



# OPEN Ionic liquid-assisted biomass-derived N, S-doped carbon dots with enhanced corrosion inhibition

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Developing efficient green corrosion inhibitors from sustainable, renewable, and cost-effective materials is a pressing challenge. Solvents impact carbon dot structures, leading to property disparities, yet the mechanism of solvent-induced structure–property modulation in carbon dots remains not deeply understood. Herein, dried dandelion leaves served as carbon, nitrogen, and sulfur sources. Using 1-butyl-3-methylimidazolium bromide (an ionic liquid, IL) and water as solvents, nitrogen- and sulfur-functionalized biomass-derived CDs, namely IL-CDs and CDs, were synthesized. The inhibition performance of IL-CDs and CDs for carbon steel in H<sub>2</sub>SO<sub>4</sub> solution was evaluated by electrochemical measurements and surface characterizations. Both IL-CDs and CDs contained abundant N- and S-functional groups, endowing them with photoluminescence and distinct UV–Vis spectral features. The IL increased the content of pyrrole-like nitrogen and C–SO<sub>x</sub> group, facilitating the formation of a denser protective film. IL-CDs (~38.3 nm) showed better inhibition efficiency (75.9%) than CDs (73.1%). Adsorption isotherms and corrosion morphology analyses indicated that the inhibition mechanism of IL-CDs and CDs mainly involved physical and chemical adsorption to form a protective film. Notably, pyrrole-like nitrogen species, through  $\pi$ -complex formation, enabled parallel adsorption onto the steel surface, playing a key role in inhibition. This study presents a green strategy for synthesizing efficient biomass-derived carbon dots with IL assistance, advancing the development of sustainable and effective inhibitors.

**Keywords** N, S doped, Biomass-derived CDs, Ionic liquid, Inhibitors, Inhibition mechanism

Reducing global steel demand through improved corrosion resistance to mitigate carbon emissions is a critical challenge<sup>1</sup>. Carbon steel is commonly used in industries, such as construction, petrochemicals, and boilers, due to its stable physical and chemical properties, as well as its low cost. However, it is susceptible to corrosion when exposed to acidic media during processes like acid pickling and oil well acidizing. Therefore, the development of efficient, cost-effective, and easily applicable corrosion inhibitors is crucial for protecting carbon steel from corrosive environments.

Organic compounds with O, N, S, and P heteroatoms, heterocycles, or conjugated unsaturated bonds, are commonly used as inhibitors in acidic environments<sup>2–5</sup>. Their inhibition mechanism typically involves the formation of protective films onto the metal surfaces via physical and/or chemical interactions, thereby slowing corrosion<sup>6</sup>. However, traditional organic inhibitors often cause environmental pollution, prompting interest in “green” alternatives, such as amino acids, ionic liquids (IL/ILs), and plant extracts<sup>7</sup>. Despite their promise, these green inhibitors possess certain inherent limitations. For example, plant extracts often require complex procedures and toxic solvents, which adversely impact the environment<sup>8,9</sup>, while some ILs and amino acids are costly and unsuitable for large-scale use. Consequently, there is growing interest in sustainable, efficient inhibitors with simple synthesis processes.

Carbon dots (CDs), a novel class of zero-dimensional carbon-based fluorescent nanomaterial discovered in 2004<sup>10</sup>, have garnered attention as promising, eco-friendly, and sustainable inhibitors due to their low cost, abundant precursors, and tunable surface properties<sup>11,12</sup>. To enhance their inhibition performance, surface modifications, including heteroatom doping (e.g., N, S, Cu, Ce), are commonly employed during preparation or post-processing<sup>13–15</sup>. The aromatic ring structures with sp<sup>2</sup>-conjugated domains in CDs, along with surface

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functional groups such as carboxyl, hydroxyl, and amino groups at their edges, serve as potential adsorption sites on metal surfaces, providing corrosion protection for metals like Fe, Cu, etc.<sup>16</sup>.

For instance, Ye et al. observed that pyridinic N and pyrrolic N in CDs facilitate chemical adsorption, while graphitic N aggregates on the metal surface to form a protective barrier through physical adsorption<sup>17</sup>. Bhargava et al. reported that polar N groups act as adsorption sites for metal interaction, whereas nonpolar R groups function as hydrophobic shields, providing physical protection<sup>18</sup>. Biomass resources, including plants and their by-products, are rich in organic components and offer significant advantages in preparing biomass-derived CDs, such as low cost, eco-friendliness, and accessibility. Biomass materials containing heteroatoms (N and S) are particularly valuable as precursors for CDs, yielding materials with lower biotoxicity compared to CDs derived from man-made carbon sources that require external heteroatom addition<sup>19</sup>.

Ionic liquids (ILs), versatile organic solvents with excellent solubility, stability, and tunable structures, influencing the physical and chemical properties of CDs, including the size, morphology, hydrophobicity, and surface functional groups<sup>20,21</sup>. ILs reduce interfacial tension and energy, promoting rapid nucleation of CDs and minimizing particle aggregation<sup>22</sup>. Wang et al. synthesized N, S-doped CDs (IL-CDs) using L-cysteine and 1-butyl-3-methylimidazolium bromide as a solvent via a solvothermal method, achieving an inhibition efficiency of up to 97% for carbon steel in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution<sup>12</sup>.

CDs can be synthesized from a wide variety of precursors, such as graphite, small molecules (e.g., citric acid, urea), polymers, and natural materials (e.g., stems, leaves, roots, petals, and fruits)<sup>7,23</sup>. Using eco-friendly, readily available raw materials to produce high-performance CDs is essential for sustainable development. Biomass, being eco-friendly, abundant, low-cost, and renewable, represents an excellent carbon source for the synthesis of biomass-derived CDs<sup>23</sup>. Numerous studies have successfully synthesized biomass-derived CDs from plants<sup>24,25</sup>, applying them in diverse fields, such as plant systems<sup>26</sup> and optoelectronics<sup>27</sup>. However, their application in corrosion protection remains underexplored<sup>7,28</sup>. Long et al. reported that CDs derived from lychee leaves, rich in O and N functional groups, achieved over 90% inhibition efficiency for carbon steel in 1 M HCl solution<sup>7</sup>. This highlights the significant potential of natural biomass-derived CDs as green inhibitors with high efficacy.

Therefore, developing simple synthesis methods for biomass-derived CDs and study their inhibition performance is of utmost importance. IL enables both physical and chemical modifications of carbon dots, effectively mitigating their inherent limitations, such as aggregation at high concentrations and suboptimal long-term corrosion inhibition performance due to weak intermolecular interactions<sup>12,29,30</sup>. Notably, the impacts of ILs and water (H<sub>2</sub>O) on the weak interactions between CDs and between CDs and metal surfaces are still not well-understood. In this study, biomass-derived CDs were synthesized from dandelion leaves, a traditional Chinese medicine, using facile, one-step solvothermal and hydrothermal methods. The influence of solvents on the inhibition performance of these CDs was comprehensively analyzed. The synthesized CDs were characterized by Fourier transform infrared spectroscopy (FTIR), UV–vis spectroscopy (UV–vis), transmission electron microscopy (TEM), and X-ray photoelectron spectroscopy (XPS). Their inhibition performance was evaluated through electrochemical impedance spectroscopy (EIS) and Tafel polarization measurements. The results indicated that dandelion leaf-derived CDs, rich in oxygen, nitrogen, and sulfur functional groups, exhibited notable inhibition in H<sub>2</sub>SO<sub>4</sub> solution. Surface analysis techniques, including scanning electron microscopy (SEM) and energy-dispersive X-ray spectroscopy (EDS), XPS, atomic force microscopy (AFM), and adsorption isotherms, provided the insights to the corrosion inhibition mechanism. This work demonstrates the effectiveness of biomass-derived CDs, especially those enhanced by IL, and underscores the role of IL in improving the dispersion and performance of CDs. These findings contribute to the development of eco-friendly, effective corrosion protection strategies.

## Experimental

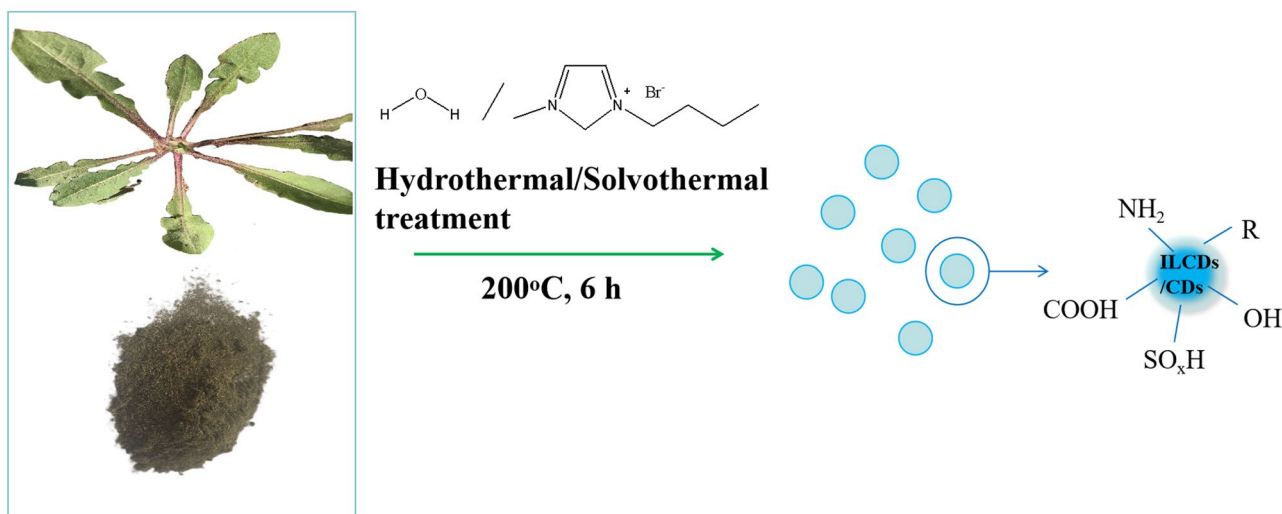
### Materials and chemicals

Carbon steel samples (10 × 10 × 2 mm) used in this study were sourced from a local supplier, and their elemental composition (by weight) consists of C 0.16–0.20%, Mn 0.31–0.46%, Si 0.05–0.20%, S 0.008–0.022%, P 0.020–0.035%, with Fe constituting the balance. Prior to each experiment, the samples were grounded with waterproof sandpaper ranging from 400 to 2000 grit, washed with deionized water, degreased with isopropanol, and air-dried. Dandelion leaves, sourced from a village in Hebei province, China, were cleaned, dried, and ground for use.

### Synthesis of IL-CDs and CDs

In contrast to conventional synthetic routes relying on precursors, such as citric acid and ammonium citrate, etc.<sup>31,32</sup>, the carbon dots in this study were synthesized from herbal-derived biomass (a widely available medicinal plant), offering advantages in environmental sustainability, cost-effectiveness, and material accessibility, compared to previously reported CD-based inhibitors, as demonstrated in prior studies<sup>33,34</sup>.

Natural dandelion leaves were selected as precursor for synthesizing carbon dots via hydrothermal/solvothermal methods using water and IL as solvents, as illustrated in Fig. 1. In a typical synthesis process, 2 g of dried dandelion powder was dissolved in 3 g of IL or 20 mL of water, and the mixture was thoroughly stirred before being placed in a 50 mL Teflon-lined stainless steel autoclave. The mixture was then heated at 200 °C for 6 h. After natural cooling to room temperature, the resulting brown-black mixture was dissolved in 40 mL of deionized water and the solution was centrifuged at 8000 rpm for 10 min. The supernatant was subsequently filtered and dialyzed in distilled water for 24 h using a dialysis membrane with a molecular weight cutoff of 1000 g/mol. After dialysis, the obtained mixture was vacuum-dried at 60 °C, yielding dark brown solid carbon dots, named IL-CDs and CDs based on the IL and water solvents used, respectively, with a yield of approximately 3–5%.



**Fig. 1.** Schematic illustration of preparation of IL-CDs (by solvothermal method) and CDs (by hydrothermal methods) from dandelion dry leaves.

### Preparation of IL-CDs and CDs solution

To investigate the inhibition behavior of IL-CDs and CDs in 0.5 M  $\text{H}_2\text{SO}_4$  solution, the concentrations of IL-CDs were selected as 15, 75, 100, and 150 mg/L, while 150 mg/L CDs was prepared for comparison. The 0.5 M  $\text{H}_2\text{SO}_4$  solution was ordered from Guangzhou Howei Pharma Technology Co., Ltd.

### Structure characterization of IL-CDs and CDs

The morphology of IL-CDs and CDs was examined using transmission electron microscope (TEM, Tecnai G2 Spirit, USA). Both IL-CDs and CDs were dispersed in ethanol, dropped onto a copper mesh, dried, and subsequently analyzed by TEM. The chemical composition and structural information were obtained using Fourier transform infrared spectroscopy (FTIR, Thermo Scientific Nicolet iS50, USA) and X-ray photoelectron spectroscopy (XPS, PHI 5000 VersaProbe III, Japan). XPS measurements were collected using an Al K $\alpha$  anode (1486.6 eV), with the binding energy reference set at the C 1s peak of carbon, 284.8 eV. Ultraviolet–visible (UV–vis) spectra of IL-CDs and CDs in deionized water were recorded using quartz cells (1 cm path length) in the 200–800 nm range with a UV–vis spectrophotometer (Agilent Cary 5000, USA).

### Electrochemical measurement

Electrochemical tests, including electrochemical impedance spectra (EIS) and Tafel polarization (PO), were conducted using a three-electrode system with an electrochemical workstation (CHI604D). A carbon steel electrode (1 cm<sup>2</sup> exposed area) was used as the working electrode, while a platinum (Pt) sheet and a saturated calomel electrode (SCE) served as the counter electrode and the reference electrode, respectively. The working electrode was immersed in a 0.5 M  $\text{H}_2\text{SO}_4$  solution, either with or without the inhibitor, for 3600 s. Open circuit potential (OCP) measurements were conducted at 298.15 K until a stable OCP was reached, followed by EIS and PO.

EIS measurements were performed with an AC signal amplitude of 5 mV over a frequency range of 10<sup>5</sup> to 0.01 Hz. Tafel polarization curves were recorded over a potential range of  $\pm 0.25$  V versus OCP at a scan rate of 0.5 mV/s. All experiments were repeated three times to ensure reproducibility. EIS data were fitted and analyzed using ZSimDemo software. The inhibition efficiency derived from EIS was calculated using Eq. 1<sup>35</sup>.

$$IE_{\text{EIS}}\% = \frac{R_{\text{ct}}^0 - R_{\text{ct}}}{R_{\text{ct}}} \times 100 \quad (1)$$

where  $R_{\text{ct}}^0$  and  $R_{\text{ct}}$  are the charge transfer resistances of carbon steel without and with the inhibitor, respectively, in  $\Omega/\text{cm}^2$ .  $IE_{\text{EIS}}\%$  refers to the inhibition efficiency gained from EIS tests.

The inhibition efficiency from PO was calculated using Eq. 2<sup>35</sup>.

$$IE_{\text{tafel}} = \frac{i_{\text{corr}}^0 - i_{\text{corr}}}{i_{\text{corr}}^0} \times 100 \quad (2)$$

where  $i_{\text{corr}}^0$  and  $i_{\text{corr}}$  represent the corrosion current densities of carbon steel in the absence and presence of the inhibitor, respectively, measured in  $\text{A}/\text{cm}^2$ .  $IE_{\text{tafel}}\%$  refers to the inhibition efficiency gained from PO tests.

## Surface characterization

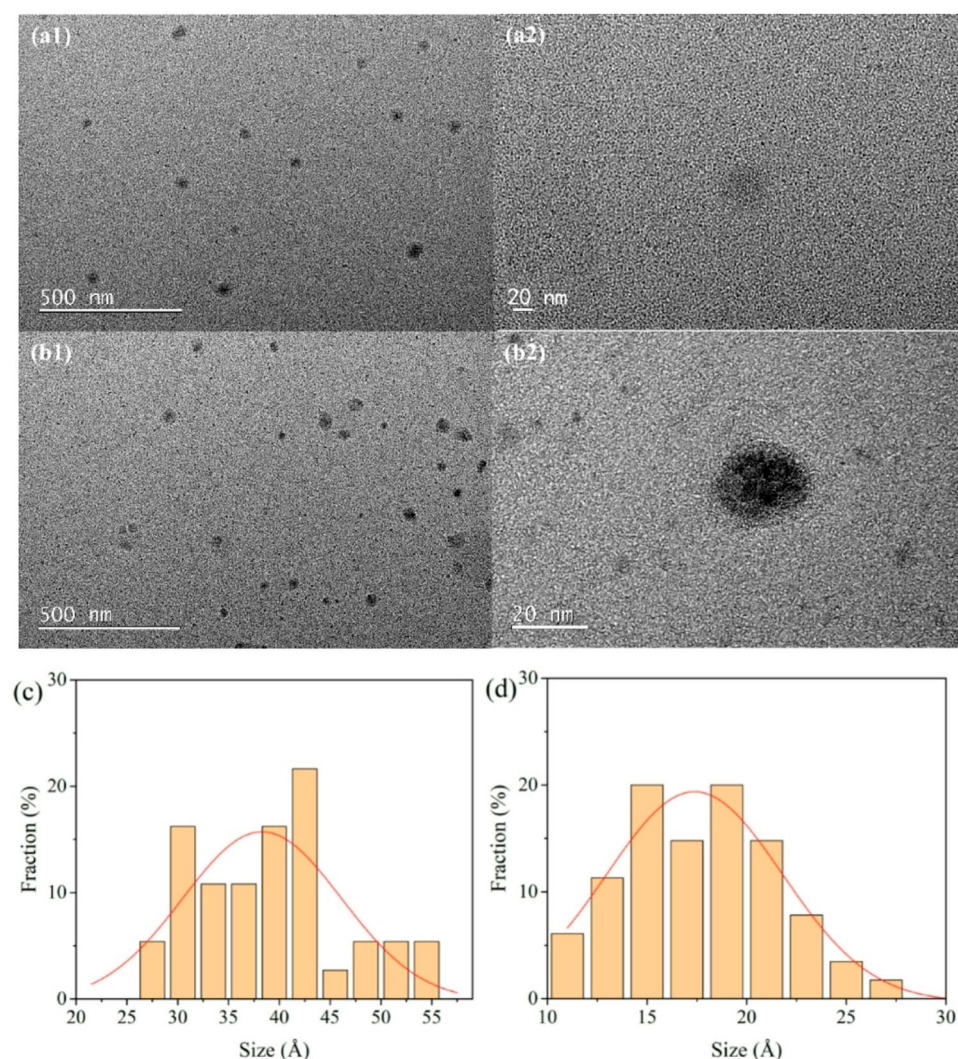
For surface characterization, carbon steel samples ( $1 \times 1 \times 0.2$  cm) were immersed in  $0.5 \text{ M H}_2\text{SO}_4$  solutions in the absence and presence of  $150 \text{ mg/L}$  of IL-CDs or CDs for 6 h. After immersion, the samples were removed from solution, cleaned with acetone, and air-dried. Surface morphology and elemental distribution after inhibitor adsorption were examined using scanning electron microscopy (SEM, (KYKY-EM8000) and energy-dispersive X-ray spectroscopy (EDS) (Oxford Instruments). X-ray photoelectron spectroscopy (XPS, PHI 5000 VersaProbe III, Japan) was used to analyze the chemical states on the steel surface. Atomic force microscopy (AFM, Bruker Dimension iCon, Germany) was employed to observe the morphology and roughness of the corroded steel.

## Results and discussion

### Preparation and characterization of IL-CDs and CDs

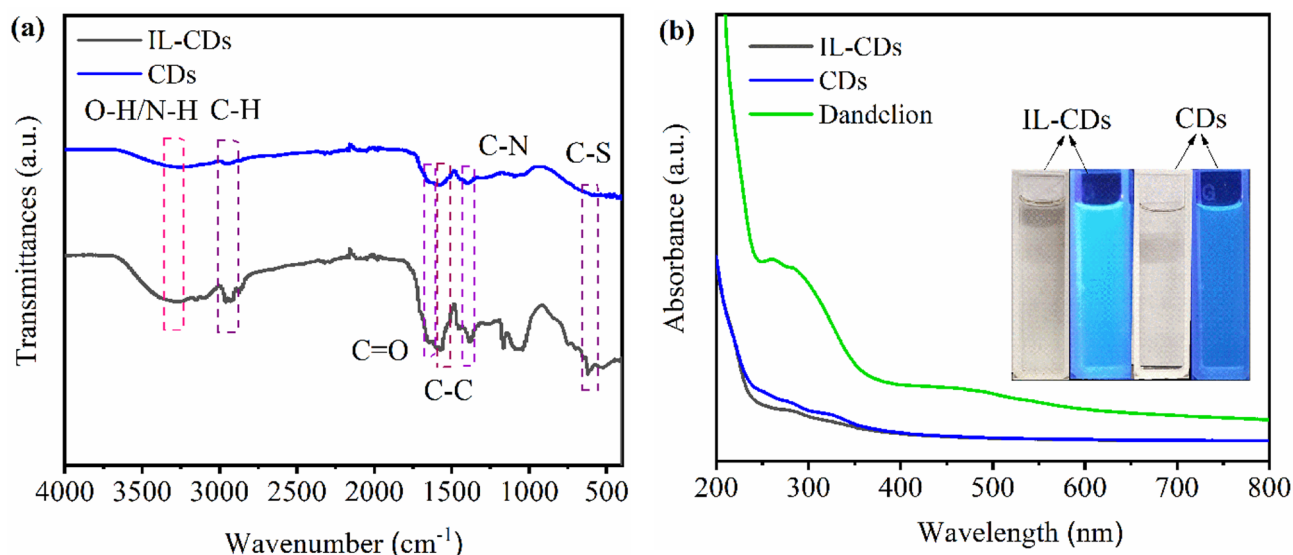
Figure 2 shows the TEM images and particle size distributions of the synthesized IL-CDs and CDs. The average size of IL-CDs is approximately  $38.3 \text{ nm}$  (Fig. 2a1,a2), while that of CDs is around  $17.5 \text{ nm}$  (Fig. 2b1,b2). Both types of carbon dots are highly dispersed and exhibit nearly quasi-spherical morphologies. The use of IL as a solvent plays a crucial role in yielding larger and more uniformly distributed carbon dots. This is because ILs do not engage in the formation of carbon nuclei during the synthesis of IL-CDs. Notably, compared to CDs, IL-CDs exhibit larger and more uniform particle sizes. The increased particle size is likely due to the IL's role in facilitating polycondensation reactions, thus promoting the growth of more homogeneous structures. TEM analysis further confirms that the carbohydrates from dandelion leaves undergo dehydration, condensation, polymerization, and subsequent carbonization under heat and pressure. This process leads to the formation of IL-CDs and CDs with  $\text{sp}^2$  carbon structures at their cores.

FTIR analysis revealed the surface chemical states of the as-prepared IL-CDs and CDs. The FTIR spectra (Fig. 3a) show characteristic peaks at  $3273$ ,  $1648$ , and  $1382 \text{ cm}^{-1}$  for IL-CDs, and at  $3262$ ,  $1634$ , and  $1401 \text{ cm}^{-1}$  for CDs. These peaks confirmed the presence of  $\text{O-H/N-H}$ ,  $\text{C=O}$ , and  $\text{C-N}$  bonds, respectively<sup>36,37</sup>. The abundant



**Fig. 2.** TEM images and grain sizes distributions of IL-CDs (a1,a2,c) and CDs2 (b1,b2,d).





**Fig. 3.** FTIR and UV-Vis spectra of as-prepared IL-CDs and CDs: FTIR spectra (a), UV-vis spectra (b), and the inset is the fluorescence images under sunlight and UV light (excited at 365 nm).

oxygen-, nitrogen-, and sulfur-containing functional groups endow IL-CDs and CDs with outstanding water solubility and dispersibility, making them ideal candidates as water-based inhibitors. Additionally, the peaks at  $2931\text{ cm}^{-1}$  for IL-CDs and  $2934\text{ cm}^{-1}$  for CDs are attributed to C-H stretching vibrations, while the peaks at  $1596\text{ cm}^{-1}$  for IL-CDs and  $1578\text{ cm}^{-1}$  for CDs correspond to C-C stretching vibrations in aromatic rings<sup>38</sup>. This confirmed the  $\text{sp}^2$  carbon structure, which is consistent with the TEM results.

The UV-visible absorption spectra (Fig. 3b) display a broad peak between 250 and 350 nm for the dandelion precursor. After the transformation into carbon dots, two weak absorption peaks appear at approximately 275 nm and 320 nm. These peaks correspond to the  $\pi-\pi^*$  transition of aromatic  $\text{sp}^2$  carbon and the  $n-\pi^*$  transition of oxygen-, nitrogen-, and sulfur-containing functional groups, respectively<sup>39</sup>. As depicted in the inset of Fig. 3b, the aqueous solutions of IL-CDs and CDs appear light yellow under visible light and emit strong blue photoluminescence (PL) when excited at 365 nm UV light. The UV-visible absorption spectra confirm the successful synthesis of the carbon dots.

The XPS survey spectrum (Fig. 4a) shows four characteristic peaks corresponding for C, O, S, and N. The high-resolution C 1s spectrum (Fig. 4b) is deconvoluted into four distinct peaks at  $284.8$ ,  $285.5 \pm 1$ ,  $286.5$ , and  $288.2\text{ eV}$ . These peaks can be assigned to C-C/C=C, C-N/C-S, C-O, and C=N/C=O bonds, respectively, thereby confirming the successful doping of N and S into the CDs<sup>40</sup>. The N 1s spectrum (Fig. 4c) is deconvoluted into three peaks at  $398.2\text{ eV}$ ,  $399.5\text{ eV}$ , and  $401.3\text{ eV}$ , which correspond to pyridinic nitrogen, pyrrolic nitrogen, and graphitic nitrogen, respectively<sup>41,42</sup>. Among them, pyrrolic nitrogen is predominant, and only minor pyridinic nitrogen is detected. The O 1s spectrum in Fig. 4d shows peaks at  $531.0\text{ eV}$  and  $531.8\text{ eV}$ , which are attributed to C=O and C-O bonds, respectively<sup>43</sup>. In Fig. 4e, the S 2p spectrum displays two peaks at  $163.9$  and  $167.7 \pm 1\text{ eV}$ , which are attributed to the C-S bond and C-SO<sub>x</sub> groups, respectively<sup>44</sup>. Sulfur predominantly exists in the form of C-SO<sub>x</sub> within the carbon dots. These findings indicate the sulfur is mainly derived from glucosinolates in the dandelion leaves. These results confirm the successful synthesis of N- and S- co-doped IL-CDs and CDs.

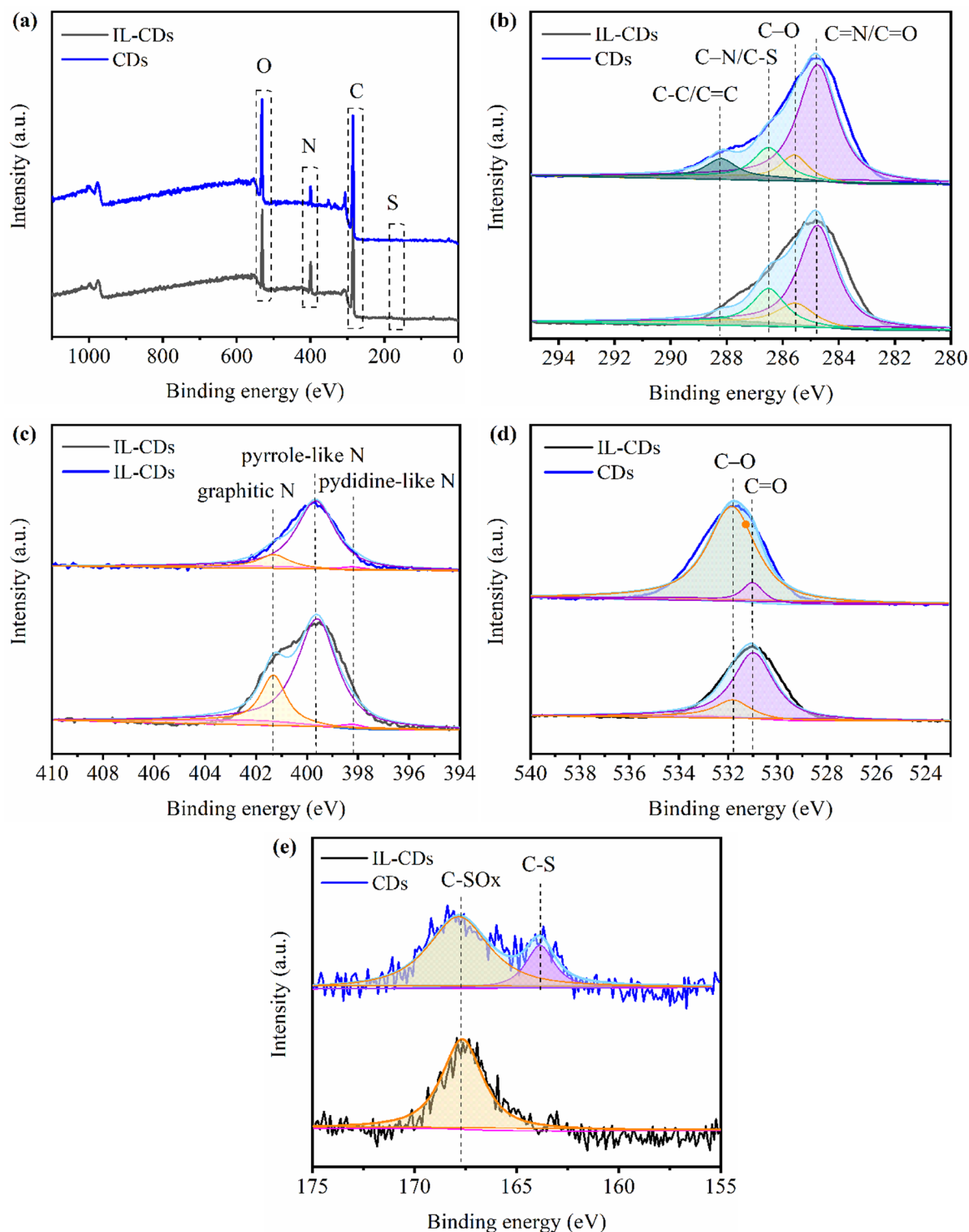
Table 1 shows a detailed summary of the nitrogen species in both IL-CDs and CDs. Pyridinic nitrogen donates lone-pair electrons to form Fe-N bonds perpendicular to the steel surface<sup>45</sup>. In contrast, pyrrolic nitrogen interacts with the steel surface via its aromatic ring, forming a  $\pi$ -complex that aligns the pyrrole ring parallel to the surface<sup>46</sup>. Thus, the adsorption of pyrrolic nitrogen significantly contributes to corrosion inhibition. As indicated in Table 1, IL-CDs, having a higher content of pyrrolic nitrogen, demonstrate superior inhibition performance.

The XPS measurements corroborate the conclusions drawn from the FTIR spectra. They strongly supporting the fact that IL-CDs and CDs are composed of an  $\text{sp}^2$  carbon core with a passivated surface rich in oxygen-, nitrogen- and sulfur-containing functional groups, which play a key role in their corrosion inhibition properties.

## Electrochemical analysis

### Electrochemical impedance spectroscopy (EIS) analysis

The Nyquist and corresponding Bode plots of carbon steel immersed in  $0.5\text{ M H}_2\text{SO}_4$  solution in the absence and presence of IL-CDs, are displayed in Figs. 5 and 6, respectively. The Nyquist plots (Fig. 5) show depressed capacitive loops, indicative of the charge transfer process occurring at the electrode/solution interface at high frequencies, both in the uninhibited and inhibited with IL-CDs and CDs. The diameter of the capacitive loop increases after the addition of IL-CDs and CDs. Moreover, the depression of the semicircle, with its center located below the x-axis, is associated with the inhomogeneity and roughness of the steel surface<sup>47,48</sup>. At low

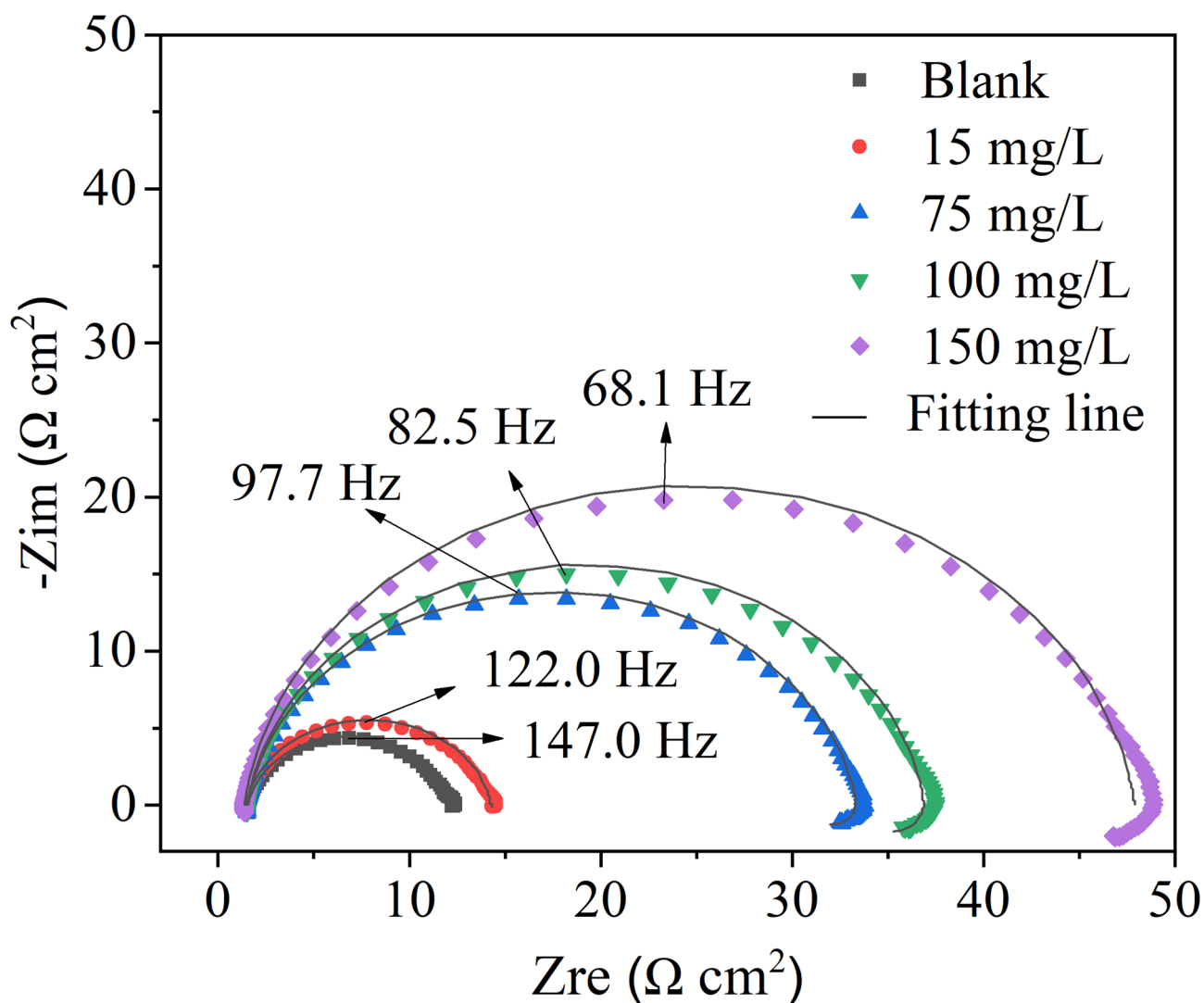


**Fig. 4.** XPS spectra of IL-CDs and CDs: (a) full spectra, (b) C 1s spectra, (c) N 1s spectra, (d) O 1s spectra, (e) S 2p spectra.

frequencies, a small inductive loop appears, which is attributed to relaxation processes, such as the adsorption of  $H^+$  and  $SO_4^{2-}$  ions onto the steel surface<sup>49,50</sup>.

The Bode plots in Fig. 6 reveal that all phase angles predominantly exhibit a single phase peak, and as the IL-CDs concentration increases, the maximum phase angle shifts towards higher values (Fig. 6a). Additionally, the impedance modulus (Fig. 6b) increases with the addition of IL-CDs, implying that IL-CDs enhance the

Samples	Pyridine-like N (%)	Pyrrole-like N (%)	Graphitic-like N (%)
IL-CDs	2.0	77.5	20.5
CDs	2.6	56.5	40.9

**Table 1.** The relative amount of N species existed in as-prepared IL-CDs and CDs.**Fig. 5.** Nyquist plots of carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with different concentrations of the IL-CDs.

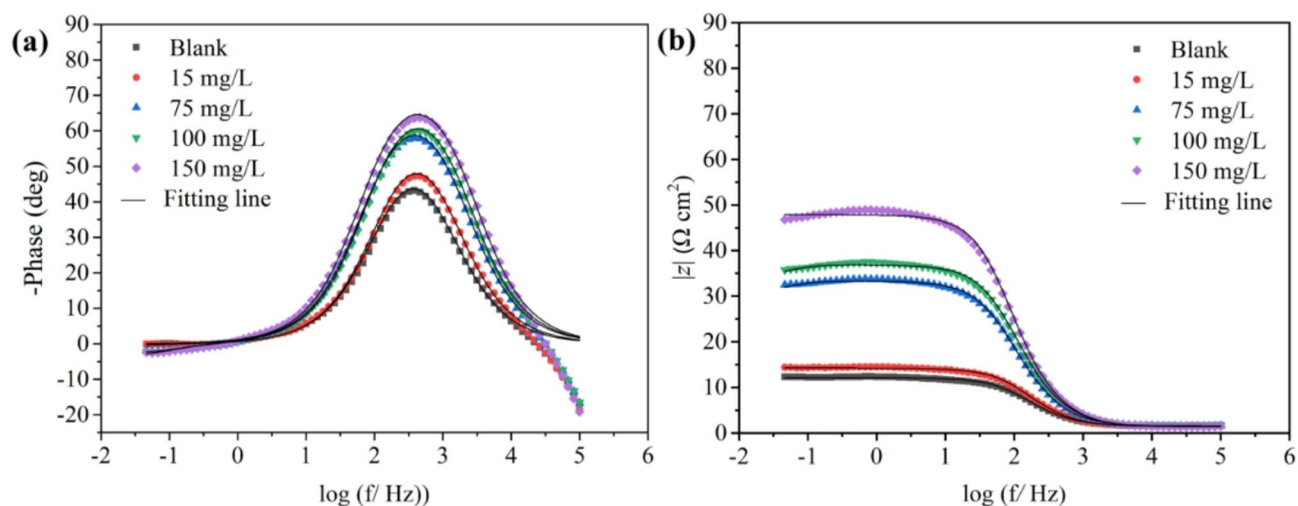
protective capability of the steel. These results indicate that the addition of IL-CDs endows the steel with effective protective.

Based on these results, a simple-constant equivalent circuit and a two-time-constant equivalent circuit were respectively used to analyze the EIS data in the absence and presence of IL-CDs (Fig. 7). These circuits consist of solution resistance ( $R_s$ ), charge transfer resistance ( $R_{ct}$ ), a constant-phase angle element (CPE/Q) to replace ideal double-layer capacitance due to the non-ideal capacitance behavior of the inhomogeneous electrode<sup>47</sup>, inductance (L), and inductive resistance ( $R_L$ ).

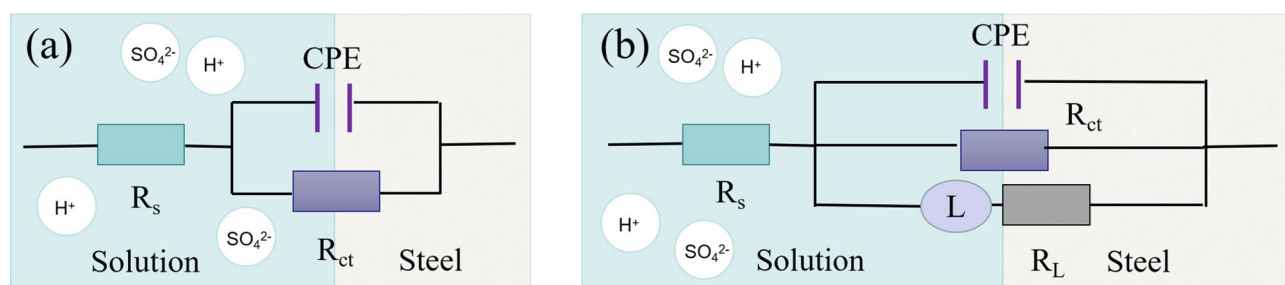
The inductance (L) is related to the relaxation process caused by adsorbed species (such as  $\text{H}^+$  and  $\text{SO}_4^{2-}$ ) on the steel surface. The impedance of CPE is described as follows<sup>51</sup>:

$$Z_{\text{CPE}} = Y_0^{-1} (j\omega)^{-n} \quad (3)$$

where  $Y_0$  is the modulus of the CPE,  $n$  ( $-1 \leq n \leq 1$ ) is the CPE exponent,  $j$  ( $j = (-1)^{1/2}$ ) refers to an imaginary number, and  $\omega$  ( $\omega = 2\pi f$ ) is the angular frequency in  $\text{rad}^{-1}$ .



**Fig. 6.** The relevant Bode plots for carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with different concentrations of the N-CDs.



**Fig. 7.** Electrochemical equivalent circuit for fitting EIS data for carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution (a) without inhibitor, and (b) with IL-CDs and CDs.

IL-CDs (mg/L)	$R_s$ ( $\Omega \text{ cm}^2$ )	CPE $Y_0$ ( $10^6 \Omega^{-1} \text{ s}^n \text{ cm}^{-2}$ )	$n$	$R_{ct}$ ( $\Omega \text{ cm}^2$ )	$L$ (H)	$R_f$ ( $\Omega \text{ cm}^2$ )	IE (%)
0	1.6	24.93	0.8979	10.6	—	—	—
15	1.5	19.46	0.9077	12.8	5305	0.1	17.2
75	1.5	12.12	0.9105	31.9	1232	364.2	66.8
100	1.5	9.69	0.9200	35.4	1344	323.9	70.0
150	1.4	8.38	0.9268	46.5	5085	308.6	77.2

**Table 2.** The fitting electrochemical parameters obtained from EIS results for carbon steel immersed in 0.5 M  $\text{H}_2\text{SO}_4$  solution in the absence and presence of different concentrations of the IL-CDs.

$C_{dl}$  can be obtained by the following equation<sup>51</sup>:

$$C_{dl} = Y_0 (2\pi f_{\max})^{n-1} \quad (4)$$

where  $f_{\max}$  refers to the maximum frequency of the impedance spectrum.

$\omega_{\max}$  can also be calculated by the following equation<sup>51</sup>:

$$C_{dl} = \epsilon^0 \epsilon S / d \quad (5)$$

where  $d$  refers to the thickness of the double layer,  $\epsilon^0$  is a constant of the permittivity of air,  $\epsilon$  is the local dielectric constant, and  $S$  represents the area of the surface of working electrode.

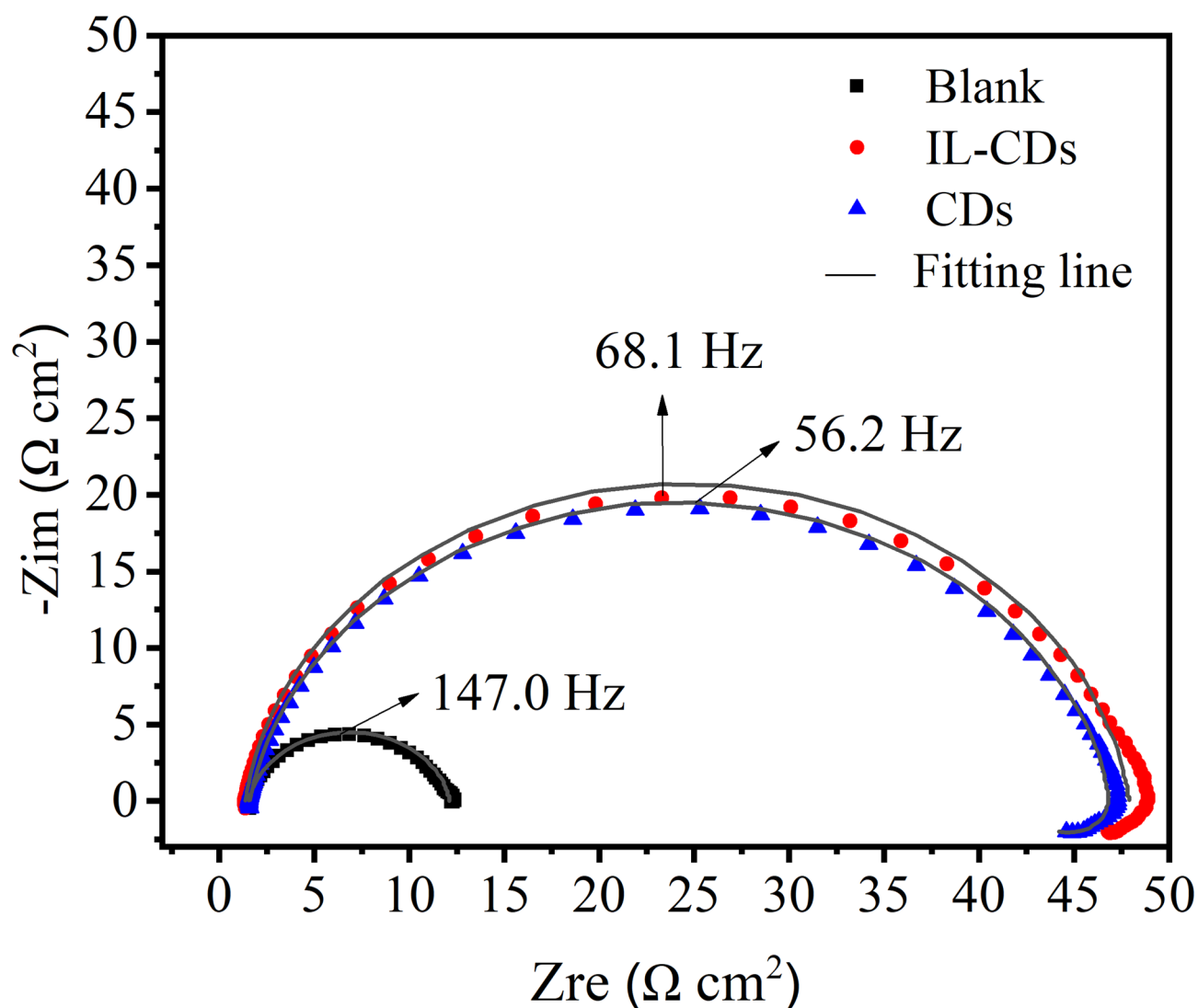
The electrochemical parameters derived from EIS data by using these circuits are summarized in Table 2. The results show that upon the addition of IL-CDs, the  $R_{ct}$  values increase while the  $C_{dl}$  values decrease, and these



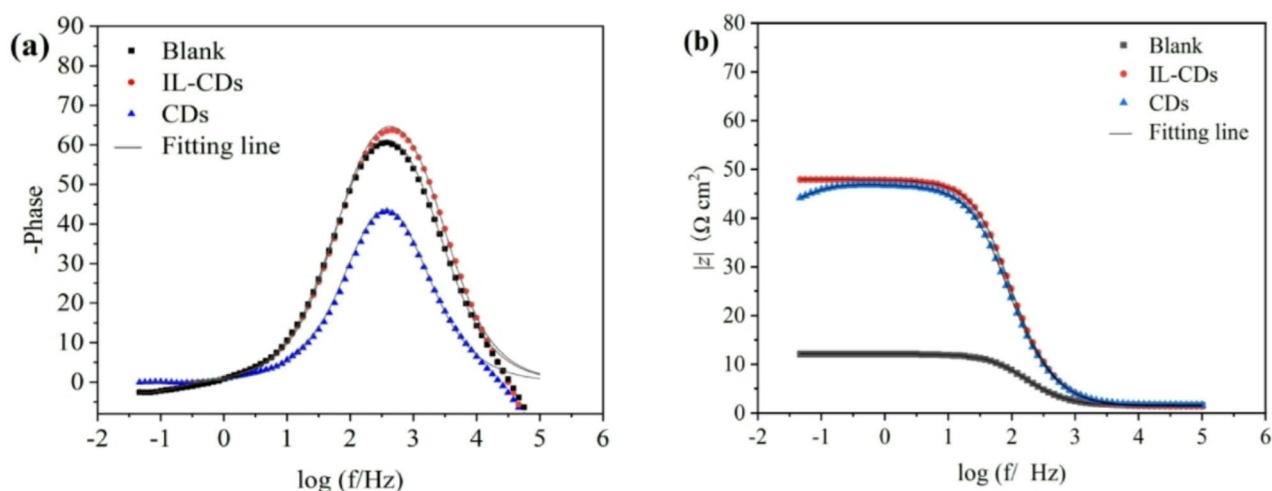
effects are more pronounced at higher IL-CDs concentrations. This indicates a decrease in the local dielectric constant  $\epsilon$  and/or an increase in the double-layer thickness  $d$ , likely attributed to the formation of an adsorption film at the metal/solution interface. It is inferred that IL-CDs molecules interact with the steel surface through adsorption, simultaneously with the displacement of  $\text{H}_2\text{O}$  molecules or other ions that were originally adsorbed on the steel surface from the bulk solution. This process effectively reduces the corrosion rate. At 298 K, it reaches a maximum of 77.2% at 150 mg/L, in contrast to 17.2% at 15 mg/L.

To compare the impact of different solvents on the performance of inhibitors, the inhibition behavior of IL-CDs and CDs for carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution is further analyzed. The Nyquist and relevant Bode plots are displayed in Figs. 8 and 9, respectively. The Nyquist plots (Fig. 8) show depressed capacitive loops. These loops, corresponding to the charge-transfer process at electrode/solution interface at the high frequencies, are present in both the blank and those with IL-CDs and CDs. The addition of IL-CDs and CDs increases the diameter of the capacitive loop. The depressed semicircle with its center below the x-axis further indicates the inhomogeneity and roughness of the steel surface<sup>52</sup>. At low frequencies, a small inductive loop appears owing to relaxation processes such as  $\text{H}^+_{\text{ads}}$  and  $\text{SO}_4^{2-}_{\text{ads}}$  on the metal substrates<sup>53,54</sup>.

The Bode plot in Fig. 9 shows that the addition of IL-CDs and CDs increases the impedance modulus and shifts the frequency at the maximum phase angle to higher values in inhibited solutions. To analyze the EIS data, the two-time-constant equivalent circuit shown in Fig. 7 was used, and the corresponding electrochemical parameters are summarized in Table 3. It is evident that with the addition of IL-CDs and CDs,  $R_{\text{ct}}$  values increase while  $C_{\text{dl}}$  values decrease, indicating the formation of a protective film at the metal/solution interface. Notably, IL-CDs shows a higher value of  $R_{\text{ct}}$  than CDs, resulting in a higher inhibition efficiency of 77.2% for IL-CDs compared to 76.6% for CDs.



**Fig. 8.** Nyquist plots for carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with 150 mg/L IL-CDs and CDs.



**Fig. 9.** The relevant Bode plots for carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with 150 mg/L IL-CDs and CDs.

IL-CDs/CDs	$R_s$ ( $\Omega \text{ cm}^2$ )	$Y_o$ ( $\mu\Omega^{-1} \text{ S}^n \text{ cm}^{-2}$ )	$CPE$ ( $\text{F cm}^{-2}$ )	$n$	$R_{ct}$ ( $\Omega \text{ cm}^2$ )	$L$ (H)	$R_f$ ( $\Omega \text{ cm}^2$ )	$IE$ (%)
0	1.6	24.93		0.8979	10.6	—	—	—
IL-CDs	1.4	8.38		0.9268	46.5	5085	308.6	77.2
CDs	1.6	10.48		0.9052	45.3	1240	433.6	76.6

**Table 3.** The fitting electrochemical parameters obtained from EIS results for carbon steel immersed in 0.5 M  $\text{H}_2\text{SO}_4$  solution in the absence and presence of 150 mg/L IL-CDs and 150 mg/L CDs.

#### Polarization (PO) analysis

Figure 10 shows the PO plots of carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution, either uninhibited or inhibited with varying concentrations of IL-CDs. Table 4 shows the corresponding electrochemical parameters, such as corrosion current density ( $i_{\text{corr}}$ ), corrosion potentials ( $E_{\text{corr}}$ ), inhibition efficiency ( $IE\%$ ), anodic slope ( $\beta_a$ ), and cathodic slope ( $\beta_c$ ). As illustrated in Fig. 10, compared to the blank solution, both the cathodic and anodic branches of the Tafel plots shift towards lower current densities, and this trend becomes more pronounced as the concentration of IL-CDs increases.

From Table 4, it is evident that, in the presence of IL-CDs, all  $i_{\text{corr}}$  values, including cathodic and anodic current densities are lower than those of the blank. This indicates that both the cathodic dissolution of Fe and the anodic hydrogen evolution reactions are inhibited. Furthermore, the corrosion current densities decrease as the IL-CDs concentration rises. The maximum shift of  $E_{\text{corr}}$  in the presence of IL-CDs is less than  $\pm 85$  mV, indicating that IL-CDs act as a mixed-type inhibitor, retarding both anodic and cathodic reaction<sup>55</sup>. Correspondingly, the  $E_{\text{corr}}$  and  $IE\%$  values increase with the IL-CDs concentration, with a maximum  $IE\%$  reaching 75.9%.

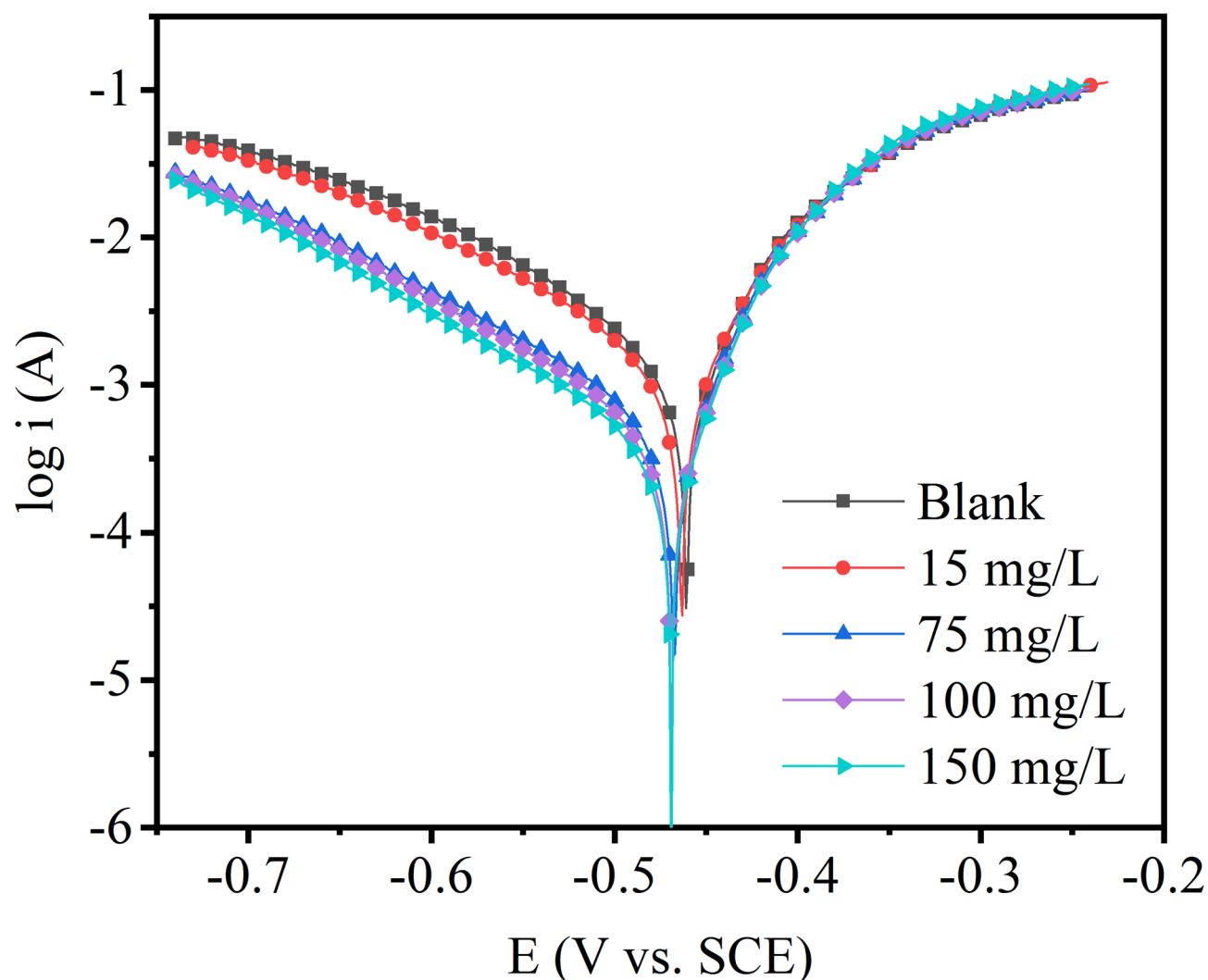
Figure 11 shows the polarization plots of carbon steel immersed in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with 150 mg/L IL-CDs and CDs. The corresponding electrochemical parameters are presented in Table 5. The Tafel plots reveal that, compared to the uninhibited steel, both the cathodic and anodic branches of IL-CDs and CDs shift towards lower current densities. Notably, IL-CDs exhibit slightly lower current densities and more positive  $E_{\text{corr}}$  values than CDs. This shift indicates that both IL-CDs and CDs act as mixed-type inhibitors, with a predominantly cathodic inhibition effect.

As shown in Table 5, at a constant concentration, IL-CDs exhibit slightly lower  $i_{\text{corr}}$ , more positive  $E_{\text{corr}}$  and higher  $IE\%$  values than CDs. This observation correlates with the higher proportion of pyrrolic N in IL-CDs, indicating that the pyrrolic N content is a crucial factor influencing the inhibition efficiency. Pyridine nitrogen typically donates lone pair electrons to form a direct Fe–N bond perpendicular to the steel surface<sup>46</sup>, while pyrrole nitrogen interacts with the steel surface via its aromatic ring, forming a  $\pi$ -complex that aligns the pyrrole ring parallel to the surface<sup>46</sup>. Therefore, the pyrrole group contributes more significantly to corrosion inhibition. As indicated in Table 1, IL-CDs have a higher concentration of pyrrolic nitrogen, which likely accounts for their superior inhibition performance. The inhibition performance is in good agreement with the results obtained from the EIS results.

#### Corrosion morphology analysis

##### SEM

Figure 12 shows the SEM images of carbon steel before and after 6 h immersion in 0.5 M  $\text{H}_2\text{SO}_4$  solution, in the absence and presence of 150 mg/L IL-CDs and CDs. The steel surface before immersion is relatively smooth, with



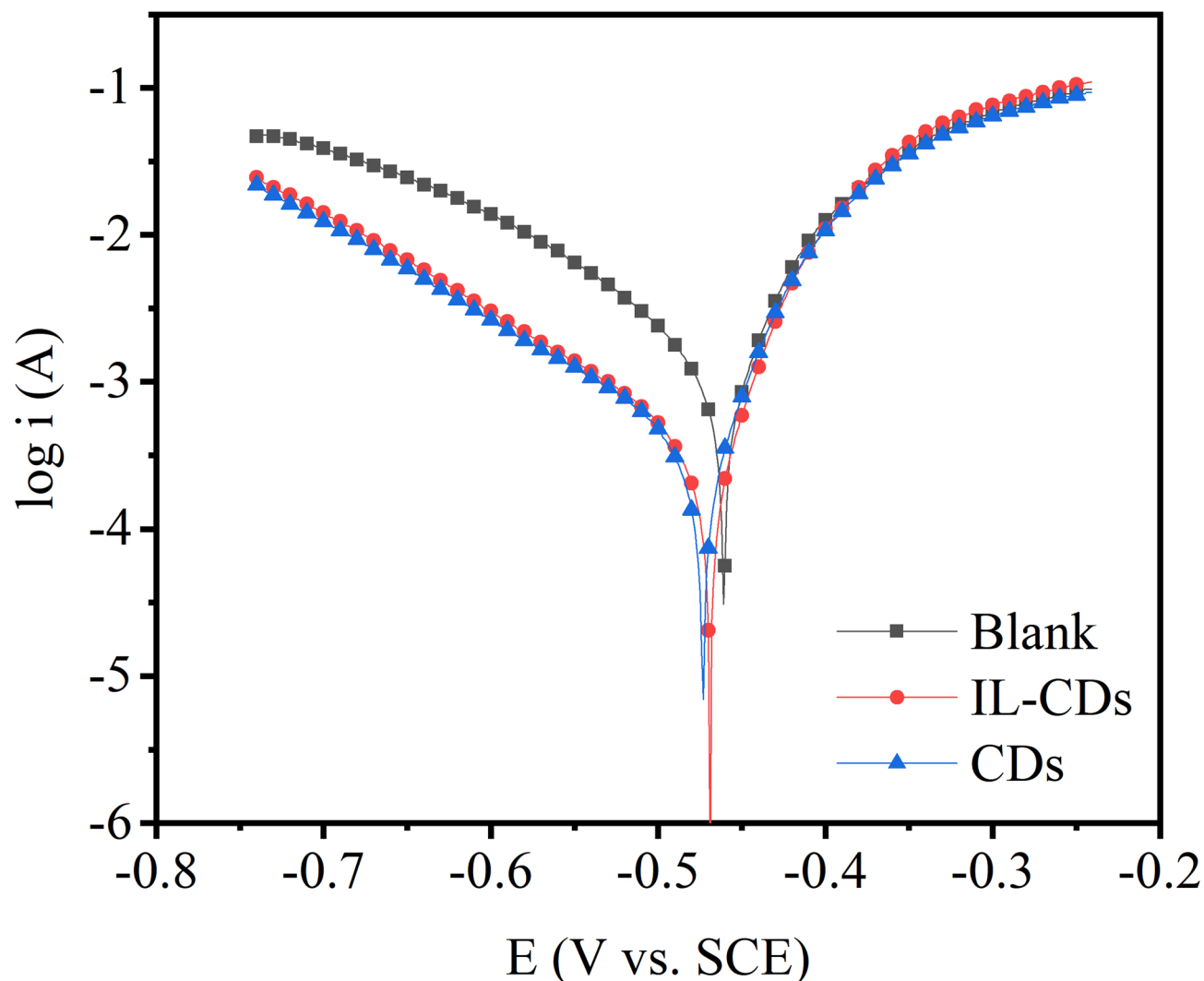
**Fig. 10.** Polarization plots of carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with varying concentrations of IL-CDs.

IL-CDs (mg/L)	$E_{\text{corr}}$ (V/SCE)	$\beta_c$ (mV dec $^{-1}$ )	$\beta_a$ (mV dec $^{-1}$ )	$i_{\text{corr}}$ ( $\mu\text{A cm}^{-2}$ )	IE (%)
0	-0.461	-7.394	9.139	2013.0	–
15	-0.463	-6.758	9.732	1760.0	12.5
75	-0.468	-6.675	11.392	707.3	64.9
100	-0.469	-6.825	11.922	579.7	71.2
150	-0.469	-6.768	12.391	484.6	75.9

**Table 4.** Electrochemical parameters obtained from polarization plots and corresponding inhibition efficiency for carbon steel in the absence and presence of IL-CDs at different concentrations.

some polishing-induced scratches (Fig. 12a). However, after immersion in the inhibitor-free  $\text{H}_2\text{SO}_4$  solution, the steel surface is severely corroded, featuring a rough structure and numerous corrosion products (Fig. 12b). In contrast, the steel surfaces treated with IL-CDs and CDs exhibit less corrosion damage, with IL-CDs-treated surface showing the least deterioration, indicating that IL-CDs provide superior inhibition compared to CDs (Fig. 12c,d).

EDS analysis of the steel without IL-CDs detected C, O, Fe and S elements (Fig. 12e). Notably, the N element was detected on the carbon steel treated with IL-CDs, indicating the successful adsorption of IL-CD molecules onto the steel surface (Fig. 12f). Additionally, compared to the blank, the intensity of S and O on the steel surface decreases in the presence of IL-CDs, further validating its inhibition ability. Elemental scanning images in Fig. 13 show a significant distribution of O on the steel surface, indicating an oxidation reaction. Meanwhile,



**Fig. 11.** Polarization plots of carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution uninhibited and inhibited with 150 mg/L IL-CDs and CDs.

IL-CDs/CDs	$E_{\text{corr}}$ (V/SCE)	$\beta_c$ ( $\text{mV dec}^{-1}$ )	$\beta_a$ ( $\text{mV dec}^{-1}$ )	$i_{\text{corr}}$ ( $\mu\text{A cm}^{-2}$ )	IE (%)
0	-0.461	-7.394	9.139	2013.0	–
IL-CDs	-0.469	-6.768	12.391	484.6	75.9
CDs	-0.473	-6.386	11.465	540.5	73.1

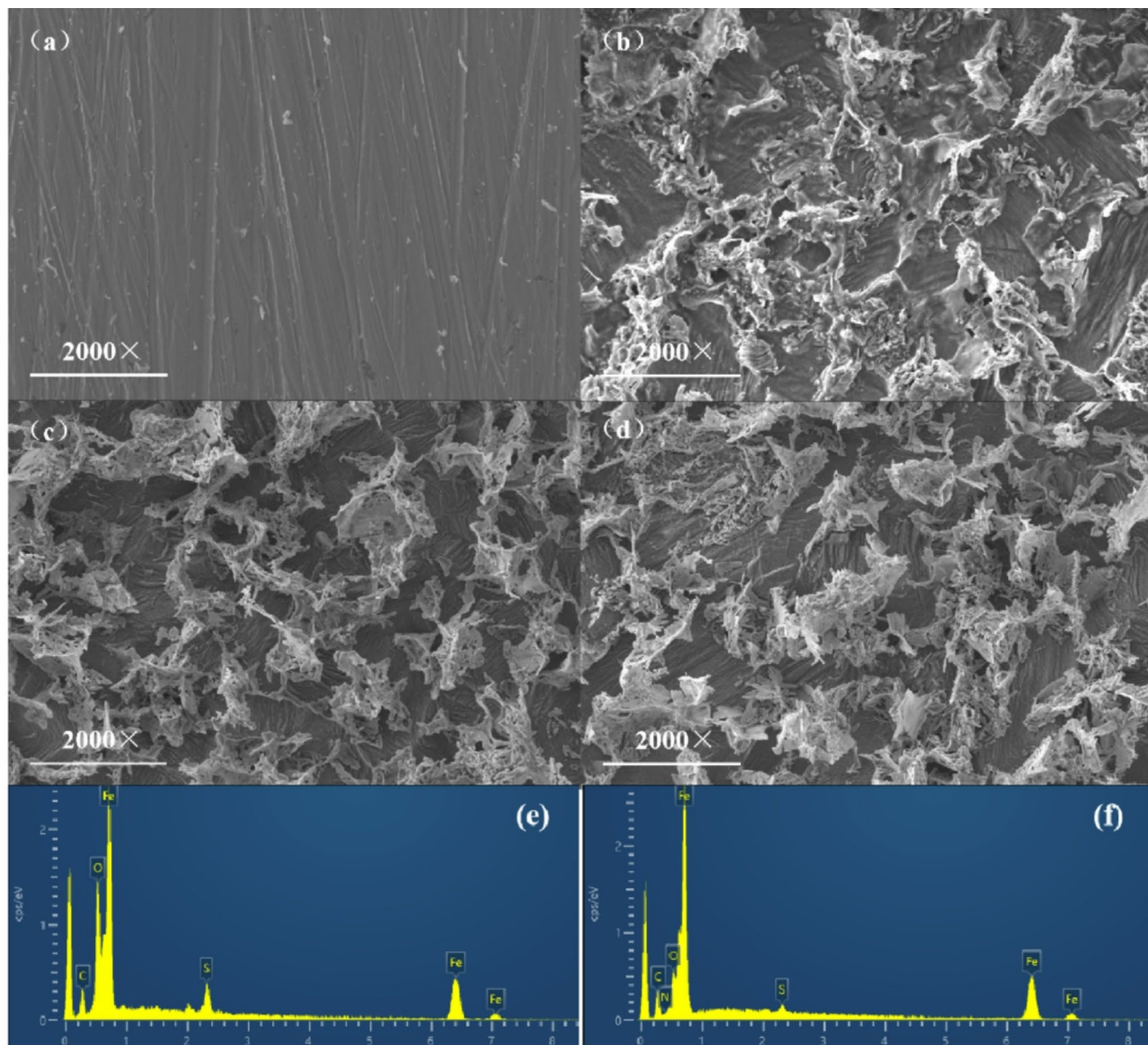
**Table 5.** Electrochemical parameters obtained from polarization plots and corresponding inhibition efficiency for carbon steel in 0.5 M  $\text{H}_2\text{SO}_4$  solution in the absence and presence of 150 mg/L IL-CDs and 150 mg/L CDs.

the N element is evenly distributed on the IL-CDs treated steel surface, illustrating a uniform distribution of IL-CDs (Fig. 13). This distribution implies that the protective film formed by IL-CDs effectively suppresses steel corrosion.

Compared to the blank  $\text{H}_2\text{SO}_4$  solution, in the presence of 150 mg/L IL-CDs, the Fe element content on the steel surface increases from 50.4 to 60.4 at%, accompanied by a decrease in the O element content from 22.1 to 7.9 at%. This evidence supports the fact the homogenous protective film formed by IL-CDs adsorption restricts corrosion caused by the blank  $\text{H}_2\text{SO}_4$  solution, resulting in fewer corrosion products on the steel surface.

#### XPS

XPS data further validate the EDS analysis findings. As seen in Fig. 14a, XPS full spectra show C, O, Fe, and S atoms on the steel surface in both uninhibited and inhibited solutions with 150 mg/L IL-CDs and CDs. Nitrogen from the IL-CDs and CDs is detected in the inhibited solution, indicating their successful adsorption onto the



**Fig. 12.** SEM images and EDS spectra of the steel surfaces before treatment (a), after 6 h of immersion in 0.5 M  $\text{H}_2\text{SO}_4$  solution without inhibitors (b), with 150 mg/L IL-CDs (c), and with 150 mg/L CDs (d), EDS-blank (e), EDS-IL-CDs (150 mg/L) (f).

steel surface to form a protective film. Figure 14b shows three peaks in the high-resolution N 1s spectrum: C–N (398.7 eV), Fe–N (400.0 eV), and  $\text{N}^+\text{H}$  (401.2 eV)<sup>56</sup>. This suggests that IL-CDs and CDs interact with the steel surface via both chemical (C–N and Fe–N) interactions and physical ( $\text{N}^+\text{H}$ ) interactions, forming a homogenous protective layer that effectively mitigates corrosion in the aggressive  $\text{H}_2\text{SO}_4$  solution.

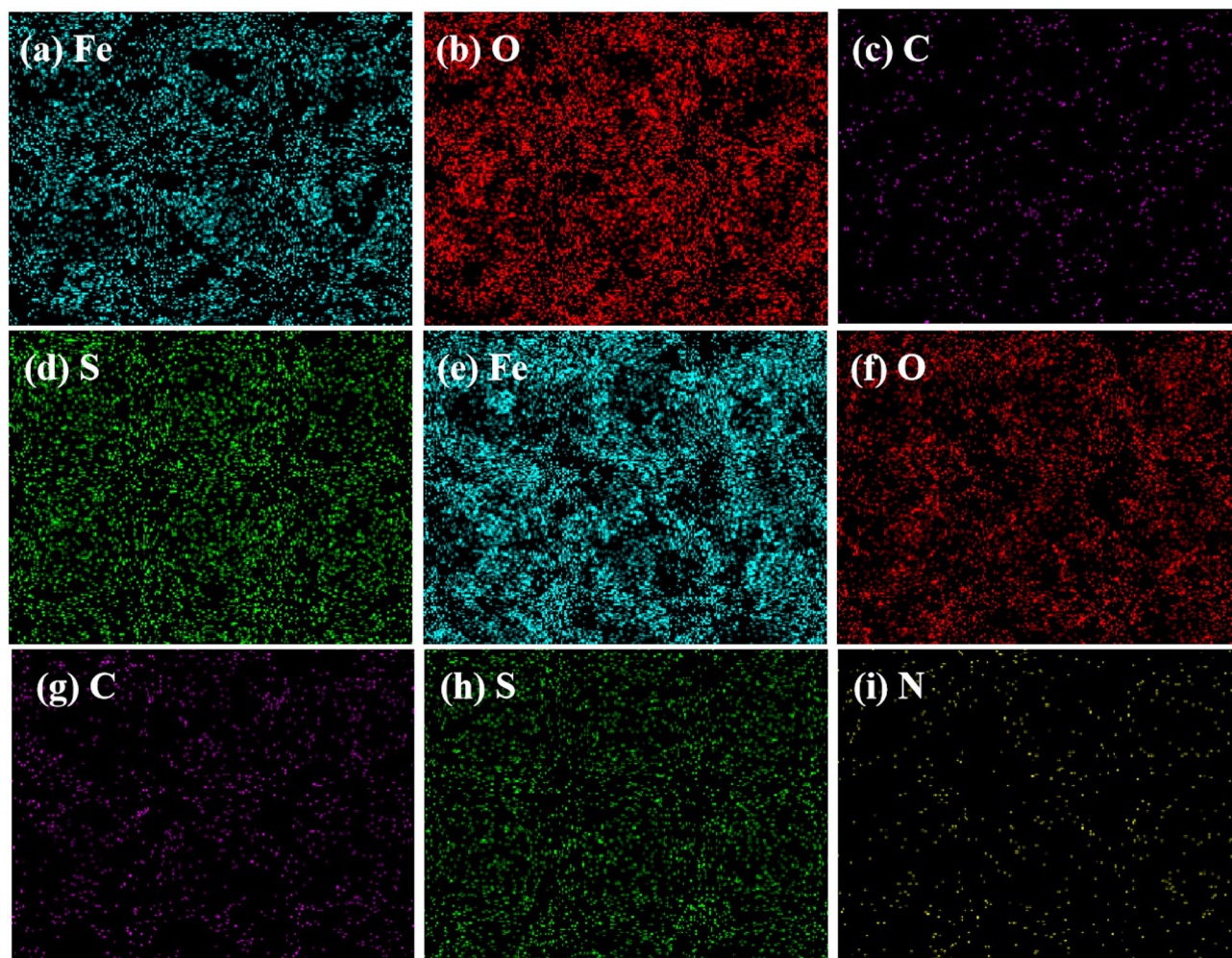
In Fig. 14c, sulfur (S) appears as Fe–S (168.3 eV) and  $\text{SO}_4^{2-}$  (169.3 eV)<sup>57</sup>. The O 1s spectrum in Fig. 14d shows three peaks at 529.6,  $531.7 \pm 1$ , and 532.8 eV, attributed to  $\text{O}^{2-}$ ,  $-\text{OH}$  of hydrous iron oxides (e.g.,  $\text{FeOOH}$ )<sup>41</sup>, and  $\text{O}=\text{C}=\text{O}$ , respectively<sup>42</sup>.

These results imply that IL-CDs adsorb onto the steel surface through chemical adsorption, complexing with Fe using lone pair electrons from N and S. Meanwhile, protonated nitrogen ( $\text{N}^+\text{H}$ ) interacts physically with the steel surface through electrostatic forces.

#### AFM analysis

The AFM images of steel surfaces, immersed in 0.5 M  $\text{H}_2\text{SO}_4$  solution with and without 150 mg/L IL-CDs and CDs for 6 h at 298 K, are illustrated in Fig. 15. After immersion in the acidic solution (Fig. 15a), the steel surface exhibits obvious depression and a high average roughness ( $R_a = 153$  nm). In contrast, the steel surfaces treated with IL-CDs and CDs (Fig. 15b,c) appear noticeably smoother, with average roughness values of 34.3, and 35.6 nm, respectively. These results demonstrate that both IL-CDs and CDs effectively mitigate steel corrosion, which is consistent with the electrochemical and SEM results.





**Fig. 13.** EDS mapping images of the steel surfaces after 6 h of immersion in 0.5 M  $\text{H}_2\text{SO}_4$  solution without IL-CDs (a) Fe; (b) O; (c) C; (d) S; and with IL-CDs: (e) Fe; (f) O; (g) C; (h) S; (i) N.

### Adsorption type analysis

In acidic media, the inhibition mechanisms of organic inhibitors are generally regarded as physical adsorption, chemisorption, or a combination of both, often accompanied by the displacement of water molecules from the metal surface. The Langmuir adsorption isotherm is commonly utilized to analyze the adsorption type, which is as follows<sup>58</sup>:

$$C_{\text{inh}}/\theta = 1/K_{\text{ads}} + C_{\text{inh}} \quad (6)$$

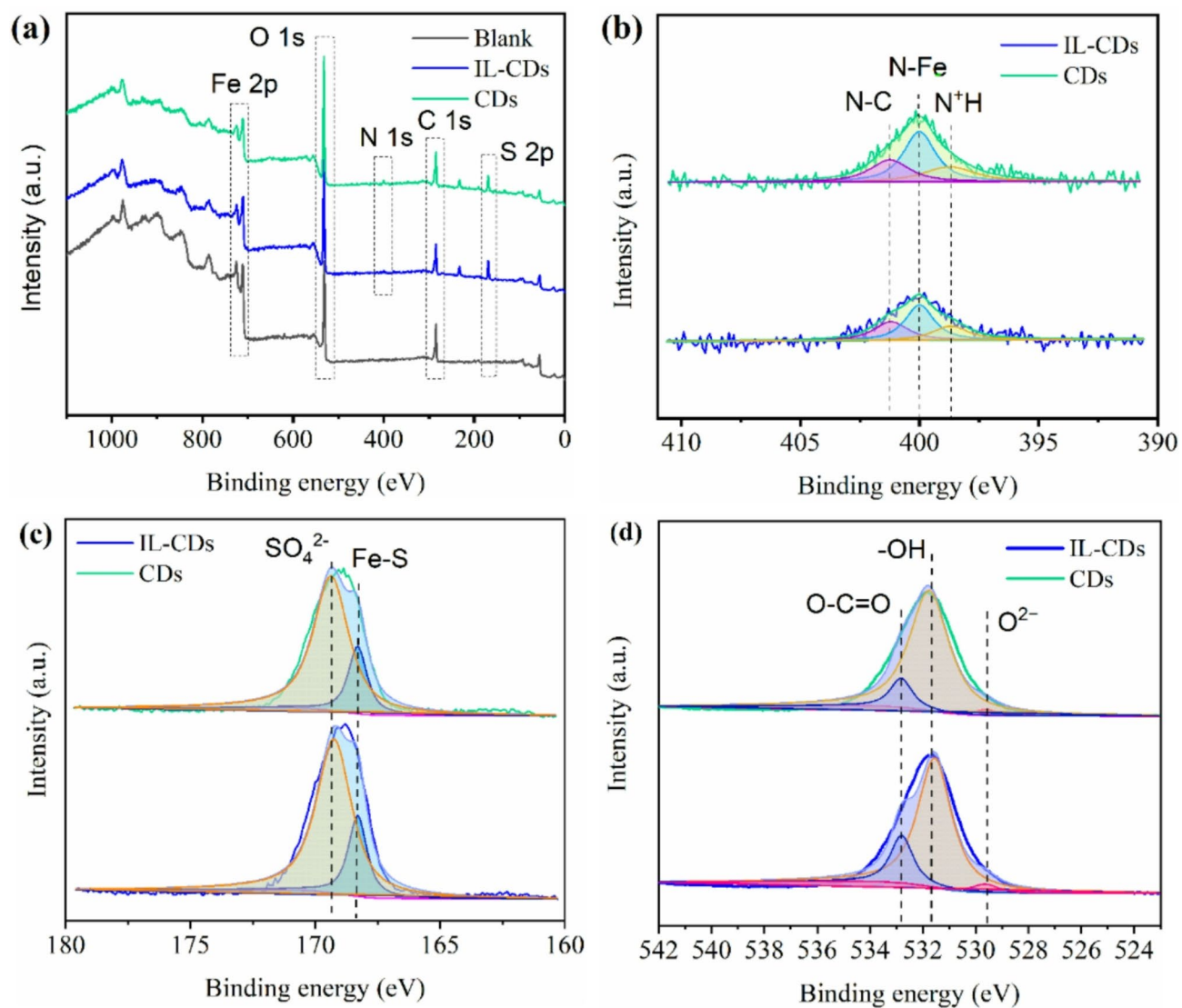
where  $C_{\text{inh}}$  is the inhibitor concentration,  $K_{\text{ads}}$  is the equilibrium constant, and  $\theta$  (defined as  $\theta = \text{IE}\%/100$ ) represents the surface coverage by the inhibitor molecules.

The relationship between  $C_{\text{inh}}$  and  $C_{\text{inh}}/\theta$  is displayed in Fig. 16. The slope of the Langmuir adsorption isotherm for IL-CDs is close to 1, with a linear correlation coefficient ( $R^2$ ) exceeding 0.93. The standard free energy ( $\Delta G_{\text{ads}}^0$ ) is obtained using the following equation<sup>58</sup>:

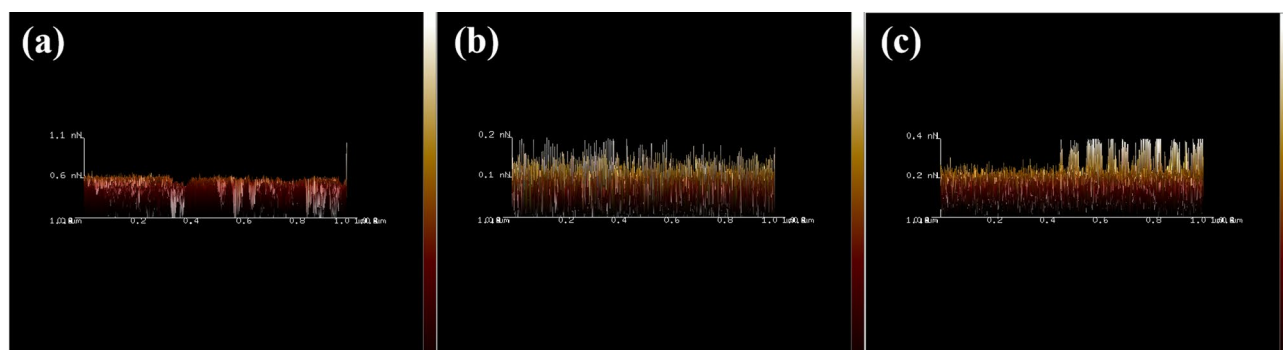
$$\Delta G_{\text{ads}}^0 = -RT \ln (C_{\text{H}_2\text{O}} K_{\text{ads}}) \quad (7)$$

where  $C_{\text{H}_2\text{O}}$  is 1000 g/L,  $R$  is the gas constant ( $8.314 \text{ J mol}^{-1} \text{ K}^{-1}$ ), and  $T$  is the temperature in Kelvin.

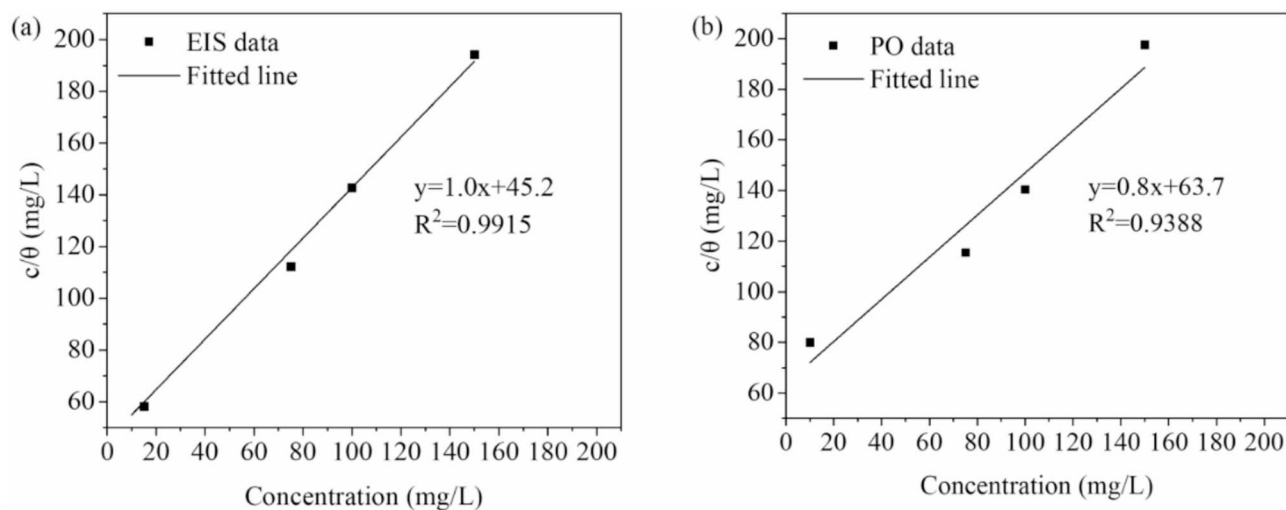
$K_{\text{ads}}$  value provides insight into the adsorption strength of an inhibitor. A higher  $K_{\text{ads}}$  value implies stronger adsorption on the metal surface<sup>59</sup>. If  $\Delta G_{\text{ads}}^0$  is negative, the adsorption process is spontaneous, vice versa. The  $\Delta G_{\text{ads}}^0$  values for IL-CDs, derived from EIS and PO results, are  $-24.8$  and  $-23.9 \text{ kJ mol}^{-1}$ , respectively, indicating that the adsorption is spontaneous and occurs at the steel/solution interface. Although the magnitude of  $\Delta G_{\text{ads}}^0$  alone may not definitively distinguish between physisorption and chemisorption<sup>28,60</sup>, it is generally accepted that if  $\Delta G_{\text{ads}}^0 > -20 \text{ kJ mol}^{-1}$ , the process is classified as physical adsorption, whereas values less than  $-40 \text{ kJ mol}^{-1}$  indicate chemical adsorption. Values between  $-40$  and  $-20 \text{ kJ mol}^{-1}$  suggests a mixed mechanism of physisorption and chemisorption<sup>58</sup>. Based on these criteria, the adsorption of IL-CDs onto the steel surface is



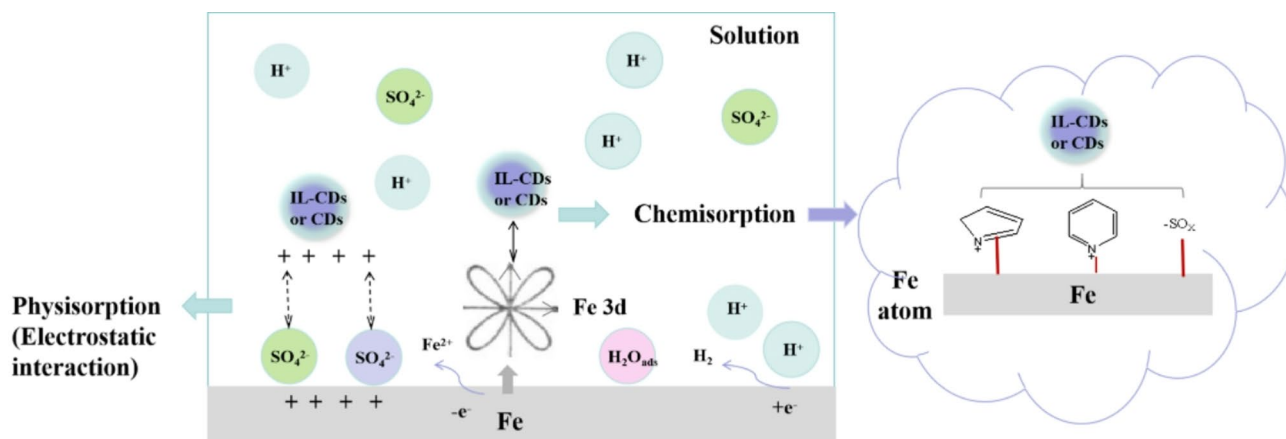
**Fig. 14.** XPS spectra (a) of carbon steel in the absence and presence of IL-CDs and CDs, and high-resolution X-ray photoelectron deconvoluted profiles of (b) N 1s, (c) S 2p, (d) O 1s for IL-CDs and CDs treated steel surface.



**Fig. 15.** 3D AFM images of the steel surface after immersion in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution in the absence and presence of IL-CDs and CDs for 6 h at 298 K: (a) without inhibitor; (b) with 150 mg/L IL-CDs; (c) with 150 mg/L CDs.



**Fig. 16.** Langmuir adsorption plots for carbon steel in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution uninhibited and inhibited by different concentrations of IL-CDs and CDs: using EIS data (a), using PO data (b).



**Fig. 17.** Proposed scheme for the adsorption and anti-corrosion mechanism of carbon dots on the Fe surface in H<sub>2</sub>SO<sub>4</sub> solution.

classified as physicochemical process at the steel/solution interface. The analysis of SEM/EDS and XPS supports the presence of both physical and chemisorption mechanisms.

### Inhibition mechanism

The inhibition mechanism of the as-prepared IL-CDs and CDs in H<sub>2</sub>SO<sub>4</sub> solution involves both physical and chemical adsorption of inhibitor molecules onto the steel surface (Fig. 17). On one hand, through electrostatic interactions, protonated IL-CDs and CDs physically adsorb onto the charged steel surface. On the other hand, pyridine-like N, pyrrole-like N, O, and S atoms in IL-CDs and CDs undergo chemical adsorption. Specifically, pyridine-N and pyrrole-like N coordinate with the unfilled 3d orbitals of Fe atoms on the steel surface, facilitating the formation of a protective layer that enhances corrosion resistance. In addition, the graphitic N atoms in IL-CDs and CDs contribute to protective shielding through electrostatic interactions. The O and S atoms, with their lone electron pairs, further enhance the adsorption, providing additional protection against the aggressive attacks of acid. Consequently, IL-CDs exhibit superior corrosion inhibition performance, attributed to their higher concentration of pyrrolic N compared to CDs.

### Conclusion

In this study, two types of biomass-derived N- and S- co-doped functionalized carbon dots (IL-CDs and CDs) were facily synthesized using dried dandelion as the precursor, with IL and H<sub>2</sub>O employed as solvents, respectively. These carbon dots were comprehensively characterized, and their inhibition performance on carbon steel in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution was systematically investigated. The mechanism of solvent-induced structure-property modulation in carbon dots is proposed. The main findings are summarized as follows:



1. Electrochemical results reveal that biomass-derived IL-CDs and CDs exhibit effective inhibition efficiency. IL-CDs provide better inhibition efficiency than CDs, which is closely associated with the higher concentration of pyrrolic-like N and the increased surface density of the C-SO<sub>x</sub> functional group in IL-CDs. The O and S atoms, with their lone electron pairs, facilitate adsorbing onto the steel surface, providing enhanced corrosion protection.
2. Surface morphological observations are consistent with electrochemical results, demonstrating that the pyrrolic-like, protonated N and sulfonate C-SO<sub>x</sub> group in IL-CDs and CDs promote adsorption onto the steel surface, forming a protective barrier against the corrosive attack of aggressive ions present in the solution.
3. The  $\Delta G_{\text{ads}}$  value indicates that the adsorption of IL-CDs onto the steel surface occurs spontaneously and follows the Langmuir adsorption isotherm model.

## Data availability

The data used to support the findings of this study are available from the corresponding authors upon request.

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## Author contributions

Shuyun Cao: Data curation, Formal analysis, Conceptualization, Writing-original draft. Funding acquisition. Yang Zhao: Resources, funding acquisition. Yubao Cao: Formal analysis. Yongwei Li: Formal analysis. Haodong Tan: Data curation, Investigation. Hong Wang: Conceptualization, Formal analysis, Supervision, funding acquisition.

## Declarations

## Competing interests

The authors declare no competing interests.



### Additional information

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