials as bone drug delivery systems: a review. *Drug Discov Today* 2010; **15**: 547–552.
- 40 Ginebra M-P, Traykova T, Planell JA. Calcium phosphate cements: competitive drug carriers for the musculoskeletal system? *Biomaterials* 2006; **27**: 2171–2177.
- 41 Perez RA, Kim HW, Ginebra MP. Polymeric additives to enhance the functional properties of calcium phosphate cements. *J Tissue Eng* 2012; **3**: 2041731412439555.
- 42 Ishikawa K. Calcium phosphate cement. Ben-Nissan B (ed.). *Advances in Calcium Phosphate Biomaterials*. Berlin: Springer. 2014, 199–227.
- 43 Chow LC. Next generation calcium phosphate-based biomaterials. *Dent Mater J* 2009; **28**: 1–10.
- 44 Murphy WL, Dennis RG, Kileny JL *et al*. Salt fusion: an approach to improve pore interconnectivity within tissue engineering scaffolds. *Tissue Eng* 2002; **8**: 43–52.
- 45 Smith BT, Santoro M, Grosfeld EC *et al*. Incorporation of fast dissolving glucose porogens into an injectable calcium phosphate cement for bone tissue engineering. *Acta Biomater* 2017; **50**: 68–77.
- 46 Assaad E, Maire M, Lerouge S. Injectable thermosensitive chitosan hydrogels with controlled gelation kinetics and enhanced mechanical resistance. *Carbohydr Polym* 2015; **130**: 87–96.
- 47 Guvendiren M, Molde J, Soares RMD *et al*. Designing biomaterials for 3D printing. *ACS Biomater Sci Eng* 2016; **2**: 1679–1693.
- 48 Do AV, Khorsand B, Geary SM *et al*. 3D printing of scaffolds for tissue regeneration applications. *Adv Healthc Mater* 2015; **4**: 1742–1762.
- 49 Bose S, Vahabzadeh S, Bandyopadhyay A. Bone tissue engineering using 3D printing. *Mater Today* 2013; **16**: 496–504.
- 50 Trombetta R, Inzana JA, Schwarz EM *et al*. 3D printing of calcium phosphate ceramics for bone tissue engineering and drug delivery. *Ann Biomater Eng* 2017; **45**: 23–44.
- 51 Butscher A, Bohner M, Roth C *et al*. Printability of calcium phosphate powders for three-dimensional printing of tissue engineering scaffolds. *Acta Biomater* 2012; **8**: 373–385.

- 52 Vacanti JP, Cima LG, Cima MJ. Vascularized tissue regeneration matrices formed by solid free form fabrication techniques. Google Patents. 2000; US Patent 6139574.
- 53 Spath S, Seitz H. Influence of grain size and grain-size distribution on workability of granules with 3D printing. *Int J Adv Manuf Technol* 2014; 70: 135–144.
- 54 Butscher A, Böhner M, Doebelin N *et al*. Moisture based three-dimensional printing of calcium phosphate structures for scaffold engineering. *Acta Biomater* 2013; 9: 5369–5378.
- 55 Zhou Z, Buchanan F, Mitchell C *et al*. Printability of calcium phosphate: calcium sulfate powders for the application of tissue engineered bone scaffolds using the 3D printing technique. *Mater Sci Eng C Mater Biol Appl* 2014; 38: 1–10.
- 56 Bertol LS, Schabbach R, dos Santos LAL. Dimensional evaluation of patient-specific 3D printing using calcium phosphate cement for craniofacial bone reconstruction. *J Biomater Appl* 2017; 31: 799–806.
- 57 Butscher A, Böhner M, Doebelin N *et al*. New depowdering-friendly designs for three-dimensional printing of calcium phosphate bone substitutes. *Acta Biomater* 2013; 9: 9149–9158.
- 58 Farzadi A, Solati-Hashjin M, Asadi-Eydivand M *et al*. Effect of layer thickness and printing orientation on mechanical properties and dimensional accuracy of 3D printed porous samples for bone tissue engineering. *PLoS ONE* 2014; 9: e108252.
- 59 Lode A, Meissner K, Luo Y *et al*. Fabrication of porous scaffolds by three-dimensional plotting of a pasty calcium phosphate bone cement under mild conditions. *J Tissue Eng Regen Med* 2014; 8: 682–693.
- 60 Luo Y, Lode A, Wu C, Gelinsky M. 3D plotting of bioceramic scaffolds under physiological conditions for bone tissue engineering. In: Wu C, Chang J, Xiao Y (eds.). *Advanced Bioactive Inorganic Materials for Bone Regeneration and Drug Delivery*. Boca Raton: CRC Press. 2013, 83–103.
- 61 Li C, Gao L, Chen F *et al*. Fabrication of mesoporous calcium silicate/calcium phosphate cement scaffolds with high mechanical strength by freeform fabrication system with micro-droplet jetting. *Journal of Materials Science* 2015; 50: 7182–7191.
- 62 Maazouz Y, Montufar EB, Guillem-Marti J *et al*. Robocasting of biomimetic hydroxyapatite scaffolds using self-setting inks. *J Mater Chem B* 2014; 2: 5378–5386.
- 63 Marques CF, Perera FH, Marote A *et al*. Biphasic calcium phosphate scaffolds fabricated by direct write assembly: mechanical, antimicrobial and osteoblastic properties. *J Eur Ceram Soc* 2017; 37: 359–368.
- 64 Ahlfeld T, Akkineni AR, Forster Y *et al*. Design and fabrication of complex scaffolds for bone defect healing: combined 3D plotting of a calcium phosphate cement and a growth factor-loaded hydrogel. *Ann Biomed Eng* 2017; 45: 224–236.
- 65 Billiet T, Gevaert E, De Schryver T *et al*. The 3D printing of gelatin methacrylamide cell-laden tissue-engineered constructs with high cell viability. *Biomaterials* 2014; 35: 49–62.
- 66 Li C, Jiang C, Deng Y *et al*. RhBMP-2 loaded 3D-printed mesoporous silica/calcium phosphate cement porous scaffolds with enhanced vascularization and osteogenesis properties. *Sci Rep* 2017; 7: 41331.
- 67 Wust S, Godla ME, Muller R *et al*. Tunable hydrogel composite with two-step processing in combination with innovative hardware upgrade for cell-based three-dimensional bioprinting. *Acta Biomater* 2014; 10: 630–640.
- 68 Park JY, Choi JC, Shim JH *et al*. A comparative study on collagen type I and hyaluronic acid dependent cell behavior for osteochondral tissue bioprinting. *Biofabrication* 2014; 6: 035004.
- 69 Sawkins MJ, Mistry P, Brown BN *et al*. Cell and protein compatible 3D bioprinting of mechanically strong constructs for bone repair. *Biofabrication* 2015; 7: 035004.
- 70 O'Neill R, McCarthy HO, Montufar EB *et al*. Critical review: injectability of calcium phosphate pastes and cements. *Acta Biomater* 2017; 50: 1–19.
- 71 Khairoun I, Driessens FC, Boltong MG *et al*. Addition of cohesion promoters to calcium phosphate cements. *Biomaterials* 1999; 20: 393–398.
- 72 Krebs J, Aebli N, Goss BG *et al*. Cardiovascular changes after pulmonary embolism from injecting calcium phosphate cement. *J Biomed Mater Res B Appl Biomater* 2007; 82: 526–532.
- 73 Bigi A, Bracci B, Panzavolta S. Effect of added gelatin on the properties of calcium phosphate cement. *Biomaterials* 2004; 25: 2893–2899.
- 74 Alkhraisat MH, Rueda C, Marino FT *et al*. The effect of hyaluronic acid on brushite cement cohesion. *Acta Biomater* 2009; 5: 3150–3156.
- 75 Liu W, Zhang J, Weiss P *et al*. The influence of different cellulose ethers on both the handling and mechanical properties of calcium phosphate cements for bone substitution. *Acta Biomater* 2013; 9: 5740–5750.
- 76 O'Hara RM, Dunne NJ, Orr JF *et al*. Optimisation of the mechanical and handling properties of an injectable calcium phosphate cement. *J Mater Sci Mater Med* 2010; 21: 2299–2305.
- 77 Khairoun I, Boltong MG, Driessens FCM *et al*. Some factors controlling the injectability of calcium phosphate bone cements. *J Mater Sci Mater Med* 1998; 9: 425–428.
- 78 Burguera EF, Xu HH, Sun L. Injectable calcium phosphate cement: effects of powder-to-liquid ratio and needle size. *J Biomed Mater Res B Appl Biomater* 2008; 84: 493–502.
- 79 Liao H, Walboomers XF, Habraken WJ *et al*. Injectable calcium phosphate cement with PLGA, gelatin and PTMC microspheres in a rabbit femoral defect. *Acta Biomater* 2011; 7: 1752–1759.
- 80 Yang HL, Zhu XS, Chen L *et al*. Bone healing response to a synthetic calcium sulfate/beta-tricalcium phosphate graft material in a sheep vertebral body defect model. *J Biomed Mater Res B Appl Biomater* 2012; 100: 1911–1921.
- 81 Babo PS, Carvalho PP, Santo VE *et al*. Assessment of bone healing ability of calcium phosphate cements loaded with platelet lysate in rat calvarial defects. *J Biomater Appl* 2016; 31: 637–649.
- 82 Hoekstra JW, Klijn RJ, Meijer GJ *et al*. Maxillary sinus floor augmentation with injectable calcium phosphate cements: a pre-clinical study in sheep. *Clin Oral Implants Res* 2013; 24: 210–216.
- 83 Verron E, Pissonnier ML, Lesoeur J *et al*. Vertebroplasty using bisphosphonate-loaded calcium phosphate cement in a standardized vertebral body bone defect in an osteoporotic sheep model. *Acta Biomater* 2014; 10: 4887–4895.
- 84 Shirakata Y, Setoguchi T, Machigashira M *et al*. Comparison of injectable calcium phosphate bone cement grafting and open flap debridement in periodontal intrabony defects: a randomized clinical trial. *J Periodontol* 2008; 79: 25–32.
- 85 Gumpert R, Bodo K, Spuller E *et al*. Demineralization after balloon kyphoplasty with calcium phosphate cement: a histological evaluation in ten patients. *Eur Spine J* 2014; 23: 1361–1368.
- 86 Larsson S, Bauer TW. Use of injectable calcium phosphate cement for fracture fixation: a review. *Clin Orthop Relat Res* 2002; 395: 23–32.
- 87 Jansen J, Ooms E, Verdonchot N *et al*. Injectable calcium phosphate cement for bone repair and implant fixation. *Orthop Clin North Am* 2005; 36: 89–95.
- 88 Lewis G. Injectable bone cements for use in vertebroplasty and kyphoplasty: state-of-the-art review. *J Biomed Mater Res B Appl Biomater* 2006; 76: 456–468.

- 89 Williams DF. On the mechanisms of biocompatibility. *Biomaterials* 2008; **29**: 2941–2953.
- 90 Bohner M. Reactivity of calcium phosphate cements. *J Mater Chem* 2007; **17**: 3980–3986.
- 91 Ruhe PQ, Hedberg-Dirk EL, Padron NT *et al*. Porous poly(DL-lactic-co-glycolic acid)/calcium phosphate cement composite for reconstruction of bone defects. *Tissue Eng* 2006; **12**: 789–800.
- 92 Takechi M, Miyamoto Y, Ishikawa K *et al*. Initial histological evaluation of anti-washout type fast-setting calcium phosphate cement following subcutaneous implantation. *Biomaterials* 1998; **19**: 2057–2063.
- 93 Guo H, Su J, Wei J *et al*. Biocompatibility and osteogenicity of degradable Ca-deficient hydroxyapatite scaffolds from calcium phosphate cement for bone tissue engineering. *Acta Biomater* 2009; **5**: 268–278.
- 94 Lanao RF, Hoekstra J, Wolke J *et al*. Porous calcium phosphate cement for alveolar bone regeneration. *J Tissue Eng Regen Med* 2014; **8**: 473–482.
- 95 del Real RP, Ooms E, Wolke JG *et al*. In vivo bone response to porous calcium phosphate cement. *J Biomed Mater Res A* 2003; **65**: 30–36.
- 96 Xu HHK, Carey LE, Simon CG *et al*. Premixed calcium phosphate cements: synthesis, physical properties, and cell cytotoxicity. *Dent Mater* 2007; **23**: 433–441.
- 97 Ueyama Y, Ishikawa K, Mano T *et al*. Initial tissue response to anti-washout apatite cement in the rat palatal region: comparison with conventional apatite cement. *J Biomed Mater Res A* 2001; **55**: 652–660.
- 98 Zhang J, Liu W, Gauthier O *et al*. A simple and effective approach to prepare injectable macroporous calcium phosphate cement for bone repair: syringe-foaming using a viscous hydrophilic polymeric solution. *Acta Biomater* 2016; **31**: 326–338.
- 99 Hench LL, Splinter RJ, Allen WC *et al*. Bonding mechanisms at the interface of ceramic prosthetic materials. *J Biomed Mater Res A* 1971; **5**: 117–141.
- 100 Kokubo T, Takadama H. How useful is SBF in predicting *in vivo* bone bioactivity? *Biomaterials* 2006; **27**: 2907–2915.
- 101 Bohner M, Lemaire J. Can bioactivity be tested *in vitro* with SBF solution? *Biomaterials* 2009; **30**: 2175–2179.
- 102 D'Onofrio A, Kent NW, Shahdad SA *et al*. Development of novel strontium containing bioactive glass based calcium phosphate cement. *Dent Mater* 2016; **32**: 703–712.
- 103 Sadiasa A, Sarkar SK, Franco RA *et al*. Bioactive glass incorporation in calcium phosphate cement-based injectable bone substitute for improved *in vitro* biocompatibility and *in vivo* bone regeneration. *J Biomater Appl* 2014; **28**: 739–756.
- 104 Albrektsson T, Johansson C. Osteoinduction, osteoconduction and osseointegration. *Eur Spine J* 2001; **10**: S96–S101.
- 105 Moreau JL, Xu HH. Mesenchymal stem cell proliferation and differentiation on an injectable calcium phosphate-chitosan composite scaffold. *Biomaterials* 2009; **30**: 2675–2682.
- 106 LeGeros RZ. Properties of osteoconductive biomaterials: calcium phosphates. *Clin Orthop Relat Res* 2002; **395**: 81–98.
- 107 Denry I, Kuhn LT. Design and characterization of calcium phosphate ceramic scaffolds for bone tissue engineering. *Dent Mater* 2016; **32**: 43–53.
- 108 Aparicio JL, Rueda C, Manchon A *et al*. Effect of physicochemical properties of a cement based on silicocarnotite/calcium silicate on *in vitro* cell adhesion and *in vivo* cement degradation. *Biomed Mater* 2016; **11**: 045005.
- 109 Kuang GM, Yau WP, Wu J *et al*. Strontium exerts dual effects on calcium phosphate cement: accelerating the degradation and enhancing the osteoconductivity both *in vitro* and *in vivo*. *J Biomed Mater Res A* 2015; **103**: 1613–1621.
- 110 Yuan H, Li Y, de Bruijn JD *et al*. Tissue responses of calcium phosphate cement: a study in dogs. *Biomaterials* 2000; **21**: 1283–1290.
- 111 LeGeros RZ. Calcium phosphate-based osteoinductive materials. *Chem Rev* 2008; **108**: 4742–4753.
- 112 Wu X, Itoh N, Taniguchi T *et al*. Requirement of calcium and phosphate ions in expression of sodium-dependent vitamin C transporter 2 and osteopontin in MC3T3-E1 osteoblastic cells. *Biochim Biophys Acta* 2003; **1641**: 65–70.
- 113 Barradas AM, Yuan H, van Blitterswijk CA *et al*. Osteoinductive biomaterials: current knowledge of properties, experimental models and biological mechanisms. *Eur Cell Mater* 2011; **21**: 407–429.
- 114 Habibovic P, Bassett DC, Doillon CJ *et al*. Collagen biomineralization *in vivo* by sustained release of inorganic phosphate ions. *Adv Mater* 2010; **22**: 1858–1862.
- 115 Zhang J, Ma X, Lin D *et al*. Magnesium modification of a calcium phosphate cement alters bone marrow stromal cell behavior via an integrin-mediated mechanism. *Biomaterials* 2015; **53**: 251–264.
- 116 Thein-Han W, Liu J, Tang M *et al*. Induced pluripotent stem cell-derived mesenchymal stem cell seeding on biofunctionalized calcium phosphate cements. *Bone Res* 2013; **4**: 371–384.
- 117 Song G, Habibovic P, Bao C *et al*. The homing of bone marrow MSCs to non-osseous sites for ectopic bone formation induced by osteoinductive calcium phosphate. *Biomaterials* 2013; **34**: 2167–2176.
- 118 Zhang J, Zhou H, Yang K *et al*. RhBMP-2-loaded calcium silicate/calcium phosphate cement scaffold with hierarchically porous structure for enhanced bone tissue regeneration. *Biomaterials* 2013; **34**: 9381–9392.
- 119 Lee GH, Makkar P, Paul K *et al*. Incorporation of BMP-2 loaded collagen conjugated BCP granules in calcium phosphate cement based injectable bone substitutes for improved bone regeneration. *Mater Sci Eng C* 2017; **77**: 713–724.
- 120 Dong J, Cui G, Bi L *et al*. The mechanical and biological studies of calcium phosphate cement-fibrin glue for bone reconstruction of rabbit femoral defects. *Int J Nanomedicine* 2013; **8**: 1317–1324.
- 121 Takechi M, Ninomiya Y, Ohta K *et al*. Effects of apatite cement containing atelocollagen on attachment to and proliferation and differentiation of MC3T3-E1 osteoblastic cells. *Materials* 2016; **9**: 283.
- 122 Liu R, Wu X, Li J *et al*. The promotion of bone tissue regeneration by BMP2-derived peptide P24-loaded calcium phosphate cement microspheres. *Ceram Int* 2016; **42**: 3177–3189.
- 123 Wang T, Wu D, Li Y *et al*. Substance P incorporation in calcium phosphate cement for dental alveolar bone defect restoration. *Mater Sci Eng C Mater Biol Appl* 2016; **69**: 546–553.
- 124 Sheikh Z, Abdallah MN, Hanafi AA *et al*. Mechanisms of *in vivo* degradation and resorption of calcium phosphate based biomaterials. *Materials* 2015; **8**: 7913–7925.
- 125 Lu J, Descamps M, Dejou J *et al*. The biodegradation mechanism of calcium phosphate biomaterials in bone. *J Biomed Mater Res A* 2002; **63**: 408–412.
- 126 Lanao RPF, Leeuwenburgh SC, Wolke JG *et al*. In vitro degradation rate of apatitic calcium phosphate cement with incorporated PLGA microspheres. *Acta Biomater* 2011; **7**: 3459–3468.
- 127 Grosfeld EC, Hoekstra JW, Herber RP *et al*. Long-term biological performance of injectable and degradable calcium phosphate cement. *Biomed Mater* 2016; **12**: 015009.
- 128 Lanao RPF, Sariibrahimoglu K, Wang H *et al*. Accelerated calcium phosphate cement degradation due to incorporation of glucono-delta-lactone microparticles. *Tissue Eng A* 2014; **20**: 378–388.

- 129 Schwarz KA, Leonard JN. Engineering cell-based therapies to interface robustly with host physiology. *Adv Drug Deliv Rev* 2016; **105**: 55–65.
- 130 Wang P, Zhao L, Liu J *et al*. Bone tissue engineering via nanostructured calcium phosphate biomaterials and stem cells. *Bone Res* 2014; **2**: 14017.
- 131 Wang P, Zhao L, Chen W *et al*. Stem cells and calcium phosphate cement scaffolds for bone regeneration. *J Dent Res* 2014; **93**: 618–625.
- 132 Villalona GA, Udelsman B, Duncan DR *et al*. Cell-seeding techniques in vascular tissue engineering. *Tissue Eng Part B Rev* 2010; **16**: 341–350.
- 133 Wang P, Song Y, Weir MD *et al*. A self-setting iPSMSC-alginate-calcium phosphate paste for bone tissue engineering. *Dent Mater* 2016; **32**: 252–263.
- 134 Mihaila SM, Popa EG, Reis RL *et al*. Fabrication of endothelial cell-laden carrageenan microfibers for microvascularized bone tissue engineering applications. *Biomacromolecules* 2014; **15**: 2849–2860.
- 135 Kang A, Park J, Ju J *et al*. Cell encapsulation via microtechnologies. *Biomaterials* 2014; **35**: 2651–2663.
- 136 Raof NA, Padgen MR, Gracias AR *et al*. One-dimensional self-assembly of mouse embryonic stem cells using an array of hydrogel microstrands. *Biomaterials* 2011; **32**: 4498–4505.
- 137 Wang L, Wang P, Weir MD *et al*. Hydrogel fibers encapsulating human stem cells in an injectable calcium phosphate scaffold for bone tissue engineering. *Biomed Mater* 2016; **11**: 065008.
- 138 Wang L, Zhang C, Li C *et al*. Injectable calcium phosphate with hydrogel fibers encapsulating induced pluripotent, dental pulp and bone marrow stem cells for bone repair. *Mater Sci Eng C Mater Biol Appl* 2016; **69**: 1125–1136.
- 139 Song Y, Zhang C, Wang P *et al*. Engineering bone regeneration with novel cell-laden hydrogel microfiber-injectable calcium phosphate scaffold. *Mater Sci Eng C Mater Biol Appl* 2017; **75**: 895–905.
- 140 Bohner M, Lemaire J, Van Landuyt P *et al*. Gentamicin-loaded hydraulic calcium phosphate bone cement as antibiotic delivery system. *J Pharm Sci* 1997; **86**: 565–572.
- 141 Le Nihouannen D, Hacking SA, Gbureck U *et al*. The use of RANKL-coated brushite cement to stimulate bone remodelling. *Biomaterials* 2008; **29**: 3253–3259.
- 142 Farokhi M, Mottaghtalab F, Ai J *et al*. Sustained release of platelet-derived growth factor and vascular endothelial growth factor from silk/calcium phosphate/PLGA based nanocomposite scaffold. *Int J Pharm* 2013; **454**: 216–225.
- 143 Parent M, Baradari H, Champion E *et al*. Design of calcium phosphate ceramics for drug delivery applications in bone diseases: a review of the parameters affecting the loading and release of the therapeutic substance. *J Control Release* 2017; **252**: 1–17.
- 144 Ghosh S, Wu V, Pernal S *et al*. Self-setting calcium phosphate cements with tunable antibiotic release rates for advanced antimicrobial applications. *ACS Appl Mater Interfaces* 2016; **8**: 7691–7708.
- 145 Lopez-Heredia MA, Kamphuis GJ, Thune PC *et al*. An injectable calcium phosphate cement for the local delivery of paclitaxel to bone. *Biomaterials* 2011; **32**: 5411–5416.
- 146 Renaudin G, Laquerriere P, Filinchuk Y *et al*. Structural characterization of sol-gel derived Sr-substituted calcium phosphates with anti-osteoporotic and anti-inflammatory properties. *J Mater Chem* 2008; **18**: 3593–3600.
- 147 Gong T, Chen Y, Zhang Y *et al*. Osteogenic and anti-osteoporotic effects of risedronate-added calcium phosphate silicate cement. *Biomed Mater* 2016; **11**: 045002.
- 148 Schumacher M, Reither L, Thomas J *et al*. Calcium phosphate bone cement/mesoporous bioactive glass composites for controlled growth factor delivery. *Biomater Sci* 2017; **5**: 578–588.
- 149 Zhu H, Guo D, Qi W *et al*. Development of Sr-incorporated biphasic calcium phosphate bone cement. *Biomed Mater* 2017; **12**: 015016.
- 150 Li X, Sogo Y, Ito A *et al*. The optimum zinc content in set calcium phosphate cement for promoting bone formation *in vivo*. *Mater Sci Eng C Mater Biol Appl* 2009; **29**: 969–975.
- 151 Li M, Liu X, Liu X *et al*. Calcium phosphate cement with BMP-2-loaded gelatin microspheres enhances bone healing in osteoporosis: a pilot study. *Clin Orthop Relat Res* 2010; **468**: 1978–1985.
- 152 Ruhe PQ, Boerman OC, Russel FG *et al*. Controlled release of rhBMP-2 loaded poly(dl-lactic-co-glycolic acid)/calcium phosphate cement composites *in vivo*. *J Control Release* 2005; **106**: 162–171.
- 153 Akkineni AR, Luo Y, Schumacher M *et al*. 3D plotting of growth factor loaded calcium phosphate cement scaffolds. *Acta Biomater* 2015; **27**: 264–274.
- 154 Schmidt-Bleek K, Schell H, Schulz N *et al*. Inflammatory phase of bone healing initiates the regenerative healing cascade. *Cell Tissue Res* 2012; **347**: 567–573.
- 155 Feng B, Jinkang Z, Zhen W *et al*. The effect of pore size on tissue ingrowth and neovascularization in porous bioceramics of controlled architecture *in vivo*. *Biomed Mater* 2011; **6**: 015007.
- 156 Xiao X, Wang W, Liu D *et al*. The promotion of angiogenesis induced by three-dimensional porous beta-tricalcium phosphate scaffold with different interconnection sizes via activation of PI3K/Akt pathways. *Sci Rep* 2015; **5**: 9409.
- 157 Yu T, Dong C, Shen Z *et al*. Vascularization of plastic calcium phosphate cement *in vivo* induced by in-situ-generated hollow channels. *Mater Sci Eng C Mater Biol Appl* 2016; **68**: 153–162.
- 158 Malhotra A, Habibovic P. Calcium phosphates and angiogenesis: implications and advances for bone regeneration. *Trends Biotechnol* 2016; **34**: 983–992.
- 159 Zhang W, Chang Q, Xu L *et al*. Graphene oxide-copper nanocomposite-coated porous CaP scaffold for vascularized bone regeneration via activation of hif-1alpha. *Adv Healthc Mater* 2016; **5**: 1299–1309.
- 160 Qiu G, Shi Z, Xu HHK *et al*. Bone regeneration in minipigs via calcium phosphate cement scaffold delivering autologous bone marrow mesenchymal stem cells and platelet-rich plasma. *J Tissue Eng Regen Med* 2017 [Epub ahead of print].
- 161 Patel ZS, Young S, Tabata Y *et al*. Dual delivery of an angiogenic and an osteogenic growth factor for bone regeneration in a critical size defect model. *Bone* 2008; **43**: 931–940.
- 162 Zhang HX, Zhang XP, Xiao GY *et al*. *In vitro* and *in vivo* evaluation of calcium phosphate composite scaffolds containing BMP-VEGF loaded PLGA microspheres for the treatment of avascular necrosis of the femoral head. *Mater Sci Eng C Mater Biol Appl* 2016; **60**: 298–307.
- 163 Thein-Han W, Xu HH. Prevascularization of a gas-foaming macroporous calcium phosphate cement scaffold via coculture of human umbilical vein endothelial cells and osteoblasts. *Tissue Eng Part A* 2013; **19**: 1675–1685.
- 164 Chen W, Thein-Han W, Weir MD *et al*. Prevascularization of biofunctional calcium phosphate cement for dental and craniofacial repairs. *Dent Mater* 2014; **30**: 535–544.
- 165 Liu X, Chen W, Zhang C *et al*. Co-seeding human endothelial cells with human-induced pluripotent stem cell-derived mesenchymal stem cells on calcium phosphate scaffold enhances osteogenesis and vascularization in rats. *Tissue Eng Part A* 2017; **23**: 546–555.
- 166 Chen W, Liu X, Chen Q *et al*. Angiogenic and osteogenic regeneration in rats via calcium phosphate scaffold and endothelial cell co-culture with human bone marrow mesenchymal stem cells (MSCs), human

- umbilical cord MSCs, human induced pluripotent stem cell-derived MSCs and human embryonic stem cell-derived MSCs. *J Tissue Eng Regen Med* 2017 [Epub ahead of print].
- 167 Jain RK, Au P, Tam J, Duda DG, Fukumura D. Engineering vascularized tissue. *Nat Biotechnol* 2005; **23**: 821–823.
- 168 Bergers G, Song S. The role of pericytes in blood-vessel formation and maintenance. *Neuro Oncol* 2005; **7**: 452–464.
- 169 Zhang C, Hu K, Liu X *et al*. Novel hiPSC-based tri-culture for pre-vascularization of calcium phosphate scaffold to enhance bone and vessel formation. *Mater Sci Eng C Mater Biol Appl* 2017; **79**: 296–304.
- 170 Laschke MW, Vollmar B, Menger MD. Inosculation: connecting the life-sustaining pipelines. *Tissue Eng Part B Rev* 2009; **15**: 455–465.
- 171 Beier JP, Horch RE, Hess A *et al*. Axial vascularization of a large volume calcium phosphate ceramic bone substitute in the sheep AV loop model. *J Tissue Eng Regen Med* 2010; **4**: 216–223.



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