

Harmonization of rapid triplet up-conversion and singlet radiation enables efficient and stable white OLEDs

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White organic light-emitting diodes (WOLEDs) hold significant promise in illumination and displays, but achieving high efficiency while maintaining stability is an ongoing challenge. Here, we strategically combine a blue donor-acceptor thermally activated delayed fluorescence (TADF) emitter featuring rapid reverse intersystem crossing rate and a yellow multi-resonance TADF emitter renowned for the fast radiative transition process to achieve warm WOLEDs with exceptional power efficiency exceeding 190 lm W^{-1} and external quantum efficiency (EQE) of 39%, setting records for WOLEDs. Meanwhile, these devices also exhibit an extended operational lifetime (LT_{80}) of 446 h at an initial luminance of 1000 cd m^{-2} . Another group of blue and yellow emitters based on our strategy achieves a standard white emission and a high EQE of 35.6%, confirming the universality of our strategy. This work presents a versatile strategy to harmonize singlet exciton radiation and triplet exciton up-conversion, thus achieving a win-win situation of efficiency and stability.

White organic light-emitting diodes (WOLEDs) with high efficiency and excellent stability have garnered increasing attention^{1–4}, especially those based on all thermally activated delayed fluorescence (TADF) emitters^{5–8}. Without the participation of expensive noble metals, all-TADF WOLEDs can achieve comparable performance with phosphorescent devices, making them more cost-effective and potential for practical application^{9–11}. Thanks to previous efforts, remarkable efficiencies have been realized in all-TADF WOLEDs with the records of power efficiency (PE) and external quantum efficiency (EQE) standing at 130 lm W^{-1} and 32.8%, respectively^{3,10}. However, their device stability remains inferior, impeding the commercialization of all-TADF WOLEDs. There is an urgent need to propose effective strategies to improve stability while maintaining high efficiency in all-TADF WOLEDs.

As widely recognized, the primary contributor to the diminished stability of TADF devices is the accumulation of long-lived triplet excitons^{7,9,12,13}. To alleviate this issue, a common approach involves

selecting donor-acceptor (D-A) type TADF emitters that exhibit a high reverse intersystem crossing rate (k_{RISC}) for fabricating all-TADF WOLEDs. This facilitates the fast conversion of “dark” triplet (T_1) excitons into “bright” singlet (S_1) excitons, suppressing exciton annihilation events such as the triplet-triplet annihilation and triplet-polaron annihilation^{14,15}. However, the twisted structure^{5,6,16} inherent in D-A type TADF emitters poses constraints on their radiative transition rate (k_r), resulting in a bottleneck for further boosting the efficiency of all-TADF WOLEDs. In contrast, multi-resonance TADF (MR-TADF) emitters, which induce the TADF property through the multi-resonance effect facilitated by *ortho*-substituted boron and nitrogen planar skeleton^{17–22}, circumvent such torsional limitations and exhibit enhanced k_r . Consequently, MR-TADF emitters not only offer a high k_r comparable to that of conventional fluorescent materials but also surpass them in exciton utilization efficiency, favoring higher device efficiency. Nonetheless, a drawback of MR-TADF emitters is their tendency to exhibit much slower k_{RISC} than D-A type TADF emitters,

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creating a trade-off scenario where efficiency may improve while device stability becomes a concern^{23,24}. Consequently, achieving a harmonious balance between efficiency and stability remains a formidable challenge when solely relying on one type of emitter.

Here, to address the contradiction, we develop an attractive approach by employing D-A type TADF emitters for the blue emission component and MR-TADF emitters for the yellow emission component in a dual-color all-TADF WOLED configuration (Fig. 1). The blue D-A type TADF emitters, characterized by high k_{RISC} , efficiently drive the up-conversion of triplet excitons. A fraction of these excitons undergo radiative transitions, leading to blue light emission, while the remaining excitons swiftly undergo Förster-type energy transfer (FET) to access the S_1 state of yellow MR-TADF emitters. Subsequently, taking advantage of the higher k_r of yellow MR-TADF emitters, they promptly engage in radiative transition, emitting the yellow component. This orchestrated process harmonizes triplet up-conversion and singlet radiation, ensuring efficient utilization of excitons to achieve dual objectives of enhancing both efficiency and stability. Specifically, we selected D-A type TADF molecule 2,3,4,5,6-pentakis(3,6-di-*tert*-butyl-9*H*-carbazol-9-yl)benzotrile (5TCzBN)^{25,26} as the blue emitter for our dual-color all-TADF WOLEDs due to its high k_{RISC} value of $6.1 \times 10^5 \text{ s}^{-1}$. As for the yellow emitter, we selected a typical MR-TADF molecule, 14,17-di-*tert*-butyl-5,5-bis(3,5-di-*tert*-butylphenyl)-5*H*-11*b*,18*b*,21*b*-triazab-boraanthra[1,9-*ab*]fluoreno[1,9-*jk*]perylene (MR-Y)²⁷, characterized by its high $k_{\text{r,s}}$ of $5.4 \times 10^7 \text{ s}^{-1}$ and remarkable quantum yield of 98%. Using these emitters, we successfully fabricated a dual-emission-layer warm WOLEDs^{8,9,28-30} with record-beating PE and EQE of exceeding 190 lm W^{-1} and 39%, respectively, accompanied by an improved efficiency roll-off compared to the device without D-A type TADF molecule as sensitizer. A detailed mechanistic analysis unveils the synergistic effect arising from the combination of D-A type TADF and MR-TADF emitters increases the rate of Förster-type energy transfer (k_{FET}) from $2.7 \times 10^7 \text{ s}^{-1}$ to $5.6 \times 10^7 \text{ s}^{-1}$, thereby augmenting the population of radiative singlet excitons. Simultaneously, the rate of Dexter-type energy transfer (k_{DET}) decreases from $2.0 \times 10^7 \text{ s}^{-1}$ to $2.4 \times 10^5 \text{ s}^{-1}$, mitigating the accumulation of dark triplet excitons at MR-TADF emitters. Consequently, the device lifetime improves by 4.5 times compared to the control device, achieving an extended LT_{80} (operational time to 80% of the initial luminance)

of 446 h at an initial luminance of 1000 cd m^{-2} , making it the most stable all-TADF warm WOLEDs^{8,13,31}. Furthermore, utilizing the proposed strategy, another combination of D-A type TADF and MR-TADF emitters achieves standard white emission with CIE (x,y) coordinates of (0.33,0.34) and a high EQE of 35.6%, confirming the universality of our strategy. Additionally, through precise optimization of exciton recombination regions, we effectively regulated the CIE coordinates, transitioning from (0.28, 0.47) to (0.35, 0.53), thus accommodating diverse illumination requirements³²⁻³⁴. This work pioneers the integration of MR-TADF materials into all-TADF WOLEDs, thereby laying a solid foundation for the design of highly efficient and stable all-TADF WOLEDs.

Results

Prior to fabricating WOLED devices, we investigated the performance of blue devices without doping MR-Y, using the following structure: indium-tin oxide (ITO)/1,4,5,8,9,11-hexaazatriphenylenehexacarbonitrile (HAT-CN, 5 nm)/1,1-bis[(di-4-tolylamino)phenyl]cyclohexane (TAPC, 30 nm)/tris(4-carbazolyl-9-ylphenyl)amine (TCTA, 15 nm)/3,3-di(9*H*-carbazol-9-yl)biphenyl (mCBP, 10 nm)/emitting layer (EML, 30 nm)/(1,3,5-triazine-2,4,6-triyl)tris(benzene-3,1-diyl)tris(diphenylphosphine oxide) (POT2T, 20 nm)/1-(4-(10-(1,1'-biphenyl)-4-yl)anthracen-9-yl)phenyl)-2-ethyl-1*H*-benzo[*d*]imidazole (ANT-BIZ, 30 nm)/8-hydroxyquinolinato lithium (Liq, 2 nm)/alumina (Al, 100 nm). In this configuration, HAT-CN serves as the hole injection layer, while both TAPC and TCTA act as the hole transporting layers. ANT-BIZ functions as the electron transporting layer. To confine exciton diffusion, mCBP and POT2T are employed as the electron-blocking layer and hole-blocking layer, respectively. The emitting layer consists of blue emitter of 5TCzBN and host of 1,3-dihydro-1,1-dimethyl-3-(3-(4,6-diphenyl-1,3,5-triazin-2-yl)phenyl)indeno[2,1-*b*]carbazole (DMIC-TRZ). These blue devices exhibit an emission peak at 490 nm with EQEs ranging from 18.3% to 22.0%. Notably, the device featuring a 20 wt% doping of 5TCzBN demonstrates the highest efficiency and minimal efficiency roll-off, as depicted in Supplementary Fig. 1 and Supplementary Table 1.

Hence, we fabricated single-emission-layer WOLEDs labeled as devices A and B, with the doping concentration of 5TCzBN fixed at 20%. The device structure is depicted in Fig. 2a, in which the EMLs of devices A and B consist of 0.3 wt% and 0.5 wt% MR-Y doped into 20 wt% 5TCzBN:DMIC-TRZ (the chemical structures of the relevant materials are shown in Fig. 2b), respectively. The current density-voltage-luminance (J - V - L) characteristics and normalized electroluminescence (EL) spectra of devices A and B at different voltages are illustrated in Supplementary Figs. 2a-2c. Due to the well-aligned energy levels of all layers, both devices could be turned on at a voltage of 2.2 V, representing the lowest value among previously reported devices (Supplementary Table 2). In comparison to device A, device B exhibits better performance (Fig. 2c), whose PE and EQE are respectively 184 lm W^{-1} and 35.9%, emitting intense warm-white light with an ultra-high luminance (L_{max}) of $127,385 \text{ cd m}^{-2}$. This enhancement of performance in device B can be attributed to the higher MR-Y concentration. However, it is worth noting that device B emitted white light with an excess of the yellow component (Fig. 2d), resulting in unsatisfactory color coordinates of (0.31, 0.51). Although reducing the MR-Y concentration to 0.3 wt% (device A) aids in improving the color coordinate (0.27, 0.46), it also brings about a sacrifice in efficiency. Furthermore, diminishing the doping ratio to 0.2% does not yield substantial chromatic enhancements, while concurrently bringing challenges in the device preparation process (Supplementary Fig. 3). Therefore, to achieve synchronous improvement of color coordinates and efficiency, further optimization of the device structure is imperative.

We further designed a dual-emission-layer device, denoted as devices C and D, with their performances depicted in Fig. 3 and

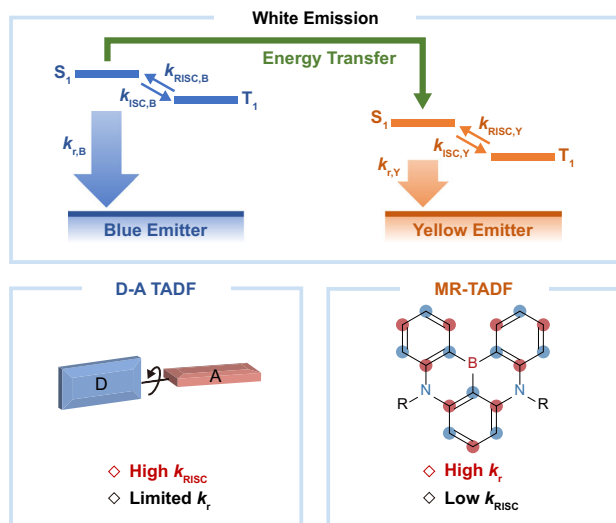


Fig. 1 | Design strategy. Schematic representation of the strategy developed in this work: employing D-A type TADF materials (lower left) with rapid k_{RISC} as the blue emitter and MR-TADF materials (lower right) with fast k_r as the yellow emitter. B, Y, D, A, k_{RISC} , k_{FET} , and k_r refer to blue emitter, yellow emitter, donor, acceptor, reverse intersystem crossing rate, intersystem crossing rate, and radiative decay rate, respectively.

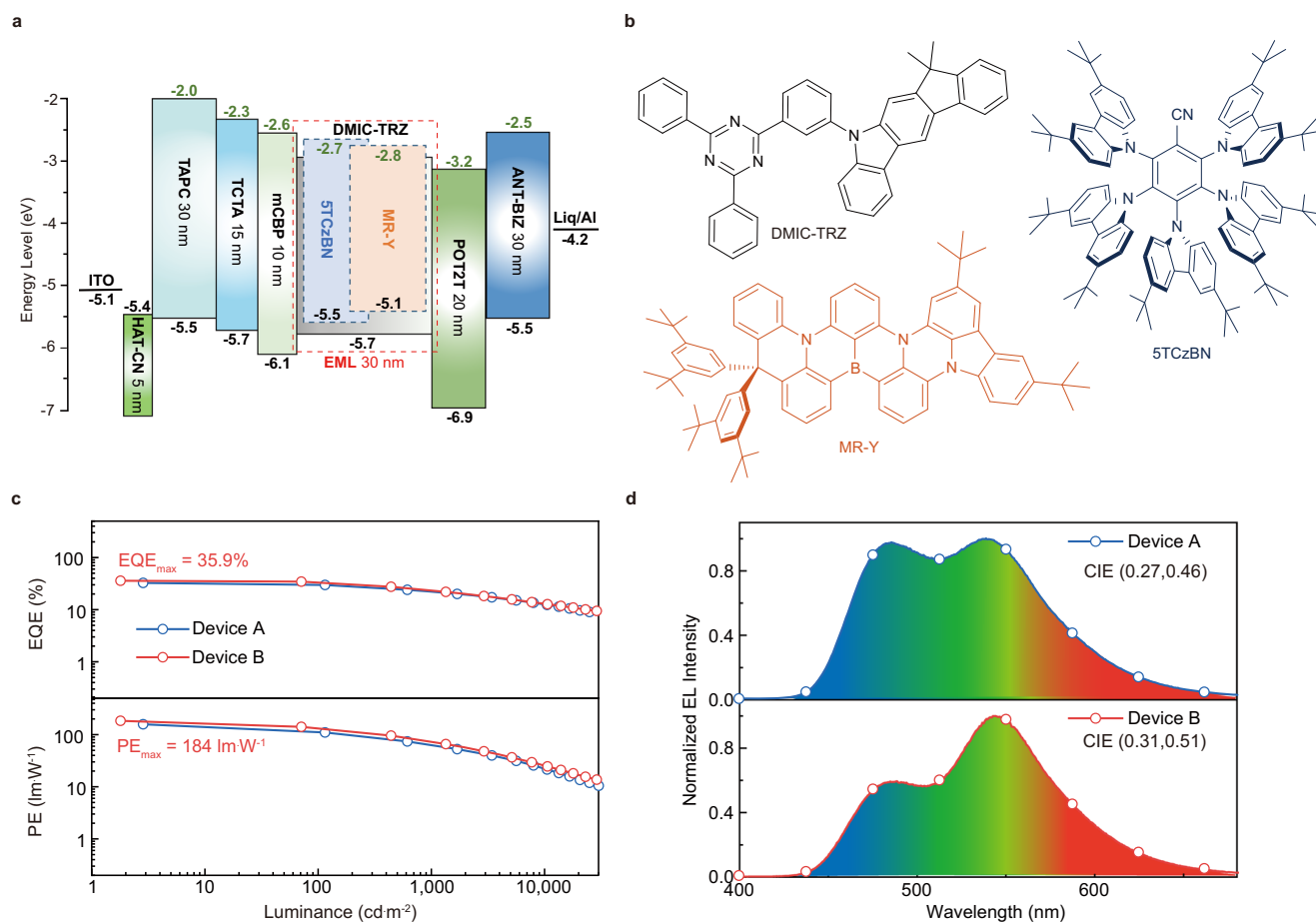


Fig. 2 | Single-emission-layer TADF WOLEDs (devices A and B). **a** The energy diagram. **b** Molecular structures of materials used in the emission layers. **c** EQE (upper panel) and PE (lower panel) versus luminance characteristics. **d** Normalized

electroluminescent spectra measured at 1000 cd m⁻². Source data are provided as a Source Data file.

Supplementary Figs. 2d–2f. In Fig. 3a, the EML of device C comprises a blue-emissive layer (B-EML) consisting of 20 wt% 5TCzBN:DMIC-TRZ and a yellow-emissive layer (Y-EML) consisting of 0.5 wt% MR-Y:20 wt% 5TCzBN:DMIC-TRZ, each with a thickness of 15 nm. This design improves CIE coordinates to (0.30, 0.49) while maintaining a remarkable PE of 190 lm W⁻¹ (confirmed at 203 lm W⁻¹ using a PR735 SpectraScan Spectroradiometer, Supplementary Fig. 4) and an unprecedented EQE of 39% (Figs. 3b and 3c). Both the EQE and PE far surpass those of previously reported white devices, including all-fluorescence WOLEDs (130 lm W⁻¹, 32.8%, ref. 8), all-phosphorescence WOLEDs (105 lm W⁻¹, 28.1%, ref. 35) and hybrid fluorescence-phosphorescence WOLEDs (71 lm W⁻¹, 25.4%, ref. 36), as shown in Figs. 3e, f and Supplementary Table 2. Importantly, device C exhibits reduced efficiency roll-off, maintaining a comparatively high PE of 82 lm W⁻¹ and an EQE of 24.1% even at the luminance of 1000 cd m⁻². In contrast, device D, fabricated without 5TCzBN in the Y-EML, demonstrates a comparable efficiency but shows a steeper roll-off at high brightness (Fig. 3c and Table 1). Furthermore, device C exhibits fast transient EL decay at 490 nm, indicating efficient energy transfer from 5TCzBN to MR-Y (Supplementary Fig. 5). Device D, however, shows slower transient EL decay at 550 nm, suggesting less effective triplet-to-singlet exciton conversion and increased triplet exciton accumulation on MR-Y. Subsequently, we evaluated the operational stabilities of devices C and D at an initial luminance of 1000 cd m⁻², as illustrated in Fig. 3d. Obviously, device C displays a significantly extended operational lifetime (LT₈₀) of 446 h, nearly 4.5 times longer than that of device D (LT₈₀ of only 99 h), which is attributed to the incorporation of

5TCzBN in the Y-EML. In addition, by extrapolating with a degradation acceleration factor (n) of 1.7, the LT₈₀ of device C at an initial luminance of 100 cd m⁻² can be estimated as 22,353 h, positioning it as one of the most stable among all-fluorescence WOLEDs (Supplementary Table 2).

To elucidate the reasons underlying the different operational lifetimes observed in devices C and D, we delved into the photo-physical properties of the materials used in the EML. As depicted in Fig. 4a, the absorption spectrum of MR-Y exhibits noticeable overlaps with the photoluminescent (PL) spectra of both DMIC-TRZ and 5TCzBN. The corresponding Förster radius (R_0) is 5.48 nm and 5.54 nm (Supplementary Note 1), respectively, suggesting the potential for energy transfer from both DMIC-TRZ and 5TCzBN to MR-Y³⁷. To decipher the energy transfer process, we fabricated four thin films: pristine DMIC-TRZ, 0.5 wt% MR-Y:DMIC-TRZ, 20 wt% 5TCzBN:DMIC-TRZ, and 0.5 wt% MR-Y:20 wt% 5TCzBN:DMIC-TRZ for collecting the transient PL decay curves (Figs. 4b–d and Supplementary Fig. 6). As illustrated in Fig. 4b, the film containing 0.5 wt% MR-Y in DMIC-TRZ displays two distinct emission peaks at 468 nm and 550 nm, corresponding to DMIC-TRZ and MR-Y emission, respectively. Comparing the transient PL decay curves at 468 nm, it is evident that the film with MR-Y displays a considerably shorter exciton lifetime (0.05 μ s) in contrast to the film without MR-Y (6.6 μ s), confirming energy transfer from DMIC-TRZ to MR-Y (Fig. 4c). Similarly, within the 20 wt% 5TCzBN:DMIC-TRZ matrix (Fig. 4d), the film containing MR-Y also demonstrates a notably reduced lifetime from 5.8 μ s to 1.6 μ s at the emission wavelength of 490 nm when compared to the film without

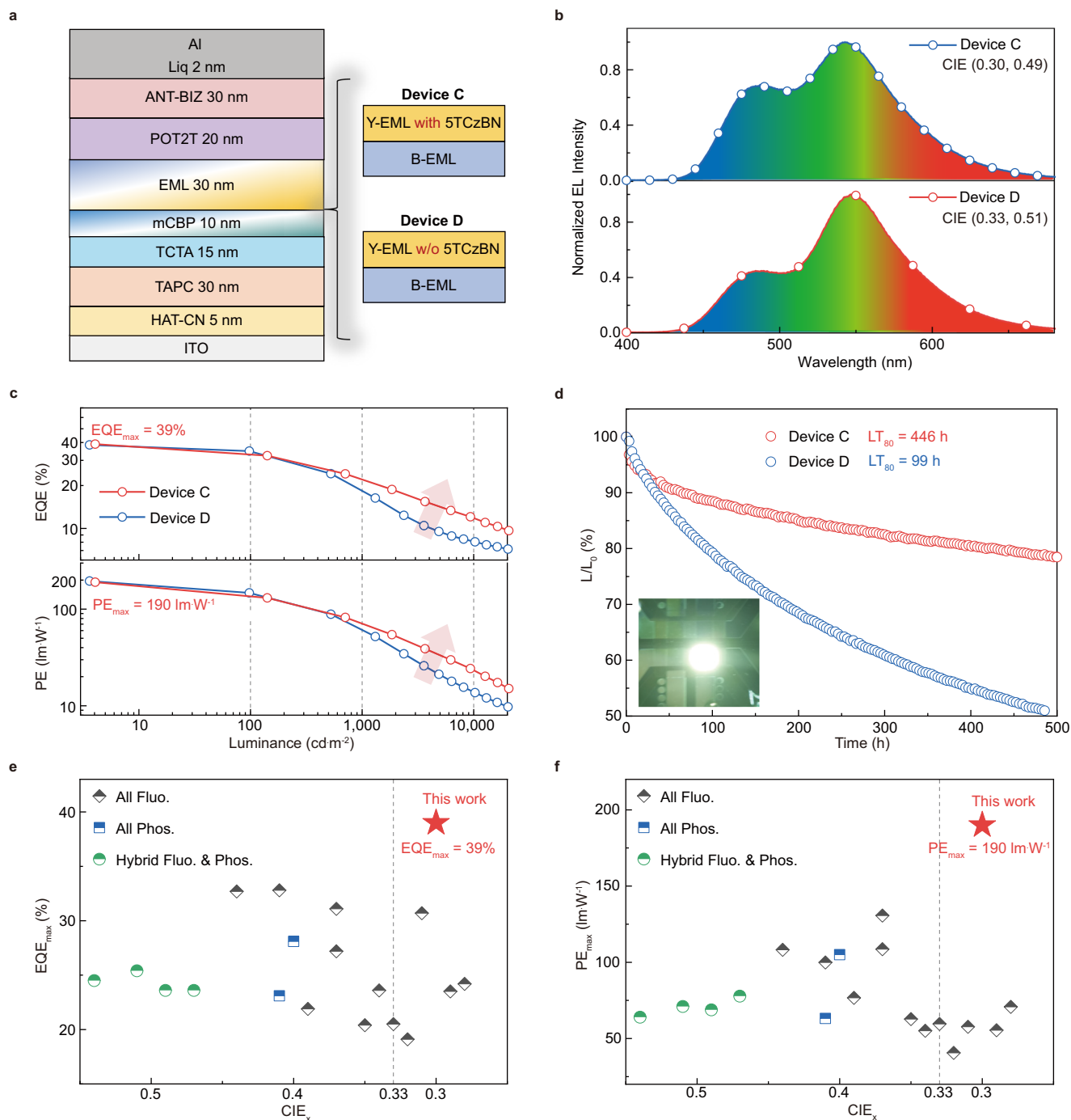


Fig. 3 | Double-emission-layer TADF WOLEDs (devices C and D). **a** Device architectures. Y-EML and B-EML refer to yellow-emissive layer and blue-emissive layer, respectively. **b** Normalized electroluminescent spectra measured at 1000 cd m^{-2} . **c** EQE (upper panel) and PE (lower panel) versus luminance characteristics. **d** Lifetimes of devices at the initial luminance of 1000 cd m^{-2} ; The insets

show the photograph of device C. **e**, **f** EQE (**e**) and PE (**f**) versus CIE_x coordinates for representative reports including all-fluorescence (black), all-phosphorescence (blue) and hybrid fluorescence-phosphorescence WOLEDs (green). Source data are provided as a Source Data file.

MR-Y. This observation provides compelling evidence for concurrent energy transfer from 5TCzBN to MR-Y.

Following the methodology outlined in the literature (relevant formulas in Supplementary Note 2)^{38–41}, we further conducted computations and summarized the k_{FET} and k_{DET} in Fig. 4e and Table 2. In the absence of 5TCzBN, the calculated k_{FET} and k_{DET} from DMIC-TRZ to MR-Y are $2.7 \times 10^7 \text{ s}^{-1}$ and $2.0 \times 10^7 \text{ s}^{-1}$, respectively. The larger k_{DET} values compared to DMIC-TRZ's intrinsic k_{RISC} value ($3.3 \times 10^5 \text{ s}^{-1}$) suggest that triplet excitons do not efficiently undergo up-conversion

in the host DMIC-TRZ, instead of populating the dark triplet state of MR-Y through a DET pathway. Moreover, considering MR-Y's relatively slow RISC process ($k_{\text{RISC}} = 1.6 \times 10^4 \text{ s}^{-1}$), a significant accumulation of T_1 excitons occurs within the device, resulting in substantial efficiency roll-off under high brightness conditions and inferior device stability in device D. With the additional doping of 20 wt% 5TCzBN, the rapid transfer of the excitons from the DMIC-TRZ to 5TCzBN occurs because only the emission of 5TCzBN in the doping film was observed. Therefore, we simplified the ternary system's energy transfer behavior into a

Table 1 | Summary of the key performance data of WOLEDs

Device	V_{on} (V) ^a	EQE (%) ^b	PE (lm W ⁻¹) ^c	L_{max} (cd m ⁻²) ^d	CIE ^e	CCT (K) ^e	CRI ^e	LT ₈₀ (h) ^f
A	2.2	32.7/24.0	158/74	84,009	(0.27, 0.46)	7217	41	-
B	2.2	35.9/27.6	184/95	127,385	(0.31, 0.51)	5954	40	-
C	2.2	39.0/24.1	190/82	98,134	(0.30, 0.49)	6343	41	446
D	2.2	38.5/16.3	196/52	103,379	(0.33, 0.51)	5569	42	99
C-1	2.2	37.4/19.6	183/65	76,662	(0.28, 0.47)	6833	42	-
C-2	2.2	38.5/22.7	194/79	105,801	(0.30, 0.49)	6310	42	-
C-3	2.2	38.8/27.6	195/100	146,016	(0.33, 0.52)	5616	40	-
C-4	2.2	37.1/30.7	194/112	132,386	(0.35, 0.53)	5218	38	-
E	2.6	32.3/22.9	101/58	104,673	(0.26, 0.46)	7444	42	-
F-1	2.4	34.5/18.9	117/44	111,200	(0.38, 0.45)	4353	54	-
F-2	2.4	35.0/20.4	117/48	109,769	(0.44, 0.46)	3362	47	-
F-3	2.2	33.6/16.7	130/41	67,392	(0.30, 0.42)	6385	56	-
F-4	2.2	37.2/18.8	144/49	87,713	(0.37, 0.45)	4632	56	-
G-1	3.0	35.8/26.0	78/35	64,647	(0.26, 0.32)	120,000	71	-
G-2	3.0	35.6/25.9	90/41	64,647	(0.33, 0.34)	5583	69	-

^aTurn-on voltage at 1 cd m⁻²;^b External quantum efficiency: maximum, values at 1000 cd m⁻²;^c Power efficiency: maximum, values at 1000 cd m⁻²;^d Maximum luminance;^e Commission Internationale de l'Éclairage coordinates, correlated color temperature and color rendering index at 1000 cd m⁻²;^f Operational time to 80% of the initial luminance of 1000 cd m⁻².

binary system to focus on the rate-determining step of energy transfer between 5TCzBN and MR-Y. In this scenario, the calculated k_{FET} and k_{DET} values become $5.6 \times 10^7 \text{ s}^{-1}$ and $2.4 \times 10^5 \text{ s}^{-1}$, respectively. Notably, the k_{DET} is significantly reduced by two orders of magnitude (from $2.0 \times 10^7 \text{ s}^{-1}$ to 2.4×10^5), which is even lower than the k_{RISC} of 5TCzBN ($6.1 \times 10^5 \text{ s}^{-1}$). Consequently, the majority of excitons within 5TCzBN undergo FET to the bright S_1 state of MR-Y, while only a negligible fraction follows the DET path to the dark T_1 state of MR-Y. This leads to a substantial decline in the population of long-lived triplet excitons. Hence, it is clear that the integration of the D-A type TADF molecule 5TCzBN into the Y-EML reconfigures energy transfer pathways for excitons to effectively mitigate detrimental long-lived T_1 excitons in the device. This alleviates the annihilation process of long-lived T_1 excitons, ultimately enhancing device stability.

To explore the spatial distribution of exciton recombination for improved control of exciton allocation in WOLEDs, we employed an ultrathin orange fluorescence emitter, 2, 8-di-*tert*-butyl-5,11-bis(4-*tert*-butylphenyl)-6,12-diphenyltetracene (TBRb), as a detector at various positions within the EML (Fig. 5a). Due to the fine overlap between the absorption spectrum of TBRb and the PL spectrum of 5TCzBN (Supplementary Fig. 7), efficient FET from 5TCzBN to TBRb could occur. The spatial distribution of excitons is characterized by the relative EL intensity of TBRb and 5TCzBN at different locations⁸. A graph illustrating relative intensity as a function of position and voltage is developed, as depicted in Fig. 5b. Our findings indicate that at lower operating voltages, the exciton recombination zone predominantly localizes near the electron-transporting layer. As the voltage increases, this recombination zone distributes more uniformly across the entire EML. Accordingly, we can infer that initially, hole-electron recombination primarily occurs within the Y-EML, adjacent to the electron-transporting side. With higher voltages, the recombination extends into the B-EML, consequently enhancing the emission intensity of both the blue and yellow light components (Supplementary Fig. 8). This deduction aligns with the results from single-carrier devices (Supplementary Fig. 9), where the hole transport is faster than the electron transport. Additionally, by modifying the sequence of the B-EML and Y-EML in the device architecture of device C, we fabricated device E as a control (Supplementary Fig. 10), which displays reduced efficiency

and a relatively dominant blue emission. This observation aligns with the above discussed exciton distribution. Thus, precise control of the order and thickness ratio between the B-EML and Y-EML is essential to achieve the desired CIE coordinates in dual-color WOLEDs.

To adjust the CIE coordinates, we fabricated four devices with varying thicknesses of the B-EML and Y-EML, ranging from 25 to 5 nm (Fig. 5c). The device structures are detailed in Supplementary Table 3. As anticipated, all four devices achieve similar PEs of approximately 190 lm W⁻¹ (Supplementary Fig. 11 and Table 1). However, these devices exhibit markedly different CIE coordinates and color temperatures, ranging from (0.28,0.47)/6833 K for device C-1, (0.30,0.49)/6310 K for device C-2, (0.33,0.52)/5616 K for device C-3 to (0.35,0.53)/5218 K for device C-4, showing a transition from warm white emission to warmer white emission (Fig. 5d). Evidently, our device design strategy provides a universal method for achieving different illumination scenarios without sacrificing efficiency, making it suitable for diverse scenarios. For example, increasing the thickness of the B-EML produces bright, cool light, which can help keep people alert and active. In contrast, decreasing the thickness of the B-EML generates a softer, warm light, potentially beneficial for human health during nighttime usage^{1,32-34,42-44}.

Building upon the effective strategy of combining blue emitters with high k_{RISC} and yellow emitters with fast k_r , we have extended this approach to enhance color quality. We incorporated the orange-yellow MR-TADF emitter named 6,9-di-*tert*-butyl-*N*¹⁶,*N*¹⁶,*N*¹⁹,*N*¹⁹-tetraphenyl-3*b*,10*b*,13*b*-triazolo-20*b*-boradifluoreno[1, 9-*ab*:1', 9'-*jk*]perylene-16,19-diamine (MR-O)²⁷, which exhibits an emission peaking at 580 nm with a high $k_{r,S}$ of $7.2 \times 10^7 \text{ s}^{-1}$, in devices F-1 - 4, to replace MR-Y and compensate for the lack of red components. As presented in Table 1 and Supplementary Fig. 12, all these devices exhibit improved CIEs compared to the MR-Y emitter, with particularly notable enhancements observed in devices F-3 and F-4 based on double-emission-layer, achieving warm-white emission with CIE coordinates of (0.31, 0.42) and (0.37, 0.45), respectively. Although the efficiencies of these devices are relatively lower than that of device C, they still surpass 130 lm W⁻¹, exceeding the highest value reported to date.

In pursuit of standard white emission targeting a CIE of (0.33, 0.33) and high color rendering index (CRI), we introduced the bluer

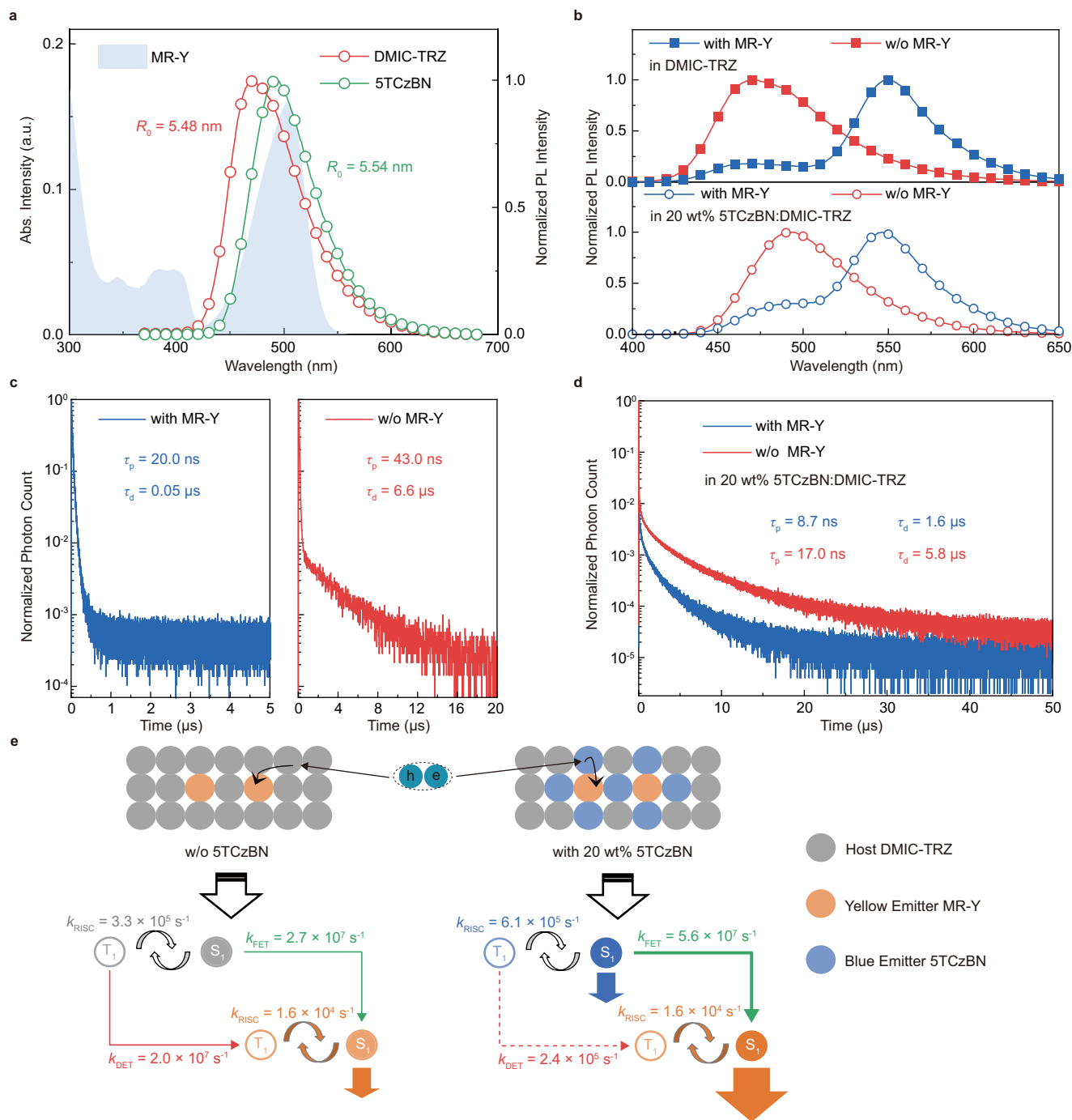


Fig. 4 | Investigation of energy transfer mechanisms. **a** Absorption spectra of MR-Y in the toluene solution and PL spectra of DMIC-TRZ (red) and 5TCzBN films (green). R_0 , Förster radius. **b** Normalized PL spectra of films with or without MR-Y doping in the DMIC-TRZ matrix (upper panel) or the 20 wt% 5TCzBN:DMIC-TRZ matrix (lower panel). **c, d** Transient PL decay curves of films with (left panel) or without (right panel) MR-Y doping in the DMIC-TRZ matrix at the emission

wavelength of 468 nm (**c**); in the 20 wt% 5TCzBN:DMIC-TRZ matrix at the emission wavelength of 490 nm (**d**). τ_p and τ_d , lifetimes of prompt fluorescent and delayed components, respectively. **e** Energy transfer pathways for the system without (left panel) or with (right panel) 20 wt% 5TCzBN. k_{RISC} , k_{FET} , and k_{DET} , rate constants of reverse intersystem crossing, Förster-type energy transfer, and Dexter-type energy transfer, respectively. Source data are provided as a Source Data file.

D-A type TADF emitter of 5-(3, 11-dimethyl-5,9-dioxo-13*b*-boranaphtho[3,2,1-*de*]anthracen-7-yl)-10,15-diphenyl-10,15-dihydro-5*H*-diindolo[3,2-*a*:3',2'-*c'*]carbazole⁴¹ (*m*MDBA-DI), which features a faster k_{RISC} of 1.5×10^6 s⁻¹ (Supplementary Fig. 13, Supplementary Fig. 14 and Table 2), for the fabrication of devices G-1 and G-2. The device structure is ITO/HAT-CN (5 nm)/TAPC (30 nm)/TCTA (15 nm)/mCBP (10 nm)/EML (30 nm)/DBFPO (10 nm)/ANT-BIZ (30 nm)/Liq (2 nm)/Al (100 nm). In devices G-1 and G-2, the EML consists of 0.3 wt% and

0.5 wt% MR-O doped into 20 wt% *m*MDBA-DI:DBFPO, where DBFPO (Dibenzo[*b,d*]furan-2,8-diylbis(diphenylphosphine oxide)), with a higher triplet energy ($T_1 = 3.1$ eV) than DMIC-TRZ ($T_1 = 2.8$ eV), serves as a suitable host for the bluer emitter *m*MDBA-DI ($T_1 = 2.9$ eV). As shown in Fig. 6a, b and Table 1, both devices exhibit a high maximum EQE of over 35%, indicating efficient exciton utilization in the devices. However, the maximum PE of 90 lm W⁻¹ is deemed unsatisfactory due to the elevated turn-on voltages (from 2.2 V to 3.0 V) necessitated by the

Table 2 | Transient PL characteristic and rate constants of films

Film	λ_{\max}^a (nm)	τ_p^b (ns)	τ_d^b (μ s)	Φ_{PL}^c (%)	Φ_p^d (%)	Φ_d^d (%)	$k_{r,s}^e \times 10^7$ (s ⁻¹)	$k_{ISC}^e \times 10^7$ (s ⁻¹)	$k_{RISC}^e \times 10^4$ (s ⁻¹)	$k_{FET}^f \times 10^6$ (s ⁻¹)	$k_{DET}^f \times 10^5$ (s ⁻¹)
20 wt% 5TCzBN:DMIC-TRZ	490	17.0	5.8	63	18	45	1.0	4.2	61	56.1	2.4
0.5 wt% MR-Y:20 wt% 5TCzBN:DMIC-TRZ	490/ 550	8.7/ 3.5	1.6/ 3.4, 110.0	95	-	-	-	-	-	-	-
0.5 wt% MR-Y:DMIC-TRZ	550	8.1	140.0	98	44	54	5.4	6.8	1.6	-	-
20 wt% mMDBA-DI:DBFPO	466	47.3	2.5	97	25	72	0.6	1.4	149	7.5	2.4
0.5 wt% MR-O:20 wt% mMDBA-DI:DBFPO	466/ 580	34.9/4.6	1.1/ 2.4, 109.0	93	-	-	-	-	-	-	-
0.5 wt% MR-O:DBFPO	580	7.1	117.0	97	51	46	7.2	6.6	1.6	-	-

^aPeak wavelength of the PL spectrum in the film;

^bLifetime of the prompt component and delayed component determined from the transient PL decay curve at 298 K;

^cAbsolute photoluminescence quantum yield measured with an integration-sphere system under argon atmosphere;

^dAbsolute photoluminescence quantum yield of prompt fluorescence and delayed fluorescence;

^eRate constants of singlet radiative decay, intersystem crossing and reverse intersystem crossing.

^fRate constants of Förster energy transfer and Dexter energy transfer.

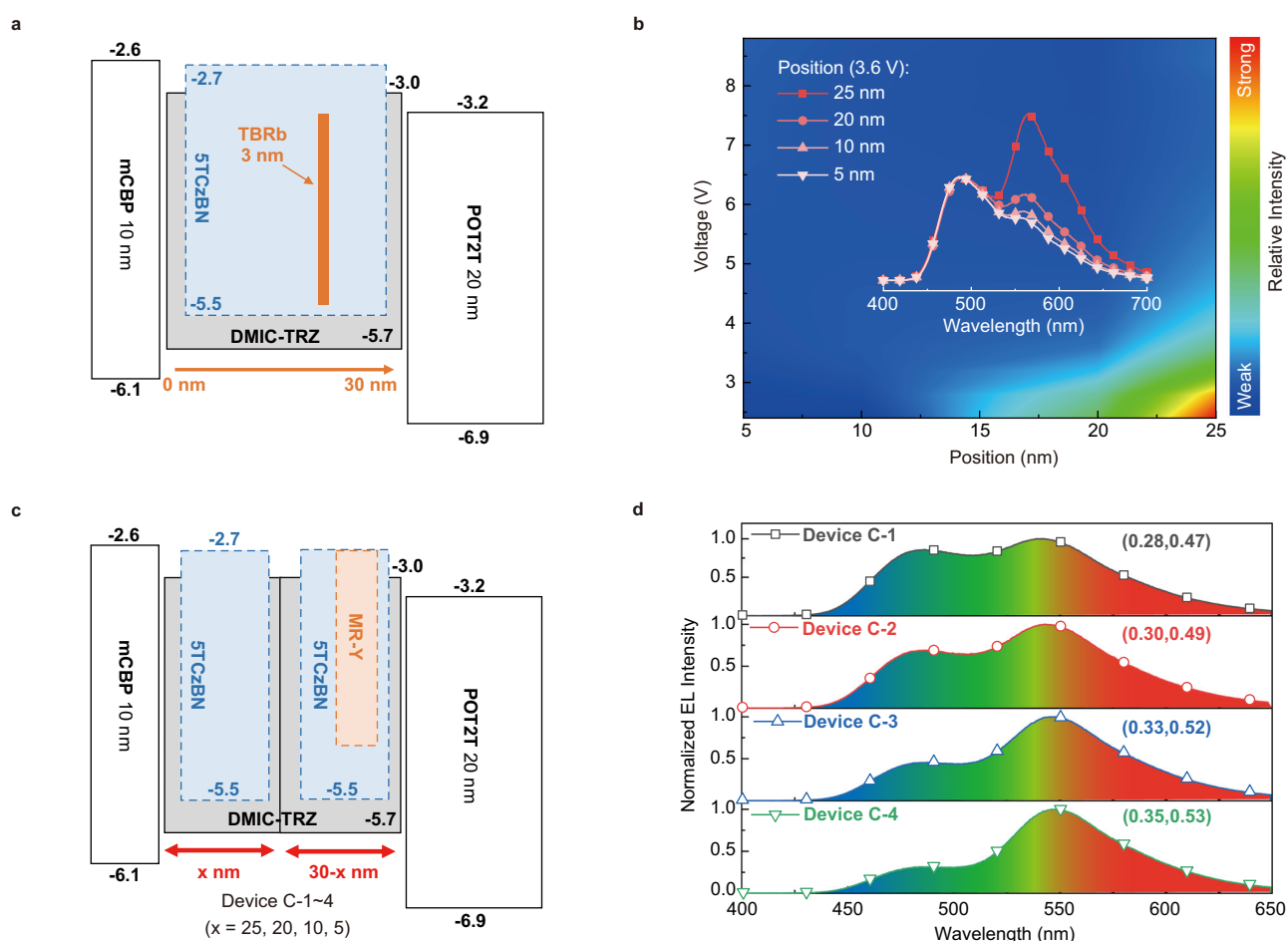


Fig. 5 | Investigation of exciton recombination region. **a** Energy-level diagrams (in eV) of devices employed for detecting exciton recombination zone. **b** Relative intensity profiles of TBRb and 5TCzBN as a function of position and voltage. **c** Energy-level diagrams (in eV) of devices C-1 - 4, with varying thicknesses of the

B-EML (25, 20, 10 and 5 nm, respectively). **d** Normalized electroluminescent spectra of devices C-1 - 4 measured at 1000 cd m⁻² (Insets depict CIE coordinates). Source data are provided as a Source Data file.

high triplet energy host (DBFPO). Particularly, device G-1 achieves a maximum EQE of 35.8% and a maximum PE of 78 lm W⁻¹ (confirmed at 84 lm W⁻¹ using a PR735 SpectraScan Spectroradiometer, Supplementary Fig. 15) at a voltage of 3.6 V (Fig. 6c), with the corresponding CIE coordinates of (0.33, 0.34) locating at blackbody radiation curve. Conversely, device G-2, under a drive voltage of 5.2 V (Fig. 6d),

manages to approach standard white light emission with CIE coordinates of (0.33, 0.34) and an EQE of 25.9% at 1000 cd m⁻², which is apparently the most efficient WOLEDs with a CIE of x and y values both close to 0.33 at the practical brightness level reported so far^{8,13,31} (Supplementary Fig. 16 and Supplementary Table 4). This demonstrates that standard white light emission can be attained through

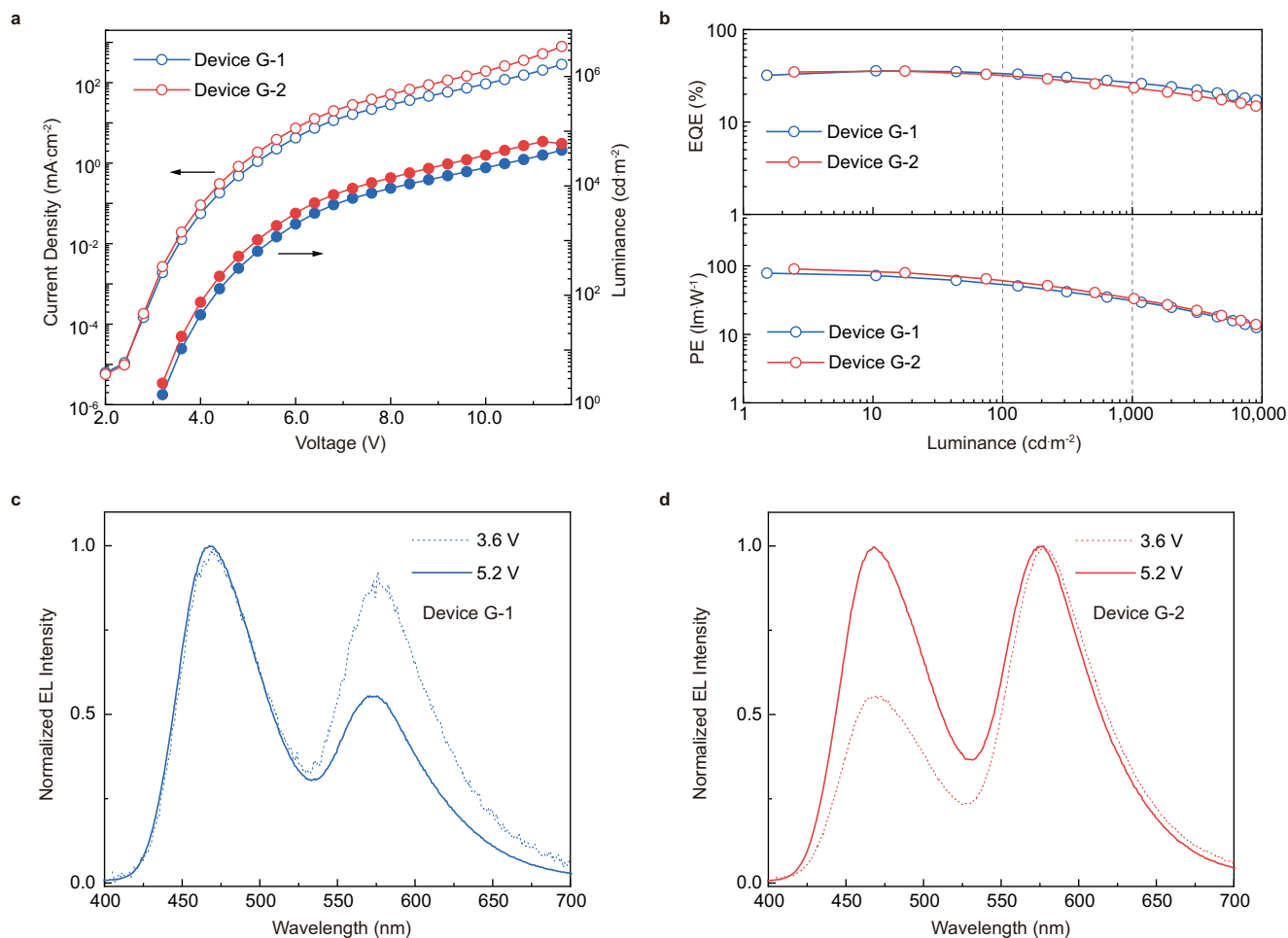


Fig. 6 | Standard white OLED (devices G-1 and G-2). **a** Current density and luminance versus voltage characteristics. **b** EQE (upper panel) and PE (lower panel) versus luminance characteristics. **c, d** Normalized electroluminescent spectra

measured at 3.6 V (dotted line) and 5.2 V (solid line) for device G-1 (blue, **c**) and device G-2 (red, **d**) (Note: The devices exhibit peak efficiency at 3.6 V and reach a brightness of 1000 cd m^{-2} at 5.2 V). Source data are provided as a Source Data file.

careful optimization of material selection, doping concentration, and voltage adjustments. Furthermore, both devices G-1 and G-2 achieve a CRI of approximately 70, indicating excellent color quality for two-color WOLEDs. These outcomes highlight the effectiveness of our approach by employing MR-TADF emitters with high k_r as yellow-light components and D-A type TADF molecules with rapid k_{RISC} as blue-light components, offering a robust platform for advancing WOLED technology.

Discussion

We have presented an effective strategy to harmonize triplet exciton up-conversion and singlet exciton radiation in WOLEDs by employing D-A type TADF emitter as the blue emission component and MR-TADF emitter as the yellow emission component. Capitalizing on the advantageous combination of a high k_{RISC} in the D-A type TADF emitter and a high k_r in the MR-TADF emitter, our warm WOLEDs have demonstrated exceptional performance with a substantial EQE of 39% and a PE exceeding 190 lm W^{-1} , accompanied by a long device lifetime with LT_{80} of 446 h at an initial luminance of 1000 cd m^{-2} . Using the strategy, we have demonstrated standard white emission with a CIE of (0.33, 0.34) and a high EQE of 35.6% by precise modulation of blue and yellow emitters. Furthermore, our detailed investigation of the exciton recombination region offers valuable insights for precisely tuning the chromaticity coordinates of WOLEDs, meeting a diverse range of illumination requirements. We believe that this strategy holds broad applicability and great potential in developing highly efficient and stable WOLEDs.

Methods

Materials

The materials used for fabricating the devices were either purchased from the company or synthesized by co-authors.

Photophysical characterization

UV-vis absorption spectra and photoluminescence (PL) spectra were measured by a Shimadzu UV-2700 spectrophotometer (Shimadzu, Japan) and a Hitachi F-7100 fluorescence spectrophotometer (Hitachi, Japan), respectively. Phosphorescence spectra were recorded on the Hitachi F-7100 fluorescence spectrophotometer at 77 K. Transient PL decay curves were obtained by FluTime 300 (PicoQuant GmbH) in conjunction with a Picosecond Pulsed UV-LASER (LASER375) serving as the excitation source. The photoluminescence quantum yields (PLQY) were measured with a Hamamatsu UV-NIR absolute PL quantum yield spectrometer (C13534, Hamamatsu Photonics) equipped with a calibrated integrating sphere. During the PLQY measurements, the integrating sphere was purged with dry argon to maintain an inert atmosphere.

WOLED fabrication and characterization

The ITO-coated glass substrates used in this study had a sheet resistance of $15 \Omega \text{ square}^{-1}$. Prior to deposition, these substrates underwent a meticulous cleaning process involving successive ultrasonication steps with acetone and ethanol, followed by thorough drying using a nitrogen gas flow. Subsequently, the cleaned substrates were

subjected to a 20-minute treatment in a UV-ozone surface processor (PL16 series, Sen Lights Corporation). Once treated, the samples were transferred to the deposition system. In the deposition process, the electron injection layer, Liq and Al were both thermally evaporated at a pressure of 5×10^{-5} Pa. The organic layers were deposited at rates ranging from 0.2 to 3 \AA s^{-1} . Following the organic film deposition, the Liq and Al layers were deposited at rates of 0.1 and 3 \AA s^{-1} , respectively. The emitting area of each device in the study was approximately 0.09 cm^2 . To characterize the devices, various measurements were conducted using specialized equipment. The current density-voltage-luminance (J - V - L), EQE/PE- L curves and electroluminescence spectra were measured using a Keithley 2400 source meter and an absolute EQE measurement system (C9920-12, Hamamatsu Photonics, Japan). The stability test of the encapsulated WOLEDs was conducted using a luminance meter (SR-3AR, TOPCON, Japan) under constant current density driving conditions with an initial luminance of 1000 cd m^{-2} . The transient electroluminescence characteristics of the devices were detected with a HAMAMATSU C16361 transient spectrometer, together with NF Corporation WF1968 to generate short-pulse excitation. The amplitude and baseline of the pulse are 4 and 0 V, respectively. All the above processes were carried out at room temperature under common laboratory conditions.

Data availability

Source data are provided with this paper. The full set of data generated in this study has been deposited in the Figshare database under the accession code <https://doi.org/10.6084/m9.figshare.24279349>. Source data are provided with this paper.

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- materials. S.G. and Y.T. characterized the device with spectroradiometers. M.H., Z.C., and C.Y. wrote the manuscript. M.H., Z.C., J.M., and C.Y. participated in discussions. All authors discussed the results and commented on the final manuscript.

Competing interests

SZU has filed patent applications on luminescent compositions and devices. C.Y., M.H., J.M. and Z.C. are the authors of the invention. CN patent application no. 2023116829470 (pending). The other authors declare no competing interests.

Additional information

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Author contributions

C.Y. supervised the projects. M.H., J.M., and C.Y. conceived the ideas. M.H., Z.C., and S.H. made the devices and carried out the characterizations. W.Y., Z.H., and Y.Z. synthesized the MR-Y, MR-O, and mMDBA-DI