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Thickness configuration optimization of $B_4C/UHMWPE$ composite armor under varying impact velocities and areal densities through numerical and experimental study

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This study aims to optimize the ceramic-to-backing thickness ratio (R_{th}) of $B_4C/UHMWPE$ composite armor to enhance the anti-penetration performance while maintaining lightweight requirements. Its primary innovation lies in systematically quantifying, through combined finite element method (FEM) and ballistic testing, the coupling mechanism of thickness ratio (R_{th} : 0.4–2.0), areal density (AD: 25.0–30.0 kg/m²), and impact velocity (V_0 : 400.0–550.0 m/s) governing the anti-penetration performance of composite armor. The results reveal that the ballistic limit velocity (V_{bl}) initially increases and then decreases as R_{th} increases from 0.4 to 2.0, peaking at $R_{th} = 1.4$ –1.6 across all AD cases. Notably, this optimal R_{th} range remains consistent across AD variations, with both projectile mass loss ratio ($R^{II}_{m,l}$) and kinetic energy loss ratio ($R^{II}_{ke,l}$) during the first two penetration stages peaking within this range, demonstrating robust design applicability. Furthermore, a key finding and significant contribution is the dynamic shift in the optimal R_{th} for minimizing projectile residual velocity (V_{re}) when V_0 exceeds V_{bl} : Under fixed AD, higher V_0 reduces the optimal R_{th} due to shortened projectile-armor interaction time, necessitating thicker UHMWPE laminate to prevent premature ceramic fracture failure and enhance the backing-plate energy dissipation. Conversely, under constant V_0 , higher AD elevates the optimal R_{th} , where AD and V_0 show opposite effects on the variation of optimal R_{th} , and the optimal R_{th} converges to 1.4–1.6 as the highest V_{bl} corresponding to given AD approaches V_0 . Critically, this study establishes a quantitative framework for V_0 -AD- R_{th} coupling effects, providing actionable guidelines for designing lightweight composite armor against diverse ballistic threats.

Keywords Ceramic composite armor, Thickness ratio, Anti-penetration performance, Areal density, Impact velocity

With the continuous development of material technology and its extensive application in the military field, the conflict between armor-piercing ammunition and armor protection technology has become increasingly intense^{1,2}. Compared to the previous homogeneous armor and metal armor, ceramic composite armor tends to exhibit higher ballistic protection efficiency³. Modern military platforms, such as armored vehicles, helicopters, and personal protection equipment, demand lightweight composite armor to meet NATO STANAG 4569 Level 3 ballistic and blast protection while ensuring mobility, such armor systems typically exhibit an areal density (AD, armor mass per unit area) range of 35.0 to 120.0 kg/m²^{4,5}. Due to the exceptional protective performance of ceramic composite armor, scholars have systematically investigated its unique ballistic mechanism⁶, and explored the effects of ceramic panel layer⁷, and backing layer⁸, and projectile impact parameters, such as impact velocity⁹ and obliquity angle¹⁰, on the ballistic performance of composite armor. Typically, ceramic composite

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armor consists of a high-hardness ceramic panel and a high-toughness backplane, and its anti-penetration performance is influenced by many factors, including ceramic panel thickness, backplane thickness, constraint condition and so on¹¹. Boron carbide (B_4C) ceramic stands out among bulletproof ceramic materials due to its exceptional characteristics. It possesses the lowest density and highest hardness when compared to other ceramics such as alumina (Al_2O_3) and silicon carbide (SiC). Moreover, it holds significant research value and exhibits promising applications in the field of bulletproof inserts and lightweight composite armor⁷. Ultrahigh molecular weight polyethylene (UHMWPE) laminate, a type of fiber reinforced polymer (FRP) laminate, is fabricated by stacking and hot-pressing multiple layers of UHMWPE fiber monolayers. Compared to metal backplane, it exhibits superior specific modulus, specific strength, fracture toughness, and anti-penetration performance. Owing to these advantages, UHMWPE laminate is widely applied in aerospace, defense, and ocean engineering, emerging as a prominent research hotspot in the protective engineering field¹²⁻¹⁴. Hence, it is essential to investigate and evaluate the anti-penetration performance of composite armor comprising B_4C ceramic panel and UHMWPE laminate to enhance its protective performance and quality protective efficiency.

Experimental testing remains a cornerstone methodology used to investigate the anti-penetration performance of composite armor. This method can provide accurate and intuitive results, whereas its high cost and time-intensive nature limit application scope. In contrast, numerical method has the advantage of low cost and comprehensive data, researchers are increasingly using the numerical method to supplement and extend the ballistic testing results¹⁵⁻¹⁷. In the ballistic testing against 7.62 mm projectiles, Den Reijer¹⁸ observed that high-hardness ceramic panels effectively fragment and decelerate projectiles, while metal backplanes provide support to prevent premature ceramic panel failure, maximizing its effectiveness in fragmenting the projectiles. Separately, Cao et al.¹⁹ investigated the damage mechanisms of ceramic composite armor against 12.7 mm armor-piercing projectiles, and found that boundary constraint conditions, backplate structural configuration have significant influence on the ballistic performance. Yu et al.²⁰ conducted ballistic testing on ceramic/metal composite armors with different thicknesses configurations using 12.7 mm projectiles, the results indicate that increasing the backplane thickness enlarged both the half-cone angle and fracture dimension of ceramic cones, while thicker ceramic panels maintained a stable half-cone angle but reduced the overall crushing size. However, neither of the above studies have quantitatively investigated the influence of structural configuration on the ballistic performance of composite armor.

There have been numerous papers exploring the anti-penetration performance under various structural configurations. Hu et al.²¹ proposed a multilayer composite armor system (metal/ceramic/UHMWPE laminate) to investigate its ballistic performance against flat-nosed projectiles. Their studies revealed that the multilayer armor systems exhibit multiple failure modes. With the optimization of structural configuration, the projectile residual velocity (V_{re}) decreased, and the energy absorption effect of armor system improved significantly. Braga et al.²² introduced a multilayered armor system comprising alumina ceramic panel, curaua fabric-reinforced composite, and aluminum alloy backplane to investigate the effect of layer thickness/areal density on the anti-penetration performance. The results revealed that aluminum alloy layer exhibits higher efficiency rather than other layers in terms of trauma absorption, and the armor system with optimized structural configurations achieved a significant thickness reduction while maintaining ballistic performance. Wang et al.²³ proposed a layered ceramic composite armor ($B_4C/C/UHMWPE$), and conducted ballistic testing using 7.62 mm projectiles. Their numerical simulations revealed that the composite armor exhibits the optimal bulletproof performance at a B_4C panel thickness of 10 mm. The above three studies primarily focus on leveraging the synergistic effects of multi-layered composite armor materials to enhance energy dissipation mechanisms, while the effect of different material layer thickness ratios on ballistic performance is not mentioned.

Si et al.²⁴ found that when the thickness ratio between the front ceramic panel and metal backplane (R_{th}) is 1:2, the projectile remains at the interface for a longer residence time, resulting in a relatively high ballistic protection efficiency. Chang et al.²⁵ studied the anti-penetration performance of lightweight ceramic/aluminum alloy composite armor under various thickness configurations, and they found that the composite armor exhibits optimal bulletproof performance at an R_{th} value of 4:1 between ceramic panel and aluminum alloy. Li et al.²⁶ utilized the FEM to simulate the penetration process of 12.7 mm projectiles into composite armor. The numerical results revealed that bulletproof performance increased with increasing component thickness, whereas the bulletproof performance initially increases and then decreases as the R_{th} value rises. The above studies explored the anti-penetration performance of ceramic composite armor under various material and thickness configurations; however, there are still some issues that need to be further investigated. Specifically, few papers have simultaneously investigated the combined effects of R_{th} , AD and projectile impact velocity (V_0) on the anti-penetration performance of composite armor. Even with the same thickness configuration, the composite armor may exhibit different ballistic behavior. The ceramic composite armor demonstrates distinct damage characteristics at different V_0 and AD conditions, and the variations of V_0 and AD significantly alter penetration resistance. Therefore, it is necessary to investigate the influence of R_{th} on the anti-penetration performance of composite armor under different V_0 and AD conditions.

In this paper, a composite armor system comprising B_4C ceramic and UHMWPE laminate with varying thickness configurations is investigated. The ballistic behavior and anti-penetration performance of composite armor system are simulated and evaluated using numerical calculation methods, and ballistic testing is conducted on $B_4C/UHMWPE$ composite targets against 7.62 mm steel-core projectiles. Specifically, this study focuses on the influence of R_{th} on the anti-penetration performance and damage characteristics, and it also discusses and analyzes the influence of V_0 and AD on the optimal R_{th} of composite armor system. This research provides a certain basis for the future structural design and thickness optimization, contributing to enhancing the protective performance of lightweight ceramic composite armor and expanding its application prospects.

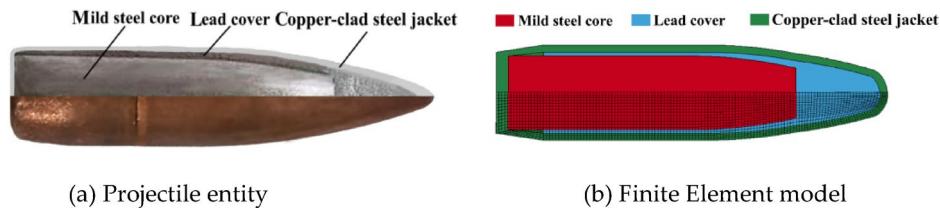


Fig. 1. Structure of 7.62 mm projectile.

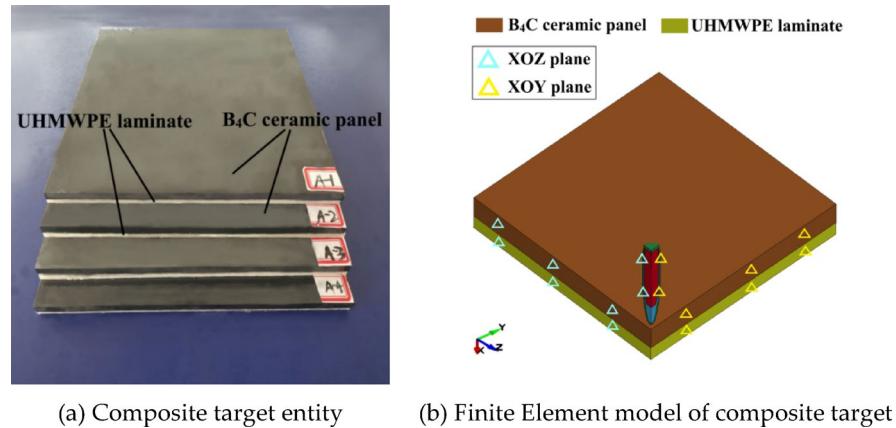


Fig. 2. $B_4C/UHMWPE$ ceramic composite target.

Ballistic testing and finite element modeling description

Projectile and target specification

In this study, the projectile is composed of mild steel core (B2F steel), lead cover and copper-clad steel jacket, as shown in Fig. 1 (a). The projectile has a length of 32.00 mm and a weight of 9.60 g, while the mild steel core measures 23.60 mm in length and weighs 4.75 g. To improve the mesh quality and computational efficiency of the finite element (FE) model, the sharp corners of projectile components are simplified, and the simplified FE model is shown in Fig. 1 (b).

As depicted in Fig. 2 (a), the $B_4C/UHMWPE$ composite armor has a lateral dimension of 300×300 mm. The B_4C ceramic panel is prepared through a hot-press sintering process and supplied by Dalian Jinma Boron Group Co., Ltd. The UHMWPE laminate is prepared in the laboratory using a fiber stacking method of $0/90^\circ/0/90^\circ$, followed by a hot-pressing process. The two components are bonded together using rubber-modified epoxy resin. The densities of B_4C ceramic and UHMWPE laminate are 2.5×10^3 kg/m 3 and 0.98×10^3 kg/m 3 , respectively. The AD value of composite armor can be calculated by summing the products of the densities and thicknesses for different armor components. In the ballistic testing, the AD of $B_4C/UHMWPE$ composite targets is 25.0 kg/m 2 , and the FE model of composite target is shown in Fig. 2 (b).

Ballistic testing method

In this study, the ballistic testing is conducted to evaluate the anti-penetration performance of $B_4C/UHMWPE$ composite target. The ballistic testing system consists of a ballistic gun, aluminum foil targets, a time interval tester, a target holder, a high-speed camera (sample rate of 3200 fps), a scale and other components. The layout diagram of testing site is shown in Fig. 3, and some images of testing facilities are presented in Fig. 4.

Specifically, 7.62×54 mm steel-core projectiles are fired from a fixed ballistic gun, with different impact velocities obtained by adjusting the propellant charge of the cartridge. A combination of aluminum foil targets, placed directly in front of the testing targets, and a time interval tester is employed to measure the projectile impact velocity. The distance between the aluminum foil targets and the ceramic composite target is approximately 1.5 m. The velocity measured by the above measuring facilities can be considered as the V_0 value. A high-speed camera are applied to record the penetration process. Meanwhile, when the testing facility positions are fixed, each pixel in the image captured by the high-speed camera represents an identical distance. The high-speed camera image and the scale are used to determine the V_{re} value in cases where the composite targets are completely penetrated.

The FE model comprises three main components: projectile, B_4C ceramic panel and UHMWPE laminate. Projectiles with initial velocity V_0 vertically impact the $B_4C/UHMWPE$ composite armour system. In addition, the structural parameters of both projectiles and composite armors are consistent with those obtained from ballistic testing.

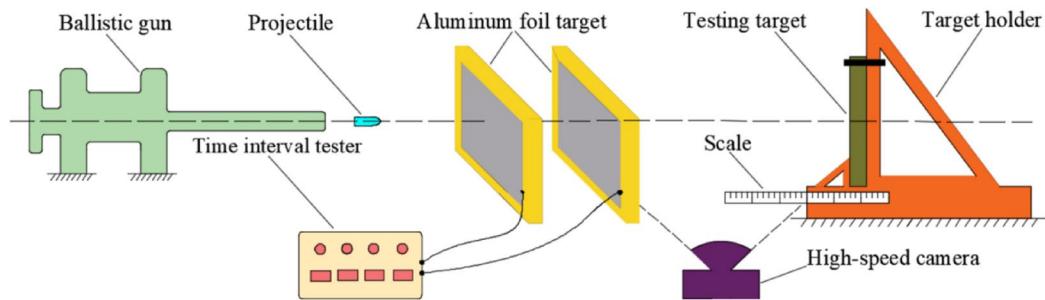


Fig. 3. Schematic diagram of the testing site layout.

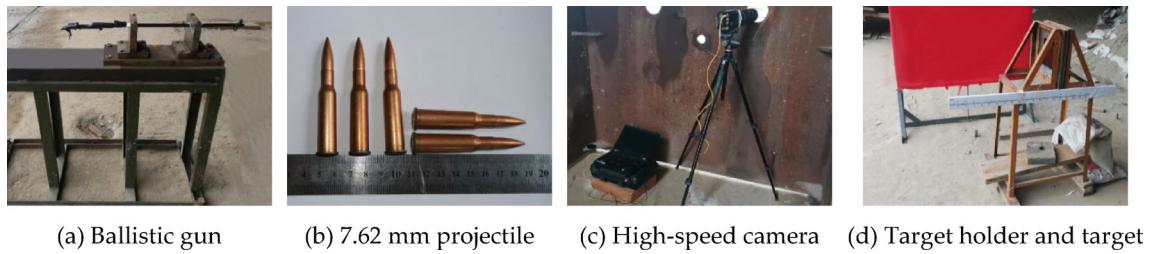


Fig. 4. Images of the ballistic testing facilities.

ρ (kg/m ³)	G (GPa)	A (MPa)	B (MPa)	N	C	M	$\dot{\varepsilon}_0$ (s ⁻¹)
7.80×10^3	75.2	293.8	230.2	0.578	0.065	0.706	2.1×10^{-3}
C_p (J/kg·K)	T_m (K)	T_r (K)	D_1	D_2	D_3	D_4	D_5
469	1795	293	0.472	18.728	-7.805	0.0193	13.017

Table 1. Material parameters for mild steel core^{29,30}.

Material constitutive model

Among the available material constitutive models, Johnson-Cook (J-C) model is particularly suitable for materials that undergo significant variations in strain rates and experience material softening due to adiabatic temperature increases caused by plastic heating^{27,28}. In the present FE model, the J-C model is utilized to describe the ballistic response behavior of three projectile components. The J-C strength model is generally represented by:

$$\sigma_y = \left[A + B \left(\bar{\varepsilon}_p \right)^N \right] \left[1 + C \ln \left(\dot{\varepsilon}^* \right) \right] \left[1 - \left(T^* \right)^M \right] \quad (1)$$

The J-C damage model is represented by:

$$D = \sum \left(\Delta \bar{\varepsilon}_p / \varepsilon^f \right) \quad (2)$$

where D is the accumulated damage factor, and fracture occurs when D reaches 1. $\Delta \bar{\varepsilon}_p$ is the incremental equivalent plastic strain during each integration cycle, while ε^f refers to the equivalent fracture strain, which is defined as:

$$\varepsilon^f = [D_1 + D_2 \exp(D_3 \sigma^*)] [1 + D_4 \ln \dot{\varepsilon}^*] [1 + D_5 T^*] \quad (3)$$

where D_1 , D_2 , D_3 , D_4 and D_5 represent the failure parameters, and the J-C model is comprehensively described in Ref²⁷. Based on experimental quasi-static and dynamic compressive/tensile mechanical properties, some scholars have fitted to obtain the J-C model parameters for the above projectile components, as presented in Tables 1 and 2²⁹⁻³².

$$\sigma^* = \sigma_i^* - D_0 (\sigma_i^* - \sigma_f^*) \quad (4)$$

Material	ρ (kg/m ³)	G (GPa)	A (MPa)	B (MPa)	N	C
Lead cover	11.34×10^3	7.0	14.0	17.6	0.685	0.035
Material	M	$\dot{\varepsilon}_0$ (s ⁻¹)	C_p (J/kg·K)	T_m (K)	T_c (K)	D_I
Lead cover	1.68	1.0	455	873	293	0.5
Copper-clad steel	1.09	1.0	169	1811	293	0.5

Table 2. Material parameters for lead cover³¹ and copper-clad steel jacket³². Based on the continuum mechanics theory and damage theory, The Johnson-Holmquist (JH-2) model relates the damage behavior of material to the reduction of strength parameters and describes the evolution of material damage by introducing damage parameters. The JH-2 model is suitable for characterizing the dynamic response behavior of brittle materials, such as ceramics and glass, under large strains, high strain rates and high pressures³³. In this study, the JH-2 model is applied to describe the dynamic response behavior of B₄C ceramic panel, and the normalized equivalent stress is expressed as:

ρ (kg/m ³)	G (GPa)	A ₀	B ₀	C ₀	N ₀	M ₀	$\dot{\varepsilon}_0$ (s ⁻¹)	K ₁ (GPa)
2.50×10^3	197	0.97	0.73	0.005	0.67	0.85	1.0	233
K_2 (GPa)	K_3 (GPa)	d_1	d_2	$\sigma_{f,max}^*$	t (GPa)	p_{HEL} (GPa)	p_{HEL} (GPa)	β
-593	2800	0.001	0.5	0.50	0.26	19.0	8.71	1.0

Table 3. Material parameters for B₄C ceramic panel³⁴.

where σ_i^* and σ_f^* represent the intact and damaged behaviors, respectively. D_0 refers to the accumulated damage factor ($0 \leq D_0 \leq 1$). The above strength parameters are given by:

$$\sigma_i^* = A_0(p^* + t^*)^{N_0} (1 + C_0 \ln \dot{\varepsilon}^*) \quad (5)$$

$$\sigma_f^* = B_0(p^*)^{M_0} (1 + C_0 \ln \dot{\varepsilon}^*) \quad (6)$$

where A_0 , B_0 , N_0 , C_0 and M_0 are the input strength parameters. p^* and t^* are the hydrostatic pressure and maximum tensile pressure strength normalized by the Hugoniot elastic limit pressure component (p_{HEL}):

$$p^* = p/p_{HEL} \quad (7)$$

$$t^* = t/p_{HEL} \quad (8)$$

In the above equation, the plastic strain to fracture is given by:

$$\varepsilon_f^p = d_1(p^* + t^*)^{d2} \quad (9)$$

where d_1 and d_2 are the fracture plastic strain parameters.

The hydrostatic pressure is expressed as:

$$p = K_1\mu + K_2\mu^2 + K_3\mu^3 + \Delta p \quad (10)$$

where K_1 , K_2 and K_3 are input pressure coefficients. μ refers to the compressibility factor ($\mu = \rho/\rho_0 - 1$). The main parameters of B₄C ceramic panel are listed in Table 3³⁴.

The composite damage model (MAT_COMPOSITE_FAILURE_SOLID_MODEL) is applied to describe the dynamic response behavior of UHMWPE laminate under the impact of projectiles. This material model includes the orthotropic elastoplastic strength model and failure criterion, and the failure criterion includes: tensile and compression failure of the fiber, tensile and compression failure of the matrix. A detailed description of the above model is in Ref³⁵. In order to prevent some instabilities, additional failure criteria are applied to delete failure elements with excessive plastic strain. Table 4 provides the main parameters of UHMWPE laminate³⁵–³⁶.

Finite element algorithm description

This subsection focuses on the mesh topology, contact formulation, and boundary conditions of the FE model. Lagrangian formulation is applied to discretize projectiles and composite targets, necessitating the definition of contacts between different components within this FE model. Specifically, all material domains are discretized using eight-node hexahedral elements (SOLID 164). The surface-to-surface erosion contact algorithm is utilized to model the interaction between projectiles and targets. Secondly, the automatic contact algorithm is utilized to capture the interfacial interaction among different projectile components. Finally, the tie contact with failure

ρ (kg/m ³)	E_a (GPa)	E_b (GPa)	E_c (GPa)	v_{ba}	v_{ca}	v_{cb}	G_{ab} (MPa)	G_{bc} (MPa)	G_{ca} (MPa)
0.98×10^3	34.257	34.257	5.1	0.0	0.013	0.013	173.8	547.8	547.8
S_{ba} (MPa)	S_{ca} (MPa)	S_{cb} (MPa)	X_C (MPa)	Y_C (MPa)	Z_C (GPa)	X_T (GPa)	Y_T (GPa)	Z_T (MPa)	
1.8	1.8	1.8	800	800	1.74	1.25	1.25	60	

Table 4. Material parameters for UHMWPE laminate^{35,36}.

Mesh size (mm)	V_0 (m/s)	V_{re} (m/s)	Computational time (min)
0.125	432.0	139.0	3750.0
0.25		138.5	276.0
0.50		104.0	85.0
1.0		82.0	3.5

Table 5. Simulation results and computational time for different mesh sizesQ.

criteria is utilized to model the interaction at the initial contact interface of different target components, and contact failure will occur once the following failure criteria are met:

$$p(\sigma_n/NFLS)^2 + (\sigma_s/SFLS)^2 \geq 1 \quad (11)$$

where σ_n is the current normal stress, and σ_s refers to the current shear stress. $NFLS$ is the normal failure stress, and $SFLS$ refers to the shear failure stress. According to the previous research, when using the rubber-modified epoxy resin for interfacial bonding, the values of $NFLS$ and $SFLS$ are set to 43.0 MPa and 25.0 MPa, respectively³⁷.

The boundary conditions in this ballistic testing system and the FE model are consistent. The ceramic composite target in the ballistic testing system is fixed to the target holder, which means that the freedom of ceramic composite target edges is constrained in all directions. A quarter FE model is generated to numerically describe the penetration process to enhance the computational efficiency. This model is symmetric about the XOY plane and the XOZ plane, as shown in Fig. 2 (b), and thus all nodes in the XOY and XOZ symmetry planes are imposed with translation and rotation constraints along the Z-axis and the Y-axis, respectively.

Mesh size sensitivity analysis

The damage behavior of materials is primarily achieved by removing elements that satisfy the failure criteria. However, it should be noted that this approach limits the ability to observe post-failure mechanisms of ceramic and fiber composites. Additionally, inappropriate mesh sizes cannot accurately represent the deformation and failure behavior of the material. The mesh sizes of both projectiles and targets affect the accuracy of the numerical simulation results. Therefore, it is necessary to conduct a convergence analysis for the penetration process.

In this FE model, projectiles with the same V_0 are used to vertically impact composite targets with the same structure. Specifically, the V_0 value is set to 432.0 m/s, and the thickness configuration of $B_4C/UHMWPE$ ceramic composite targets is 8.0 mm + 5.0 mm. To maintain the accuracy and stability of simulation calculation, it is critical to ensure similar mesh sizes at the projectile/target contact interface. Ballistic testing results from Ref^{38,39} indicate that ceramic panel and projectile fragment sizes are predominantly in the range of 0.10–1.0 mm. According to the verification and validation methodology prescribed in Ref⁴⁰, convergence verification was performed using four geometrically refined meshes with a constant refinement ratio of 2.0: 1.0, 0.50, 0.25, and 0.125 mm.

Table 5 details the values of V_0 , V_{re} and computational time for different mesh sizes. The simulation results show that the V_{re} values differ significantly when mesh sizes are 0.25 mm, 0.5 mm and 1.0 mm. In contrast, the V_{re} values are similar for mesh sizes of 0.125 mm and 0.25 mm. The results indicate that the simulation results have limited dependency when the mesh size is below 0.25 mm, and the computation time increases dramatically at this level. To balance the simulation accuracy and computational efficiency, the impact center region is meshed using a 0.25 mm size, while a 1.0 mm size is used for regions farther away from the impact center region. The FE model after meshing is shown in Fig. 5.

Validation of finite element method (FEM)

Ballistic testing results

The $B_4C/UHMWPE$ composite targets utilized in this ballistic testing have an AD of 25.0 kg/m². These targets have two different thickness configurations. “Target A” comprises an 8.0 mm B_4C ceramic panel and a 5.0 mm UHMWPE laminate, while “Target B” comprises a 6.0 mm B_4C ceramic panel and a 10.0 mm UHMWPE laminate. The high-speed camera records several moments of the projectile penetrating Target A at 432.0 m/s, as shown in Fig. 6. The image obtained from the high-speed camera clearly shows that the target is impacted vertically by the projectile. The ballistic testing results for Target A and Target B are presented in Table 6, mainly including the penetration results, projectile impact velocity (V_0), and residual velocity (V_{re}).

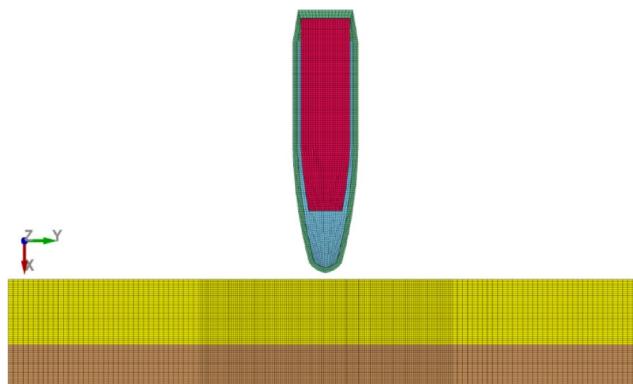


Fig. 5. Finite Element model after meshing.

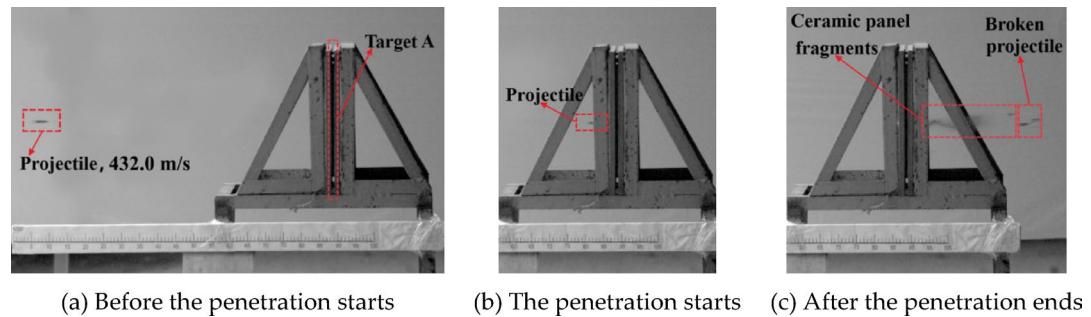


Fig. 6. Images of the projectile penetrating Target A.

Thickness configuration	V_0 (m/s)	Penetration results		V_{re} (m/s)		
		Ballistic testing	Simulation	Ballistic testing	Simulation	Error (%)
Target A (8.0 mm B_4C + 5.0 mm UHMWPE)	344.0	PP	PP	-	-	-
	400.0	PP	PP	-	-	-
	408.0	PP	PP	-	-	-
	432.0	CP	CP	144.0	138.5	-3.8
Target B (6.0 mm B_4C + 10.0 mm UHMWPE)	312.0	PP	PP	-	-	-
	380.0	PP	PP	-	-	-
	384.0	PP	PP	-	-	-
	400.0	CP	CP	92.0	95.0	3.3

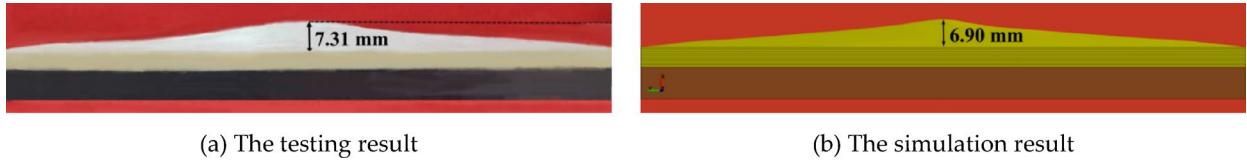
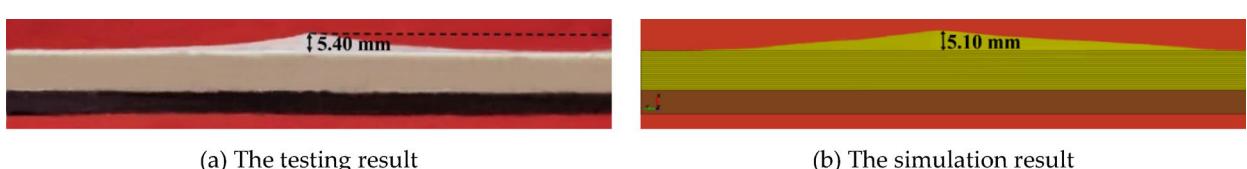
Table 6. Comparison of ballistic testing and numerical simulation results. Note: partial penetration (PP); complete penetration (CP).

FEM accuracy analysis

In this study, extensive numerical calculations are performed to evaluate the ballistic performance of B_4C /UHMWPE composite targets, it is necessary to verify the validity of numerical calculation method firstly. The FE model needs to maintain the same initial conditions as the ballistic testing. Specifically, the projectiles in the FE model have the same V_0 as those used in the ballistic testing, and the composite targets in the FE model have the same thickness configurations as Target A and Target B in the ballistic testing (described in Sect. 3.1). The penetration results of numerical calculations are also presented in Table 6. For both Target A and Target B, simulated penetration results and V_{re} values are in good agreement with ballistic testing results. Moreover, the V_{re} errors between ballistic testing and numerical calculations are within 3.8% when the projectiles completely penetrate ceramic composite targets.

When the projectiles fails to completely penetrate the composite armor, the bulge height of UHMWPE laminate increases gradually as the V_0 rises. Fragments of broken projectile and ceramic panel are not fully recycled. Both Target A and Target B are not completely penetrated when the two targets are struck by projectiles with impact velocities of 408.0 m/s and 384.0 m/s, respectively. The accuracy of numerical method can be verified by comparing the bulge height error in the ballistic testing and numerical simulations at the same V_0 . Table 7

Thickness configuration	V_0 (m/s)	Bulge heights of backplane (mm)		
		Ballistic testing	Simulation	Error (%)
Target A	408.0	7.31	6.90	-5.60
Target B	384.0	5.40	5.10	-5.55

Table 7. Bulge heights of UHMWPE laminates for both target A and target B.**Fig. 7.** Comparison of backplane bulge height between testing and simulation for Target A.**Fig. 8.** Comparison of backplane bulge height between testing and simulation for Target B.

presents the bulge heights of UHMWPE laminates for both Target A and Target B. In Fig. 7, the bulge heights of UHMWPE laminates for Target A are 7.31 mm and 6.90 mm in ballistic testing and simulation, respectively. For Target B, the bulge heights in ballistic testing and simulation are 5.40 mm and 5.10 mm, respectively, as shown in Fig. 8. In general, the discrepancy in bulge heights between ballistic testing and simulation for both targets does not exceed 5.60%. Overall, Target A exhibits similar anti-penetration performance in ballistic testing and numerical simulation, as does Target B.

The FEM framework used in this study, while validated for the global ballistic parameters, has inherent limitations in simulating post-failure behavior. The JH-2 model (for B_4C ceramic) and the composite damage model (for UHMWPE laminate) use element deletion to simulate material failure, preventing numerical instability but limiting the modeling of post-fragmentation effects such as secondary fragment interactions, debris friction, and the fine-scale flow of pulverized ceramic. Additionally, constitutive models based on the continuum assumptions approximate material responses phenomenologically and cannot resolve micro-fracture mechanisms. Although the above mesh sensitivity analysis confirmed the convergence of global ballistic parameters, local damage morphology remains influenced by element size. Despite these limitations, they primarily affect detailed failure visualization rather than core conclusions. Experimental validation confirms that the FEM can reliably predict the optimal ceramic-to-backing thickness ratio (R_{th}) and anti-penetration performance under varying V_0 and AD conditions. Here, the ballistic limit velocity (V_{bl}) is generally applied to quantitatively evaluate the anti-penetration performance of a protective structure against a given threat.

Results and discussion

The aforementioned numerical simulation method is utilized to investigate the combined effect of R_{th} (between B_4C ceramic panel and UHMWPE laminate), V_0 and AD values on the anti-penetration performance.

Influence of R_{th} on the V_{bl}

Existing research demonstrates that higher areal density (AD) can enhance the anti-penetration performance of composite armor, but increases weight and cost. Optimizing the thickness configuration without altering materials or AD can also further improve its anti-penetration performance^{10,41}. The AD range of 25.0–30.0 kg/m² based on standard requirements and practical limits of lightweight armor design^{4,5,10}, and this R_{th} range of 0.4–2.0 balances comprehensiveness with realistic design limits and provides a robust parameter space for our investigation^{24,25}. The FEM was employed to calculate the V_{bl} of B_4C /UHMWPE composite targets with various R_{th} values, where V_{bl} is defined as the minimum velocity required for a projectile to completely penetrate a target plate with a 50% probability. This paper focuses on composite armor with moderate-to-low AD values. The AD range is selected as 25.0–30.0 kg/m² for systematic investigation, and the R_{th} ranges from 0.4 to 2.0 with an interval of 0.2.

Figure 9 illustrates the relationship between V_{bl} and R_{th} for B_4C /UHMWPE composite targets. The V_{bl} exhibits a non-monotonic dependence on R_{th} values. For an AD of 25.0 kg/m², V_{bl} increases gradually as R_{th} ranges from 0.4 to 1.4, peaks at $R_{th} = 1.4$ –1.6, and then decreases as R_{th} further rises from 1.6 to 2.0. To further

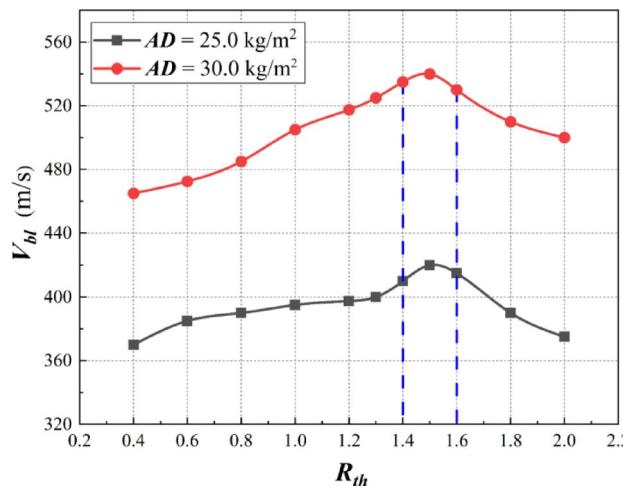


Fig. 9. The variation curves of V_{bl} vs. R_{th} ($AD = 25.0$ and 30.0 kg/m^2).

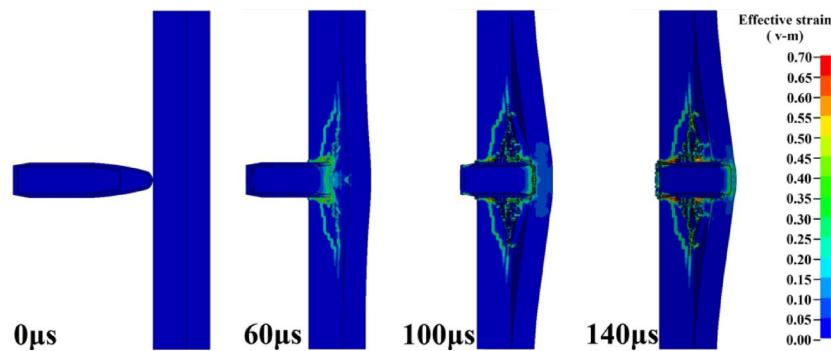


Fig. 10. Effective strain of projectile and target at several moments ($AD = 25.0 \text{ kg/m}^2$, $R_{th} = 1.5$).

determine the optimal R_{th} corresponding to the highest V_{bl} , additional calculations at $R_{th} = 1.3$ and 1.5 reveal that the highest V_{bl} is 420.0 m/s at $R_{th} = 1.5$. Likewise, for the AD of 30.0 kg/m^2 , V_{bl} increases gradually as the R_{th} ranges from 0.4 to 1.5 , and followed by a decrease as the R_{th} further ranges from 1.5 to 2.0 . Overall, considering the two aforementioned AD s, the optimal R_{th} range for achieving the highest V_{bl} is 1.4 – 1.6 . Compared to the results of previous studies, Si et al.²⁴ found that a SiC ceramic/4340 steel composite armor achieved the highest V_{bl} when the R_{th} was 2.0 (with a total thickness of 30.0 mm). Lu et al.⁴¹ reported that an Al_2O_3 ceramic/UHMWPE laminate exhibited optimal energy dissipation performance at an R_{th} of 1.25 , and that the thickness of front layer significantly influences the optimal R_{th} . Different ceramic/backing combinations require distinct optimal configurations. This study investigates B_4C panel (hardest monolithic ceramic) paired with UHMWPE laminate (highest specific strength) based on ballistic mechanisms, addressing prior limitations in AD control. Our findings reveal an optimal $\text{B}_4\text{C}/\text{UHMWPE}$ R_{th} of 1.4 – 1.6 at medium-to-low AD ranges (25.0 – 30.0 kg/m^2), demonstrating superior application potential.

The penetration process and damage characteristics are investigated using a $\text{B}_4\text{C}/\text{UHMWPE}$ composite target with ($AD = 25.0 \text{ kg/m}^2$, $R_{th} = 1.5$), and the V_0 is 420.0 m/s , which corresponds to the V_{bl} value. Figure 10 shows the effective strain distribution of the projectile and target at several critical moments during the penetration process (0– $140 \mu\text{s}$). In addition, Fig. 11 quantifies the temporal evolution curve of the projectile residual mass ratio (R_m = residual mass / initial mass) and projectile residual kinetic energy ratio (R_{ke} = residual kinetic energy / initial kinetic energy). The penetration process into composite targets is briefly divided into three stages: 0 – $60 \mu\text{s}$, 60 – $100 \mu\text{s}$, and 100 – $140 \mu\text{s}$. In the first stage (Stage I, 0 – $60 \mu\text{s}$), projectile impacts ceramic panel, and the resulting compressive stress waves are propagated toward the projectile and ceramic panel, respectively. The projectile begins to fracture and erode, and the ceramic panel backside develops axial and radial cracks under the action of reflected tensile wave, preliminary forming a fractured ceramic cone, while the UHMWPE backplane exhibits only minor deformation. After Stage I, the R_m and R_{ke} values are measured at 61.3% and 27.6% , respectively, along with measurable losses of projectile mass and velocity. During the second stage (Stage II, 60 – $100 \mu\text{s}$), the projectile continues penetrating the ceramic panel until the ceramic cone zone was completely perforated. Meanwhile, the UHMWPE laminate undergoes substantial flexural and tensile deformation induced by both fragmented projectiles and ceramic fragments. Until $100 \mu\text{s}$, the R_m and R_{ke} values remain only 49.0% and 6.5% . In the final stage (Stage III, 100 – $140 \mu\text{s}$), the fragmented projectile and ceramic fragments continue to penetrate

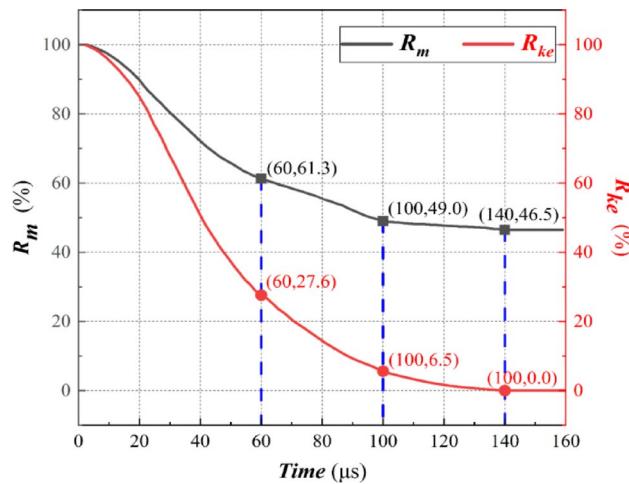


Fig. 11. The variation curves of R_m and R_{ke} vs. time ($AD = 25.0 \text{ kg/m}^2$, $R_{th} = 1.5$).

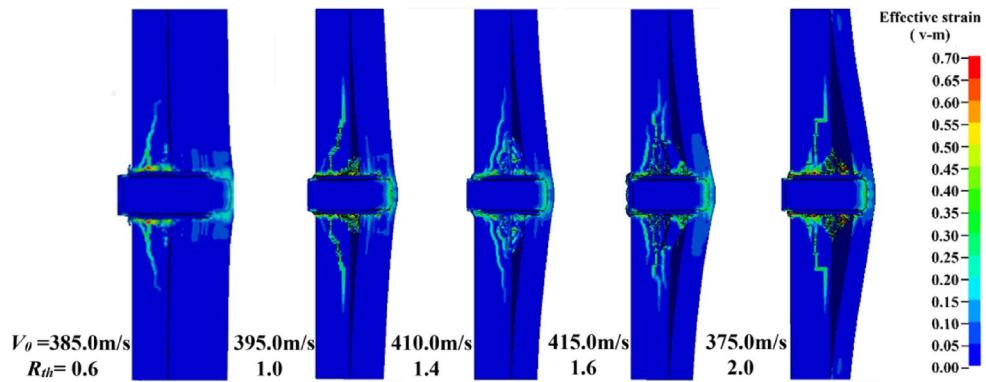


Fig. 12. Final effective strain of projectiles and targets for several R_{th} values ($AD = 25.0 \text{ kg/m}^2$).

UHMWPE laminate until the R_{ke} is reduced to zero. Notably, the projectile exhibits a mass loss ratio of 2.5% and a kinetic energy loss ratio of 6.5%, much lower than the 51.0% and 93.5% ratios in Stages I-II. During the penetration process, the ceramic panel predominantly dissipates projectile kinetic through fragmentation and erosive wear mechanisms in Stages I-II, effectively reducing the projectile mass and velocity, thereby establishing a dual energy-attenuation pathway. Meanwhile, the UHMWPE laminate serves as a support structure to protect the ceramic panel in Stages I-II, and transforms to absorb the projectile residual kinetic energy in Stage III.

For composite targets with R_{th} values of 0.6, 1.0, 1.4, 1.6, and 2.0, the V_{bl} values are 385.0, 395.0, 410.0, 415.0, 375.0 m/s, respectively. Each composite target was subjected to projectile impact at its corresponding V_{bl} value. The effective strain of projectiles and composite targets after the penetration was simulated for each R_{th} condition, as shown in Fig. 12. The corresponding curves in Fig. 13 depict the projectile mass loss ratio during the Stages I-II ($R_{m,l}^{II}$), the projectile mass loss ratio during the third penetration ($R_{m,l}^{III}$), and the projectile kinetic energy loss ratio during the Stages I-II ($R_{ke,l}^{II}$). The projectile-target interaction time refers to the duration of contact between projectile and target components, during which the projectile kinetic energy is converted into stress waves, thermal energy and so on, culminating in fracture failure of ceramic panel.

The $R_{m,l}^{II}$ and $R_{ke,l}^{II}$ values increase gradually as R_{th} ranges from 0.6 to 1.4, peak when R_{th} reaches 1.4–1.6, and then decrease gradually. R_{th} continues to rise from 1.6 to 2.0. When R_{th} is below 1.4, increasing the ceramic panel thickness prolongs projectile-target interaction time, which enhances the crushing and erosive effect on the projectile. For the case of $V_0 < V_{bl}$, the projectile remains in interaction with ceramic panel for a relatively longer duration, allowing cracks to expand in a quasi-static manner. This process generates characteristic conical crack patterns while dissipating most of the projectile's kinetic energy through crack formation and frictional sliding along fracture surfaces, demonstrating the ceramic layer's primary role in energy absorption. As a result, the final residual projectile length progressively decreases. Additionally, UHMWPE laminate within the corresponding thickness range effectively supports the ceramic panel while absorbing the projectile residual kinetic energy. When R_{th} is within the range of 1.4–1.6, the ceramic cone is fully developed, with both axial and radial cracks propagating extensively throughout the ceramic cone region. This crack propagation mechanism maximized the dissipation of projectile kinetic energy during the penetration process. Notably, the formation process of ceramic cone constitutes the primary energy dissipation stage throughout the penetration process. As

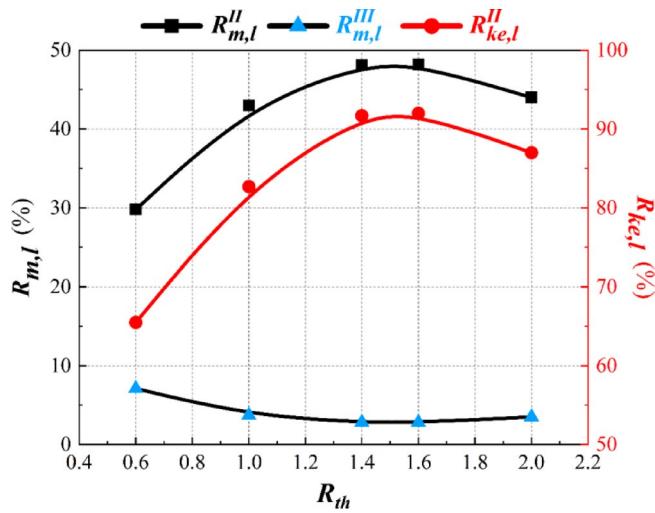


Fig. 13. The variation curves of $R_{m,l}^{\text{II}}$ and $R_{ke,l}^{\text{II}}$ vs. R_{th} ($AD = 25.0 \text{ kg/m}^2$).

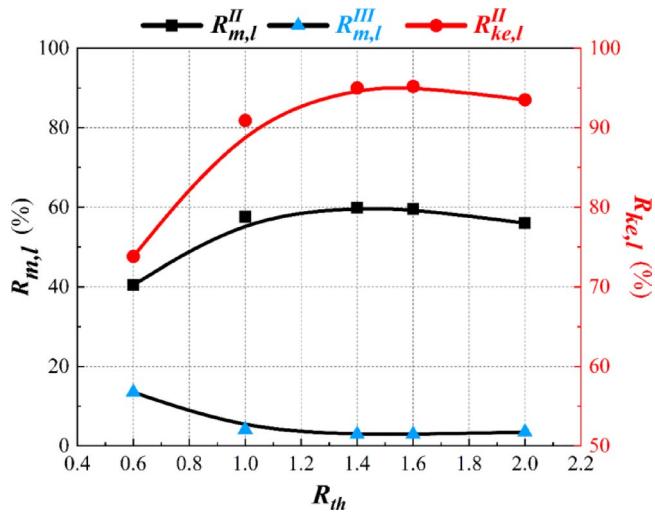


Fig. 14. The variation curves of $R_{m,l}^{\text{II}}$ and $R_{ke,l}^{\text{II}}$ vs. R_{th} ($AD = 30.0 \text{ kg/m}^2$).

R_{th} continues to rise from 1.6 to 2.0, the UHMWPE laminate progressively becomes insufficient to support and protect the ceramic panel, and crack propagation in the ceramic cone region is limited, resulting in the decline of $R_{m,l}^{\text{II}}$ and $R_{ke,l}^{\text{II}}$ values. Figure 13 shows that the projectile mass loss ratio ($R_{m,l}^{\text{III}}$) remains largely unchanged with increased UHMWPE laminate thickness, but decreases with reduced ceramic panel thickness. Overall, the projectile final mass loss ratio ($R_{m,l}^{\text{F}} = R_{m,l}^{\text{II}} + R_{m,l}^{\text{III}}$) first increased and then decreased as R_{th} ranges from 0.6 to 2.0. A similar rise-fall trend is observed for the projectile length loss ratio, as shown in Fig. 12. Therefore, within the optimal R_{th} range (1.4–1.6), adequate projectile-target interaction time facilitates controlled crack propagation in the ceramic cone region, thereby enhancing the $R_{m,l}^{\text{II}}$ and $R_{ke,l}^{\text{II}}$ values. The ceramic-dominated energy dissipation mechanism in Stages I-II exhibits greater efficiency than the backplate-governed Stages III, thereby increasing the anti-penetration performance (V_{bl}) of composite targets.

For composite targets ($AD = 30.0 \text{ kg/m}^2$) with R_{th} values of 0.6, 1.0, 1.4, 1.6, and 2.0, V_{bl} values are 475.0, 505.0, 535.0, 530.0, 500.0 m/s, respectively. Each target was subjected to projectile impact at its corresponding V_{bl} value. Figure 14 presents the variation curves of $R_{m,l}^{\text{II}}$, $R_{m,l}^{\text{III}}$, and $R_{ke,l}^{\text{II}}$ with respect to R_{th} . Both $R_{m,l}^{\text{II}}$ and $R_{ke,l}^{\text{II}}$ values unimodal distributions within the R_{th} range of 0.4–2.0, peaking at 1.4–1.6. Appropriately increasing the ceramic panel thickness decreases $R_{m,l}^{\text{III}}$ but elevates the projectile total mass loss ratio ($R_{m,l}^{\text{F}} = R_{m,l}^{\text{II}} + R_{m,l}^{\text{III}}$). Insufficient thickness in either ceramic panel or UHMWPE laminate (i.e., caused by R_{th} being too low or too high) will reduce the $R_{m,l}^{\text{II}}$ and $R_{ke,l}^{\text{II}}$ values. Similarly, the $R_{ke,l}^{\text{II}}$ value is much higher than that in Stage III ($R_{ke,l}^{\text{III}} = 1 - R_{ke,l}^{\text{II}}$). The above discussion demonstrates that maintaining an optimal thickness ratio is critical for achieving synergistic protective effects, and the optimal R_{th} value for achieving the highest V_{bl} is 1.4–1.6 when the ADs are 25.0 and 30.0 kg/m^2 , demonstrating superior application potential.

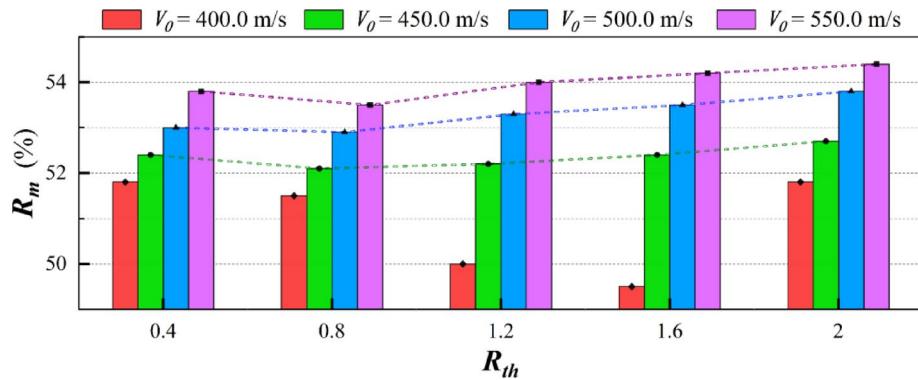


Fig. 15. R_m for different R_{th} and V_0 values ($AD = 25.0 \text{ kg/m}^2$).

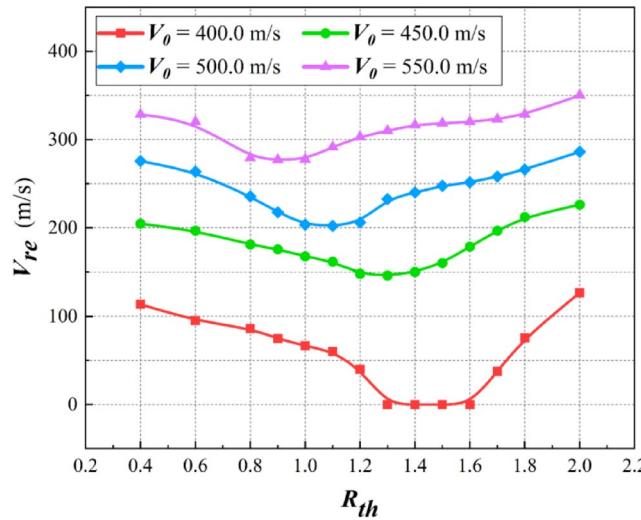


Fig. 16. The variation curve of V_{re} vs. R_{th} ($AD = 25.0 \text{ kg/m}^2$).

Influence of V_0 on the optimal R_{th}

The previous section has discussed the influence of R_{th} on the V_{bl} value of ceramic composite targets. Results reveal that optimally designed targets ($R_{th} = 1.4\text{--}1.6$) can effectively prevent projectile penetration when V_0 remains below the V_{bl} threshold. To extend this protective capacity to lower AD conditions, the influence of V_0 ($V_0 > V_{bl}$) on the optimal R_{th} is systematically investigated.

For composite targets with an AD of 25.0 kg/m^2 , projectiles at 400.0 m/s can completely penetrate composite armors within the R_{th} range of $0.4\text{--}1.3$ and $1.6\text{--}2.0$, excluding the range of $1.3\text{--}1.6$. The highest V_{bl} value reaches 420.0 m/s at $R_{th} = 1.5$. Therefore, projectiles at 400.0 , 450.0 , 500.0 and 550.0 m/s are utilized to penetrate composite armors with various R_{th} values, ranging from 0.4 to 2.0 with an 0.1 or 0.2 increment. Figure 15 shows the projectile final residual mass ratio after the penetration ($R_m = 1 - R_{m,l}^{\text{II}} - R_{m,l}^{\text{III}}$) at various V_0 and R_{th} values. When V_0 is set to 400.0 , 450.0 , 500.0 and 550.0 m/s , the maximum differences in R_m among various R_{th} values ($0.4\text{--}2.0$) are 2.3% , 0.6% , 0.9% and 0.9% , respectively. For cases where V_0 exceeds V_{bl} , the variation of R_m relative to R_{th} does not exceed 0.9% . Consequently, post-penetration V_{re} values are utilized to evaluate the anti-penetration performance of composite armor. Lower V_{re} value correlate with superior anti-penetration performance at a given V_0 . The selected V_0 range of $400.0\text{--}550.0 \text{ m/s}$ is both practically attainable and representative of real-world threats, ensuring that our findings are relevant for application.

As presented in Fig. 16, for composite armors with an AD of 25.0 kg/m^2 under 400.0 m/s projectile impact, the V_{re} initially decreases and then increases as R_{th} varies from 0.4 to 2.0 , and the optimal R_{th} range for achieving the lowest V_{re} is $1.3\text{--}1.6$. Within this R_{th} range, projectiles at 400.0 m/s fail to completely penetrate composite armors, resulting in a V_{re} value of zero. When V_0 increases to 450.0 m/s , 500.0 m/s and 550.0 m/s , the V_{re} - R_{th} trend persists, while the optimal R_{th} ranges are revised to $1.2\text{--}1.4$, $1.0\text{--}1.2$ and $0.8\text{--}1.0$, respectively.

For the $\text{B}_4\text{C}/\text{UHMWPE}$ composite target ($AD = 25.0 \text{ kg/m}^2$, $R_{th} = 1.6$) with a V_{bl} of 415.0 m/s , projectile impacts were simulated at V_0 values of 400.0 , 450.0 , 500.0 , and 550.0 m/s . Figure 17 depicts the variation curves of $R_{m,p}^{\text{II}}$, $R_{m,l}^{\text{III}}$ and $R_{m,l}^{\text{T}}$ values versus V_0 , along with the projectile-target interaction time in Stages I-II (T_{it}^{II}). For the case of $V_0 > V_{bl}$, the interaction time is drastically reduced. The fracture mode of ceramic panel shifts from the gradual quasi-static crack growth to a rapid, dynamic brittle failure mode. We observe that cracks propagate much faster

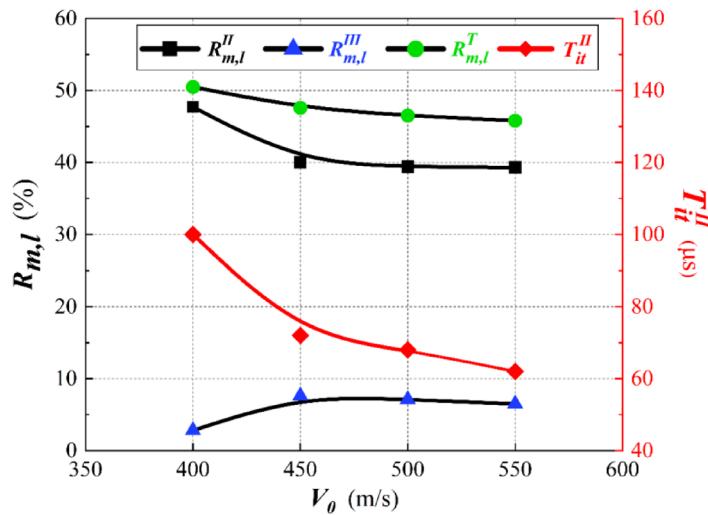


Fig. 17. The variation curves of $R_{m,l}^{\text{II}}$, $R_{m,l}^{\text{III}}$, $R_{m,l}^{\text{T}}$ and T_{it}^{II} vs. V_0 ($AD = 25.0 \text{ kg/m}^2$, $R_{th} = 1.6$).

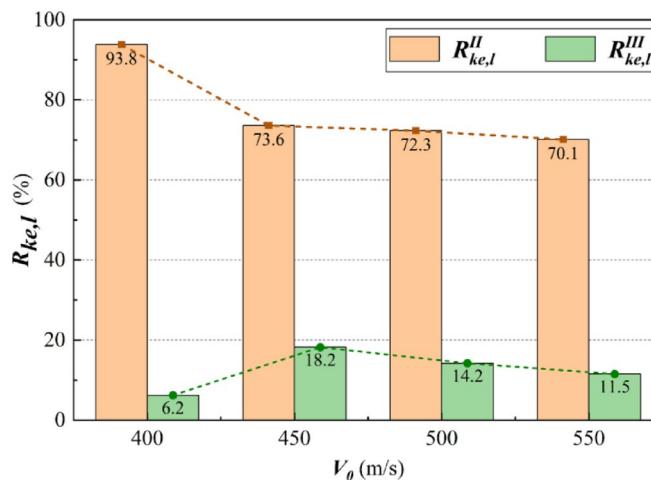


Fig. 18. The variation curves of $R_{ke,l}^{\text{II}}$ and $R_{ke,l}^{\text{III}}$ vs. V_0 ($AD = 25.0 \text{ kg/m}^2$, $R_{th} = 1.6$).

and the ceramic tends to shatter and fragment under these high-rate loading conditions. Consequently, the ceramic panel's energy dissipation decreases, shifting the optimal ceramic-to-backing thickness ratio (R_{th}) due to altered energy distribution mechanisms. When V_0 exceeds the V_{bl} , the variation of $R_{m,l}^{\text{T}}$ relative to V_0 remains within 1.8% across the R_{th} range of 0.4–2.0, as depicted in Figs. 15 and 17, indicating that V_0 has a minimal effect on the $R_{m,l}^{\text{T}}$ under these conditions. When V_0 exceeds the V_{bl} , the fracture mode of ceramic panel transitions from quasi-static propagation to dynamic brittle fracture, accompanied by a substantial acceleration in crack propagation speed. Concurrently, the UHMWPE laminate fails to adequately support the B_4C ceramic panel and absorb the residual kinetic energy, resulting in a sharp decline in T_{it}^{II} which subsequently causes significant reductions in $R_{m,l}^{\text{II}}$ and $R_{m,l}^{\text{T}}$. Figure 18 illustrates the variation of $R_{ke,l}^{\text{II}}$ and $R_{ke,l}^{\text{III}}$ with respect to various V_0 values. The $R_{ke,l}^{\text{II}}$ values remain significantly higher than the $R_{ke,l}^{\text{III}}$ values during the penetration process. When V_0 exceeds the V_{bl} , $R_{ke,l}^{\text{II}}$ decreases dramatically while the $R_{ke,l}^{\text{III}}$ increases significantly, again indicating that B_4C ceramic panel fails to fully fragment the projectile. In this case, increasing the UHMWPE laminate thickness contributes to the kinetic energy absorption capacity in Stages I-II by extending the interaction time T_{it}^{II} and the effective load-bearing area, while also enhancing residual kinetic energy dissipation in Stage III. Consequently, the optimal R_{th} range gradually decreases as the V_0 increases.

For composite targets with an AD of 30.0 kg/m^2 , the highest V_{bl} reaches 540.0 m/s , slightly below 550.0 m/s . Projectile penetration was simulated at V_0 values of 550.0 , 600.0 , 650.0 and 700.0 m/s , to penetrate composite targets with various R_{th} values. Figure 19 presents the variation of post-penetration V_{re} with respect to R_{th} . Similar to the case of $AD = 25.0 \text{ kg/m}^2$, V_{re} exhibits a trend of first decreasing and then increasing across all tested V_0 values, and the optimal R_{th} ranges are identified as follows: 1.4 – 1.6 , 1.2 – 1.4 , 1.1 – 1.3 , 1.0 – 1.2 . Under fixed AD s, the optimal R_{th} for minimizing the V_{re} decreases with increasing V_0 , it can also be said that the optimal R_{th} decreases as the V_{bl} – V_0 differential (at optimal R_{th}) widens. Overall, the anti-penetration performance of

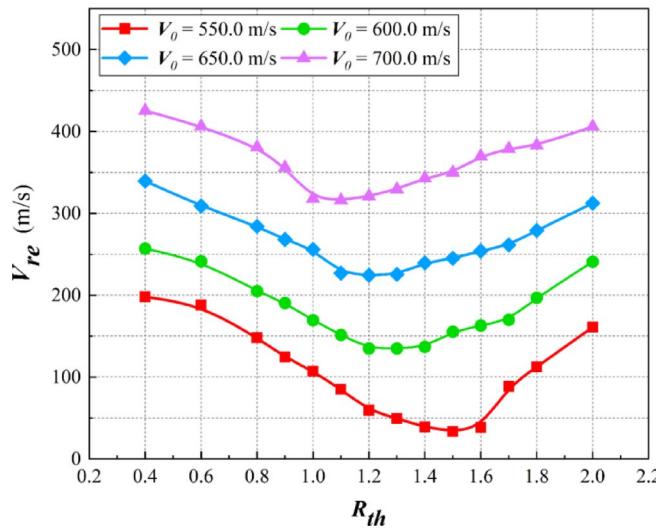


Fig. 19. The variation curves of V_{re} vs. R_{th} for different V_0 values ($AD = 30.0 \text{ kg/m}^2$).

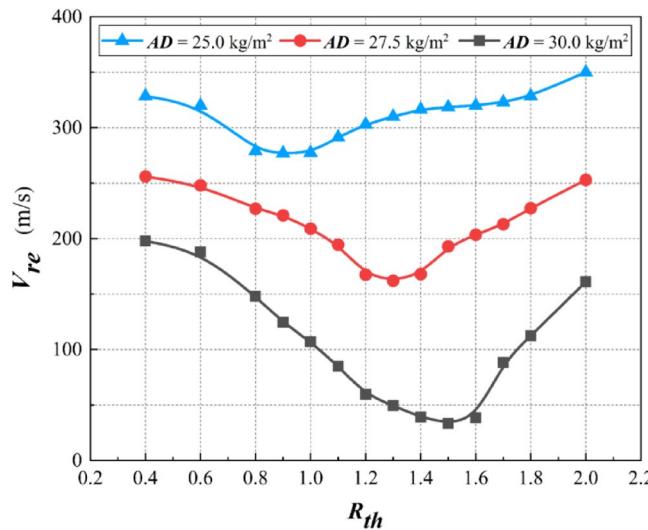


Fig. 20. The variation curves of V_{re} vs. R_{th} for different AD s ($V_0 = 550.0 \text{ m/s}$).

composite armor dynamically correlated with R_{th} and V_0 values. Zhang et al.⁴² and Wu et al.⁴³ demonstrated that V_0 significantly affects the ballistic performance of composite armor, consequently altering its optimal thickness configuration. Prakash et al.⁴⁴ further revealed that impact velocity governs internal energy distribution, with enhanced energy absorption observed until reaching a critical velocity threshold. However, these studies did not quantitatively investigate velocity-dependent optimal R_{th} s. This work systematically quantifies how V_0 (exceeding V_{bl}) influences the optimal R_{th} at given areal densities and elucidates the underlying mechanisms.

Influence of AD on the optimal R_{th}

The preceding discussion demonstrates that for AD s ranging from 25.0 to 30.0 kg/m^2 , the optimal R_{th} consistently remains within 1.4–1.6 for achieving the highest V_{bl} . However, the composite armor exhibits distinct failure modes and energy dissipation distribution patterns when V_0 exceeds the V_{bl} . Notably, V_{bl} varies significantly with AD , necessitating research on the optimal R_{th} when V_0 exceeds the V_{bl} . Simulations are conducted for three AD values: 25.0, 27.5 and 30.0 kg/m^2 , and V_0 is set to 550.0 m/s, exceeding the V_{bl} across all tested AD cases. Figure 20 illustrates the correlation between V_{re} and R_{th} under different AD s. For the above AD s, the corresponding optimal R_{th} ranges for achieving the lowest V_{re} are 0.8–1.0, 1.2–1.4, and 1.4–1.6, respectively. Furthermore, the optimal R_{th} increases as the AD increases when V_0 exceeds the V_{bl} .

For the AD of 30.0 kg/m^2 , the highest V_{bl} of composite armor fluctuates between 530.0 and 540.0 m/s, slightly below 550.0 m/s. The optimal R_{th} range remains 1.4–1.6 for the V_0 of 550.0 m/s. As the AD decreases from 30.0 to 27.5 kg/m^2 , the V_{bl} value drops significantly below 550.0 m/s. When composite armor is fully penetrated, the projectile-target interaction time decreases and the ceramic panel crack propagation accelerates, collectively

reducing the projectile energy dissipation efficiency during the first two phases. For a given *AD*, the variation of R_m relative to R_{th} remains below 0.9% when V_0 exceeds V_{bp} while reducing the ceramic panel thickness appropriately exhibits negligible effect on both $R_{m,l}^T$ and $R_{ke,l}^II$ values. Within the V_{bp} , ceramic panel dominates projectile energy dissipation in penetration stages I-II, maintaining optimal R_{th} at 1.4–1.6 for the *ADs* of 25.0–30.0 kg/m². When V_0 exceeds V_{bp} , the $V_{bl}-V_0$ differential (at optimal R_{th}) widens with *AD* reduction, equivalent to the increase of V_0 . Consequently, increasing the UHMWPE laminate thickness enhances its ability to dissipate the projectile kinetic energy, the optimal R_{th} range is 1.2–1.4 for the *AD* of 27.5 kg/m². When the *AD* further decreases to 25.0 kg/m², the optimal R_{th} range further decreases to 0.8–1.0. In summary, for composite armor with *AD* ranging from 25.0 to 30.0 kg/m², the optimal R_{th} value gradually increases with increasing *AD* when the projectile penetrates the composite armor at the same V_0 (exceeding the V_{bp}), and the optimal R_{th} converges to 1.4–1.6 as the highest V_{bl} of composite armor approaches V_0 .

Bendor et al.⁴⁵ applied Florence's model to determine the optimal configuration of two-component ceramic-faced lightweight armor against normal ballistic impacts. Fawaz et al.⁴⁶ further developed this approach by integrating Florence's model with a hybrid evolutionary algorithm for enhanced armor design optimization. Both studies showed that the R_{th} (corresponding to the V_{bl}) decreases with increasing *AD*, the optimal R_{th} obtained from the former studies showed minimal variation at 25.0–30.0 kg/m², which aligns with our Sect. 4.1 findings. Crucially, this section further demonstrate that when V_0 exceeds V_{bp} , the optimal R_{th} increases with rising *AD*, which is a previously unreported relationship.

Conclusions

To guide the structural optimization of composite armor with low-to-moderate *ADs* (25.0–30.0 kg/m²), numerical simulations of 7.62 mm steel-core projectile through B₄C/UHMWPE ceramic composite target were conducted using the FEM, validated through the ballistic testing results. The influence of R_{th} , V_0 , and *AD* on the anti-penetration performance is systematically investigated. The following conclusions are drawn from this study:

(1) The FEM is utilized to quantify the $R_{m,l}$ and $R_{ke,l}$ values at various penetration stages, aiming to evaluate the ballistic performance of composite armor. Simulation results exhibit a maximum deviation of 5.60% compared to the ballistic testing results.

(2) The V_{bl} is significantly affected by *AD* and R_{th} values. When R_{th} increases from 0.4 to 2.0, the V_{bl} initially increases and then decreases, with the optimal R_{th} range stabilizing at 1.4–1.6 for *ADs* of 25.0 and 30.0 kg/m². Within this range, ceramic cone cracks fully propagate, with $R_{m,l}^T$ and $R_{ke,l}^II$ values peaking during penetration stages I-II, establishing the optimal protection mechanism of ceramic-dominated fragmentation and backing-plate synergistic energy dissipation.

(3) When V_0 exceeds V_{bp} , V_{re} initially decreases and then increases with rising R_{th} , and R_{th} shows negligible effect on the $R_{m,l}^T$. Furthermore, the optimal R_{th} for achieving the lowest V_{re} decreases with rising V_0 , indicating that a thicker backing laminate is required to compensate the shortened projectile-target interaction time and improve the $R_{ke,l}^III$ value.

(4) Under constant V_0 conditions where V_0 exceeds V_{bp} , the optimal R_{th} range increases with increasing *AD*, and the optimal R_{th} converges to 1.4–1.6 as the highest V_{bl} corresponding to given *AD* approaches V_0 . Specifically, at a V_0 of 550.0 m/s, the optimal R_{th} ranges are 0.8–1.0, 1.2–1.4 and 1.4–1.6 for *ADs* of 25.0, 27.5 and 30.0 kg/m², respectively.

(5) This study quantifies the V_0 -*AD*- R_{th} coupling effects, providing actionable guidelines for designing lightweight composite armor against diverse ballistic threats. Given the limitations of the FEM in capturing microscale damage evolution, future work may incorporate more advanced modeling approaches, such as FEM-SPH or multiscale simulation, to better resolve the underlying mechanisms of this coupling behavior. Moreover, extending the framework to include complex loading conditions, such as varying projectile calibers, oblique impact angles, and temperature effects, would further enhance its applicability and engineering relevance.

Data availability

No datasets were generated or analysed during the current study.

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Declarations

Competing interests

The authors declare no competing interests.

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