

ARTICLE

Received 7 Mar 2012 | Accepted 13 Jun 2012 | Published 17 Jul 2012

DOI: 10.1038/ncomms1956

Microporous metal-organic framework with potential for carbon dioxide capture at ambient conditions

Shengchang Xiang^{1,2}, Yabing He², Zhangjing Zhang^{1,2}, Hui Wu^{3,4}, Wei Zhou^{3,4}, Rajamani Krishna⁵ & Banglin Chen²

Carbon dioxide capture and separation are important industrial processes that allow the use of carbon dioxide for the production of a range of chemical products and materials, and to minimize the effects of carbon dioxide emission. Porous metal-organic frameworks are promising materials to achieve such separations and to replace current technologies, which use aqueous solvents to chemically absorb carbon dioxide. Here we show that a metal-organic frameworks (UTSA-16) displays high uptake ($160\text{ cm}^3\text{ cm}^{-3}$) of CO_2 at ambient conditions, making it a potentially useful adsorbent material for post-combustion carbon dioxide capture and biogas stream purification. This has been further confirmed by simulated breakthrough experiments. The high storage capacities and selectivities of UTSA-16 for carbon dioxide capture are attributed to the optimal pore cages and the strong binding sites to carbon dioxide, which have been demonstrated by neutron diffraction studies.

¹ Fujian Provincial Key Laboratory of Polymer Materials, College of Materials Science and Engineering, Fujian Normal University, Fuzhou, Fujian 350007, China. ² Department of Chemistry, University of Texas at San Antonio, One UTSA Circle, San Antonio, Texas 78249-0698, USA. ³ NIST Center for Neutron Research, Gaithersburg, Maryland 20899-6102, USA. ⁴ Department of Materials Science and Engineering, University of Maryland, College Park, Maryland 20742, USA. ⁵ Van 't Hoff Institute for Molecular Sciences, University of Amsterdam, Science Park 904, Amsterdam 1098 XH, The Netherlands. Correspondence and requests for materials should be addressed to B.C. (email: Banglin.Chen@utsa.edu).

Carbon dioxide (CO_2) emissions have become one of the most serious issues and environmental concerns facing our civilization today¹. These emissions are mainly generated from the combustion of coal, oil and natural gas, the main energy resources for our daily life, economic growth and industrial development². Before we realize the cleaner alternative energy and establish the energy infrastructure for the implementation of new materials and technologies in the future, CO_2 emissions will continue to increase; it is thus very important and mandatory for us to capture and separate CO_2 to minimize their environmental impact.

The state-of-the-art technology to capture CO_2 makes use of aqueous alkanolamine absorbents to chemically absorb CO_2 (ref. 3). This technology costs a large amount of regeneration energy, because the CO_2 can only be released at quite a high elevated temperature, owing to the strong interactions of the CO_2 with these absorbents (the typical enthalpy of absorption lies the range of -50 to -100 kJ mol^{-1} at 296 K and low CO_2 loadings) and the high boiling points of these alkanolamine absorbents; in fact, CO_2 captured from a post-combustion flue gas using this technology results in an energy penalty of about 30% of the output of the power plant⁴. Furthermore, this technology will also lead to some environmental issues owing to the corrosive nature of these absorbents.

The emerging porous metal-organic frameworks (MOFs) are promising as the cost-effective and efficient materials for CO_2 capture and separation^{5–10}. The regeneration energy cost to utilize these porous materials for CO_2 capture by the implementation of temperature swing adsorption, pressure swing adsorption (PSA) and vacuum swing adsorption is significantly lower than the above-mentioned alkanolamine technology. More importantly, the rapid development over the past decade in this research field to target some porous MOFs for their extremely high uptake of CO_2 at high pressure^{11,12} and to immobilize functional sites, such as open metal sites^{13–23}, $-\text{NH}_2$ and $-\text{OH}$ organic sites into the pore surfaces to enhance their interactions and thus enforce their efficient CO_2 separation selectivity have principally ensured the feasibility of porous MOFs for CO_2 capture^{24–38}.

Although porous MOFs are principally feasible for CO_2 capture, realization of some practically useful MOF materials for its implementation is still challenging. The flue gas generated from the combustion of coal in the air consists of low content of CO_2 (15–16%) and high content of N_2 , and is released at a total pressure of approximately 1 bar, although the CO_2 capture from methane in biogas streams should be realized at ambient conditions, which means that the ideal MOF materials for these CO_2 capture should be those exhibiting extraordinarily high CO_2 uptake and selectivity at near-ambient pressures. Furthermore, they should be highly stable, particularly in the small amount of moisture, and can be easily regenerated for long-term usage. For practical applications, high volumetric CO_2 adsorption capacities are even more important than gravimetric CO_2 adsorption capacities for the temperature swing adsorption, PSA and vacuum swing adsorption post-combustion CO_2 capture and separation to fully utilize the fixed-bed space and minimize the energy regeneration cost.

To screen a series of porous MOFs synthesized in our own laboratory having different pore structures, surface areas and pore-surface functionalities, herein we report the performance of an adsorbent $[\text{K}(\text{H}_2\text{O})_2\text{Co}_3(\text{cit})(\text{Hcit})]^{39}$ (we term this MOF UTSA-16; UTSA = University of Texas at San Antonio; H_4cit = citric acid) for efficient CO_2 capture at room temperature and 1 bar. Its potential for CO_2 capture has been further examined by simulated PSA for the breakthrough separation of $\text{CO}_2(50\%)/\text{CH}_4(50\%)$ mixture (biogas streams) and $\text{CO}_2(15\%)/\text{N}_2(85\%)$ flue gas mixture (post-combustion CO_2 capture), highlighting UTSA-16 as a promising material with high separation selectivity and capacity for CO_2 capture applications at ambient conditions. Neutron diffraction studies reveal that both suitable pore/cage space and the terminal water molecules

have enabled their strong interactions with CO_2 molecules, and thus collaboratively enforced the high separation selectivity and capacity of this MOF material for CO_2 capture and separation at ambient conditions.

Results

Examined MOF materials. The examined porous MOFs are general ones ($\text{Zn}_4\text{O}(\text{FMA})_3$, UTSA-33a) without any specific functional sites, those with open $\text{Cu}(\text{II})$ sites (UTSA-20a and UTSA-34b), open $\text{Zn}(\text{II})$ sites ($\text{Zn}_4(\text{OH})_2(1,2,4\text{-BTC})_2$ and $\text{Zn}_5(\text{BTA})_6(\text{TDA})_2$) and open $\text{Yb}(\text{III})$ sites ($\text{Yb}(\text{BPT})$), those with functional $-\text{OH}$ sites on the pore surfaces (UTSA-15a, UTSA-25a and $\text{Cu}(\text{BDC-OH})$) and one with the terminal water molecules (UTSA-16)³⁹. Their structural features, pore characteristics and adsorption isotherms are shown in Supplementary Table S1 and Supplementary Figs S1 and S2. As established by Krishna and Long⁴⁰, the separation performance of porous materials in PSA separation of gas mixtures is strongly influenced by capacity considerations. For this reason, we systematically use volumetric CO_2 uptake capacities to characterize the performance of the reported MOF materials. As shown in Supplementary Fig. S1, UTSA-16 takes up significantly large amount of CO_2 ($160\text{ cm}^3\text{ cm}^{-3}$) at 296 K and 1 bar, which is at least three times higher than the rest of the porous MOFs (even some of these porous MOFs exhibit higher porosities, as shown Supplementary Table S1). That is really remarkable. The high CO_2 adsorption of UTSA-16 has featured it to be the second highest porous MOF for CO_2 capture and separation at 296 K and 1 bar (the best known MOF is Mg-MOF-74 (ref. 18) with storage capacity of $162\text{ cm}^3\text{ cm}^{-3}$). For comparing the separation performance of these MOFs synthesized in our group, we selected a number of MOFs for which the separation data has been reported, and for which the required isotherm data are available in the literature. The chosen MOFs are as follows: CuBTC⁴¹, Mg-MOF-74 (refs 15,17,42), $\text{Zn}(\text{bdc})(\text{dabco})_{0.5}$ (ref. 43), MIL-101 (ref. 41), Cu-TDPAT⁴⁴, bio-MOF-11 (ref. 45), MOF-177 (ref. 17), ZIF-78 (ref. 28) and ZnMOF-74 (refs 20,46,47). Supplementary Table S2 provides the structural data and pore characteristics for these chosen MOFs. The measured experimental data on pure component isotherms for CO_2 , CH_4 and N_2 at 273 and 296 K in our synthesized MOFs, and those literature data of temperature-dependent single component CO_2 , CH_4 and N_2 sorption isotherms for the chosen MOFs were first converted to absolute loadings, using the Peng–Robinson equation of state for estimation of the fluid densities. Depending on the guest–host combination, the choice of the isotherm is either a 1-site Langmuir or a 2-site Langmuir, or a 2-site Langmuir–Freundlich model. The isotherm models selected along with the fit parameters are specified in Supplementary Tables S3–33. For comparison, the absolute loading of CO_2 on UTSA-16 and the chosen MOFs at 296 K is shown in Fig. 1, indicating that the CO_2 loading on UTSA-16 is slightly lower than Mg-MOF-74, but higher than the rest of MOFs. The loading dependence of Q_{st} for CO_2 in different MOFs, determined using the pure component isotherm fits, is compared in Fig. 2a and Supplementary Tables S34.

IAST calculations for binary CO_2/CH_4 mixtures. We compared the performance of our MOFs with several MOFs (Mg-MOF-74, Cu-TDPAT, CuBTC, MIL-101 and $\text{Zn}(\text{BDC})(\text{DABCO})_{0.5}$) for CO_2 capture and separation from $\text{CO}_2(1)/\text{CH}_4(2)$ mixtures containing 50 mole% CO_2 , and maintained at isothermal conditions at 296 K and pressures up to 200 kPa . For comparison purposes, three zeolites (NaX ^{42,48}, JBW and MFI)⁴⁹ with high selectivity were also considered for analysis.

Using the pure component isotherm fits, the adsorption selectivities defined by

$$S_{\text{ads}} = \frac{q_1/q_2}{p_1/p_2} \quad (1)$$

were determined using the IAST of Myers and Prausnitz⁵⁰. The accuracy of the IAST calculations for estimation of the component loadings for several binary mixtures in a wide variety of zeolites and MOFs has been well established by comparison with the Configurational-Biast Monte Carlo simulations of mixture adsorption⁴⁹.

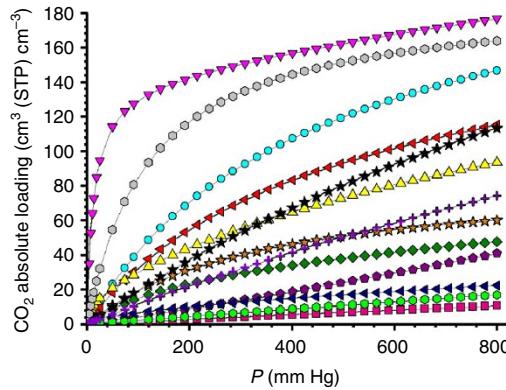
We proceeded further to determine the CO_2/CH_4 adsorption selectivities for the different MOFs at 296 K. In these calculations, the partial pressures of CO_2 and CH_4 are taken to be equal to each other, that is, $p_1 = p_2$. As shown in Fig. 2b,c, UTSA-16 has a higher selectivity than the other MOFs reported from our own lab and most

reported MOFs in the literature, although it has a lower CO_2/CH_4 selectivity than Mg-MOF-74. The specific orbital interactions of CO_2 molecules with the open Mg atoms accounts for the high selectivities offered by Mg-MOF-74 (refs 20,51–53). Among all the pore materials, JBW has the highest selectivity, followed by Mg-MOF-74, NaX and UTSA-16. For ease of comparison, Supplementary Table S35 presents the numerical data on the adsorption CO_2/CH_4 selectivities at 200 kPa and 296 K in different MOFs.

Besides separation selectivity, capacity considerations are important in determining the performance of any given adsorbents in a PSA unit. The loading of CO_2 in the adsorbed phase within the porous structures in equilibrium with a binary CO_2/CH_4 gas mixture maintained at isothermal conditions at 296 K were determined for a range of pressures up to 200 kPa. For a total pressure of 200 kPa, the data on gravimetric uptake capacities are listed in Supplementary Table S35. At this pressure, we note that CO_2 uptake capacity of Mg-MOF-74 is about twice as high as any other material. JBW has a low uptake capacity because of its low pore volume⁵⁴. We will see later that JBW will have poor performance in a PSA unit.

Because of the wide differences in the framework densities, the hierarchy of CO_2 uptake capacities is quite different when expressed as mol l^{-1} of adsorbent; the corresponding data are presented in Fig. 2d. We now note that UTSA-16 is only slightly lower than Mg-MOF-74 in terms of their volumetric uptake capacities and both of them exhibit significantly higher volumetric storage capacities than the rest of MOFs. The volumetric uptake capacity of UTSA-16 is comparable to that of the NaX zeolite.

Figure 1 | Comparison of the absolute loading of CO_2 at 296 K. From top to bottom, MgMOF-74 (magenta down triangle), UTSA-16 (grey hexagon), ZnMOF-74 (cyan circle), bio-MOF-11 (red left triangle), CuBTC (black star), Cu-TDPAT (yellow up triangle), UTSA-20a (violet cross), ZIF-78 (orange star), $\text{Zn}_5(\text{BTA})_6(\text{TDA})_2$ (olive diamond), $\text{Zn}(\text{bdc})(\text{dabco})_0.5$ (purple pentagon), MIL-101 (navy left triangle), Yb(BPT) (green hexagon) and MOF-177 (pink square).



Breakthrough calculations for PSA adsorber. Most commonly, the adsorption selectivity on its own is used to screen MOFs for any separation task. The performance of an MOF in a PSA unit is governed by both selectivity and capacity factors. The proper combination of both of these factors is obtained by use of breakthrough calculations⁴⁰.

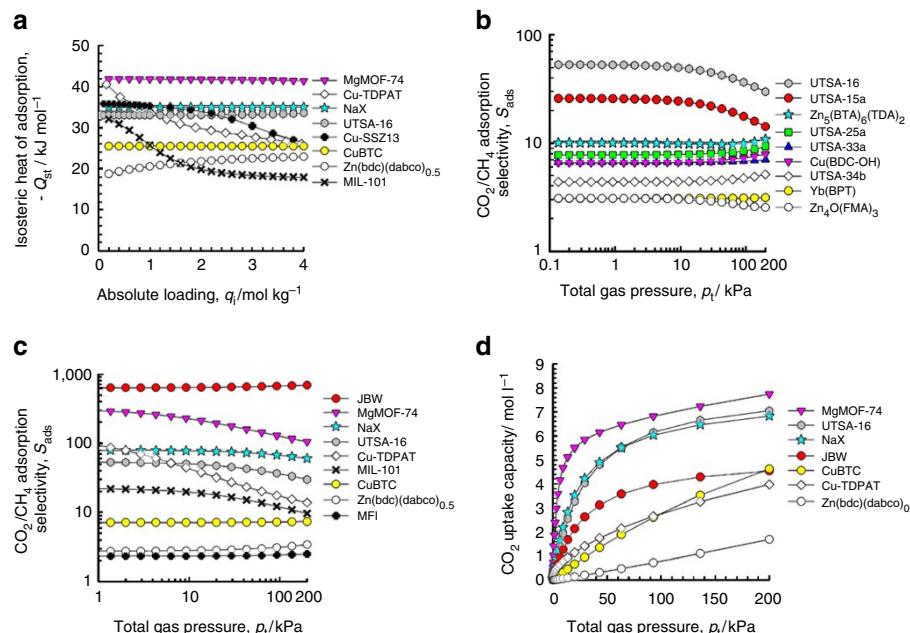


Figure 2 | IAST calculations for binary CO_2/CH_4 mixtures on the MOFs. (a) Comparison of the loading dependence of the isosteric heats of pure CO_2 adsorption on the porous materials. (b,c) CO_2/CH_4 adsorption selectivity for the variety of MOFs considered in this work. (d) CO_2 uptake volumetric capacity of some important MOFs. In these calculations, the partial pressures of CO_2 and CH_4 are taken to be equal to each other, that is, $p_1 = p_2 = 100$ Pa. All calculations are for 296 K, using the T -dependent isotherm parameters reported in Supplementary Tables S3–S33. The calculations of the Q_{st} of Cu-TDPAT are those reported in the paper by Li et al.⁴⁴

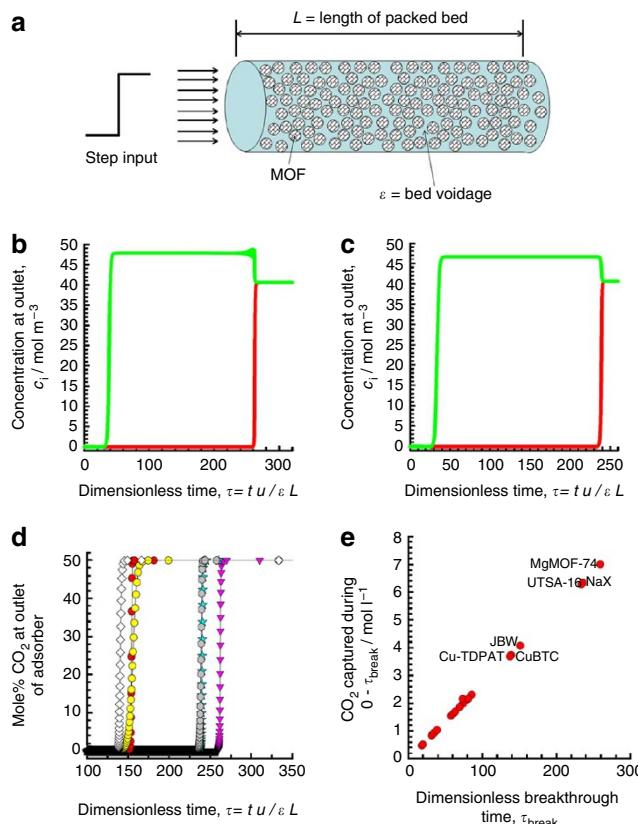


Figure 3 | Breakthrough calculations for PSA adsorber. (a) Schematic of a packed bed adsorber. Breakthrough characteristics of an adsorber packed with (b) Mg-MOF-74 and (c) UTSA-16 and maintained at isothermal conditions at 296 K. In these calculations, the partial pressures of CO₂ and CH₄ at the inlet is taken to be $p_1 = p_2 = 100$ kPa. (d) Plot of the number of moles of CO₂ captured per litre of adsorbent material during the time interval 0– τ_{break} against the breakthrough time τ_{break} for packed bed adsorber with step-input of a 50/50 CO₂/CH₄ mixture at 296 K and total pressures of 200 kPa. The breakthrough times, τ_{break} , correspond to those when the outlet gas contains 0.05 mole% CO₂. The considered MOFs are MgMOF-74 (purple down triangle), NaX (cyan star), UTSA-16 (grey hexagon), JBW (red circle), CuBTC (yellow circle) and Cu-TDPAT (open diamond). (e) Plot of the number of moles of CO₂ captured from CO₂/CH₄ mixture per litre of adsorbent material, during the time interval 0– τ_{break} against the breakthrough time τ_{break} for packed bed adsorber with step-input of a 50/50 CO₂/CH₄ mixture at 296 K and total pressures of 200 kPa. The breakthrough times, τ_{break} , correspond to those when the outlet gas contains 0.05 mole% CO₂.

Figure 3a shows a schematic of a packed bed adsorber. The breakthrough calculations were performed using the following methodologies developed and described in earlier works^{40,54,55}, the breakthrough simulation methodology has been validated by direct comparison with experimental breakthrough data in the work of Bloch⁵⁶. Figure 3b and present typical breakthrough curves for Mg-MOF-74 and UTSA-16, respectively. The x axis in Fig. 3b–d is a dimensionless time, τ , defined by dividing the actual time, t , by the characteristic time, Le/u . Figure 3d shows the gas composition, expressed as mole% CO₂ in outlet gas as a function of the dimensionless time for a selection of porous adsorbent materials. On the basis of these calculations, the value of the breakthrough time, τ_{break} , can be determined for a specified purity of CO₂ in the outlet gas stream. For an arbitrarily chosen purity of 0.05 mole% CO₂, the data of dimensionless breakthrough times are presented in

Supplementary Table S36. It shows that Mg-MOF-74 and UTSA-16 have τ_{break} of 259 and 234, respectively, which are significantly higher than for Cu-BTC (139), Cu-TDPAT (138), ZIF-78 (81) and rest of the MOFs. UTSA-16 has the comparable τ_{break} with the best zeolite NaX (235). A longer τ_{break} is desirable because the frequency of regeneration is reduced. It needs to be mentioned that although JBW zeolite has the highest CO₂/CH₄ adsorption separation selectivity, its breakthrough time is much shorter than UTSA-16 (Fig. 3d).

The economics of a PSA unit for post-combustion CO₂ capture will be dictated primarily by the amount of CO₂ captured during the adsorption cycle, that is, during the time interval 0– τ_{break} ; this amount can be determined from a material balance; the data on the amount adsorbed is also presented in Supplementary Table S36. The CO₂ capture presented in Supplementary Table S36 is expressed both in mol kg⁻¹ of adsorbent and mol l⁻¹ of adsorbent material. The proper metric for comparing different adsorbents for use in a PSA unit is on the basis of moles captured per litre of adsorbent. This is because an existing adsorber will have a fixed volume. When considering the choice of the best MOF to replace the existing adsorbent, we should compare these on the capture capacity per litre of adsorbent. Figure 3e presents plots of the number of moles of CO₂ captured per litre of adsorbent against the breakthrough time τ_{break} . It is remarkable to note that per litre of adsorbent material, the number of moles of CO₂ capture is almost perfectly linearly related to the dimensionless breakthrough time. This is a rational result. The breakthrough calculations for different materials were performed with the same volume of adsorber, containing identical volumes of adsorbent materials. Expressed differently, Fig. 3e demonstrates that the proper metric that determines the separation characteristics of a PSA adsorber is the dimensionless breakthrough time, τ_{break} , which in turn depends on both selectivity and capacity metrics. From a practical point of view, the volumetric capture capacity is more relevant for screening and choosing the suitable MOFs. Apparently, UTSA-16 is a superior adsorbent for CO₂/CH₄ separation at ambient conditions than most MOFs, other than MgMOF-74.

Comparison of MOFs for CO₂/N₂ separation. We further compared the performance of UTSA-16 with several MOFs (mmen-CuBTri, Mg-MOF-74, MOF-177, ZIF-78, Cu-TDPAT, bioMOF-11, Zn-MOF-74) for CO₂ capture from CO₂(1)/N₂(2) mixtures containing 15 mole% CO₂, and maintained at isothermal conditions at 296 K, which represent conditions relevant for flue gas processing. For comparison, five representative zeolites (NaX^{42,48}, Cu-SSZ13⁵⁷, H-SSZ13⁵⁷, JBW⁴⁹ and MFI⁴⁹) exhibiting high CO₂/N₂ separation selectivity are also included. Figure 4a presents the IAST calculations for CO₂/N₂ adsorption selectivity. We note that the selectivities for different MOF materials vary by about two orders of magnitude. Among the MOFs, amine-functionalized mmen-CuBTri has the highest adsorption selectivity followed closely by UTSA-16 and Mg-MOF-74. The lowest adsorption selectivity is obtained with MOF-177 that has the most open structure of all MOF materials considered here, which is apparently not favourable for low-pressure CO₂ capture. JBW has the highest adsorption selectivity but a poor uptake capacity because of its low pore volume⁵⁴.

Figure 4b shows data on the CO₂ uptake capacity expressed as mol l⁻¹ of adsorbent. The highest uptake capacity is afforded by Mg-MOF-74, and this is closely followed by UTSA-16. The lowest uptake is offered by MOF-177. The data on the selectivity and capacity for operation at a total pressure of 100 kPa are also summarized in Supplementary Table S37.

To evaluate the different adsorbents for use in a PSA unit, breakthrough calculations were performed. Figure 4c compares the breakthrough characteristics in terms of mole% CO₂ in outlet gas as a function of the dimensionless time for operation at a total pressure

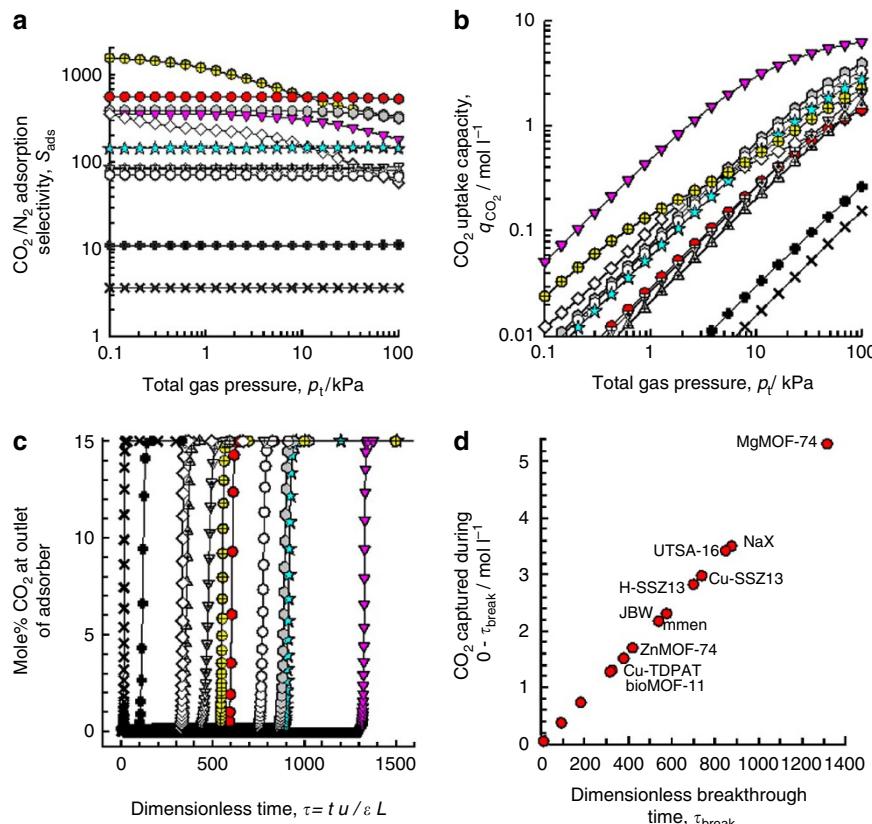


Figure 4 | Comparison of MOFs for CO₂/N₂ separation. Calculations using Ideal Adsorbed Solution Theory (IAST) of Myers and Prausnitz⁵⁰ for (a) CO₂/N₂ adsorption selectivity, and (b) volumetric CO₂ uptake capacity in equilibrium with a binary CO₂/N₂ gas mixture maintained at isothermal conditions at 296 K. In these calculations, the partial pressures of CO₂ and N₂ are taken to be p₁/p₂ = 15/85. (c) CO₂/N₂ mixture breakthrough characteristics in terms of mole% CO₂ in outlet gas as a function of the dimensionless time in an adsorber packed with different adsorbents and maintained at isothermal conditions at 296 K. In these calculations, the partial pressures of CO₂ and N₂ are taken to be p₁ = 15 kPa and p₂ = 85 kPa. The considered MOFs from a to c are MgMOF-74 (purple down triangle), NaX (cyan star), UTSA-16 (grey hexagon), JBW (red circle), mmen-CuBTTri (yellow plus-centered circle), Cu-SSZ13 (open circle), ZnMOF-74 (plus-centered down triangle), bio-MOF-11 (plus-centered up triangle), Cu-TDPAT (open diamond), MFI (black plus) and MOF-177 (black cross). (d) Plot of the number of moles of CO₂ captured from CO₂/N₂ mixture per litre of adsorbent material, during the time interval 0–τ_{break} against the breakthrough time τ_{break} for packed bed adsorber with step-input of a 15/85 CO₂/N₂ mixture at 296 K and total pressures of 100 kPa. The breakthrough times, τ_{break}, correspond to those when the outlet gas contains 0.05 mole % CO₂.

of 100 kPa. Supplementary Table S38 summarizes the data on dimensionless breakthrough times. The breakthrough times are a reflection of the combination of the selectivity and capacity metrics. The longest breakthrough time is obtained with Mg-MOF-74. The reason for this can be traced to a combination of high selectivities and high capacities. This is followed by NaX zeolite and UTSA-16. It is particularly noteworthy that JBW and mmen-CuBTTri that exhibit extremely high selectivities have significantly shorter breakthrough times. This is because both have low uptake capacities. The shortest breakthrough time is obtained with MOF-177; ZIF-78 has only a slightly higher breakthrough time.

Figure 4d presents plots of the number of moles of CO₂ captured per litre of adsorbent material, during the time interval 0–τ_{break} against the breakthrough time τ_{break}. We note that there is a perfect linear correlation between the amount captured per litre of adsorbent and τ_{break}, in line with the findings of this study reported in Fig. 4d. This re-emphasizes the conclusion that the dimensionless breakthrough time is the proper metric to reflect the separation performance of any MOF in a PSA unit. Mg-MOF-74 ends up at the desirable top-right corner of Fig. 4d. It is interesting to note that Cu-TDPAT does not possess the required combination of selectivity and capacity to be a viable candidate for post-combustion CO₂ capture in PSA operations. The lowest capture capacity is with

MOF-177, with ZIF-78 displaying only slightly better performance. This point is being stressed here because of the special attention given by Bae and Snurr⁵⁸ to ZIF-78 in their recent review as the best adsorbent for CO₂ capture. This further underlines the need for a proper comparison of different MOFs using the breakthrough analysis presented here. Another interesting point to note is that the amount of CO₂ captured per litre of adsorbent material follows the hierarchy Mg-MOF-74 > UTSA-16 ≈ NaX, similar to the hierarchy determined for separation of CO₂/CH₄ mixtures; compare Figs 3e and 4d.

Comparison with Cu-(bpy-1)₂. During the writing of this manuscript, the paper of Burd *et al.*⁵⁹ appeared online, in which the authors present the uptake data for CO₂, CH₄ and N₂ at 296 K in Cu-(bpy-1)₂. We scanned their experimental data presented in Fig. 2 of their paper, converted to absolute component loadings, fitted the pure component isotherms and calculated the adsorption selectivities, using IAST, of Cu-(bpy-1)₂ for separation of CO₂/CH₄ and CO₂/N₂ mixtures. For CO₂/CH₄ mixtures, our IAST calculations agree very well with those reported in Fig. 3 of their paper. Supplementary Fig. S3 presents a comparison of the CO₂/CH₄ adsorption selectivity of Cu-(bpy-1)₂ with that for the variety of MOFs considered in this work. We note that UTSA-16 has a

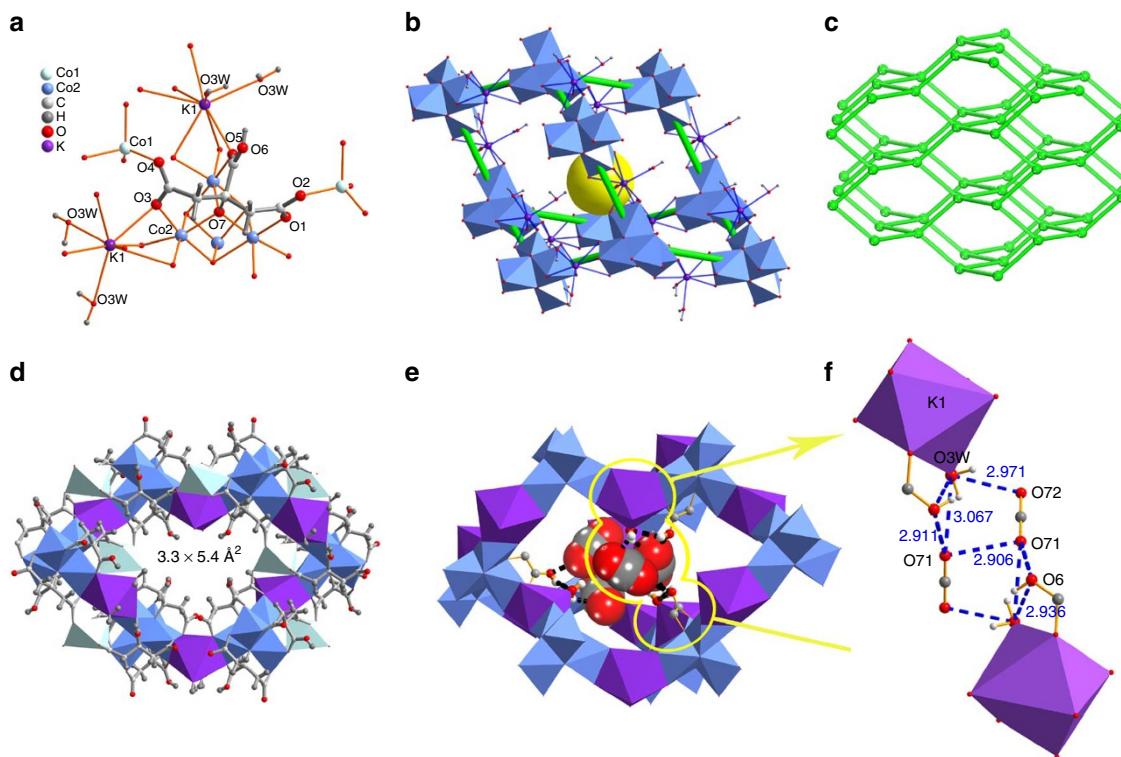


Figure 5 | The structure for CO₂-loaded UTSA-16. (a) The coordination mode of citrate ligand. Each citrate chelates one cubic (Co(2)₄O₄) cluster, two Co(1) tetrahedral and two K ions. There are two water (O3w) coordinated to each K ions. (b) Each cubic (Co(2)₄O₄) cluster node is linked by four K polyhedral linkers to give a diamondoid cage (yellow ball of about 4.5 Å in diameter). There are four crystallographically independent O3w from 4 K polyhedral toward the cage. For clarity, Co1, C and H on citrate ligand are omitted. The linkages between the cubic clusters are present with the green sticks (not the real bond) to illustrate the framework topology. (c) The resulting dia network. (d) The diamondoid cage has a small widow with the dimensions of 3.3×5.4 Å². All metal ions are present as the polyhedral with the colours same as those for the atoms in a. (e) A couple of CO₂ dimers are trapped within the cage; (f) The cooperative interactions between CO₂ molecules and the framework (O71(CO₂)···O71(CO₂) and O71(CO₂)···O6 short contact; O72(CO₂)···H-O3w, O71(CO₂)···H-O3w and O3w···H-O6 hydrogen-bonding interactions). The angle between the two linear CO₂ molecules is of 64.93(2)°.

significantly higher selectivity than Cu-(bpy-1)₂. Supplementary Fig. S4 presents a comparison of the CO₂/N₂ adsorption selectivity of Cu-(bpy-1)₂ with that for the variety of MOFs considered in this work. We note, once again, that UTSA-16 has a significantly higher selectivity than Cu-(bpy-1)₂.

Neutron diffraction studies. The extraordinary performance of UTSA-16 for the CO₂ capture at ambient conditions has motivated us to rationalize the crucial factors to enforce the high CO₂ adsorption selectivity and capacity. To experimentally verify the preferential adsorption sites and thus to elucidate the highest density of CO₂ trapped into its pore, neutron-powder diffraction (NPD) data were collected on UTSA-16 loaded with 0.45 CO₂ per coordinated H₂O, using the high-resolution neutron powder diffractometer (BT-1) at the NIST Center for Neutron Research. The Rietveld refinement was performed on the NPD data using the programme EXGUI, yielding the agreement factors of $R_{wp} = 0.0345$, $R_p = 0.0292$ and $\chi^2 = 1.17$. The Rietveld fit to the diffraction pattern is shown in Supplementary Fig. S5. From the refinement, CO₂ is only adsorbed at one site with 0.45 CO₂ per co-ordinated H₂O loading, that is, the total amount of CO₂ molecules obtained from the refinement is in good agreement with the target value for the CO₂ loading in the host lattice. The crystal structure of the CO₂-loaded UTSA-16 is shown in Fig. 5.

The structure of CO₂-loaded UTSA-16 is a three-dimensional framework in which the coordination environments and framework topology are shown in Fig. 5a–c. As shown in Fig. 5d, the pore window is about 3.3×5.4 Å². The diamondoid cage of about 4.5 Å

in diameter can trap two pairs of CO₂ molecules (Fig. 5e) through UTSA-16···K-OH₂···O₂C hydrogen-bonding interactions (O3w···H···O72 = 2.971(1) Å, <O3w-H···O72 = 102.05(4)°; O3w-H···O71 = 3.067(1) Å, <O3w-H···O71 = 135.98(4)°) and O···O close contact (O71···O6 = 2.911(2) Å; O71···O71 = 2.906(1) Å), as shown in Fig. 5f. The angle between the two linear CO₂ molecules is of 64.93°. On the basis of the stoichiometric ratio 1:1 of CO₂ to coordinated water, the density and uptake capacity of trapped CO₂ will be 0.768 g cm⁻³ and 118 cm³(STP) cm⁻³. The remaining uptake of 42 cm³(STP) cm⁻³ might be attributed from the empty pore space, which accounts for 26% total CO₂ uptake capacity, highlighting the crucial contribution from the interactions between the O atom of CO₂ adsorbate and the O atoms (O3w and O6) on the pore surfaces. The nearest intermolecular O···O contact between CO₂ molecules trapped in UTSA-16 is shorter than those reported in other CO₂-loaded MOFs (Supplementary Table S39). The density of the CO₂ in UTSA-16 is approximately equal to the liquid CO₂ density of 1.032 g cm⁻³ (ref. 60), estimated according to the amount of gas adsorbed and the Platon pore volume (30.2%) of the framework of 1.037 g cm⁻³ at 1 atm and 296 K, which is the highest ever reported on the MOFs for CO₂ capture at the same condition. The highest CO₂ density means UTSA-16 can trap CO₂ in the most efficient mode into its pore to reach saturation even at 296 K and 1 atm (Supplementary Fig. S6), featuring UTSA-16 as the very promising material for the highly selective separation of CO₂ from the light gases. In summary, the uniqueness of UTSA-16 for its highly selective separation of CO₂ from the light gases is attributed to the following factors: (1) stable diamondoid

network comprising of (Co_4O_4) clusters and K-polyhedra linkers via face-sharing, which is supported by the infinite three-dimensional heteronuclear M-O-M connections, (2) diamondoid cage of 4.5\AA in diameter to efficiently trap two pairs of CO_2 molecules, (3) small windows of $3.3 \times 5.4\text{\AA}^2$ for its potentially high size-selective effect (the kinetic molecule sizes for CO_2 , N_2 and CH_4 are 3.30 , 3.64 and 3.80\AA , respectively), and (4) four crystallographically dependent coordinated water molecules within each cavity for their hydrogen-bonding interactions with the CO_2 gas molecules.

Discussion

The microporous MOFs are promising materials for CO_2 capture and separation. As we can rationally design and synthesize some highly porous MOFs, which can take up significant amount of CO_2 under high pressure (a few extremely highly porous MOFs with BET surface areas over $6,000\text{ m}^2\text{ g}^{-1}$ have been constructed; The CO_2 storage capacities at room temperature and high pressure are basically proportional to the surface areas of the MOF materials)¹², it is quite straightforward and feasible to realize MOF materials for high-pressure CO_2 capture and separation. The CO_2 storage capacities at room temperature and low pressure (1 bar; essential conditions for CO_2 capture and separation in post-combustion mixtures and biogas streams), however, are not dependent on their pore surface areas, which have been clearly demonstrated in our examined MOF materials and literatures. Such a situation poses a great challenge to synthetic materials chemists on how to target some unique porous MOF materials, for which CO_2 capture can be significantly high at ambient conditions. Ideally, we already have the resolution, that is, to immobilize specific sites such as open metal sites and $-\text{NH}_2$ sites to induce their strong interactions with CO_2 , and to optimize the pore/cage sizes to maximize the van der Waals interactions between the pore surfaces and CO_2 . However, only a few porous MOFs have been realized for high CO_2 capture at ambient conditions. The fact that Mg-MOF-74 takes up much more CO_2 than other M-MOF-74 ($\text{M} = \text{Ni}^{2+}$, Co^{2+} , Zn^{2+} and Fe^{2+}), and some other porous MOFs with open metal sites (for example, UTSA-20a and Yb(BPT) in this report) do not necessarily exhibit high CO_2 uptake at room temperature and 1 bar, indicates that both different type of open-metal sites and MOF structures have important roles to enforce their strong interactions with CO_2 at ambient conditions. Mg-MOF-74 is very unique because of its extremely high CO_2 uptake at room temperature and 1 bar. However, because the interactions of Mg-MOF-74 with water molecules are very strong, Mg-MOF-74 is highly sensitivity to the moisture, leading to its significantly reduced CO_2 storage capacities once it is exposed to the air. UTSA-16 exhibits repeatable and reversible CO_2 sorption (Supplementary Fig. S7). More importantly, the re-activated UTSA-16 from the used sample, which has been exposed in the air for 3 days, still keeps the same sorption capacities for CO_2 , indicating that UTSA-16 is not air sensitive. Such properties are apparently important for its potential industrial usage. It needs to be mentioned that the existence of the coordinated water molecules is essential for UTSA-16 to have high CO_2 capture capacity. In fact, once all the water molecules within UTSA-16 are removed at elevated temperature, the resulting anhydrous material is non-porous. Another advantage of UTSA-16 for its CO_2 capture and separation is its cheap cost of the raw materials, particularly the very common organic linker citric acid, which should allow us to enlarge the synthesis straightforwardly. Our accidental discovery of UTSA-16 for high CO_2 capture and separation at ambient conditions might facilitate the extensive research on porous MOF materials for CO_2 capture in the near future.

Methods

Synthesis and gas adsorption. All reagents and solvents employed in synthetic studies were commercially available and used as supplied without further purification. As-synthesized sample UTSA-16 was synthesized, activated

accordingly to our reference procedures³⁹, and confirmed by powder X-ray diffraction study and N_2 adsorption isotherms. The detailed synthetic procedures are in the following: the hydrothermal reaction of $\text{Co}(\text{OAc})_2 \cdot 4\text{H}_2\text{O}$ (0.249 g 1 mmol), $\text{C}_6\text{H}_8\text{O}_7 \cdot \text{H}_2\text{O}$ (0.21 g , 1 mmol), KOH (0.168 g , 3 mmol), H_2O (2.5 mL) and $\text{C}_2\text{H}_5\text{OH}$ (2.5 mL) in the molar ratio 1:1:3:139:43 is conducted in a teflon-lined 23 mL Parr acid digestion bomb. The bomb is then placed in a furnace that is heated from room temperature to 120°C in half an hour, kept at 120°C for 2 days, and then slowly cooled to room temperature at a rate of 4°C h^{-1} . And violet prismatic samples were obtained, which were filtered, washed with ethyl ether and dried at room temperature. The sample is degassed for 24 h at 363 K and $5\mu\text{m Hg}$ to obtain the activated UTSA-16. Overactivation at higher temperature and vacuum will lead to the collapse of the framework and non-porous material. The preparation and activation methods for other MOFs synthesized in our lab were followed our references (Supplementary Table S1). CO_2 , CH_4 and N_2 adsorption isotherms for our MOFs were measured on Micromeritics ASAP 2020 Accelerated Surface Area and Porosimetry analyzer. As the centre-controlled air condition was set up at 23.0°C , a water bath of 23.0°C was used for gas adsorption isotherms at 296.0 K .

Fits of pure component isotherms for the pore materials. The measured experimental data on pure component isotherms for CO_2 , CH_4 and N_2 at 273 K and 296 K in the MOFs synthesized in our lab, and the literature data of temperature-dependent single component CO_2 , CH_4 and N_2 sorption isotherms for the chosen MOFs and the representative zeolites were first converted to absolute loadings, using the Peng–Robinson equation of state for estimation of the fluid densities. The pore volume data given in Supplementary Tables S1 and S2 were used for this purpose. Depending on the guest–host combination the choice of the isotherm is either a 1-site Langmuir, or a 2-site Langmuir, or a 2-site Langmuir–Freundlich model. The isotherm models selected along with the fit parameters are specified in Supplementary Tables S3–S33.

Isosteric heat of adsorption. The isosteric heat of adsorption, Q_{st} , defined as

$$Q_{\text{st}} = RT^2 \left(\frac{\partial \ln p}{\partial T} \right)_q \quad (2)$$

were determined using the pure component isotherm fits for the 11 MOFs synthesized in this study. For some guest–host combinations for which either the dual-site Langmuir or dual-site Langmuir–Freundlich fits were used, Q_{st} is a function of the loading. The Q_{st} values for CO_2 and CH_4 for the 11 different MOFs synthesized in this work are reported in Supplementary Table S34. The values are for the loadings corresponding to that in equilibrium with a bulk gas pressure of 100 kPa at 296 K . The loading dependence of Q_{st} for CO_2 in different MOFs is compared in Fig. 2a for some important MOFs that emerge in this study as promising candidates for CO_2/CH_4 and CO_2/N_2 separations.

Breakthrough calculations for PSA adsorber. The breakthrough calculations were performed using the following the methodologies developed and described in earlier works^{40,54,55}. Assuming plug flow of $\text{CO}_2(1)/\text{CH}_4(2)$ or $\text{CO}_2(1)/\text{N}_2(2)$ gas mixture through a fixed bed maintained under isothermal conditions and negligible pressure drop, the partial pressures in the gas phase at any position and instant of time are obtained by solving the following set of partial differential equations for each of the species i in the gas mixture.

$$\frac{1}{RT} \varepsilon \frac{\partial p_i}{\partial t} = -\frac{1}{RT} \frac{\partial (up_i)}{\partial z} - (1-\varepsilon) \rho \frac{\partial q_i}{\partial t}; \quad i=1,2 \quad (3)$$

In equation (3), t is the time, z is the distance along the adsorber, ρ is the frame-work density, ε is the bed voidage, and u is the superficial gas velocity. The molar loadings of the species i , q_i , at any position z , and time t is determined from the IAST calculations. Details of the numerical procedures used are available in earlier works^{40,54,55}. Specifically, the calculations presented here were performed taking the following parameter values: bed length, $L = 0.3\text{ m}$; voidage of bed, $\varepsilon = 0.4$; superficial gas velocity and $u = 0.04\text{ m s}^{-1}$ (at inlet). When comparing different materials, the fractional voidage is held constant at $\varepsilon = 0.4$. This implies the volume of adsorbents remains the same. The total mass of the adsorbents used is governed by the framework density.

NPD studies. NPD data were collected on the high-resolution neutron powder diffractometer (BT-1) at the NIST Center for Neutron Research with a $\text{Cu}(311)$ monochromator and using in-pile collimation of 15 min of arc, corresponding to a wavelength of 1.5403\AA . The activated UTSA-16 sample was first degassed under vacuum and transferred to a cylindrical vanadium can equipped with a capillary gas line and a packless valve, and sealed with an indium O-ring in a helium-filled glove box. The sample can was then mounted onto a sample stick, equipped with a stainless steel gas line with an additional valve for a top-loading closed-cycle helium refrigerator (CCR). Targeted amount of CO_2 , that is 0.45 CO_2 per

coordinated H_2O , was then loaded into the UTSA-16 sample at room temperature. The sample was slowly cooled to 210 K at which point the CO_2 was completely adsorbed. Once the system was equilibrated at 210 K, the sample was further cooled to 4 K before the NPD measurement. No evidence of solid CO_2 was observed on the structural refinement, indicating all CO_2 was adsorbed into the UTSA-16.

NPD data were analysed using the Rietveld refinement method with the programme EXPGUI. The CO_2 molecule was treated as a linear rigid body with a C–O bond length of 1.15 Å. Atomic positional and rotational coordinates, a thermal factor and occupancy were refined for the CO_2 site. The final refinements on all the parameters, including atomic positions, thermal factors, occupancies for both framework and adsorbed CO_2 molecules, and lattice parameters yielded very good agreement factors of $R_{\text{wp}} = 0.0345$, $R_{\text{p}} = 0.0292$ and $\chi^2 = 1.17$ (Supplementary Fig. S5). Also, the total amount of CO_2 molecules obtained from the refinement is in very good agreement with the target values for the CO_2 loading in the UTSA-16 host lattice. This is another indication of the high quality of the refined structure.

References

1. Pachauri, R. K. & Reisinger, A. *IPCC Fourth Assessment Report* (Intergovernmental Panel on Climate Change, 2007).
2. Quadrelli, R. & Peterson, S. The energy-climate challenge: recent trends in CO_2 emissions from fuel combustion. *Energ. Policy* **35**, 5938–5952 (2007).
3. Rochelle, G. T. Amine scrubbing for CO_2 capture. *Science* **325**, 1652–1654 (2009).
4. Haszeldine, R. S. Carbon capture and storage: how green can black be? *Science* **325**, 1647–1652 (2009).
5. D'Alessandro, D. M., Smit, B. & Long, J. R. Carbon dioxide capture: prospects for new materials. *Angew. Chem. Int. Ed.* **49**, 6058–6082 (2010).
6. Sumida, K. *et al.* Carbon dioxide capture in metal-organic frameworks. *Chem. Rev.* **112**, 724–781 (2012).
7. Drage, T. C. *et al.* Materials challenges for the development of solid sorbents for post-combustion carbon capture. *J. Mater. Chem.* **22**, 2815–2823 (2012).
8. Zhang, J.- P. & Kitagawa, S. Supramolecular isomerism, framework flexibility, unsaturated metal center, and porous property of $\text{Ag(I)}/\text{Cu(I)}$ 3,3',5,5'-tetramethyl-4,4'-bipyrazolate. *J. Am. Chem. Soc.* **130**, 907–917 (2008).
9. Li, J.- R. *et al.* Carbon dioxide capture-related gas adsorption and separation in metal-organic frameworks. *Coord. Chem. Rev.* **255**, 1791–1823 (2011).
10. Vaidhyanathan, R. *et al.* Direct observation and quantification of CO_2 binding within an amine-functionalized nanoporous solid. *Science* **330**, 650–653 (2010).
11. Millward, A. R. & Yaghi, O. M. Metal-organic frameworks with exceptionally high capacity for storage of carbon dioxide at room temperature. *J. Am. Chem. Soc.* **127**, 17998–17999 (2005).
12. Furukawa, H. *et al.* Ultrahigh porosity in metal-organic frameworks. *Science* **329**, 424–428 (2010).
13. Vimont, A. *et al.* Investigation of acid sites in a zeotypic giant pores chromium(III) carboxylate. *J. Am. Chem. Soc.* **128**, 3218–3227 (2006).
14. Dincă, M. & Long, J. R. Hydrogen storage in microporous metal-organic frameworks with exposed metal sites. *Angew. Chem. Int. Ed.* **47**, 6766–6779 (2008).
15. Dietzel, P. D. C., Besikiotis, V. & Blom, R. Application of metal-organic frameworks with coordinatively unsaturated metal sites in storage and separation of methane and carbon dioxide. *J. Mater. Chem.* **19**, 7362–7370 (2009).
16. Chen, B., Xiang, S. & Qian, G. Metal-organic frameworks with functional pores for recognition of small molecules. *Acc. Chem. Res.* **43**, 1115–1124 (2010).
17. Mason, J. A., Sumida, K., Herm, Z. R., Krishna, R. & Long, J. R. Evaluating metal-organic frameworks for post-combustion carbon dioxide capture *via* temperature swing adsorption. *Energ. Environ. Sci.* **4**, 3030–3040 (2011).
18. Caskey, S. R., Wong-Foy, A. G. & Matzger, A. J. Dramatic tuning of carbon dioxide uptake via metal substitution in a coordination polymer with cylindrical pores. *J. Am. Chem. Soc.* **130**, 10870–10871 (2008).
19. Llewellyn, P. L. *et al.* High uptakes of CO_2 and CH_4 in mesoporous metal-organic frameworks MIL-100 and MIL-101. *Langmuir* **24**, 7245–7250 (2008).
20. Yazaydin, A. Ö. *et al.* Screening of metalorganic frameworks for carbon dioxide capture from flue gas using a combined experimental and modeling approach. *J. Am. Chem. Soc.* **131**, 18198–18199 (2009).
21. Yazaydin, A. Ö. *et al.* Enhanced CO_2 adsorption in metal-organic frameworks via occupation of open-metal sites by coordinated water molecules. *Chem. Mater.* **21**, 1425–1430 (2009).
22. Liu, J. *et al.* $\text{CO}_2/\text{H}_2\text{O}$ adsorption equilibrium and rates on metal-organic frameworks: HKUST-1 and Ni/DOBDC. *Langmuir* **26**, 14301–14307 (2010).
23. Aprea, P., Caputo, D., Gargiulo, N., Iucolano, F. & Pepe, F. Modeling carbon dioxide adsorption on microporous substrates: comparison between Cu-BTC metalorganic framework and 13X zeolitic molecular sieve. *J. Chem. Eng. Data* **55**, 3655–3661 (2010).
24. Phan, A. *et al.* Synthesis, structure, and carbon dioxide capture properties of zeolitic imidazolate frameworks. *Acc. Chem. Res.* **43**, 58–67 (2010).
25. Furukawa, H., Kim, J., Ockwig, N. W., O'Keefe, M. & Yaghi, O. M. Control of vertex geometry, structure dimensionality, functionality, and pore metrics in the reticular synthesis of crystalline metal-organic frameworks and polyhedra. *J. Am. Chem. Soc.* **130**, 11650–11651 (2008).
26. Lin, Q., Wu, T., Zheng, S.- T., Bu, X. & Feng, P. Single-walled polytetrazolate metal-organic channels with high density of open nitrogen-donor sites and gas. *J. Am. Chem. Soc.* **134**, 784–787 (2012).
27. Kim, T. K. & Suh, M. P. Selective CO_2 adsorption in a flexible non-interpenetrated metal-organic framework. *Chem. Commun.* **47**, 4258–4260 (2011).
28. Banerjee, R. *et al.* Control of pore size and functionality in isoreticular zeolitic imidazolate frameworks and their carbon dioxide selective capture properties. *J. Am. Chem. Soc.* **131**, 3875–3877 (2009).
29. Morris, W. *et al.* A combined experimental-computational investigation of carbon dioxide capture in a series of isoreticular zeolitic imidazolate frameworks. *J. Am. Chem. Soc.* **132**, 11006–11008 (2010).
30. Banerjee, R. *et al.* High-throughput synthesis of zeolitic imidazolate frameworks and application to CO_2 capture. *Science* **319**, 939–943 (2008).
31. Thallapally, P. K. *et al.* Flexible (breathing) interpenetrated metal-organic frameworks for CO_2 separation applications. *J. Am. Chem. Soc.* **130**, 16842–16843 (2008).
32. Liu, J., Thallapally, P. K., McGrail, B. P., Brown, D. R. & Liu, J. Progress in adsorption-based CO_2 capture by metal-organic frameworks. *Chem. Soc. Rev.* **41**, 2308–2322 (2012).
33. Gadzikwa, T., Farha, O. K., Mulfat, K. L., Hupp, J. T. & Nguyen, S. T. A Zn-based, pillared paddlewheel MOF containing free carboxylic acids *via* covalent post-synthesis elaboration. *Chem. Commun.* **3720**–3722 (2009).
34. Inubushi, Y. *et al.* Modification of flexible part in Cu^{2+} interdigitated framework for CH_4/CO_2 separation. *Chem. Commun.* **46**, 9229–9231 (2010).
35. Mohideen, M. I. H. *et al.* Protecting group and switchable pore-discriminating adsorption properties of a hydrophilic-hydrophobic metal-organic framework. *Nat. Chem.* **3**, 304–310 (2011).
36. Tian, Y.- Q. *et al.* Cadmium imidazolate frameworks with polymorphism, high thermal stability, and a large surface area. *Chem. Eur. J.* **16**, 1137–1141 (2010).
37. Neofotistou, E., Malliakas, C. D. & Trikalitis, P. N. Unprecedented sulfone-functionalized metal-organic frameworks and gas-sorption properties. *Chem. Eur. J.* **15**, 4523–4527 (2009).
38. Liu, X.- M., Lin, R.- B., Zhang, J.- P. & Chen, X.- M. Low-dimensional porous coordination polymers based on 1,2-bis(4-pyridyl)hydrazine: from structure diversity to ultrahigh CO_2/CH_4 selectivity. *Inorg. Chem.* **51**, 5686–5692 (2012).
39. Xiang, S.- C. *et al.* A 3D canted antiferromagnetic porous metal-organic framework with anatase topology through assembly of an analogue of polyoxometalate. *J. Am. Chem. Soc.* **127**, 16352–16353 (2005).
40. Krishna, R. & Long, J. R. Screening metal-organic frameworks by analysis of transient breakthrough of gas mixtures in a fixed bed adsorber. *J. Phys. Chem. C* **115**, 12941–12950 (2011).
41. Chowdhury, P., Mekala, S., Dreisbach, F. & Gumma, S. Adsorption of CO , CO_2 and CH_4 on Cu-BTC and MIL-101 metal organic frameworks: Effect of open metal sites and adsorbate polarity. *Microporous Mesoporous Mater.* **152**, 246–252 (2012).
42. Belmabkhout, Y., Pirngruber, G., Jolimaitre, E. & Methivier, A. A complete experimental approach for synthesis gas separation studies using static gravimetric and column breakthrough experiments. *Adsorption* **13**, 341–349 (2007).
43. Mishra, P., Mekala, S., Dreisbach, F., Mandal, B. & Gumma, S. Adsorption of CO_2 , CO , CH_4 and N_2 on a zinc based metal organic framework. *Sep. Purif. Technol.* **94**, 124–130 (2012).
44. Li, B. *et al.* Enhanced binding affinity, remarkable selectivity, and high capacity of CO_2 by dual functionalization of a rht-type metal-organic framework. *Angew. Chem. Int. Ed.* **51**, 1412–1415 (2012).
45. An, J., Geib, S. J. & Rosi, N. L. High and selective CO_2 uptake in a cobalt adeninate metal-organic framework exhibiting pyrimidine- and amino-decorated pores. *J. Am. Chem. Soc.* **132**, 38–39 (2010).
46. Simmons, J. M., Wu, H., Zhou, W. & Yildirim, T. Carbon capture in metal-organic frameworks—a comparative study. *Energ. Environ. Sci.* **4**, 2177–2185 (2011).
47. Dickey, A. N., Yazaydin, A. Ö., Willis, R. R. & Snurr, R. Q. Screening CO_2/N_2 selectivity in metal-organic frameworks using Monte Carlo simulations and ideal adsorbed solution theory. *Can. J. Chem. Eng.*, doi:10.1002/cjce.20700 (2012).
48. Cavenati, S., Grande, C. A. & Rodrigues, A. E. Adsorption equilibrium of methane, carbon dioxide, and nitrogen on zeolite 13X at high pressures. *J. Chem. Eng. Data* **49**, 1095–1101 (2004).
49. Krishna, R. & van Baten, J. M. In silico screening of metal-organic frameworks in separation applications. *Phys. Chem. Chem. Phys.* **13**, 10593–10616 (2011).
50. Myers, A. L. & Prausnitz, J. M. Thermodynamics of mixed gas adsorption. *AIChE J.* **11**, 121–130 (1965).
51. Herm, Z. R., Swisher, J. A., Smit, B., Krishna, R. & Long, J. R. Metal-organic frameworks as adsorbents for hydrogen purification and pre-combustion carbon dioxide capture. *J. Am. Chem. Soc.* **133**, 5664–5667 (2011).

52. Herm, Z. R., Krishna, R. & Long, J. R. CO₂/CH₄, CH₄/H₂ and CO₂/CH₄/H₂ separations at high pressures using Mg₂(dobdc). *Microporous Mesoporous Mater.* **151**, 481–487 (2012).
53. Ruthven, D. M., Farooq, S. & Knaebel, K. S. *Pressure Swing Adsorption* (VCH Publishers, 1994).
54. Walton, K. S. & LeVan, M. D. Consistency of energy and material balances for bidisperse particles in fixed-bed adsorption and related applications. *Ind. Eng. Chem. Res.* **42**, 6938–6948 (2003).
55. Krishna, R. & Baur, R. Modelling issues in zeolite based separation processes. *Sep. Purif. Technol.* **33**, 213–254 (2003).
56. Bloch, E. D. *et al.* Hydrocarbon separations in a metal-organic framework with open iron(II) coordination sites. *Science* **335**, 1606–1610 (2012).
57. Hudson, M. R. *et al.* Unconventional and highly selective CO₂ adsorption in zeolite SSZ-13. *J. Am. Chem. Soc.* **134**, 1970–1973 (2012).
58. Bae, Y.-S. & Snurr, R. Q. Development and evaluation of porous materials for carbon dioxide separation and capture. *Angew. Chem. Int. Ed.* **50**, 11586–11596 (2011).
59. Burd, S. D. *et al.* Highly selective carbon dioxide uptake by [Cu(bpy-n)₂(SiF₆)] (bpy-1=4,4'-bipyridine; bpy-2=1,2-bis(4-pyridyl)ethene). *J. Am. Chem. Soc.* **134**, 3663–3666 (2012).
60. CRC Handbook of Chemistry and Physics 74th edn (The Chemical Rubber Co., 1993).

Acknowledgements

This work was supported by the Award AX-1730 from Welch Foundation (B.C.) and the Award 'Min-Jiang Scholar Program' in Fujian Province, China (S.X.).

Author contributions

S.X., R.K., B.C. and W.Z. designed the study, analysed the data and wrote the paper. Y.H. and Z.Z. prepared part of MOFs samples and collected the gas adsorption data. H.W. and W.Z. constructed the work of neutron diffraction. R.K. performed the breakthrough analysis. All authors discussed the results and commented on the manuscript.

Additional information

Accession codes: The X-ray crystallographic coordinates for structures reported in this Article have been deposited at the Cambridge Crystallographic Data Centre (CCDC), under deposition number CCDC 869399. These data can be obtained free of charge from The Cambridge Crystallographic Data Centre via www.ccdc.cam.ac.uk/data_request/cif. A summary table of crystal data is also given in the Supporting Information.

Supplementary Information accompanies this paper at <http://www.nature.com/naturecommunications>

Competing financial interests: The authors declare no competing financial interests.

Reprints and permission information is available online at <http://npg.nature.com/reprintsandpermissions/>

How to cite this article: Xiang, S. *et al.* Microporous metal-organic framework with potential for carbon dioxide capture at ambient conditions. *Nat. Commun.* **3**:954 doi: 10.1038/ncomms1956 (2012).