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UV-shielding photochromic composite films based on nanocrystalline cellulose modified with CeO_2 and WO_3 nanoparticles

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The UV-shielding composite films based on nanocrystalline cellulose (CNC) modified with both CeO_2 and WO_3 nanoparticles were prepared by solvent casting method. In the composites, both CeO_2 and WO_3 nanoparticles showed redox activity under UV-irradiation, and the joint effects of that metal oxides in the composite films were evidenced. The WO_3 nanoparticles provided reversible photochromic properties to the WO_3 /CNC film under UV-irradiation while CeO_2 nanoparticles inhibited the photochromic effect. The mechanism of this inhibition in the CeO_2/WO_3 /CNC films was investigated using UV-vis and FTIR spectroscopy. The photodegradation of CNC catalyzed by CeO_2 nanoparticles inherent in CeO_2 /CNC films was prevented by the addition of WO_3 nanoparticles. The CeO_2 /CNC and CeO_2/WO_3 /CNC films exhibited strong UV-shielding property. The synthesized photostable UV-shielding CeO_2/WO_3 /CNC films with photochromic property which can easily be tuned by the CeO_2 to WO_3 ratio can be useful for various applications, including protection of UV-sensitive dyes and production of UV-shielding packages.

Keywords UV-shielding films, Photochromism, Ceria, Tungsten oxide, Cellulose, Nanocomposite

Inorganic nanoparticles are widely used in UV-protective materials because of their greater stability compared to organic UV-absorbers, which are often prone to photodegradation. Among the most popular materials, TiO_2 , ZnO , ZrO_2 and CeO_2 nanoparticles effectively block UV-A or UV-B radiation, depending on the particle size^{1–4}. The photocatalytic activity of some inorganic materials, e.g. TiO_2 or ZnO can limit their use as photoprotectors. Absorption of UV radiation leads to charge separation and trigger redox reactions on the surface of semiconductor metal oxide nanoparticles, which is advantageous for catalytic or photocatalytic applications^{1,5–10}. However, with regard to photoprotective materials, redox surface activity is undesirable because it can result in the material's degradation and even in the formation of toxic products^{1,2}.

Nanoscale cerium dioxide (CeO_2) is considered as a promising photoprotective material as it effectively absorbs UV light and often shows negligible photocatalytic activity^{1,11,12}. CeO_2 with particle size less than 10 nm can be prepared by various methods, including hydrothermal¹³, solvothermal¹⁴, microwave¹⁵, spray pyrolysis¹⁶, room-temperature synthesis in biopolymer matrices^{17,18} and others. The CeO_2 nanoparticles possess fluorite-type crystal structure, where cerium is surrounded by eight oxygen atoms, the band gap of this semiconductor is about 3.2 eV^{8,9,15}.

The CeO_2 nanoparticles are considered to be low- or non-cytotoxic^{12,16,19}, and thus attract a great deal of attention as an alternative component of cosmetic sunscreens that doesn't cause whitish appearance of skin in contrast to TiO_2 and ZnO ¹. Therefore, cerium oxide is well studied as a UV-protective component in dispersions^{1,12}. Availability of synthesis methods of fine CeO_2 nanoparticles that can easily disperse in a polymer matrix enables the use of cerium dioxide as a UV-protective component in various composite bulk materials, including electrospun cellulose nanofibers¹³, shielding for wood²⁰, automotive lacquers⁸, mesostuctured layers for solar cells²¹. Immobilizing cerium oxide in biopolymers is particularly interesting for producing safe materials for wearable textiles, medical applications, and product packaging.

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Nanocrystalline cellulose or cellulose nanocrystals (CNC) is a widely used nontoxic, biodegradable and commercially available material, which allows to produce flexible and transparent films^{22,23}. Chemical modification and functionalization of cellulose-based materials with inorganic nanoparticles improve their barrier²⁴, photoprotective²⁵, luminescence²⁶, magnetic²⁷, antimicrobial²⁸, adsorption²⁹ and other properties.

Nanocrystalline cellulose is a transparent material, which can be provided with UV-protective properties by introduction of UV absorbers. Although various nanoparticles (e.g., metal oxides and lignin) significantly improve the UV-protective properties of CNC films, they often reduce transparency in the visible range and increase light scattering (haze) of the composite films. Recently, CNC films with different percentages of lignin nanoparticles and ZnO nanoparticles demonstrated a strong UVC and UVB blocking performance³⁰. Increasing of nanoparticles content from 1 to 5% gradually increased UV-blocking properties, however, at the same time reduced transparency in the visible light region. The PVA/CNC/TiO₂ composite exhibited high mechanical strength, which was provided by the CNC, and excellent UPF values, which were provided by the TiO₂ nanoparticles³¹. Incorporation of ZnO nanoparticles into CNC/pectin nanocomposite films reduced transmissions at 600 nm and 280 nm from 70.11 to 29.93% and 28.27 to 4.93%, respectively³². Hybrid films composed by cellulose nanocrystals and carboxymethylated cellulose nanofibrils loaded with montmorillonite clay, despite their low transparency, possessed UV-blocking features³³.

However, only a few studies have focused on bulk UV-protective composites based on cellulose or nanocrystalline cellulose modified with cerium dioxide. These studies have shown that cerium dioxide nanoparticles provide cellulose materials with strong photoprotective properties. The *in situ* synthesis of CeO₂ nanoparticles on chitosan-treated linen fabrics improved the UV-protective properties of the material, increasing the UPF sevenfold compared to untreated fabric³⁴. Nanofibers of natural cotton cellulose were prepared by electrospinning and decorated with CeO₂ nanoparticles of 40–60 nm in size by hydrothermal method to improve UV-protective properties of the material³⁵. The UV-blocking hybrid coatings with CeO₂ and SiO₂ were prepared using cellulose nanocrystals and cellulose nanofiber as a biopolymer matrix³⁵. CeO₂ of 8 nm in size and SiO₂ of 5 nm in size nanoparticles were added to improve UV screening and hardness properties, respectively. The coatings were highly transparent in visible light region, while absorption in UVA region increased with CeO₂ content. The transparent UV-shielding composite films based on regenerated cellulose were fabricated by *in situ* synthesis of CeO₂ from cerium nitrate precursor³⁶. The diameters of rod-like cerium oxide nanoparticles in composites vary from 20 to 50 nm depending on the precursor concentrations. The transmittance of the nanocomposite films in the visible light region decreased with increasing precursor concentrations.

Therefore, this work focuses on a novel bulk composites based on nanocrystalline cellulose and CeO₂. Using the hydrothermal method described in our previous work allows for the production of ultrafine CeO₂ particles³⁷, which can be easily introduced into nanocrystalline cellulose films via solvent casting. Additionally, the use of small particles is expected to produce films transparent in the visible range.

Tungsten trioxide (WO₃) semiconductor nanoparticles demonstrate strong UV-absorption, photochromic, photocatalytic and electrochromic properties^{38–42}. Tungsten trioxide exhibits photocatalytic properties under UV-irradiation, and WO₃ composites are used for water splitting systems and waste water pollutant treatment^{43,44}, since an efficient charge transfer is provided at the liquid–solid interface. On the other hand, the UV-induced electron transfer in WO₃ nanoparticles can lead to reduction of the colorless W⁺⁶ to the blue-colored W⁺⁵³⁹. The subsequent oxidation by oxygen molecules in dark results in bleaching of blue color³⁹, thus the reversible photochromic properties make tungsten trioxide a photostable and reusable marker of UV-radiation. Adjusting of WO₃ photochromic performance in nanocomposites reveal opportunity to produce smart UV-protective and UV-sensitive materials. For example, influence of the preparation method on coloring contrast and photochromism reversibility of TiO₂–WO₃ nanoparticles was recently reported⁴⁵. The highly reversible photochromism of the composite films of nanocrystalline cellulose with WO₃ nanoparticles stabilized by polyvinylpyrrolidone was reported in our previous work³⁹. Ultrafine tungsten oxide nanoparticles are colorless, unlike larger particles, which are yellow in color. Therefore, a nanocrystalline cellulose film containing ultrafine WO₃ nanoparticles was transparent in the visible range³⁹. Due to the photochromic properties of tungsten oxide, the WO₃ composite films can also be used as UV sensors. The current study aims to use WO₃ and CeO₂ nanoparticles to develop UV-protective composite nanocellulose films and examine the effect of the material's composition on the photochromic properties of tungsten oxide.

In this study, we produced a novel UV-shielding composite film based on nanocrystalline cellulose (CNC) modified with CeO₂ and WO₃ nanoparticles with tunable photochromism dependent on the CeO₂/WO₃ ratio and enhanced photostability. Transparent films were obtained by synthesizing ultrasmall CeO₂ and WO₃ nanoparticles, that were immobilized in CNC without additional surfactants. For the first time, the mutual influence of CeO₂ and WO₃ nanoparticles under the action of UV irradiation was investigated. We propose an electron transfer mechanism, whereby CeO₂ quenches the photochromic response of WO₃, while WO₃ inhibits the pro-oxidant photocatalytic degradation of the cellulose matrix typically catalyzed by CeO₂ under UV light. The prepared CeO₂/WO₃/cellulose composite film overcomes the individual limitations of each component, offering an alternative to organic filters and CeO₂-based filters for protecting light-sensitive materials. UV-vis and FTIR spectroscopy have been applied to study the UV-shielding properties of the composite films using tungsten trioxide as a photochromic marker and β-carotene as a photodegradable natural dye, and to monitor the kinetics of WO₃ photochromism and the photodegradation of the cellulose film.

Experimental Materials

Ceric ammonium nitrate (NH₄)₂Ce(NO₃)₆ (Sigma Aldrich), sodium tungstate (VI) dihydrate Na₂WO₄·2H₂O (Sigma Aldrich), 1-(4-tert-Butylphenyl)-3-(4-methoxyphenyl)-1,3-propanedione – “avobenzone” (Sigma Aldrich), β-carotene (0.30%, Luxomix), isopropanol (purity 98%), ethanol (purity 95%), and H₂SO₄ (purity

98%) were used as starting reagents. Filter paper “blue ribbon” was utilized as a cellulose source. Cation exchange resin (Amberlite® IR120) in acidic form was used in WO_3 synthesis.

Preparation of CNC

The dispersion of nanocrystalline cellulose (CNC) was prepared by a commonly used method described elsewhere⁴⁶. Filter paper (blue ribbon) was shredded and soaked in distilled water for 30 min, then sulfuric acid (98%) was added slowly with vigorous stirring in an ice bath until the acid concentration reached 65%. After 50 min stirring at 47 °C, 1.5 L of cold distilled water was added to the obtained pulp to stop hydrolysis. The resulting nanocrystalline cellulose was rinsed with distilled water, separated by centrifugation, and the residual acid was removed by dialysis. Finally, the nanocrystalline cellulose was sonicated in an ice bath for 2 h to obtain a homogeneous dispersion. The concentration of CNC in the dispersions, measured gravimetrically, was 1–2 wt.%.

Preparation of the metal oxide sols

A facile method of synthesis of surfactant-free CeO_2 nanoparticles with narrow size distribution from ceric ammonium nitrate $(\text{NH}_4)_2\text{Ce}(\text{NO}_3)_6$ was described in details previously³⁷. $(\text{NH}_4)_2\text{Ce}(\text{NO}_3)_6$ was dissolved in deionized water to obtain a 0.28 M solution. The solution was heated in an autoclave at 95 °C for 24 h to obtain a yellow precipitate of cerium dioxide. The precipitate was separated by centrifugation and rinsed with isopropanol. The prepared cerium dioxide readily dispersed in deionized water to form a stable, transparent yellow sol with a pH of about 3. The concentration of the sol, measured gravimetrically, was 2.9 wt.%.

The method of preparation of ultra-small WO_3 nanoparticles described elsewhere^{47,48} was modified to obtain WO_3/CNC composites. Transparent solution of tungstic acid was prepared from 0.033 M solution of $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ in distilled water, using cation exchange resin in H^+ form. The obtained sol of hydrated tungsten trioxide was added to the CNC dispersion immediately after the preparation. The concentration of WO_3 in the sol was 2.4 wt.% as measured gravimetrically.

Preparation of the composite films

The flexible transparent colorless films were prepared by the solvent casting method. The ratios of metal oxides to CNC in the films varied as follows: CeO_2/CNC (1/10, 0.5/10, 0.1/10, and 0.01/10), $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10, 0.5/1/10, 0.1/1/10, and 0.01/1/10), WO_3/CNC (1/10), avobenzene/CNC (0.1/10 and 0.01/10), and avobenzene/ WO_3/CNC (0.1/1/10 and 0.01/1/10), the molar ratios were calculated per formula units of oxides and cellulose monomer. The corresponding wt.% of CeO_2 , WO_3 , and CNC in the prepared films are summarized in Table 1.

To prepare the film, the required amount of CeO_2 or WO_3 sols were added to the CNC dispersion and stirred to obtain homogeneous mixtures. As avobenzene is virtually insoluble in water, it was dissolved in a small amount of ethanol before mixing with cellulose or WO_3/CNC dispersions. The required amount of distilled water was added to achieve CNC concentration of 1.2% in each resulting mixture. Then 4 mL of the mixtures were poured into plastic Petri dishes of 40 mm diameter and dried at room temperature for 5 days. After additional drying at 50 °C, the weight loss due to residual water elimination was 3–7% for all the samples. The thickness of the films was $40 \pm 2 \mu\text{m}$.

Analysis methods

X-ray diffraction (XRD) patterns were recorded on a Bruker D8 Advance diffractometer (CuK α radiation). Crystallite sizes were estimated using Scherrer equation. The crystallinity index of cellulose nanocrystals was calculated by Segal's method using the intensity of the reflection at 23° (I_{200}) and the intensity in the minima at 19° between $\langle 110 \rangle$ and $\langle 200 \rangle$ reflections as the amorphous part of the signal (I_{AM}) by Eq. (1)²⁶:

$$\text{CI} = (I_{200} - I_{\text{AM}}) / (I_{200}) * 100\% \quad (1)$$

Scanning electron microscopy (SEM) images were obtained on Carl Zeiss NVision 40 electron microscope at an accelerating voltage of 0.5 kV, elements content and distribution in samples were measured by energy-dispersive X-ray (EDX) analysis (X-Max detector, Oxford Instruments). Transmission electron microscopy (TEM) images were obtained on a Leo-912 AB OMEGA microscope with an accelerating potential of 60–120 kV. Fourier transform infrared spectra (FTIR) were measured with resolution of 2 cm^{-1} in the range of 400–4000 cm^{-1} using an INFRALUM FT-08 spectrometer. The hydrodynamic radii and ζ -potentials were assessed using a Photocor Compact Z analyzer with a thermally stabilized 638 nm semiconductor laser.

Composition	Molar ratios calculated per formula units	wt.% of CeO_2	wt.% of WO_3	wt.% of CNC
WO_3/CNC	1/10	-	12.5	87.5
CeO_2/CNC	1/10	9.5	-	90.5
$\text{CeO}_2/\text{WO}_3/\text{CNC}$	1/1/10	8.5	11.5	80
$\text{CeO}_2/\text{WO}_3/\text{CNC}$	0.5/1/10	4.5	12	83.5
$\text{CeO}_2/\text{WO}_3/\text{CNC}$	0.1/1/10	1	12.5	86.5
$\text{CeO}_2/\text{WO}_3/\text{CNC}$	0.01/1/10	0.1	12.5	87.4

Table 1. The list of the prepared composite films.

UV-visible reflection spectra of the films were measured using an Ocean Optics QE65000 spectrometer with HPX-2000 xenon lamp. Kinetic features of photochromic reactions were studied using a xenon lamp of spectrometer as a UV-light source. Beta-carotene photodegradation was studied using UV-camera with the emission maximum at 312 nm, UV-visible transmittance spectra of solutions were measured on an OKB SPEKTR SF-2000 spectrometer.

Results and discussion

Characterization of the composite films

The X-ray diffraction data for the CNC film (Fig. 1a) showed signals at $2\theta = 16^\circ$ and 23° corresponding to $<1-10>$ overlapped with $<110>$ and $<200>$ reflections of the $\text{I}\beta$ crystalline cellulose structure, respectively^{46,49}. Crystallinity index of cellulose material calculated by Segal's method was 80%, which is in accordance with literature data for sulfuric acid hydrolyzed cellulose (60–90%)²². TEM image of the CNC sample clearly shows rod-like disordered particles (SI, Fig. S1). The measurements taken on 100 individual particles shows, that the hydrolysis of cellulose with sulfuric acid resulted in the formation of cellulose nanorods with a length of 240 ± 65 nm and a width of 9 ± 2 nm, which corresponds to the typical size range for nanocrystalline cellulose²². EDX data shows that sulfur is present in the material due to the partial esterification of CNC during hydrolysis (SI, Table S1). The ζ -potential of the CNC dispersion at pH 6.5 was -28 mV, and the charge measured using NaOH titration was 0.4 mmol/g.

The prepared cerium dioxide surfactant-free sol is stabilized by electrostatic interaction between nanoparticles, as was described in details previously³⁷. The sol shows a pH about 3, and the ζ -potential of the CeO_2 nanoparticles is about +30 mV, due to the presence of nitric acid formed during hydrothermal treatment of ceric ammonium nitrate. The TEM image reveals ultrasmall CeO_2 nanoparticles within 2–3 nm (Fig. S2). The average hydrodynamic radius of CeO_2 was about 7 nm (SI, Fig. S3). The XRD patterns of the CeO_2/CNC and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films, in addition to the CNC reflections show weak reflections attributed to CeO_2 nanoparticles of fluorite-type (PDF2 #34–394) crystal structure (Fig. 1a), the average size of the crystallites was estimated as 3 nm. The deconvolution of the peaks in the CeO_2/CNC XRD pattern is shown in Fig. S4 (see SI).

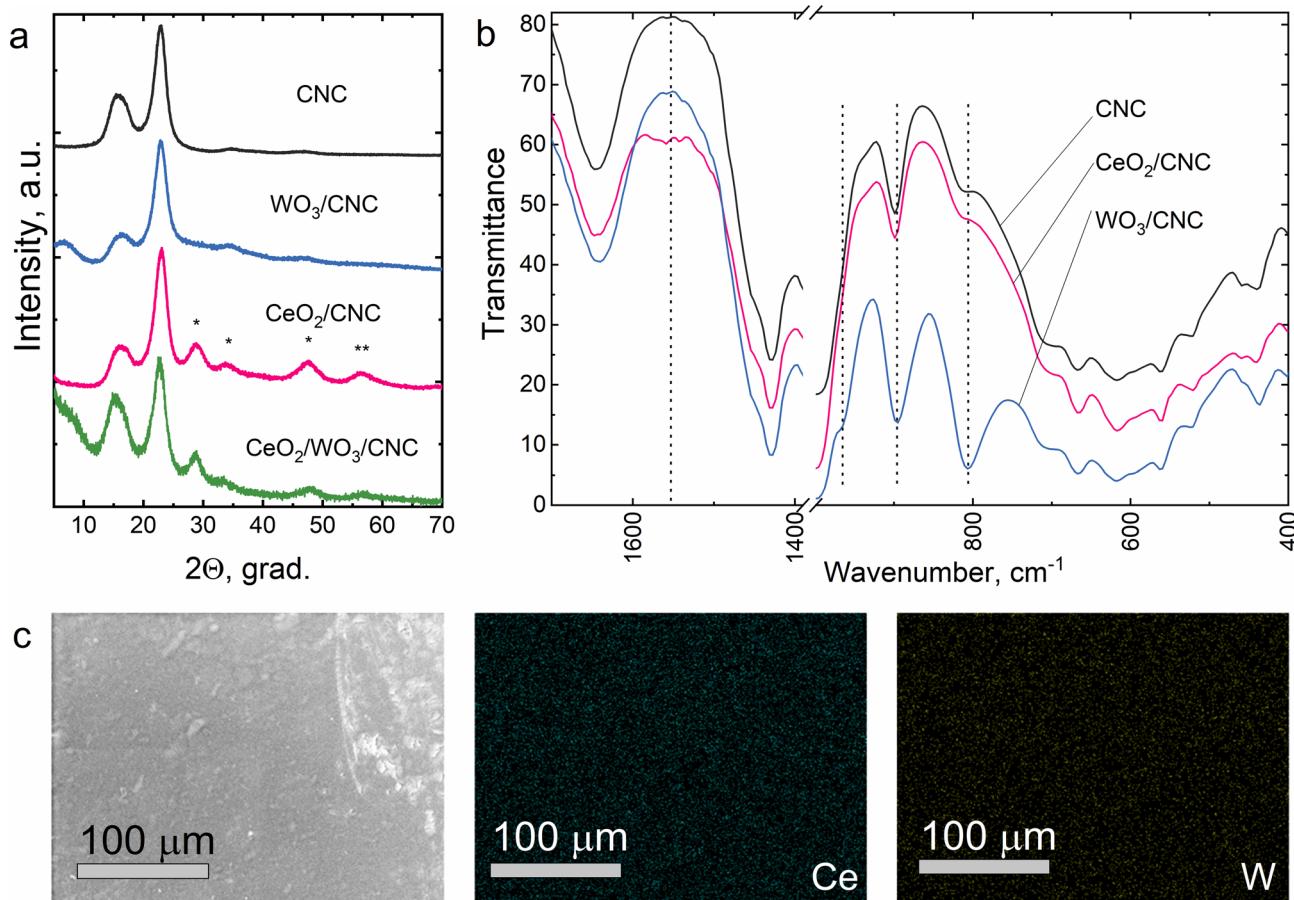


Fig. 1. (a) XRD patterns of the films: CNC, WO_3/CNC 1/10, CeO_2/CNC 1/10 (*asterisks mark reflections of CeO_2) and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10). (b) Fragments of FTIR spectra of the CNC (upper), CeO_2/CNC 1/10 (middle), and WO_3/CNC 1/10 (lower) films. (c) SEM image of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) film and distributions of Ce and W elements.

The as prepared WO_3 nanoparticles with average hydrodynamic radius about 1.4 nm possess ζ -potential of about -20 mV and the bare sol remains visually transparent and stable during several hours (SI, Fig. S3). In the XRD patterns of the WO_3/CNC and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films, the wide signal is present at 2θ angles $< 10^\circ$ due to scattering on ultra-small amorphous particles of WO_3 (Fig. 1a)³⁹. The TEM image shows ultrasmall nanoparticles of WO_3 within 2–6 nm (Fig. S2), that is in consistence with previously data reported for the WO_3 synthesized by the similar method^{47,48}.

Figure 1b shows FTIR spectra of the composite films. The broad band at 3500–3100 cm^{-1} is related to stretching vibrations of -OH groups of cellulose, the band at 2900 cm^{-1} is due to the stretching vibrations of $\text{CH}-$ and CH_2- groups, the band at 1650 cm^{-1} is attributed to $\text{C}=\text{O}$ vibrations of CNC^{23,50}. The band at 1430 cm^{-1} corresponds to CH_2 deformation vibrations, and broad band at 1200–1000 cm^{-1} is due to the stretching vibrations of $\text{C}-\text{O}-\text{C}$ and $\text{C}-\text{O}$ groups⁵⁰. In addition to the CNC absorption bands, the bands at 803 and 947 cm^{-1} assigned to $\text{W}-\text{O}-\text{W}$ and $\text{W}=\text{O}$ vibrations, respectively, are present in FTIR spectrum of the WO_3/CNC composite film^{39,51,52}. The absorption band at 899 cm^{-1} corresponds to $\text{C}-\text{O}$ vibration in CNC film⁵⁰. In the WO_3/CNC film the more intensive absorption band with maximum at 895 cm^{-1} appears due to overlapping of the $\text{C}-\text{O}$ vibration of CNC with the absorption band related to WO_3 .

The very weak absorption band related to $\text{Ce}-\text{O}$ stretching vibrations is present at 550–450 cm^{-1} in the FTIR spectrum of the CeO_2/CNC film⁵³. In addition, FTIR spectrum of the CeO_2/CNC film has the absorption band at 1570 cm^{-1} in the region of carboxyl group vibrations.

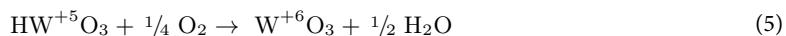
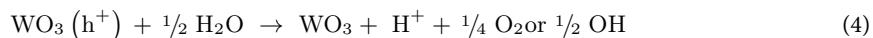
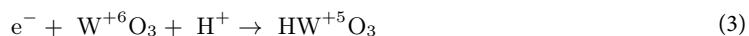
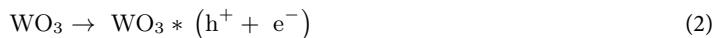
Cerium and tungsten oxides were uniformly distributed in the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ composite films (Fig. 1c), and the molar ratio of W and Ce atoms in the prepared composite film was also confirmed by EDX data (SI, Table S2). Thermal analysis of the CNC and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) films showed 4–5% weight loss at temperatures below 200 $^\circ\text{C}$ due to water desorption, the degradation of the films occurred at 250–450 $^\circ\text{C}$, and oxide nanoparticles did not significantly affect the thermal stability of the films (SI, Fig. S5–S6).

Optical and photochromic properties of the composites

The CNC film was transparent in UV and visible region. Cerium dioxide semiconductor nanoparticles caused strong absorption below 430 nm in UV-vis spectrum of the CeO_2/CNC composite films (Fig. 2a). The band gap about 2.8 eV estimated using Tauc plot is typical for CeO_2 nanoparticles (SI, Fig. S7)⁸.

The prepared WO_3/CNC and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ composite films were also colorless and possessed a strong absorption in UV-region. UV light exposure caused an intense blue coloration of the WO_3/CNC film, while the blue coloration of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ film was very weak.

Photochromic properties of WO_3 nanoparticles are related to UV-induced reduction of W^{+6} in colorless WO_3 nanoparticles to W^{+5} and W^{+4} species which absorb light in the visible region^{38,39,54–56}. Absorption of a UV-photon with an energy higher than the band gap of WO_3 nanoparticles (2.6–3.0 eV, depending on particle size⁵⁷) initiates electron transition to conduction band and formation of positively charged hole in valence band (2). Reduction of the W^{+6} ion by the electrons results in blue coloration of the tungsten trioxide nanoparticles (3). Interaction between positive holes and water molecules absorbed on the surface of WO_3 nanoparticles weakens O-H bonds, resulting in a release of protons H^{+6} and highly reactive oxygen species, which can recombine to form O_2 molecules or produce OH^\bullet radicals (4)^{38,48}. The reverse discoloration occurs in the dark due to oxidation of tungsten ions to colorless W^{+6} species by oxygen molecules (5).



The efficiency of electron–hole recombination and alternative redox reactions affect the photochromic properties of tungsten oxide. Various ligands with -OH or =NH groups enhance the photochromism in WO_3 composites. Apparently, complexation with ligands through hydrogen and donor–acceptor bonds promotes the reduction of W^{+6} under UV light. For example, Adachi et al. showed strong photochromic properties of $\text{WO}_3/\text{cellulose}$ films, while surfactant-free WO_3 nanoparticles and $\text{WO}_3/\text{triacetyl cellulose}$ hybrid films were not photochromic³⁸. Polyvinylpyrrolidone and polycarbonate have also been used to enhance the photochromic properties of tungsten oxide^{39,54}.

Although cellulose bear three -OH groups in each glucopyranose fragment, the aqueous dispersion containing both WO_3 and CNC did not exhibit photochromic properties. The -OH groups on cellulose surface probably cannot substitute water molecules in the coordination sphere of the hydrated WO_3 nanoparticles. On the contrary, the WO_3/CNC films obtained by the solvent casting method were strongly photochromic, indicating coordination bonds between tungsten trioxide nanoparticles and -OH groups on the surface of the cellulose nanoparticles.

Exposure of the composite films to UV-light resulted in the increase in the intensity of absorption bands near 780 and 620 nm corresponding to W^{+5} and W^{+4} species, respectively (Fig. 2b and c)^{55,56}. The absorption at 780 nm appeared first under UV irradiation, while the absorption band at 620 nm grew rapidly and dominated in the spectra of the films after several minutes of UV-exposure. Notably, the absorption band intensities in the WO_3/CNC spectrum increased significantly faster than in the corresponding $\text{CeO}_2/\text{WO}_3/\text{CNC}$ film spectra. This effect depends on the composition of the composite films, resulting in slower coloration of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films with higher CeO_2 content (Fig. 2e). In order to estimate the excitation wavelengths causing

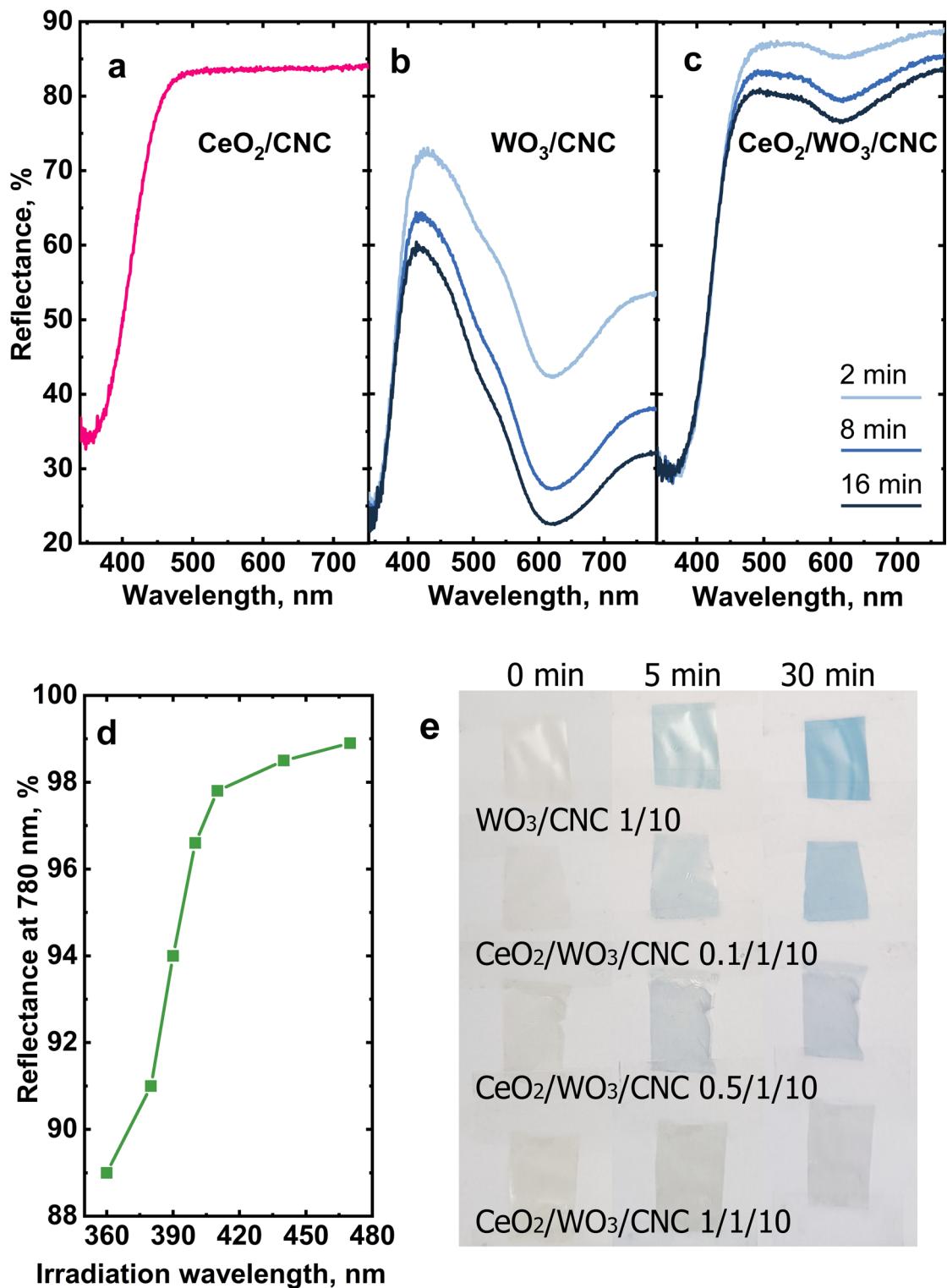


Fig. 2. (a) UV-vis spectrum of the CeO_2/CNC 1/10 film. (b and c) Changes in UV-vis. spectra of the WO_3/CNC (1/10) and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) films, respectively, under UV-irradiation for 2-, 8- and 16-min. (d) Reflectance at 780 nm in spectra of the WO_3/CNC (1/10) under irradiation with different wavelengths (as measured after 5 min of irradiation of the initially colorless sample). e) The appearance of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films with different component ratios before and after UV-irradiation.

photochromic reaction, the WO_3 /cellulose film was exposed to narrow-band UV light (bandwidth ~ 10 nm) having wavelengths ranging from 470 to 380 nm. The absorption band in the UV spectra at 780 nm indicating the photochromic transition appeared under irradiation with wavelengths shorter than 390 nm (see Fig. 2d).

Effect of CeO_2 on photochromic properties of the composite films

In order to study the effect of CeO_2 on the photochromic properties of the synthesized composite films, the coloration kinetics under irradiation with the Xe-lamp of a spectrometer were measured for WO_3/CNC , $\text{CeO}_2/\text{WO}_3/\text{CNC}$ and WO_3/CNC top covered with CeO_2/CNC film (Fig. 3).

The CeO_2/CNC film placed onto the WO_3/CNC sample showed a strong shielding effect and inhibited the coloration of tungsten species upon Xe-lamp irradiation, because of the strong UV-absorption of CeO_2 nanoparticles. The UV-absorption band of CeO_2 (< 430 nm) fully overlapped the spectral region initiating coloration of WO_3 (< 390 nm). The CeO_2/CNC (0.01/10) film with very low content of ceria did not affect the WO_3/CNC coloration kinetics (Fig. 3a). In turn, the samples with higher CeO_2/CNC ratios (0.1/10, 0.5/10, and 1/10) decreased coloration rate of the WO_3/CNC film (Fig. 3a).

The photochromic property of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films depended on the CeO_2 content in the composite. Upon the UV-irradiation of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films, the intensity of the absorption band at 620 nm increased slowly (Fig. 3b). The kinetic colouration curves for the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (0.01/1/10 and 0.1/1/10) samples demonstrated minor differences from the kinetics for the reference WO_3/CNC (1/10) film. For the composite films with higher CeO_2 concentration, significantly slower coloration was observed. The $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) film remained transparent and nearly colorless during prolonged UV-exposure (over 7 h). As shown in

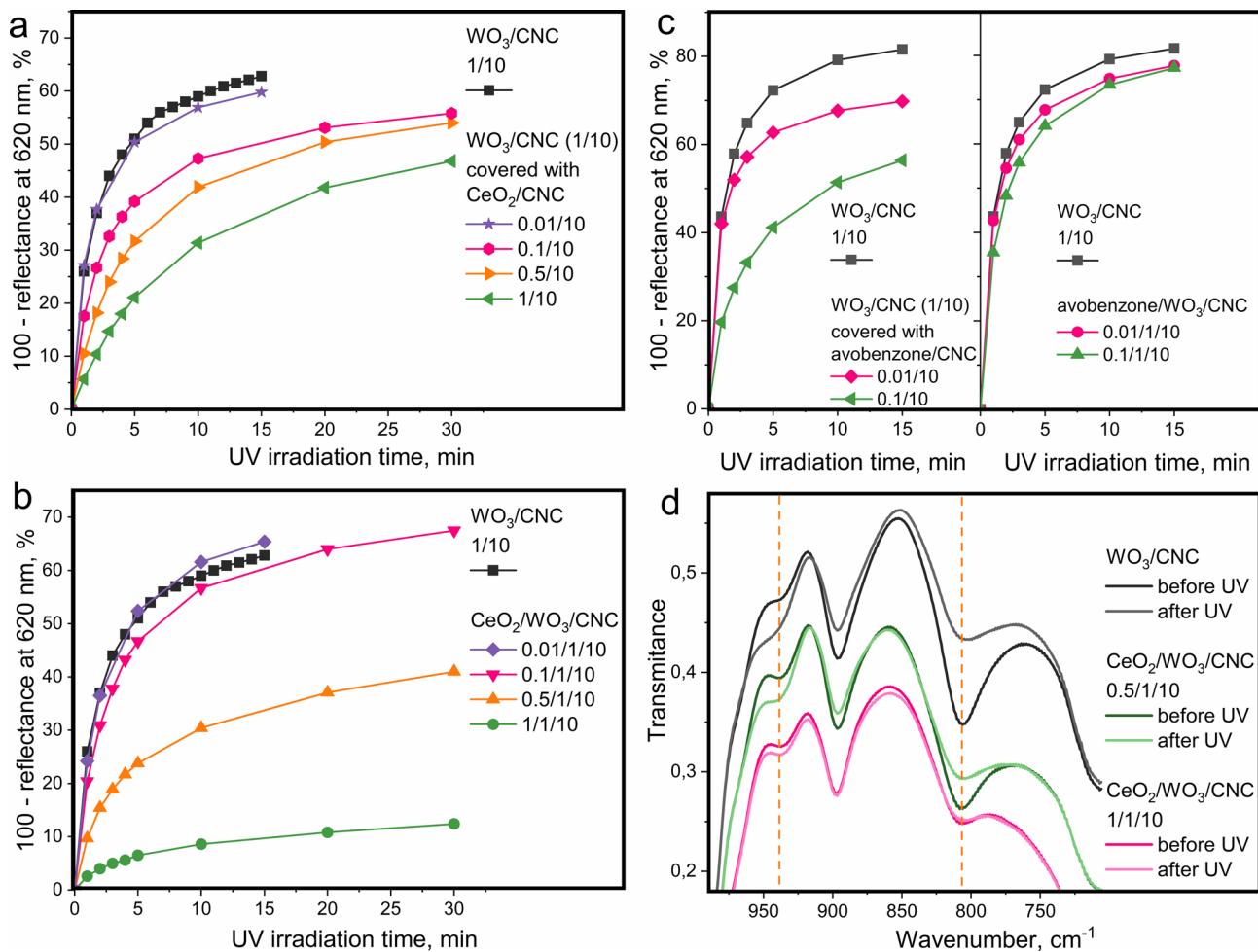


Fig. 3. (a) The coloration kinetic of the WO_3/CNC (1/10) film and the WO_3/CNC (1/10) top covered with the CeO_2/CNC films with different CeO_2 content. (b) The coloration kinetic of the WO_3/CNC (1/10) film and the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films with different component ratios. (c) The coloration kinetic of the WO_3/CNC (1/10) film, the WO_3/CNC (1/10) film top covered with the avobenzone/cellulose films with different component ratios, and the avobenzone/ WO_3/CNC films with different component ratios. The kinetic curves were measured using the relative intensities of the 620 nm band in UV-vis diffuse reflectance spectra. d) The fragments of the FTIR spectra before and after UV-irradiation of the films: WO_3/CNC (1/10), $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (0.5/1/10) and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10).

Fig. 3a and b, the addition of CeO_2 in the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ composite films inhibits photochromism of WO_3 in a more pronounced manner than top covering the WO_3/CNC film with the CeO_2/CNC film.

The same experiments were performed to compare the UV-protecting effect of CeO_2 with that of one of the most common organic UV-filters, avobenzone (absorption maximum at 357 nm)^{58,59}. Both the avobenzone/ WO_3/CNC (0.01/1/10 and 0.1/1/10) films and the WO_3/CNC film top covered with the avobenzone/CNC (0.01/10 and 0.1/10) films were exposed to UV-light. As shown on Fig. 3c, the avobenzone/CNC films effectively shielded WO_3/CNC films and inhibited their coloration, and the UV-shielding efficiency correlated with the avobenzone concentration in the avobenzone/CNC films. On the other hand, the use of avobenzone as a component in composite films (avobenzone/ WO_3/CNC samples) showed only a negligible effect on the coloration of WO_3 (Fig. 3c), and thus the shielding effect of CeO_2 nanoparticles in the composite $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films was much more prominent than that of avobenzone.

The rate of non-chain photochemical processes linearly depends on intensity of absorbed light, and each UV-absorber reduces a light intensity according to the following Eq. (6):

$$I(\text{UV}) = I^0(\text{UV}) \cdot C \cdot \varepsilon \cdot l \quad (6)$$

where $I(\text{UV})$ and $I^0(\text{UV})$ – intensities of transmitted and incident UV-light; C – concentration of UV-absorber; ε – extinction coefficient of UV-shielding material; l – optical path length.

The UV-shielding CeO_2/CNC and avobenzone/CNC films had the same thickness and the optical path lengths were also the same. Thus, the intensity of UV-light which reaches the surface of the WO_3/CNC film depended only on the concentration of the UV-filter (CeO_2 or avobenzone) in the CeO_2/CNC or avobenzone/CNC films.

When avobenzone or CeO_2 were added directly in the photochromic films (avobenzone/ WO_3/CNC and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ samples, respectively) the WO_3 nanoparticles in the composites were exposed to UV-light along with avobenzone and CeO_2 . In the case of avobenzone/ WO_3/CNC films this expectedly resulted in a quite low UV-shielding effect of avobenzone. In contrast, the strong inhibition of photochromism of $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films cannot be fully explained by CeO_2 UV-absorption. Since cerium has two stable oxidation states, it can participate in additional redox reactions that affect the photochromism of WO_3 .

The FTIR spectra of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ and WO_3/CNC films demonstrated changes after UV-exposure (Fig. 3d). The absorption band of WO_3 in the WO_3/CNC film at 803 cm^{-1} faded under UV-irradiation, while another band of WO_3 at 947 cm^{-1} became more intense. The observed changes in the vibrational spectra are related to the changes in the chemical bonding in the WO_3/CNC composite due to the reduction of tungsten ions in the course of the reaction (3). For the non-irradiated $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (0.5/1/10 and 1/1/10) samples, the intensity of the IR-absorption band at 803 cm^{-1} negatively correlated with the CeO_2 content. Note that UV-irradiation caused only minor changes in the absorption band intensities at 803 and 947 cm^{-1} of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (0.5/1/10) film, and almost negligible changes in the absorption band intensities of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) film. These results indicated that CeO_2 and WO_3 nanoparticles in the composite films are able to interact chemically, preventing the coloration of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films. Such an interaction between CeO_2 and WO_3 nanoparticles may provide effective electron transfer from excited tungsten trioxide to CeO_2 thus leading to the partial reduction of cerium ions instead of the reduction of tungsten from colorless W^{+6} to blue W^{+5} (reaction (3)).

Blue WO_3/CNC films became colorless when oxidized with atmospheric oxygen in the dark by reaction (5). Discoloration kinetic curves of the WO_3/CNC (1/10) and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (0.1, 0.5, and 1/1/10) films are presented in Fig. 4a. The discoloration kinetic of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) film cannot be compared to those of the other samples due to the very low intensity of the band at 620 nm. Comparison of the discoloration kinetics of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (0.1/1/10 and 0.5/1/10) films showed that the higher CeO_2 content promotes faster discoloration process. The oxidation of tungsten ions by CeO_2 by the reaction (7) may facilitate the discoloration of WO_3 :



Effect of WO_3 on the photostability of the composite films

Metal oxide nanoparticles are photostable, unlike organic photoprotectors such as avobenzone⁵⁹. However, due to their photocatalytic properties, they can affect the photodegradation of the polymer matrix in which they are embedded.

The prolonged UV-irradiation of the CeO_2/CNC film resulted in the growth of a new band at 1570 cm^{-1} in the FTIR spectrum (Fig. 4b), which corresponds to asymmetric vibrations of $-\text{COO}^-$ carboxylic groups. This indicates that the photo-induced oxidation of CNC can proceed in the presence of CeO_2 nanoparticles. In some cases, nanoscale CeO_2 can show catalytic effects and can exhibit prooxidant properties which are often regarded as oxidoreductase-like activity^{60,61}. CeO_2 also caused oxidation of the chelating molecules under light exposure, as was observed for dextran coated and citrate coated cerium oxide nanoparticles^{20,62}. In the CeO_2/CNC films the hydroxyl groups of cellulose coordinated with cerium ions possibly undergo oxidation under UV-light.

On the contrary, the FTIR spectra of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films didn't change significantly after even prolonged UV-exposure (over 7 h) and the band inherent in asymmetric vibrations of $-\text{COO}^-$ carboxylic groups was not detected (Fig. 4b). This observation confirms the protective effect of WO_3 nanoparticles against possible cellulose oxidation in the presence of CeO_2 . Photochromic WO_3 sol can show photoreductive properties and act as photoprotector in photooxidation processes, as it was reported elsewhere⁶³. Thus, the effect of WO_3 on the photostability of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ composite may involve photoreduction mechanisms that prevents oxidation of cellulose under UV-irradiation.

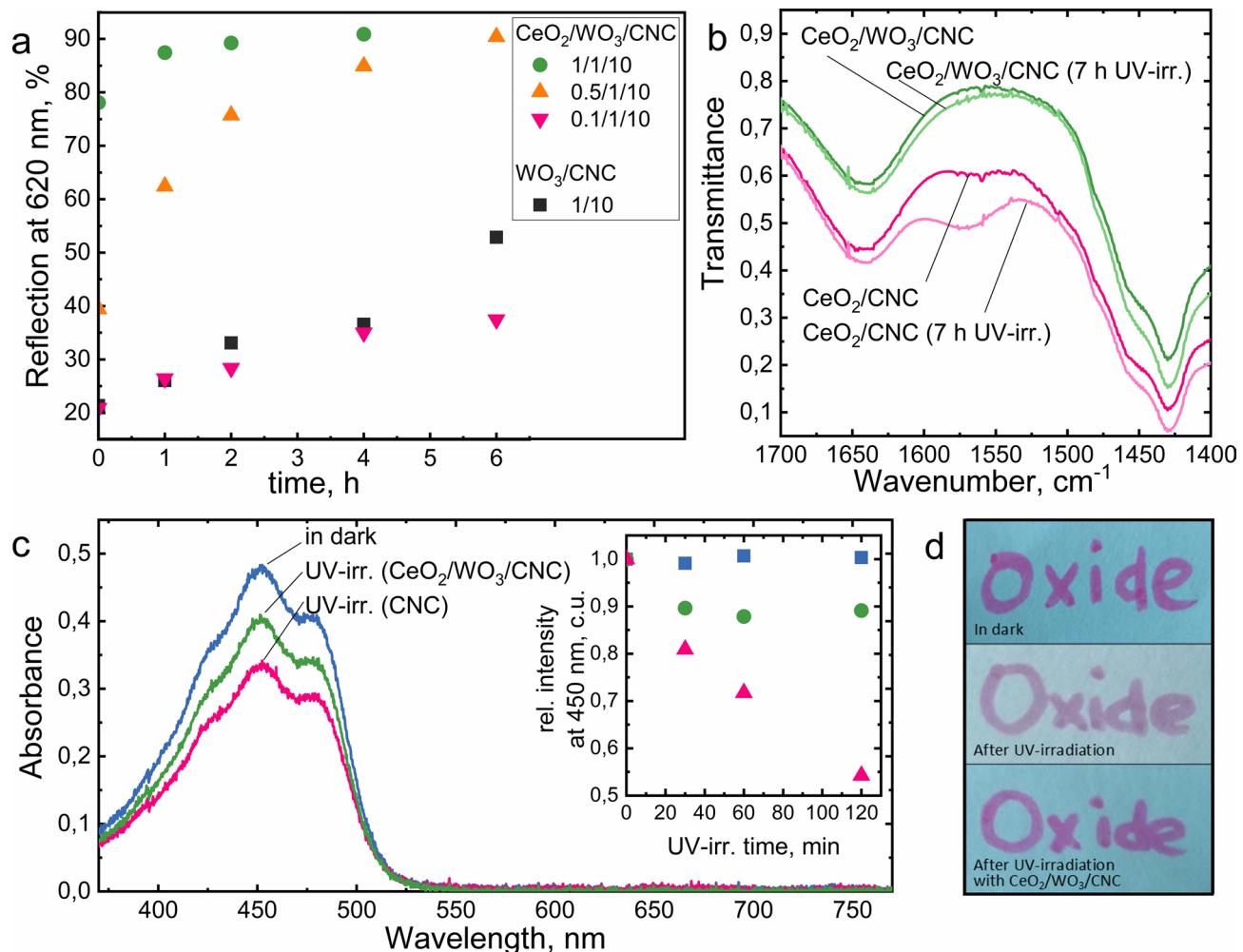


Fig. 4. (a) The discoloration kinetic of the WO_3/CNC (1/10) film and the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films with different component ratios, as measured using intensities of an absorption band at 620 nm in the diffuse reflectance spectra. (b) The fragments of FTIR spectra of the CeO_2/CNC (1/10) and $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/10) films before and after 7 h UV-exposure. (c) The effect of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ and CNC shielding films on the UV-photodegradation of beta-carotene. (d) The effect of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ and CNC protective films on the UV-photodegradation of dyes used in office supplies.

UV-shielding property of the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ composite films

All the prepared $\text{CeO}_2/\text{WO}_3/\text{CNC}$ films demonstrated strong absorption in UV range, while the photochromic properties depended on the CeO_2 to WO_3 ratio. The film with the highest cerium dioxide content $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) showed the least photochromic changes in comparison with the other films and remained nearly colorless when exposed to UV radiation. As the transparency in visible range and no color changes is preferable for UV-shielding materials, the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) has the most preferable composition among others.

The ultraviolet protection factor (UPF) and percentages of UVA and UVB radiation blocking for the $\text{CeO}_2/\text{WO}_3/\text{CNC}$ (1/1/10) film were calculated using equations 8–10^{64,65}:

$$\text{UVA blocking} (\%) = 100 - \frac{100 \int_{320}^{400} T(\lambda) d\lambda}{\int_{320}^{400} d\lambda} \quad (8)$$

$$\text{UVB blocking} (\%) = 100 - \frac{100 \int_{320}^{400} T(\lambda) d\lambda}{\int_{320}^{400} d\lambda} \quad (9)$$

$$\text{UPF (UVA)} = \frac{\int_{320}^{400} E(\lambda) S(\lambda) d\lambda}{\int_{320}^{400} E(\lambda) S(\lambda) T(\lambda) d\lambda} \quad (10)$$

where $E(\lambda)$ is the relative erythema spectral effectiveness, $S(\lambda)$ is the spectral irradiance ($\text{W}/\text{m}^2 \text{ nm}^{-1}$), $T(\lambda)$ is the experimental spectral transmittance of the sample, and λ is wavelength. The values of $E(\lambda)$ and $S(\lambda)$ were obtained from the National Oceanic and Atmospheric Administration database (NOAA). Transmittance spectra

Sample	UVA blocking	UVB blocking	UPF (UVA)
CNC	68 ± 1	90 ± 6	3.4 ± 0.1
CeO ₂ /WO ₃ /CNC (1/1/10)	99 ± 0.1	100 ± 0.001	450 ± 33

Table 2. The ultraviolet protection factor (UPF) and percentages of UVA and UVB radiation blocking.

were measured on UV-vis spectrometer using air as a reference. The sample was rotated for 90° each time, resulting in four spectra.

The CeO₂/WO₃/CNC (1/1/10) demonstrated UVA and UVB shielding of 99% and 100%, respectively (Table 2). UPF is typically used to characterize UV-protective textiles, with values above 50 being considered excellent. While the UPF in the CNC film was only 3.4 in the 320–400 nm range, it was about 450 for the CeO₂/WO₃/CNC (1/1/10) composite film, which demonstrates the excellent UV-protection properties of the material.

The UV shielding properties of the synthesized films were tested with respect to a solution of beta-carotene—the well-known antioxidant^{66,67}. Cuvettes containing β-carotene solutions in heptane were covered with the CNC or CeO₂/WO₃/CNC (1/1/10) films and further exposed to UV light with a wavelength maximum at 312 nm. The reference sample of beta-carotene was stored in the dark. Upon the UV-irradiation of β-carotene sample covered with unmodified CNC film, its optical absorption band at 450 nm rapidly faded and its intensity showed a twofold decrease within 2 h. In turn, CeO₂/WO₃/CNC (1/1/10) composite film effectively prevented the bleaching of β-carotene, and only a 10% decrease in the intensity of the absorption band was observed after 2 h of UV-irradiation (Fig. 4c). The UV-shielding properties of the CeO₂/WO₃/CNC (1/1/10) composite were also demonstrated for the common office notepaper sheets (Fig. 4d).

Conclusions

The CeO₂/CNC, WO₃/CNC, and CeO₂/WO₃/CNC composite films were produced from nanocrystalline cellulose and metal oxide sols containing no additional organic stabilizers by the solvent casting method. The UV-shielding and photochromic properties of the films were studied.

The WO₃/CNC films turned blue under UV-irradiation with wavelengths below 390 nm due to the reduction of W⁶⁺ to W⁵⁺ and W⁴⁺ and became colorless upon exposure to atmospheric oxygen in the dark, demonstrating reversible photochromic properties. The addition of CeO₂ nanoparticles inhibited the photochromic properties of the CeO₂/WO₃/CNC films in a concentration-dependent manner. In addition to the absorption of UV light by CeO₂, its redox interactions with WO₃ can impact the inhibition of the photochromic property of the composites.

The CeO₂/CNC and CeO₂/WO₃/CNC films demonstrated strong UV-shielding properties, and protected both photochromic films and photodegradable organic dyes from the colour changes. The UV-exposure of the CeO₂/CNC films caused photodegradation of CNC, while the addition of WO₃ nanoparticles protected CNC in the CeO₂/WO₃/CNC composites.

Therefore, the mutual influence of CeO₂ and WO₃ nanoparticles in CeO₂/WO₃/CNC composite films was studied for the first time, and the transparent UV-shielding CeO₂/WO₃/CNC films with enhanced photostability were produced. The photochromic properties of these films can easily be tuned by the CeO₂ to WO₃ ratio change. The CeO₂/WO₃/CNC films can be useful for various applications, including protection of photodegradable dyes, paper documents, and production of UV-shielding packages.

Data availability

Data analyzed during the current study are available from the corresponding author upon request.

Received: 15 September 2025; Accepted: 9 December 2025

Published online: 24 December 2025

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Acknowledgements

The research was supported by the Russian Science Foundation (project 24-13-00370). The experiments were performed using the equipment of the Joint Research Centre of IGIC RAS. We would like to acknowledge Dr. Alexander B. Shcherbakov and Prof. Sergey V. Ryazantsev for the valuable discussions they provided during the planning and development of this work.

Author contributions

S.K., A.B., and V.I. contributed to the study conception and design. Material preparation, data collection and analysis were performed by S.K., M.P., T.K., A.L., A.Y. The first draft of the manuscript was written by S.K. and M.P. All the authors commented on the previous versions of the manuscript. All the authors have given approval to the final version of the manuscript.

Funding

The research was supported by the Russian Science Foundation (project 24-13-00370).

Declarations

Competing interests

The authors declare no competing interests.

Additional information

Supplementary Information The online version contains supplementary material available at <https://doi.org/10.1038/s41598-025-32327-w>.

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