



OPEN A neutron dark-field imaging study towards guiding magnetic flux in electric steel sheets through laser shock peening

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Well-designed guidance of the magnetic field through electric steel sheets forming the rotor of electrical motors and generators is of utmost importance with regards to efficiency. Typically, this is achieved through tailored cutouts in the sheets. While fulfilling the magnetic guidance needs this strategy structurally weakens the rotor and ultimately limits the possible rotation speeds due to the mechanical forces these imply. Recently an alternative approach has been reported, in which the material is embossed in respective regions. The, thus, induced stresses act, based on the inverse magnetostrictive effect, as flux barriers without, unlike traditional cutouts, weakening the structure significantly. Here we propose to replace the embossing through laser shock peening, which can be controlled digitally and hence be applied flexibly, without the need for mechanical changes in the production lines, like in the case of a punch.

The current trend and environmental relevance of electric vehicles create a growing demand for highly efficient electromobility and electric machinery in general. Efficiency for rotating electrical machines implies foremost to optimize magnetic flux and avoid stray flux related losses. This can be achieved by well-designed magnetic flux guidance which thus is of utmost importance for e.g. synchronous reluctance machines and permanent magnet synchronous machines^{1,2}. To date such flux guidance is achieved through specifically designed holes in the rotor fabricated of non-grain oriented electric steel sheets³. This means the electric steel sheets are cut into the desired shapes before assembling the rotor from the sheet material. While the cutouts function as flux barriers, they weaken the structure of the rotor which has to withstand high centrifugal forces during high frequency rotation. This in turn limits the maximum rotation speed that can be reached and therefore decreases the best energy density and efficiency that can be achieved^{4,5}. Recently a novel approach has been demonstrated aiming to overcome these competing influences on the efficiency of electric engines. In this approach the cutouts are replaced by corresponding embossing patterns, which imply compressive stresses in treated areas. According to the magneto-mechanical Villari effect^{6,7} these stresses pin the magnetic domains, which leads to a reduced magnetic permeability in the worked area of the NGO sheet material^{8–10}. This way flux barriers of desired shapes can be created without the detrimental weakening effect on the rotor, which is implied by cutouts. This consequently has a significant potential to further increase the efficiency of electric machinery.

Here we present an alternative approach towards generating flux barriers based on the inverse magnetostrictive effect, namely through applying laser shock peening (LSP) instead of a mechanical punch. This process can be fully controlled digitally and can be integrated with laser cutting¹¹, laser scribing¹² and laser healing¹³ in a future advanced digital manufacturing process of high efficiency electric machinery. In addition, it can also be applied to the thin side of the electric steel laminates, with the aim to reduce magnetic conductivity in the leakage web between the tooth tips in induction machines with closed rotor slots. In order to demonstrate the effect of laser shock peening and thus the feasibility of the approach we apply neutron grating interferometry to visualize the local effect of the process. This approach is in accordance with earlier works on the effect of different cutting and laser techniques on the local magnetic domain structure and permeability of electric steels^{11–13} but also the works on embossing to create magnetic flux barriers^{10,14}.

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sample	#1	#2	#3	#4	#5	#6	#7a	7b	#8	#9
laser energy	1.5 J	1.5 J	1.5J	0.5 J	0.5 J	0.5 J	1.5 J	1 J	0.5 J	0.5 J
frequency	10 Hz	10 Hz	10 Hz	10 Hz	10 Hz	10 Hz	5 Hz	5 HZ	5 Hz	5 Hz
Al layer	yes	yes	yes	no	no	no	no	no	yes	yes
spot overlap	80%	80%	80%	80%	80%	80%	80%	80%	80%	80%
side treatments	no	no	no	yes	yes	yes	yes	yes	no	yes
remark		failed							half	

Table 1. List of samples and treatment parameters.

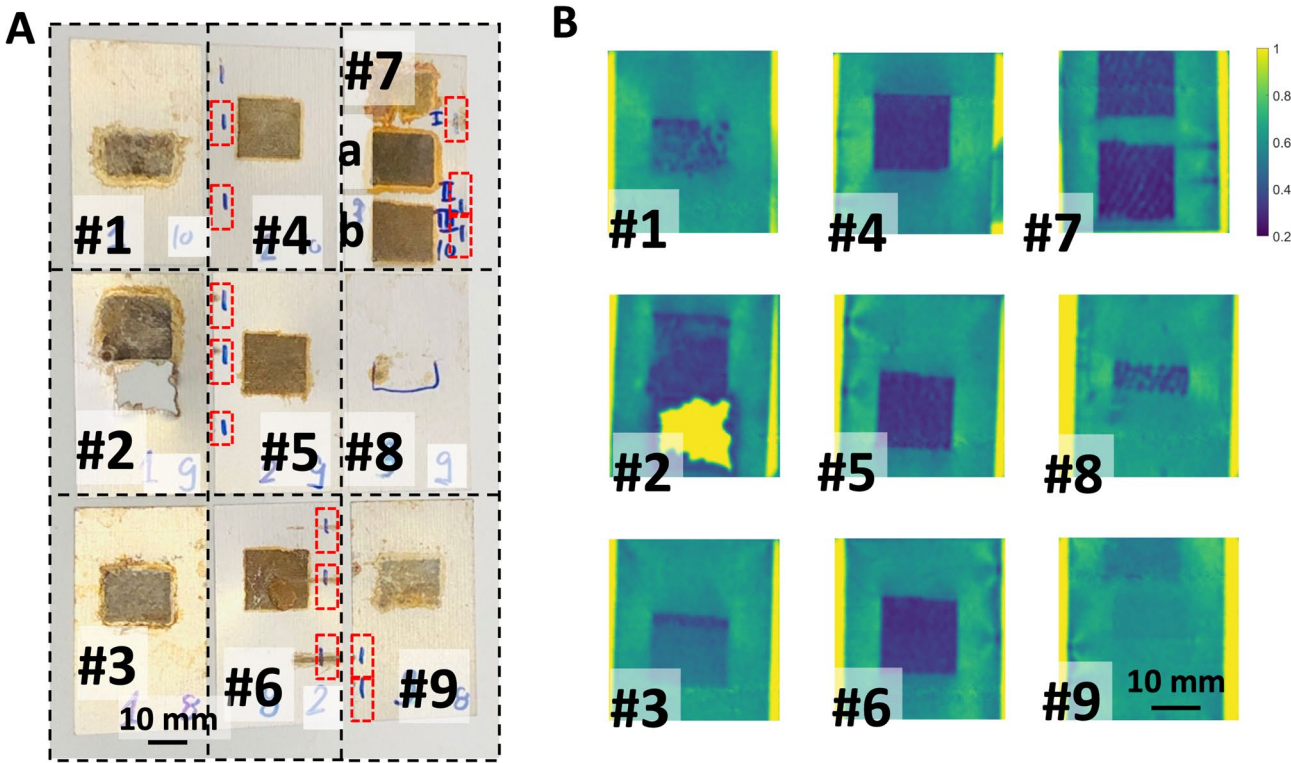


Fig. 1. (A) photograph of NGO electric steel sheet samples #1 - #9 after LSP treatment; note that the red frames mark LSP treatments done in TD onto the slim side of the sheets; sample #2 features a hole due to high energy laser LSP treatment, sample #7 has three different treatments, of which only two are considered successful and the treatment of sample #8 conveys a smaller area; (B) corresponding dark-field contrast neutron images displaying most treated regions; note also the visibility of the side treatments in areas corresponding to the red frames in (A).

Experimental Samples and sample treatment

The samples are 300 micrometer thick non-grain oriented (NGO) electric steel (280-30AP) lamination sheets. These sheets are coated with glassy magnesium silicate. From the as received 30 mm x 300 mm sheets pieces of 30 mm x 60 mm were cut for LSP treatments. Rectangular central regions of about 15 mm x 15 mm of nine 30 mm x 60 mm pieces were LSP treated with different parameters, including laser energies from 0.5 to 1.5 J, at pulse frequencies of 5–10 Hz and with or without the use of a protective sacrificial aluminum layer. The spot overlap of the LSP procedure was 40%. Some selected samples (#2,#3,#5,#8 and #9) were additionally treated from the side, i.e. on the 300 micrometer thick edge. These treatments were done with 1.5 J and 10 Hz. A list of the samples as well as corresponding images are shown in Table 1; Fig. 1, respectively.

Measurements

Dark-field contrast measurements^{15–17} were performed at the BOA beamline at the SINQ neutron source of the Paul Scherrer Institute in Switzerland¹⁸. Dark-field contrast imaging is sensitive to small angle neutron scattering in the ultra-small and very-small angle regime, i.e. for scattering from structures in a size range from some 10 nm to some 10 micrometers, depending on the applied set-up and parameters. A symmetric Talbot Lau grating interferometer¹⁹ was employed for the measurements and the samples were mounted in

an electromagnetic yoke with tunable applied field. Dark-field contrast is generated by the effect of magnetic domains on the refractive index of neutrons, respectively by their contribution to the scattering length density contrast. These contributions are opposite for spin-up and spin-down neutrons¹⁶. Thus, the neutron dark-field contrast is sensitive to the domain wall density in the material and the respective magnetisation. According to the magnetomechanical effect the induced stress in the material increases the magnetic anisotropy and domain wall density. This directly affects the magnetic permeability and the dark-field contrast, which increases correspondingly as the permeability is decreasing. This has been used in numerous studies to investigate the local effect of stress on the magnetic permeability^{10–14,20}, e.g. at cutting edges with respect to different cutting processes, but also with regards to the creation of artificial flux barriers through intentionally induced stresses.

Samples have been measured in two configurations, with the magnetic field applied in vertical direction, i.e. perpendicular to the scattering sensitivity of the Talbot Lau grating interferometer, and with the magnetic field applied horizontally, perpendicular to the grating lines. As the samples were rotated together with the magnetic yoke, the field was always applied in the same direction with respect to the NGO steel sheet, i.e. in rolling direction (RD).

This implies that when the sample is oriented with RD vertical, domain density in transversal direction (TD) is probed, and when RD is horizontal, domain density along RD is probed. While all samples have been screened with dark-field contrast imaging (Fig. 1) some selected samples, namely samples #5 and #7 (a, b) as well as an untreated reference have been exposed to step-wise hysteretic magnetization. This was done in the neutron imaging instrument with a yoke installed in the sample position and enabled to record corresponding dark-field contrast image series in analogy to other similar reported investigations of stress-affected NGO steel sheets¹⁴. The applied magnetic fields correspond to 0, 1500, 3000 and 4500 A/m.

Results

The measurement of the untreated reference sample reveals an anisotropy in the magnetic domain structure, and thus of the material. This is typically found in NGO steels due to anisotropic production conditions^{21,22}. The results (Fig. 2) suggest that the domain wall density is higher in TD of the NGO strip (Fig. 2A & B), than in RD, i.e. the longer direction (Fig. 2C & D). The relative response to the applied field is, however, comparable and is showing a decrease in dark field contrast (increasing dark-field values) of about 20% in both cases for an applied field of 4500 A/m.

The measurement of sample #5 clearly displays the effect of the LSP with considerably increased dark field contrast in the treated area, as is expected from domain pinning due to residual stresses and deformation (Fig. 3). Note that the green rectangles in Fig. 3A and C denote the region of the line profile impacted by the side treatments with LSP, which are all clearly visible in the dark-field images (Fig. 3E & F). The red rectangles on the other hand mark an area where the electric steel sheet is not fully contacted with the yoke for geometrical reasons. This artefact can be ignored to a first approximation and impacts the results insofar as a certain asymmetry can be observed in the dark-field profiles and in particular under the applied magnetic field condition.

A comparison of the 4 types of dark-field contrast profiles presented in Fig. 3 alongside with the reference profiles of the untreated sample under the respective same conditions reveal some remarkable details. When first comparing the TD profiles for the vertical and horizontal sample RD/field orientation given in Fig. 3A and B, respectively, significant differences are observed. This is in particular the case for the region adjacent to the treated area. While both cases reveal a nearly constant dark-field contrast with respect to measurements with and without applied field in the treated area, the adjacent regions differ significantly. These seem to be unaffected with regards to the TD domain wall density (Fig. 3A), where these regions behave very similar to the untreated sample, but appear strongly affected with regards to the RD domain wall density, i.e. along the field direction (Fig. 3B). In the latter case the dark-field signal reaches values comparable to the 4500 A/m applied field case already without applied field, in particular where the sample is additionally affected by the side treatment. At the applied field of 4500 A/m the values reach those of the untreated sample in both cases. Notably a similar effect is observed for the opposite direction, where the regions in front and behind the treated area with respect to the field direction (Fig. 3C). Here the domain wall density across the strip (TD), i.e. perpendicular to the field direction appears to be reduced already before applying an external magnetic field.

Along the strip (RD) it is seen (Fig. 3D), on the other hand, that at maximum applied field of 4500 A/m the dark field values do not reach the level of that of the reference sample, implying a remaining higher domain wall density in RD and, thus, field direction in this case as compared to the untreated reference. This appears again to be related to the flux barrier, which reduces the magnetic flux in this direction in that area, as is part of the desired effect of the treatment. Comparable trends to those described here for sample #5 can be found for treated areas in other samples of the measurement series, and these align well with dark-field imaging results for embossing treatments²³.

Sample #7 features two LSP treated areas that differ in the laser power used, which was 1.5 J for the area #7a and 1 J for the area 7b (Fig. 4). In Fig. 4A we compare the response of the treated areas to increasing applied external fields. It becomes obvious, that the treatment with the higher laser power (#7a) is more sustainable in resisting the applied field than the area (#7b) treated with two thirds of the laser power, i.e. 1 J. When comparing with sample #5 which has been treated with only 0.5 J but 10 Hz, in contrast to 5 Hz in the sample #7 treatments, one finds sample 5 would range between sample #7a and #7b.

Sample #7 also features treatments from the side which are clearly visible in Fig. 4B at the right side of the #7b treatment area at the edge of the NGO electric steel sheet. These treatments leave traces of their effect in the dark-field contrast image for at least 5 mm into the sheet area. The enlarged detail in Fig. 4B reveals a complex pattern around needle like straight and fading signatures. To the sides a V-shaped region of less dark-field contrast, comparable to a non-affected sheet are found, and subsequently in the triangular regions next/below the V darker, i.e. more affected areas. Figure 4C displays slight differences between the two side treatments

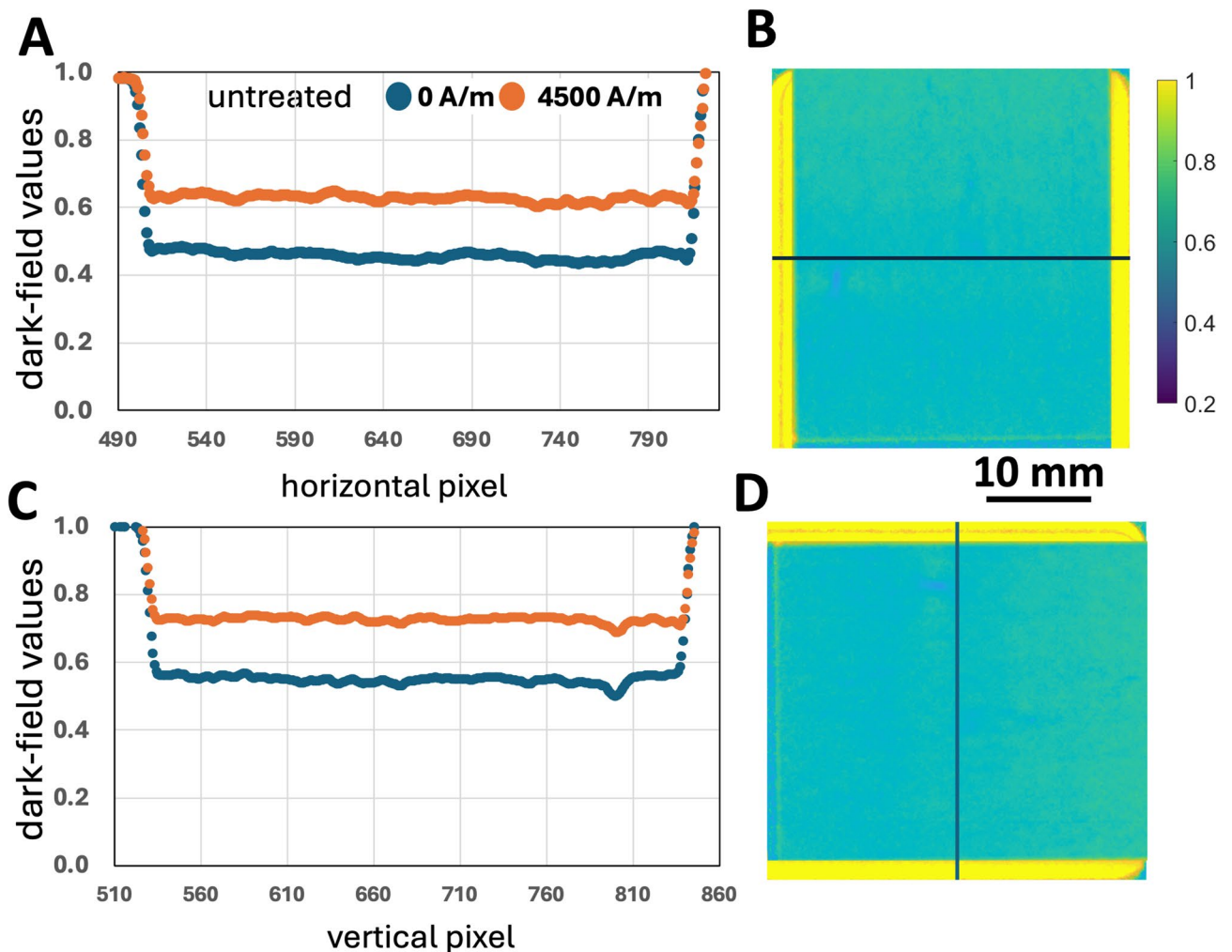


Fig. 2. Untreated reference sample: **A** and **C** show the TD dark-field profiles of the untreated reference electric steel sheet for the vertical and horizontal field/sample (with respect to RD) measurement, respectively. **B** and **D** display corresponding dark field contrast images.

in line profiles indicated in the detail in Fig. 4B by a black line. The upper treatment appears to have a stronger and more sustainable effect. At this stage it cannot be concluded whether this is due to the accuracy of the treatments, which were in principle done with the same parameters, or whether the relation to the treated areas on the surface have an impact.

Discussion

We applied laser shock peening to NGO electric steel sheets and could demonstrate, utilizing neutron dark-field contrast imaging, the effect and success of the treatments. Neutron dark-field contrast imaging has been established in recent years for corresponding spatially resolved observations of magnetisation, domain structure and domain pinning. Here it clearly shows that the induced stresses through the laser treatment deteriorate the domain structure similar to mechanical punching treatments and, thus, affects the local magnetic permeability. In contrast we found through EBSD that the microstructure of the electric steel sheets is not affected, i.e. not noticeably altered, by the successful treatments. In addition we find that sheets treated with same parameters display coinciding results, which implies repeatability. In analogy to the mechanical punching approach the LSP treatment, thus, appears promising in guiding magnetic flux around such treated areas, which can replace the currently dominating approach of material cut-outs in laminates of electrical machinery. In contrast to such, the here demonstrated treatments can avoid the mechanical weakening of the material, as no cut-outs are necessary. Therefore, the approach can be a promising route for processing aiming at high performance electric motors. One can envision a laser-based manufacturing line, where lasers cut the required shapes, repair unwanted domain deterioration e.g. from cutting and induce stressed areas for magnetic field guidance, as proposed here etc. The advantage is a flexible programable digitally controlled process.

We acknowledge that the presented work is a mere individual first attempt in this direction, and many issues have to be addressed still. In the context of the here proposed laser shock peening treatments a prevailing issue appears to be the obvious damage to the isolating coating of the electric steel material. This might be avoided in

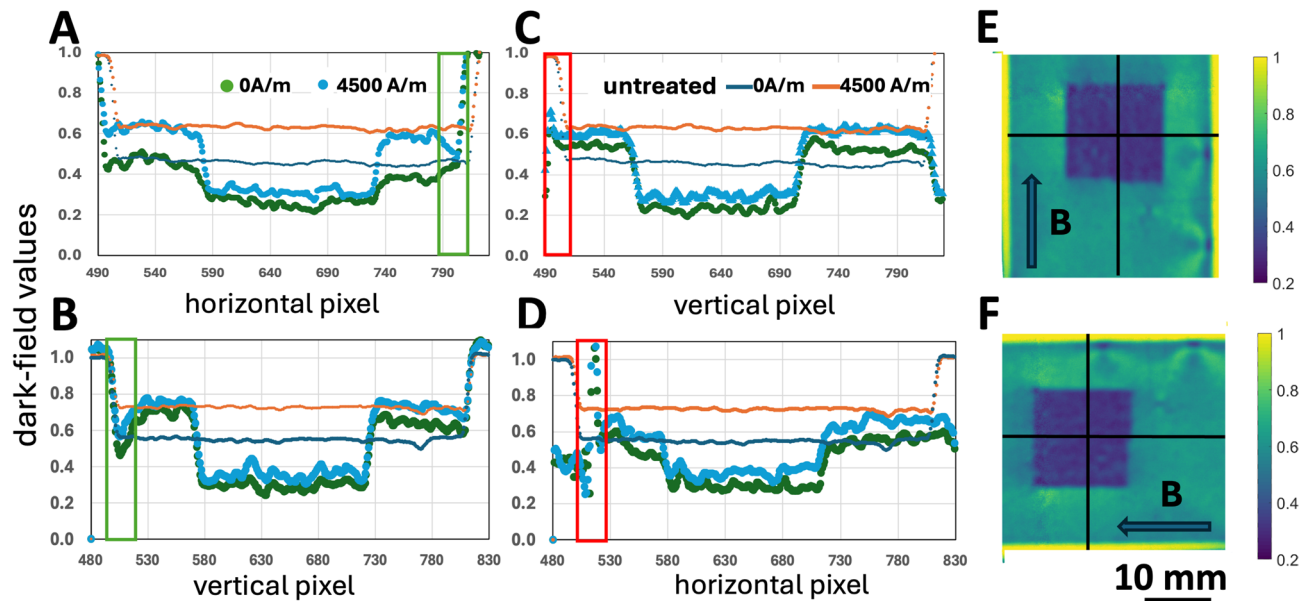


Fig. 3. Sample #5: **A** and **B** show the transverse (TD) dark-field profiles across the electric steel sheet #5 for the vertical and horizontal RD/field orientation measurement, respectively, and respective reference profiles of the untreated steel sheet sample. **C** and **D** show respective profiles along the RD of the steel strip for the two field orientations, respectively, and **E** and **F** show the corresponding dark-field contrast images.

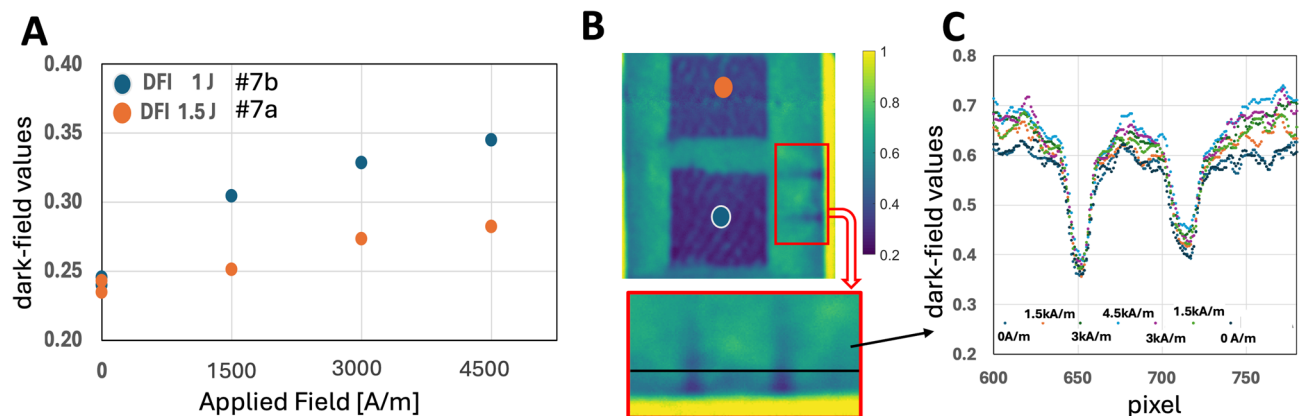


Fig. 4. Sample#7: **(A)** Neutron dark-field contrast in the two treated areas, as indicated in **(B)**, for increasing applied external magnetic fields; **(B)** dark-field contrast image of the sheet and a detail including the signature of two side treatments; **(C)** dark-field contrast line profiles at the position indicated in **(B)** for different applied external magnetic fields in a hysteresis scan;

an optimized laser shock peening process, which is beyond the scope of the current work introducing the mere potential of such treatments and a method to characterize its effects with the required spatial image resolution.

Methods

Dark-Field contrast imaging

Dark-field contrast imaging was performed at the BOA beamline¹⁸ at the SINQ neutron source of the Laboratory of Neutron Scattering and Imaging at the Paul Scherrer Institut in Switzerland. The utilized symmetric Talbot Lau interferometer consisted of three gratings. A source grating (G0), a phase grating (G1) and an analyser grating (G2). All gratings have a period of 25 μm . G0 and G1 are absorption gratings consisting of Gd lines with a thickness of 20 μm on a quartz waver with 100 mm diameter. G1 is a Si phase grating inducing a $\pi/2$ phase shift through a line height of 43.8 μm . This implies that the grating interferometer is optimised for a wavelength of 3.5 \AA , while a cold neutron spectrum with a Maxwellian distribution with a peak at 2.8 \AA and a weighted mean at 3.8 \AA was applied. The distance between gratings is about 44.6 cm, which corresponds for G1 to G2 to the first fractional Talbot distance, considering a magnification factor $M = 2$. Dark-field measurements consist of phase stepping scans¹⁵ and 13 phase steps over one period were accounted for. Every step was recorded taking three

images which were subsequently reduced through median filtering to single images per step. From these images the dark-field contrast image was calculated. The dark-field contrast image evaluates the modulation amplitude in each pixel with regards to the phase scan normalized to the modulation offset, i.e. the transmission image. These images $V(x, y)$ are normalised to the open beam visibility $V_0(x, y)$ which ranges on pixel average at about 28%.

A PSI midibox detector²⁴ was utilized featuring a LiFZnS(Ag) scintillator screen (from RC Tritec AG, Switzerland) with a thickness of 100 μm from RC Tritec, a Zeiss MAKRO-PLANAR 2/100 mm ZF.2 objective lens and an ANDOR-iKon-M CCD camera. The instrument pinhole was set to $20 \times 20 \text{ mm}^2$ and was 6.5 m upstream of the sample position. The exposure time per image was 30 s, leading to a total exposure per dark-field contrast image of about 20 min. All images were background and dose corrected.

Data availability

Raw data files used to produce the figures and tables in this report is available upon request from the corresponding author.

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Author contributions

JZ and MS conceived the experiments. MS conceived the methodical approach. JZ, MB and MS analysed the

data. JZ, JV, M.B. and GN conducted the experiments. PR provided the material and supported the experiments and analyses. JZ and MS wrote the initial manuscript. All authors reviewed the manuscript.

Declarations

Competing interests

The authors declare no competing interests.

Additional information

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