

ARTICLE

Open Access

Polarity control of 2D semiconductor for reconfigurable electronics

Xiaoqian He^{1,2}, Kejie Guan^{2,3}, Fuqin Sun², Xiaoshuang Gou², Lin Liu², Yingyi Wang², Weifan Zhou^{2,3}, Yang Xia^{2,4}, Cheng Zhang^{1,2,4}, Hao Dai^{2,3}, Zhanxia Zhao¹✉, Xiaowei Wang^{2,3}✉ and Ting Zhang^{2,3,5,6}✉

Abstract

The controllable modulation of carrier polarity in semiconductors is essential for enabling dynamic configurations in reconfigurable devices. Ambipolar two-dimensional (2D) semiconductors, characterized by their atomic-scale thickness and excellent gate modulation efficiency, have emerged as highly promising channel materials for such devices. However, existing methods for polarity control encounter challenges in achieving reversible modulation during device operation. Here, we report a novel strategy for reversibly modulating the polarity of ambipolar 2D semiconductors through gate-controlled charge trapping. We demonstrate a double-gate $\text{TaO}_x/\text{WSe}_2/\text{h-BN}$ field-effect transistor, which can reversibly switch between n-type and p-type transport characteristics via electric-field-driven bipolar charge trapping at the $\text{TaO}_x/\text{WSe}_2$ interface. With this method, an electrically configurable complementary inverter is created with a single WSe_2 flake, exhibiting a power consumption of just 0.7 nW. Additionally, a programmable p-n/n-p diode is realized with a $> 100,000$ -fold change in the rectification ratio. These results demonstrate the great potential of gate-controlled bipolar charge trapping for advancing reconfigurable electronics.

Introduction

Reconfigurable electronic devices offer the technology with run-time reconfigurability at the device level, allowing electronic circuits to be programmed according to specific requirements and diverse functionalities^{1,2}. Recent developments in such devices, including field-effect transistors (FETs)^{3,4}, gate-tunable p-n junctions⁵, and photodetectors⁶, have exhibited great potential to enhance circuit functions. Semiconductor polarity control serves as a pivotal technology for advancing these devices⁷, where reversible polarity modulation enables the dynamic reconfiguration of circuit logic functions and operational modes². Ambipolar two-dimensional (2D) semiconductors, such as molybdenum ditelluride (MoTe_2)^{8,9}, tungsten diselenide (WSe_2)^{10,11}, and black phosphorus (BP)^{12,13}, are particularly well-suited for such applications due to their exceptional gate-controlled

capability to rapidly and reversibly switch carriers between holes and electrons¹⁴.

To date, polarity control in ambipolar 2D semiconductors has primarily relied on two distinct strategies. The first strategy involves extrinsic doping techniques—including substitutional doping^{15,16}, chemical intercalation¹⁷, and surface charge transfer^{18,19}—which modulate semiconductor polarity through charge transfer between the dopants and the semiconductor material. However, these methods preclude reversible dynamic modulation after fabrication²⁰. In contrast, the second strategy utilizes electric field gating^{9,21,22} to achieve dopant-free reversible polarity control. This approach, however, requires a continuous power input to maintain field effects during operation, leading to ongoing power consumption that hinders the practical implementation of reconfigurable multifunctional devices²³.

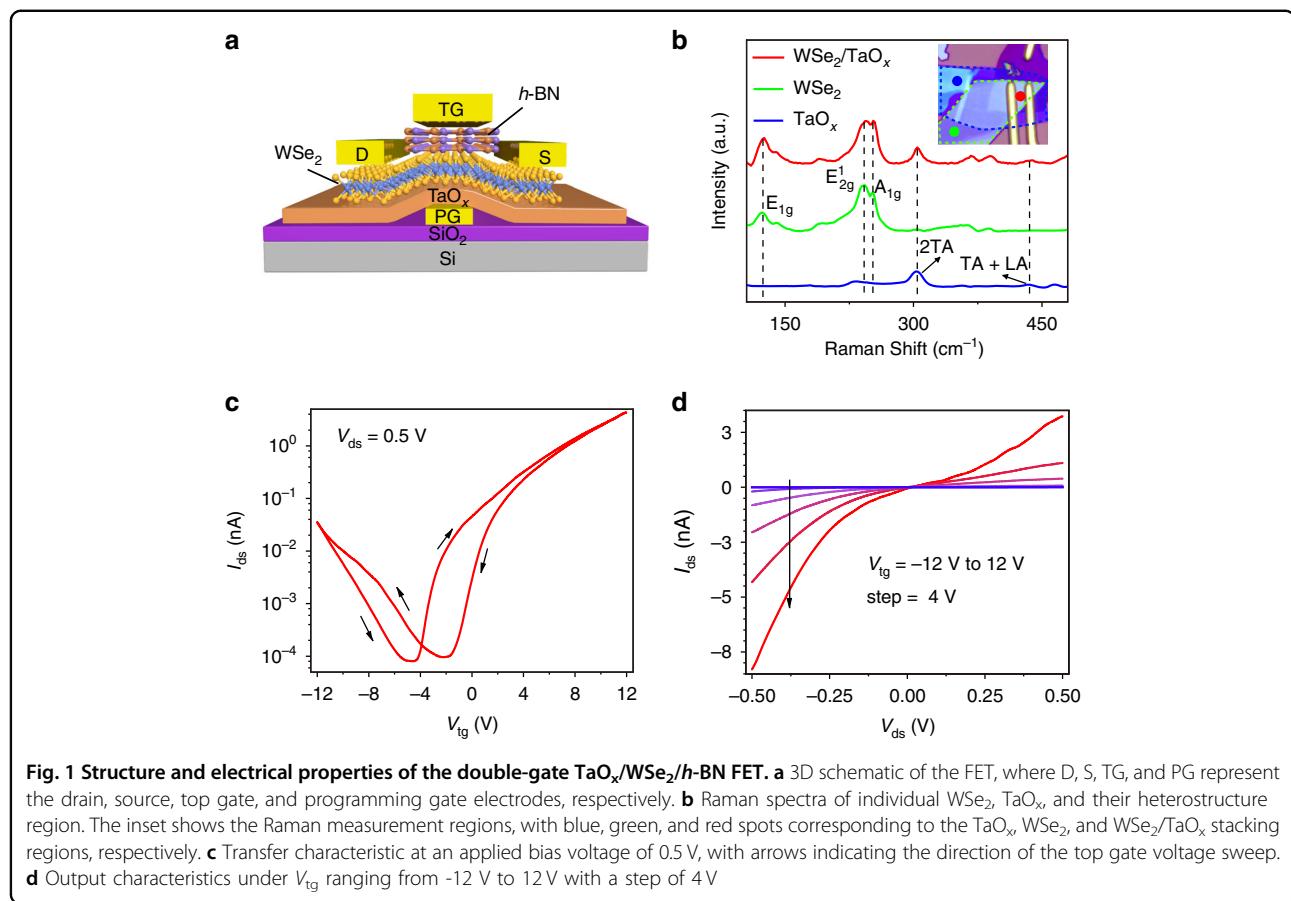
Beyond the aforementioned polarity modulation approaches, recent studies have demonstrated defect engineering strategies for developing reconfigurable electronic devices. Two primary methodologies have emerged: gate-stack-integrated interfacial layers and environmentally responsive approaches. The former

Correspondence: Zhanxia Zhao (zhaozhanxia@shu.edu.cn) or Xiaowei Wang (xwwang2022@sinano.ac.cn) or Ting Zhang (tzhang2009@sinano.ac.cn)

¹College of Sciences, Shanghai University, Shanghai 200444, PR China

²i-Lab, Suzhou Institute of Nano-Tech and Nano-Bionics (SINANO), Chinese Academy of Sciences (CAS), Suzhou, Jiangsu 215123, PR China

Full list of author information is available at the end of the article



utilizes materials such as PbI_2 ²⁴, AlO_x ²⁵, and $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ ²⁶ in gate stacks, where an electric field-modulated charge trapping/detrapping mechanism at the channel-defect layer interface enables the reconfigurability of the devices. The latter employs either gas-phase exposure, leveraging defect-mediated O_2 adsorption and desorption on channel surfaces for reversible polarity switching, or photon-assisted interfacial charge trapping to achieve non-volatile device operation^{27–29}. Crucially, the intrinsic material defects in crystalline structures and the controllable introduction of defects through processing demonstrate that defect engineering is a critical strategy for advancing reconfigurable electronics. The synergistic combination of inevitable crystalline imperfections and tunable defect engineering provides essential guidelines for semiconductor polarity control in reconfigurable electronic devices.

In this work, we demonstrate dynamically controllable polarity modulation of the double-gate $\text{TaO}_x/\text{WSe}_2/h\text{-BN}$ FETs through the gate-controlled bipolar charge trapping at the $\text{TaO}_x/\text{WSe}_2$ interface. By manipulating the polarity of the gate voltage, electrons or holes in the WSe_2 channel can be selectively captured, resulting in electrostatic doping and polarity modulation of the channel. Using this

method, we successfully constructed an electrically configurable complementary inverter within the same WSe_2 flake, representing a power consumption of only 0.7 nW. Additionally, programmable diodes were fabricated based on this strategy, exhibiting reversible transitions between p-n and n-p configurations with a rectification ratio change of over 10^5 .

Results and discussion

The schematic structure of a double-gate $\text{TaO}_x/\text{WSe}_2/h\text{-BN}$ FET is shown in Fig. 1a, consisting of a few-layer WSe_2 as the channel material, TaO_x as the interface layer, $h\text{-BN}$ flake as the gate dielectric and Cr/Au as the gate electrodes and source/drain contacts (the detailed fabrication process is illustrated in Fig. S1). The TaO_x layer was obtained by oxidizing TaS_2 through UV exposure followed by a subsequent heating treatment (see Methods section). The UV treatment enables the rapid formation of oxide layers with nanometer-scale interface uniformity³⁰. It can be observed from atomic force microscopy (AFM) characterizations that the surface roughness of the sample increases slightly with increasing duration of UV exposure while remaining at the nanometer level (Fig. S2a, b), demonstrating interface uniformity. At the same time, the

analysis of sample thickness with different oxidation durations indicates that oxidation reached self-limitation after 25–30 min of UV treatment (Fig. S2c). Thus, the subsequent thermal oxidation process effectively circumvents the intrinsic self-limiting oxidation kinetics of TaS_2 , facilitating the complete conversion to amorphous TaO_x while preserving its structural integrity³¹. Furthermore, this approach simultaneously induces surface reconstruction to reduce roughness (Fig. S2d) and promotes defect formation through high thermal energy input³².

The optical differences between TaO_x and TaS_2 are given in Fig. S3a, b. Due to the larger band gap, TaO_x exhibits greater transparency than TaS_2 ^{33,34}. The Raman spectra depicted in Fig. S3c provide additional evidence for the oxidation of TaS_2 into TaO_x , as demonstrated by the observed shifts in the characteristic peaks^{35,36}. Given that a volume decrease of approximately 25% would be expected if the TaS_2 was transformed into crystalline TaO_x , the observed thickness reduction of only ~2% (Fig. S3d-f) with unchanged lateral dimensions suggests that the TaO_x is a likely amorphous structure, exhibiting a significantly lower density compared to crystalline TaO_x ^{37,38}. It is this lower-density non-crystalline structure that enables atmospheric oxygen to penetrate deeply into the material, facilitating uniform and complete thermal oxidation throughout the sample. Raman characterization is also employed to identify the WSe_2 , TaO_x , and heterostructure region (Fig. 1b). For TaO_x , the Raman peaks observed at 302 cm^{-1} and 435 cm^{-1} correspond to the 2TA and TA + LA modes, respectively, which arise from two-phonon scattering processes³⁵. The Raman spectra of WSe_2 exhibit three distinct peaks at 137 cm^{-1} , 248 cm^{-1} , and 251 cm^{-1} , corresponding to the E_{1g} , E_{2g} , and A_{1g} vibrational modes of the WSe_2 flakes^{39,40}. Notably, the Raman spectrum of the $\text{WSe}_2/\text{TaO}_x$ heterostructure region contains all the characteristic peaks of both materials, confirming the successful stacking of the TaO_x and WSe_2 flakes.

Figure 1c, d show the electrical performance of this double-gate FET. The transfer characteristic in Fig. 1c shows a typical ambipolar behavior with an on/off ratio of approximately 10^5 for n-type transport and over 10^2 for p-type transport, indicating that n-type behavior dominates in the WSe_2 FET. The observed hysteresis between the forward and backward sweeps can mainly be attributed to charge trapping effects at the interface of $\text{TaO}_x/\text{WSe}_2$ or $\text{WSe}_2/h\text{-BN}$ ^{41,42}. The output characteristics in Fig. 1d show a significant increase in current as the gate voltage rises, demonstrating the effective modulation of the double-gate FET by the gate electric field.

To investigate the hysteresis mechanism observed in the double-gate FET, we first compared the hysteresis width (ΔV) of FETs with and without the stacked TaO_x layer. The ΔV is extracted by calculating the voltage difference

between forward and backward sweeps at a current level of 10^{-3} nA for the electron transport branch. As shown in Fig. S4, the transfer characteristics of the WSe_2 FET with TaO_x stacking exhibit more significant hysteresis, indicating that the hysteresis behavior is related to the TaO_x layer. Then, a comparison of the ΔV in double-gate FETs based on TaO_x prepared by different UV exposure durations (all followed by $300\text{ }^\circ\text{C}/3\text{ h}$ thermal treatment) reveals a larger hysteresis in the 1.5 h device (7.90 V) versus the 5 min (1.68 V) and 30 min (6.72 V) devices (Fig. S5), once again proving that the hysteresis behavior of the device originates from the TaO_x layer. Therefore, we ascribe the hysteresis in our devices to the presence of TaO_x under the WSe_2 channel.

Additionally, transfer characteristics under varying top-gate voltage (V_{tg}) sweep ranges and rates are obtained to further research this mechanism, as illustrated in Fig. 2a, b. It can be found that the ΔV enlarges with the increase of gate voltage sweeping range or with the decrease of sweeping rate, as shown in Fig. 2c, d, respectively, suggesting that the hysteresis observed in our devices originates from charge trapping under the gate electric stress^{42,43}. In particular, as shown in Fig. 2e, when $V_{tg} < 0$, the electric field at the $\text{WSe}_2/\text{TaO}_x$ interface is oriented downward, which promotes the migration of holes from the WSe_2 channel to the TaO_x surface. These holes are subsequently trapped by defects, creating a positive electrostatic field that shifts the threshold voltage negatively for the forward sweep. Conversely, when $V_{tg} > 0$, electrons are driven and captured, resulting in a negative electrostatic field that causes a positive shift of the threshold voltage for the backward sweep. This bipolar charge trapping behavior, driven by the top-gate electric field, thus generates a clockwise hysteresis for the electron transport branch and a counterclockwise hysteresis for the hole transport branch. As the gate voltage sweeping range increases or the sweeping rate decreases, a greater number of carriers will migrate to and get trapped at the $\text{WSe}_2/\text{TaO}_x$ interface, thus leading to an enlarged hysteresis.

Leveraging the established thickness dependence of WSe_2 carrier polarity¹¹, we explore the potential of this bipolar charge trapping in polarity modulation of double-gate FETs fabricated with p-type (~4 nm), ambipolar (~7 nm), and n-type (~10 nm) WSe_2 flakes (Figure S6), respectively. To investigate dynamic modulation effects, voltage pulses with different polarities are applied to the PG electrode. As shown in Fig. 3a, the conductivity type of the WSe_2 FET changes gradually from its initial n-type to a programmed p-type after being programmed by a positive voltage pulse ($V_{PG} > 0$). And a negative voltage pulse ($V_{PG} < 0$) can reverse this change. Conversely, the pristine p-type WSe_2 FET undergoes a transition to a programmed n-type under a negative program voltage

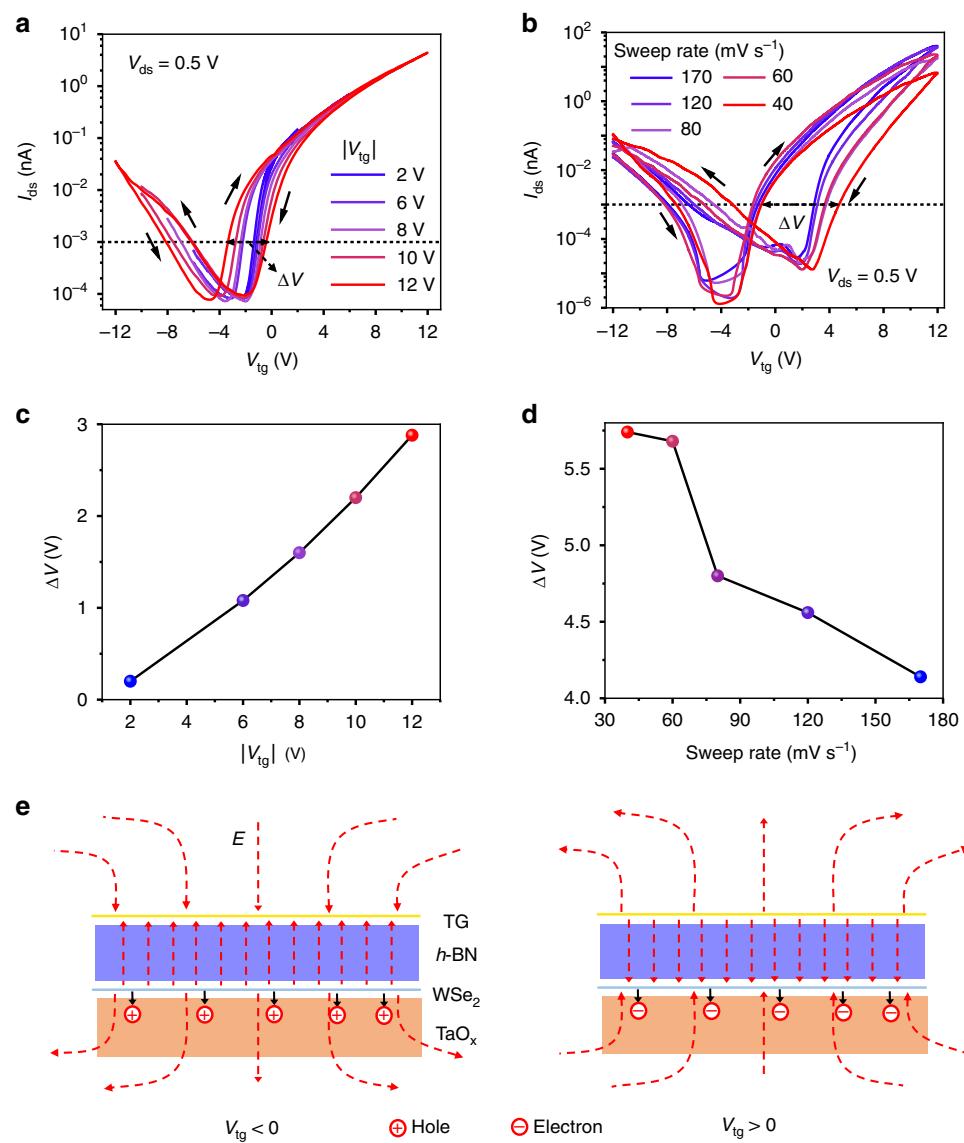
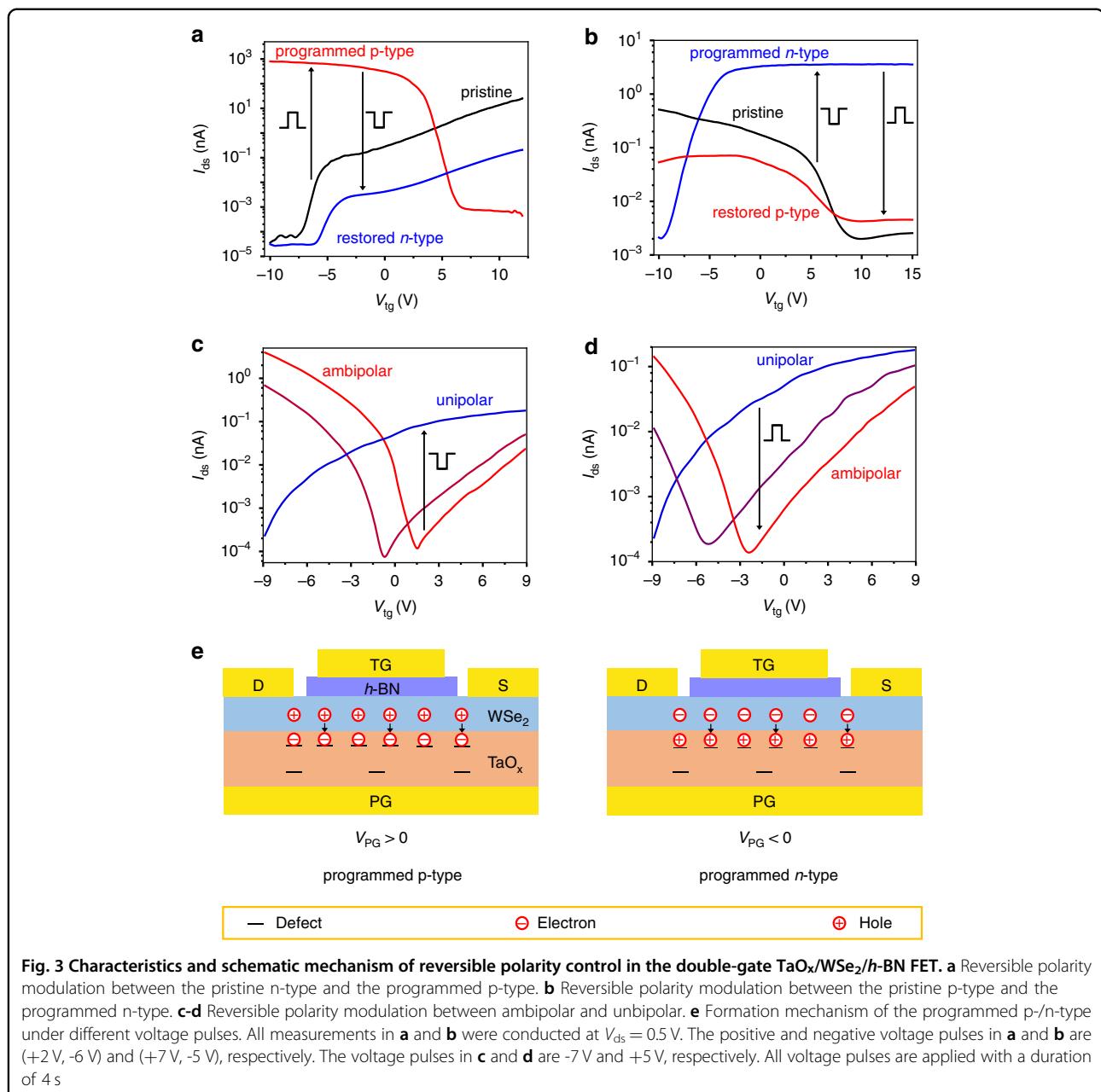


Fig. 2 Characteristics and principal mechanism of the hysteresis in the double-gate TaO_x/WSe₂/h-BN FET. Transfer characteristics at different V_{tg} **a** sweeping ranges and **b** sweeping rates. Variation of hysteresis at different V_{tg} **c** sweeping ranges **d** sweeping rates. **e** Principal mechanism underlying the hysteresis behavior. The solid arrows in **a** and **b** indicate the direction of the top-gate voltage sweep. ΔV and the dashed line represent the size and the current level of the hysteresis window, respectively. The dashed arrows in **e** show the direction of the top gate electric field

pulse, which can also be reversed by altering the pulse polarity, as shown in Fig. 3b. Furthermore, reversible changes between ambipolar and unipolar transport characteristics are also achieved in the WSe₂ FETs through the gate-controlled bipolar charge trapping (Fig. 3c, d). The detailed processes corresponding to Fig. 3a–d are provided in Fig. S7. This polarity control exhibits a retention time exceeding 10³ s (Fig. S8). Meanwhile, prolonged pulse durations consistently enhance the efficacy of WSe₂ polarity modulation, indicating an effect similar to that observed when manipulating the amplitude of the pulse voltage (Fig. S9). These findings confirm the

great advantage of gate-controlled bipolar charge trapping in realizing reversible polarity control for the ambipolar 2D semiconductor.

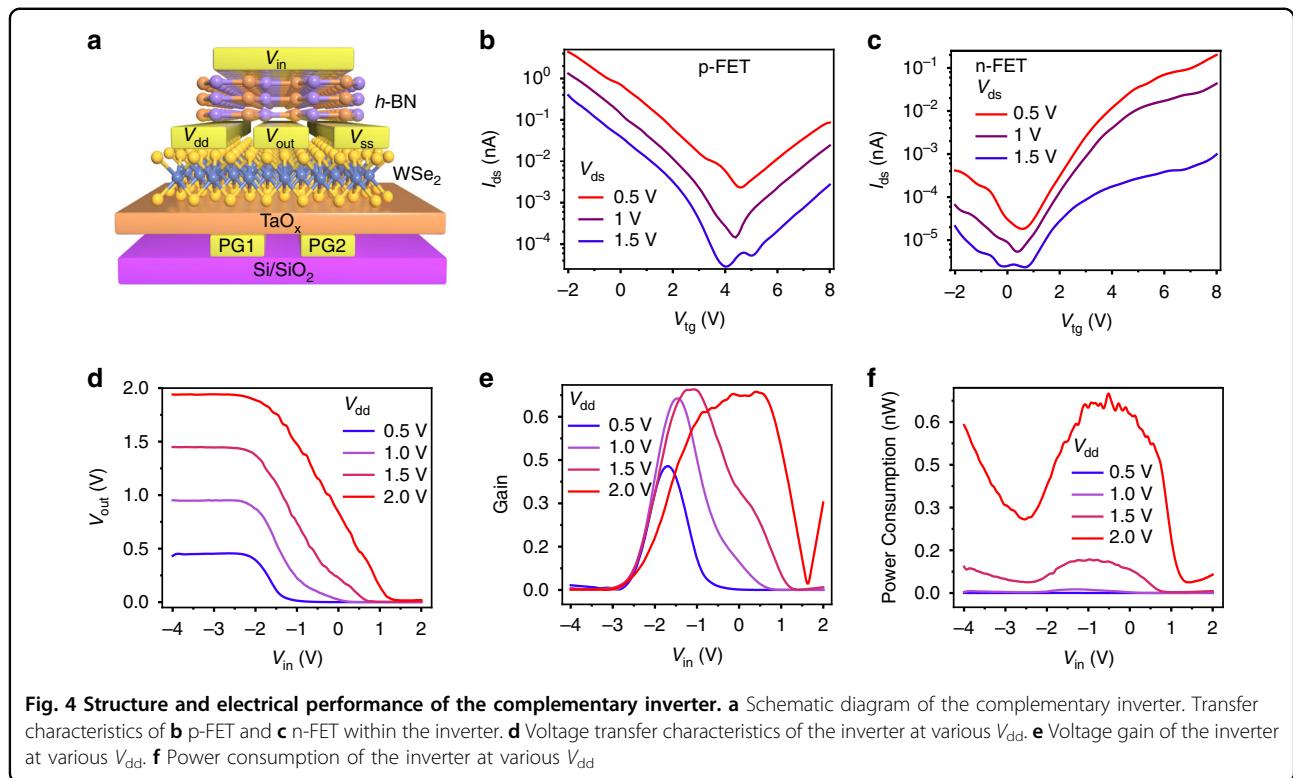
The detailed mechanism underlying the reversible polarity modulation of the double-gate TaO_x/WSe₂/h-BN FET is schematized in Fig. 3e. As discussed earlier regarding the hysteresis mechanism, when $V_{PG} > 0$, electrons, which are driven and captured at the TaO_x/WSe₂ interface by the gate electric field, induce electrostatic doping to accumulate holes in the WSe₂ channel, resulting in a programmed p-type polarity. Conversely, when $V_{PG} < 0$, holes are trapped to induce electron



accumulation and thus form a programmed n-type polarity. Therefore, the polarity of such a double-gate WSe_2 FET can be reversibly modulated by gate-controlled bipolar charge trapping through altering the polarity of the PG voltage.

A complementary inverter was fabricated based on a single WSe_2 flake to investigate the feasibility of this polarity regulation method for reconfigurable devices. As illustrated in Fig. 4a, the complementary inverter consists of two WSe_2 transistors connected in series, with each transistor controlled by an independent PG (PG1 and PG2). Positive and negative voltage pulses are applied to PG1 and PG2, respectively, to establish a p-type and an

n-type FET, as evidenced by their distinct transfer characteristics shown in Fig. 4b, c. Subsequently, the voltage transfer curves, plot of input (V_{in}) versus output voltage (V_{out}), of the complementary inverter under different supply voltages (V_{dd}) were measured, as shown in Fig. 4d. Signal inversions are observed with high V_{out} at negative V_{in} and low V_{out} at positive V_{in} , which further demonstrates that both p- and n-type FETs are successfully achieved through polarity modulation. Figure 4e presents the voltage gain of the programmable inverter, defined as $|\Delta V_{out}/\Delta V_{in}|$, with a maximum gain of 0.7. Notably, the complementary inverter exhibits a peak power consumption of 0.7 nW at $V_{dd} = 2.0$ V (Fig. 4f), indicating



that gate-controlled charge trapping is an effective strategy for constructing low-power logic devices.

To further explore the potential of this strategy in reconfigurable electronic devices, we created a programmable diode using a single WSe₂ flake. Figure 5a, b shows that the programmable diode has two PGs, PG1, and PG2, which are designed to simultaneously modulate the transport type of the WSe₂ channel by programming with opposite voltage pulses. Figure 5c shows that the transfer behavior gradually changes from p-n type to n-p type after being programmed by negative and positive voltage pulses applied to the PG1 and PG2 electrodes, respectively. This transition can be accurately quantified by the change of the current rectification ratio $|I_+/I_-|$, where I_+ and I_- represent the current values at bias voltages of +2 V and -2 V, respectively. As shown in Fig. 5d, the rectification ratio of the diode changes from 6672 to 0.06, indicating a transition from a p-n diode to an n-p diode after programming. Notably, the formed n-p diode can be reversed into the p-n mode with an over 10^5 -fold change in rectification ratio by exchanging the polarity of the voltage pulses applied to PG1 and PG2 electrodes, as shown in Fig. 5e. These results demonstrate reversible conversions between p-n and n-p configurations of the WSe₂ channel and highlight the great potential of the gate-controlled charge trapping strategy in reconfigurable electronics.

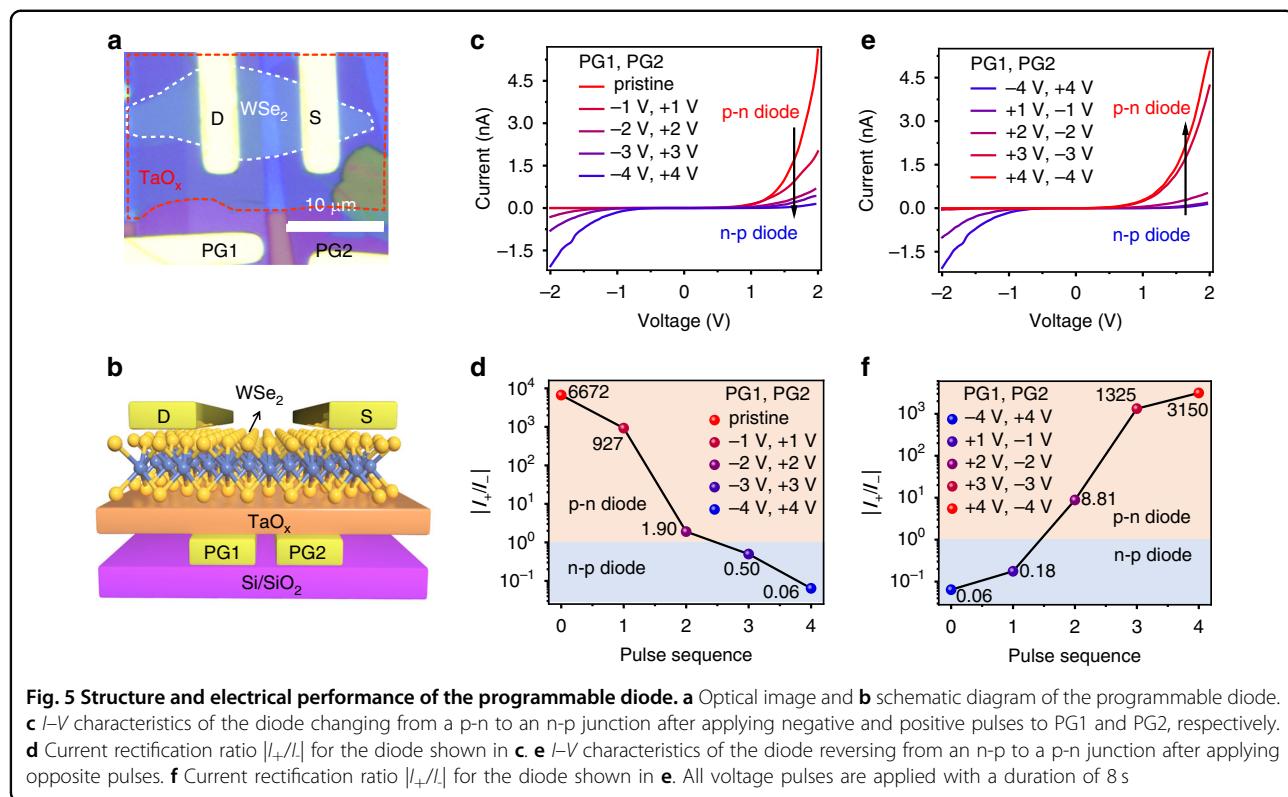
Conclusion

A strategy for reversible polarity modulation in ambipolar 2D FETs via the gate-controlled bipolar charge trapping mechanism is proposed. By inserting a TaO_x layer into the gate stack of WSe₂ FET, the conductive characteristics can be reversibly switched between n-type and p-type behavior by controlling the polarity of the PG voltage pulses. Based on this strategy, an electrically configurable complementary inverter with a power consumption of 0.7 nW and a programmable diode with a 100,000-fold change in rectification ratio were successfully fabricated using a single WSe₂ flake, respectively. Our work demonstrates that gate-controlled bipolar charge trapping is a promising method for reconfigurable electronics.

Materials and methods

Fabrication of the double-gate TaO_x/WSe₂/h-BN FET

The WSe₂, TaS₂, and h-BN flakes were obtained by mechanical exfoliation from their bulk crystals. The thickness of p-type, ambipolar, and n-type WSe₂ flakes are ~4 nm, ~7 nm, and ~10 nm, respectively. The TaS₂ thickness was controlled in the range of 20–50 nm. The stacked TaS₂ flake and WSe₂/h-BN heterostructure were produced with a dry transfer technique, as reported. TaO_x was obtained from TaS₂ through 1.5 h of UV exposure and 3 hours of heating at 300 °C. All electrodes, including the PG (5 nm Cr/15 nm Au), the S/D (5 nm Cr/50 nm



Au), and the TG (5 nm Cr/70 nm Au), were defined using the MicroWriter ML 3 Pro system (Durham Magneto Optics) followed by metal thermal evaporation and lift-off process. A detailed schematic of the fabrication process for the double-gate $\text{TaO}_x/\text{WSe}_2/\text{h-BN}$ FET is provided in Fig. S1.

Characterization and measurement of the double-gate $\text{TaO}_x/\text{WSe}_2/\text{h-BN}$ FET

Raman spectra were acquired using an Invia Qontor instrument (Renishaw) with a 532 nm laser of 5 W power under ambient conditions. The measure setting parameters of samples are shown as follows: 1. accumulations: 50; 2. acquire time: 5 s; 3. grating: 1800 (532 nm); 4. ND filter: 10%; 5. Objective: 50x_LWD. The sample thicknesses were measured by atomic force microscopy (AFM, Oxford Instruments MFP-3D AFM) using an Otespa-R3 probe in tapping mode, with optimized parameters: 1. scan rate: 1.0 Hz; 2. samples/line: 256; 3. integral gain: 0.5; 4. proportional gain: 5.0; 5. drive frequency: 300 kHz; 6. amplitude setpoint: 120 mV; 7. drive amplitude: 120 mV. All electrical measurements were conducted in a vacuum probe station using a Keithley 4200A-SCS system.

Acknowledgements

We acknowledge the funding support from the National Natural Science Foundation of China (Grant No. 62371448, 62125112, U24A20228, 62204254, 62301553, 62401563, 52270130), the Strategic Priority Research Program of the

Chinese Academy of Science (XDB0520301), the Natural Science Foundation of Jiangsu Province (BK20243004), the Foundation Research Project of Jiangsu Province (BK20230234), the Basic Research Development Program of Suzhou (SJC2023004) and the China Postdoctoral Science Foundation (2023M742556). The authors are grateful for the technical support of AFM and Photolithography from the Nano-X Vacuum Interconnected Workstation of Suzhou Institute of Nano-Tech and Nano-Bionics (SINANO), Chinese Academy of Sciences (CAS).

Author details

¹College of Sciences, Shanghai University, Shanghai 200444, PR China. ²i-Lab, Suzhou Institute of Nano-Tech and Nano-Bionics (SINANO), Chinese Academy of Sciences (CAS), Suzhou, Jiangsu 215123, PR China. ³School of Nano-Tech and Nano-Bionics, University of Science and Technology of China, Hefei, Anhui 230026, PR China. ⁴Key Laboratory of Efficient Low-carbon Energy Conversion and Utilization of Jiangsu Provincial Higher Education Institutions, School of Physical Science and Technology, Suzhou University of Science and Technology, Suzhou 215009, PR China. ⁵Key Laboratory of Semiconductor Display Materials and Chips, Suzhou Institute of Nano-Tech & Nano-Bionics (SINANO), Chinese Academy of Sciences (CAS), Suzhou, Jiangsu 215123, PR China. ⁶Nano-X Vacuum Interconnected Workstation, Suzhou Institute of Nano-Tech & Nano-Bionics (SINANO), Chinese Academy of Sciences (CAS), Suzhou, Jiangsu 215123, PR China

Conflict of interest

Ting Zhang is an Editor for the journal, no other author has reported any competing interest.

Supplementary information The online version contains supplementary material available at <https://doi.org/10.1038/s41378-025-01029-8>.

Received: 28 February 2025 Revised: 3 June 2025 Accepted: 5 July 2025
Published online: 29 September 2025

References

1. Zhang, H. T. et al. Reconfigurable perovskite nickelate electronics for artificial intelligence. *Science* **375**, 533–539 (2022).
2. Fei, W., Trommer, J., Lemme, M. C., Mikolajick, T. & Heinzig, A. Emerging reconfigurable electronic devices based on two-dimensional materials: a review. *InfoMat* **4**, e12355 (2022).
3. Zhao, Z., Kang, J., Rakheja, S. & Zhu, W. Control-gate-free reconfigurable transistor based on 2D MoTe₂ with asymmetric gating. *Appl. Phys. Lett.* **124**, 073506 (2024).
4. Tsai, M. Y. et al. A reconfigurable transistor and memory based on a two-dimensional heterostructure and photoinduced trapping. *Nat. Electron.* **6**, 755–764 (2023).
5. Li, D., Wang, B., Chen, M., Zhou, J. & Zhang, Z. Gate-controlled BP-WSe₂ heterojunction diode for logic rectifiers and logic optoelectronics. *Small* **13**, 1603726 (2017).
6. Huang, J. et al. Reconfigurable WSe₂ Schottky heterojunctions for logic rectifiers and ultrafast photodetectors. *Sci. China Mater.* **66**, 4711–4722 (2023).
7. Peng, R. et al. Programmable graded doping for reconfigurable molybdenum ditelluride devices. *Nat. Electron.* **6**, 852–861 (2023).
8. Sun, X. et al. Contact and injection engineering for low SS reconfigurable FETs and high gain complementary inverters. *Sci. Bull.* **65**, 2007–2013 (2020).
9. Larentis, S. et al. Reconfigurable complementary monolayer MoTe₂ field-effect transistors for integrated circuits. *ACS Nano* **11**, 4832–4839 (2017).
10. Resta, G. V. et al. Doping-free complementary logic gates enabled by two-dimensional polarity-controllable transistors. *ACS Nano* **12**, 7039–7047 (2018).
11. Wang, Z. et al. Controllable doping in 2D layered materials. *Adv. Mater.* **33**, 2104942 (2021).
12. Wu, P. et al. Complementary black phosphorus tunneling field-effect transistors. *ACS Nano* **13**, 377–385 (2018).
13. Wu, P., Reis, D., Hu, X. S. & Appenzeller, J. Two-dimensional transistors with reconfigurable polarities for secure circuits. *Nat. Electron.* **4**, 45–53 (2021).
14. Hu, W. et al. Ambipolar 2D semiconductors and emerging device applications. *Small Methods* **5**, 2000837 (2021).
15. Nipane, A., Karmakar, D., Kaushik, N., Karande, S. & Lodha, S. Few-layer MoS₂ p-type devices enabled by selective doping using low energy phosphorus implantation. *ACS Nano* **10**, 2128–2137 (2016).
16. Dolui, K., Rungger, I., Das Pemmaraju, C. & Sanvito, S. Possible doping strategies for MoS₂ monolayers: an ab initio study. *Phys. Rev. B* **88**, 075420 (2013).
17. Gong, Y. et al. Spatially controlled doping of two-dimensional SnS₂ through intercalation for electronics. *Nat. Nanotechnol.* **13**, 294–299 (2018).
18. Chang, Y. M. et al. Reversible and precisely controllable p/n-type doping of MoTe₂ transistors through electrothermal doping. *Adv. Mater.* **30**, 1706995 (2018).
19. Fang, H. et al. High-performance single layered WSe₂ p-FETs with chemically doped contacts. *Nano Lett.* **12**, 3788–3792 (2012).
20. Hu, H. et al. Recent advances in doping engineering of black phosphorus. *J. Mater. Chem. A* **8**, 5421–5441 (2020).
21. Beck, M. E. & Hersam, M. C. Emerging opportunities for electrostatic control in atomically thin devices. *ACS Nano* **14**, 6498–6518 (2020).
22. Wang, J. et al. Recent progress on localized field enhanced two-dimensional material photodetectors from ultraviolet-visible to infrared. *Small* **13**, 1700894 (2017).
23. Luo, P. et al. Doping engineering and functionalization of two-dimensional metal chalcogenides. *Nanoscale Horiz.* **4**, 26–51 (2019).
24. Liu, C. et al. Highly reliable van der Waals memory boosted by a single 2D charge trap medium. *Adv. Mater.* **36**, 2305580 (2024).
25. Tan, Z. D. et al. Reconfigurable logic and in-memory computing based on electrically controlled charge trapping in dielectric engineered 2D semiconductor transistors. *Adv. Funct. Mater.* **35**, 2417887 (2025).
26. Wang, X., Liu, X., Zhou, Y., Yuan, K. & Zhou, D. Z. Reconfigurable Quasi-nonvolatile memory/subthermionic FET functions in ferroelectric–2D semiconductor vdW architectures. *Adv. Mater.* **34**, 2200032 (2024).
27. Chen, F. et al. Reversible charge-polarity control for multioperation-mode transistors based on van der Waals heterostructures. *Adv. Sci.* **9**, 2106016 (2022).
28. Kim, S. W. et al. Nonvolatile reconfigurable logic device based on photo-induced interfacial charge trapping in van der Waals gap. *ACS Appl. Mater. Interfaces* **16**, 22131–22138 (2024).
29. Zhang, W. et al. Synchronous regulation of hydrophobic molecular architecture and interface engineering for robust WORM-type memristor. *Adv. Funct. Mater.* **34**, 2404625 (2024).
30. Zhang, Y. et al. A single-crystalline native dielectric for two-dimensional semiconductors with an equivalent oxide thickness below 0.5 nm. *Nat. Electron.* **5**, 643–649 (2022).
31. Chamlagain, B. et al. Thermally oxidized two-dimensional TaS₂ as a high-k gate dielectric for MoS₂ field-effect transistors. *2D Mater.* **4**, 031002 (2017).
32. Li, Y. et al. PbP-level NO₂ sensing properties at room temperature of ultra-thin SnS₂ nanopetals annealed at different temperatures. *Sens. Actuators B: Chem.* **370**, 132398 (2022).
33. Shvets, V. A. et al. Electronic structure and charge transport properties of amorphous Ta₂O₅ films. *J. Non-crystalline Solids* **354**, 3025–3033 (2008).
34. Zhao, R. et al. Tuning phase transitions in 1T-TaS₂ via the substrate. *Nano Lett.* **17**, 3471–3477 (2017).
35. Gritsenko, V. A. et al. Nanoscale potential fluctuations in nonstoichiometric tantalum oxide. *Nanotechnology* **29**, 425202 (2018).
36. Fan, S. et al. Excitations of intercalated metal monolayers in transition metal dichalcogenides. *Nano Lett.* **21**, 99–106 (2020).
37. Revelli, J. F. & Disalvo, F. J. Tantalum disulfide (TaS₂) and its intercalation compounds. *Inorg. Syntheses: Nonmolecular Solids* **30**, 155–169 (1995).
38. Reisman, A., Holtzberg, F., Berkenblit, M. & Berry, M. Reactions of the group VB pentoxides with alkali oxides and carbonates. III. Thermal and X-Ray phase diagrams of the system K₂O or K₂CO₃ with Ta₂O₅. *J. Am. Chem. Soc.* **78**, 4514–4520 (1956).
39. Terrones, H. et al. New first order Raman-active modes in few layered transition metal dichalcogenides. *Sci. Rep.* **4**, 4215 (2014).
40. Li, Y., Xiao, J., Cao, X., Gu, Z. & Zhang, W. Lateral WSe₂ homojunction through metal contact doping: excellent self-powered photovoltaic photodetector. *Adv. Funct. Mater.* **33**, 2213385 (2023).
41. Yang, F. S. et al. Oxidation-boosted charge trapping in ultra-sensitive van der Waals materials for artificial synaptic features. *Nat. Commun.* **11**, 2972 (2020).
42. Lan, C. et al. The origin of gate bias stress instability and hysteresis in monolayer WS₂ transistors. *Nano Res.* **13**, 3278–3285 (2020).
43. Xiao, B. & Watanabe, S. Oxygen vacancy effects on an amorphous-TaO_x-based resistance switch: a first principles study. *Nanoscale* **6**, 10169–10178 (2014).